



# RPPR Final Report

## as of 31-Aug-2021

Agency Code: 21XD

Proposal Number: 70871MSYIP

Agreement Number: W911NF-17-1-0398

### INVESTIGATOR(S):

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Country: USA

DUNS Number: 041027822

EIN: 020222111

**Report Date:** 14-Jun-2021

Date Received: 31-Aug-2021

**Final Report** for Period Beginning 15-Sep-2017 and Ending 14-Mar-2021

**Title:** Magneto-electronic Sensors for Gasotransmitters

**Begin Performance Period:** 15-Sep-2017

**End Performance Period:** 14-Mar-2021

**Report Term:** 0-Other

Submitted By: Katherine Mirica

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Phone: (603) 646-8188

**Distribution Statement:** 1-Approved for public release; distribution is unlimited.

**STEM Degrees:** 2

**STEM Participants:**

**Major Goals:** This proposal focuses on addressing the analytical challenge of detecting gasotransmitters. The experimental approach relies on developing new materials and a new paradigm in signal transduction and amplification in solid-state chemical sensors that takes advantage of two fundamental properties of electrons: charge and spin. The molecular design towards this goal employs bottom-up synthesis of multifunctional responsive two-dimensional (2D) nanomaterials capable of changing their electrical resistance in response to a chemical stimulus due to perturbations of both electronic and magnetic properties.

The research in the proposal is organized into three major goals. Each major goal is subdivided into a set of sub-goals, which are then executed using a discrete set of tasks.

Major Goal 1: Molecular design and synthesis of magneto-electronic materials having stimuli-responsive magnetic and electronic properties.

#### 1.1. Molecular Design of Receptors

1.1.1. Computational assessment of analyte-MOF interactions using molecular analog of metal bis(dioxolene) receptors. Milestone: Computed minimized structures, binding affinities, and spin densities for molecular MOF analogs interacting with analytes. Progress: 100 % complete (Fig. 1-2).

1.1.2. Computational assessment of analyte-COF interactions using molecular analog of metal phthalocyanine (MPc) receptors. Milestone: Computed minimized structures, binding affinities, and spin densities for MPc units interacting with analytes. Progress: 100 % complete (Fig. 3).

#### 1.2. Molecular Design of Stimuli-Responsive Materials

1.2.1. Computational examination of the influence of metal and ligand on band gap of MOF monolayers. Milestone: Computed minimized structures, band structures, and band gaps for MOF monolayers. Progress: 100 % complete (Fig. 4).

1.2.2. Computational assessment of the influence of MPc subunit and covalent linkers on the band gap of COFs monolayers. Milestone: Computed minimized structures, band structures, and band gaps for COF monolayers. Progress: 100 % complete. (Fig. 5-6).

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### 1.3. Design and Synthesis of Stimuli-Responsive MOFs

1.3.1. Optimization of morphological control for producing free-standing MOF nanowires and nanosheets.

Milestone: Publication quality and statistically significant SEM images and X-Ray diffractograms characterizing MOF nanowires and nanosheets. Progress: 100 % complete (Fig. 7-8).

1.3.2. Computational assessment of the influence of metal and ligand on the anisotropy of charge transport in MOF nanorods. Milestone: Modeled crystalline substructures of different MOF analogs in Material Studio to compute electron transport through-stack and through-sheet. Progress: 100 %. (Fig. 9)

1.3.3. Computational assessment of the influence of metal and ligand on the anisotropy of charge transport in MOF nanosheets. Milestone: Modeled crystalline substructures of different MOF analogs in Material Studio to compute electron transport through-stack and through-sheet. Progress: 100 %. (Fig. 9)

### 1.4. Design and Synthesis of Stimuli-Responsive COFs

1.4.1. Synthesis and characterization of MPC-based molecular precursors to COFs. Milestone: Optimized synthetic access and rigorous characterization using the combination of NMR, IR, Mass Spectrometry, and Elemental Analysis. Progress: 100 % complete (Fig. 10-12).

1.4.2. Synthesis and characterization of MPC-based COFs. Milestone: Optimized synthetic access and rigorous materials characterization using pXRD, BET, SEM, and TEM. Progress: 100 % complete (Fig. 13-14).

1.4.3. Optimization of morphological control within MPC-based COFs. Publication quality and statistically significant SEM images and X-Ray diffractograms characterizing COF morphology as a function of synthetic conditions. Progress: 100 % complete (Fig. 15).

Major Goal 2: Studies of structure-property relationships in these materials in solid-state devices.

### 2.1. Studies of Electrical Properties of MOFs and COFs

2.1.1. Measurement of conductivity of MOFs having highly tuned morphological control. Milestone: Measured conductivity of MOF nanowires and nanosheets. Progress: 100% Complete (Fig. 16).

2.1.2. Examination of the impact of the electrodes on device performance in MOFs. Milestone: Measurements of the influence of the electrode type upon sensing performance of MOF in response to target analytes. Progress: 100% complete (Fig. 17).

2.1.3. Measurement of conductivity of COFs having tuned morphological control. Measured conductivity of COF nanocrystals. Progress: 100 % complete (Fig. 18).

2.1.4. Examination of the impact of the electrodes on device performance in COFs. The progress on this task was disrupted due to COVID-19 pandemic.

### 2.2. Studies of Magnetic Properties of MOFs and COFs

2.2.1. EPR studies of analyte binding in MOFs. Milestone: Measured perturbations to the concentration and chemical environment of unpaired spins in conductive MOFs as they bind gaseous analytes using electron paramagnetic spectroscopy and X-ray photoelectron spectroscopy. Progress: 100 % complete (Fig. 19-23).

2.2.2. EPR studies of analyte binding in COFs. Measured perturbations to the concentration and chemical environment of unpaired spins in conductive COFs as they bind gaseous analytes electron paramagnetic spectroscopy. Progress: 100 % complete (Fig. 24-25).

2.2.3. SQUID magnetometry of MOFs. Milestone: Characterize magnetic ordering in MOFs. Progress: 100 % complete (Fig. 26).

2.2.4. SQUID magnetometry of COFs. Milestone: Characterize magnetic ordering in COFs. Due to COVID-19 pandemic, magnetic properties were assessed computationally Progress: 100 % complete (Fig. 27).

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Major Goal 3: Magneto-electronic signal amplification and selective detection of CO, NO, and H<sub>2</sub>S using solid state devices.

### 3.1. Studies of Coupled Conductivity and Magnetism in MOFs and COFs.

3.1.1. Measurements of microwave conductivity in MOFs. Milestone: Measured microwave conductivity in MOFs in the presence and absence of analytes. Due to travel restrictions, necessary instrumentation was not accessible to complete this task as anticipated. (Table 1)

3.1.2. Measurements of microwave conductivity in COFs. Milestone: Measured microwave conductivity in COFs in the presence and absence of analytes. Due to travel restrictions, necessary instrumentation was not accessible to complete this task as originally anticipated. (Table 2)

### 3.2. Coupling Chemiresistance with Magnetoresistance

3.2.1. Fabrication of Hall Bar devices with MOFs. Milestone: Reliable fabrication of Hall Bar devices with embedded MOFs. Progress: 20 % complete (Fig. 28-30).

3.2.2. Measurements of charge mobility of MOFs. Milestone: Optimized measurements of charge mobility in MOFs. Progress: 50% complete (Fig. 28, 31).

3.2.3. Measurements of field effect in MOFs. Milestone: Optimized measurements of electric and magnetic field effect in MOFs. Progress: 50% complete (Fig. 28-29).

3.2.4. Measurements of the perturbations of charge mobility and field effect upon CO, NO, and H<sub>2</sub>S binding to MOFs. Milestone: Optimized measurements of the perturbations to the charge mobility and field effects in MOFs upon analyte exposure. Progress: 100 % complete. (Fig. 32-33)

3.2.5. Fabrication of Hall Bar devices with COFs. Milestone: Reliable fabrication of Hall Bar devices with embedded COFs. This task was disrupted due to COVID-19 restrictions.

3.2.6. Measurements of charge mobility of COFs. Milestone: Optimized measurements of charge mobility in COFs. Progress: 20% complete. (Fig. 34)

3.2.7. Measurements of field effect in COFs. Milestone: Optimized measurements of electric and magnetic field effect in COFs. Progress: 20% complete. (Fig. 34). Progress on this task was disrupted due to COVID-19 restrictions.

3.2.8. Measurements of the perturbations of charge mobility and field effect upon CO, NO, and H<sub>2</sub>S binding to COFs. Milestone: Optimized measurements of the perturbations of analytes to the charge mobility. Progress: 100% complete. (Fig 35-37).

**Accomplishments:** please see attached pdf document

## RPPR Final Report as of 31-Aug-2021

**Training Opportunities:** Training opportunities supported by this grant involved professional development of graduate research assistants in the form of research mentoring and presentations. Specific highlights include:

Trainee Name: Robert M. Stolz  
Title of event: Institute seminar, Frontier Institute for Research in Sensor Technologies (FIRST), University of Maine  
Role: Presenter  
Presentation Title: Host–Guest Interactions and Redox Activity in Layered Conductive Metal–Organic Frameworks  
Date: 07/31/2020

Trainee Name: Robert M. Stolz  
Title of event: American Chemical Society Spring National Meeting, 2020, San Francisco CA  
Role: Presenter  
Presentation Title: Understanding the Surface Chemistry of 2D Conductive MOFs: insight into the role of the interface in electrically-transduced chemical sensing  
Date: 08/17/2020–08/20/2020

Trainee Name: Robert M. Stolz  
Title of event: Frühjahrssymposium, 2021, Leipzig, Germany  
Your role: Presenter and member of the American Delegation  
Presentation Title: Mechanisms of Electroanalysis with Conductive Metal–Organic Frameworks  
Date: 03/29/2021–04/01/2021

Trainee Name: Robert M. Stolz  
Title of event: Northeastern Section Younger Chemists Committee (Northeastern Section of the American Chemical Society), 2021, Boston, MA  
Role: Presenter (poster)  
Presentation Title: Mechanisms of Electroanalysis with Conductive Metal–Organic Frameworks  
Date: 04/24/2021

Trainee Name: Robert M. Stolz  
Title of event: JACS in Session with ACS Applied Materials & Interfaces: webinar series, 2021  
Role: Attendee  
Presentation Title: N/A  
Date: 06/08/2021

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**Results Dissemination:** The progress on this project has been disseminated in 13 peer reviewed publications, since the start of this award:

1. Ko, M.; Aykanat, A.; Smith, M. K.; Mirica, K. A.\* "Drawing Sensors with Ball-Milled Blends of Metal-Organic Frameworks and Graphite" *Sensors* 2017, 17, 2192. Invited paper for a special issue "Chemiresistive Sensors: Status and the Future" co-edited with Prof. Timothy M. Swager.
2. Smith, M.; Mirica, K. A.\* "Self-Organized Frameworks on Textiles (SOFT): Conductive Fabrics for Simultaneous Sensing, Capture, and Filtration of Gases" *J. Am. Chem. Soc.* 2017, 139, 16759–16767.
3. Mendecki, L.; Mirica, K. A.\* "Conductive Metal–Organic Frameworks as Ion-to-Electron Transducers in Potentiometric Sensors" *ACS Appl. Mater. Interfaces* 2018, 10, 19248–19257.
4. Ko, M.; Mendecki, L.; Mirica, K. A.\* "Conductive two-dimensional metal–organic frameworks as multifunctional materials" *Chem. Commun.* 2018, 54, 7873-7891.
5. Meng, Z.; Stolz, R. M.; Mendecki, L.; Mirica, K. A.\* "Electrically-Transduced Chemical Sensors Based on Two-Dimensional Nanomaterials" *Chem. Rev.* 2019, 119, 478–598.
6. Meng, Z.; Aykanat, A.; Mirica, K. A.\* "Proton Conduction in 2D Aza-Fused Covalent Organic Frameworks" *Chem. Mater.* 2019, 31, 819–825.
7. Meng, Z.; Aykanat, A.; Mirica, K. A.\* "Welding Metallophthalocyanines into Bimetallic Molecular Meshes for Ultrasensitive, Low-Power Chemiresistive Detection of Gases" *J. Am. Chem. Soc.* 2019, 141, 2046–2053.
8. Meng, Z.; Stolz, R.; Mirica, K. A.\* "Two-Dimensional Chemiresistive Covalent Organic Framework with High Intrinsic Conductivity" *J. Am. Chem. Soc.* 2019, 141, 11929–11937.
9. Aykanat, A.; Meng, Z.; Bendetto, G.; Mirica, K. A.\* "Molecular Engineering of Multifunctional Metallophthalocyanine-Containing Framework Materials" *Chem. Mater.* 2020, 32, 5372–5409.
10. Ko, M.; Mendecki, L.; Eagleton, A.; Durbin, C. G.; Stolz, R. M.; Meng, Z.; Mirica, K. A.\* "Employing Conductive Metal–Organic Frameworks for Voltammetric Detection of Neurochemicals" *J. Am. Chem. Soc.* 2020, 142, 11717–11733
11. Stolz, R. M.; Mahdavi-Shakib, A.; Frederick, B. G.\*; Mirica, K. A.\* "Host–Guest Interactions and Redox Activity in Layered Conductive Metal–Organic Frameworks" *Chem. Mater.* 2020, 32, 7639–7652.
12. Meng, Z.; Luo, J.; Li, W.\*; Mirica, K. A.\* "Hierarchical Tuning of the Performance of Electrochemical Carbon Dioxide Reduction Using Conductive Two-Dimensional Metallophthalocyanine Based Metal–Organic Frameworks" *J. Am. Chem. Soc.* 2020, 142, 21656–21669.
13. Meng, Z.; Mirica, K. A.\* "Two-Dimensional d-pi Conjugated Metal-Organic Framework Based on Hexahydroxytrinaphthylene" *Nano Research* 2021, 14, 369–375.

Additionally, the research has been disseminated in 52 research presentations presented by PI.

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**Honors and Awards:** Awards to Katherine Mirica (PI) during 2017-2021:

Karen E. Wetterhahn Memorial Award for Distinguished Creative or Scholarly Achievement (2021)  
Gordon Russell 1955 Fellowship, Dartmouth College (2021)  
NIH Maximizing Investigators' Research Award (2020)  
Camille Dreyfus Teacher Scholar Award (2020)  
NSF CAREER Award (2020)  
Cottrell Scholar Award (2019)  
Alfred P. Sloan Research Fellowship in Chemistry (2018)  
3M Non-Tenured Faculty Award (2018)  
American Chemical Society PMSE Young Investigator Award (2018)  
Thieme Chemistry Journal Award (2018)  
Enhanced Junior Faculty Fellowship, Dartmouth College (2018)

### Protocol Activity Status:

**Technology Transfer:** 1. monthly update call with Todd Miller and Natalie Pomerantz at Natick Soldier Center (Natick, MA)  
2. collaborative discussions with Caitlyn Callahan at CRREL about other projects (Hanover, NH)  
3. collaborative discussions and sample exchange with Gregory Peterson regarding testing metal-organic frameworks on textiles for another project (Aberdeen Proving Ground, MD)

### PARTICIPANTS:

**Participant Type:** PD/PI

**Participant:** Katherine Mirica

**Person Months Worked:** 6.00

Project Contribution:

National Academy Member: N

**Funding Support:**

**Participant Type:** Graduate Student (research assistant)

**Participant:** Robert Stolz

**Person Months Worked:** 15.00

Project Contribution:

National Academy Member: N

**Funding Support:**

**Participant Type:** Graduate Student (research assistant)

**Participant:** Michael Ko

**Person Months Worked:** 15.00

Project Contribution:

National Academy Member: N

**Funding Support:**

**Participant Type:** High School Student

**Participant:** Katherine MacVeagh

**Person Months Worked:** 3.00

Project Contribution:

National Academy Member: N

**Funding Support:**

**Participant Type:** Undergraduate Student

**Participant:** Adelaide Levenson





## RPPR Final Report as of 31-Aug-2021

**Publication Type:** Journal Article      Peer Reviewed: Y      **Publication Status:** 1-Published

**Journal:** Chemistry of Materials

Publication Identifier Type: DOI

Publication Identifier: 10.1021/acs.chemmater.8b03897

Volume:                      Issue:

First Page #:

Date Submitted: 1/25/19 12:00AM

Date Published: 1/7/19 5:00AM

Publication Location:

**Article Title:** Proton Conduction in 2D Aza-Fused Covalent Organic Frameworks

**Authors:** Zheng Meng, Aylin Aykanat, Katherine A. Mirica

**Keywords:** proton transport; covalent organic frameworks

**Abstract:** This paper describes the first demonstration of a general concept for achieving facile proton conduction within a class of layered two-dimensional aza-fused  $\pi$ -conjugated covalent organic frameworks (COFs). The built-in phenanthroline moieties promote efficient adsorption of water and acidification by H<sub>3</sub>PO<sub>4</sub> to produce excellent proton conductivities of 10<sup>(-5)</sup> and 10<sup>(-3)</sup> S/cm for the pristine and acidified analogues, respectively. These molecular design concepts are poised for development of highly stable proton-conducting materials.

**Distribution Statement:** 3-Distribution authorized to U.S. Government Agencies and their contractors

Acknowledged Federal Support: Y

**Publication Type:** Journal Article      Peer Reviewed: Y      **Publication Status:** 1-Published

**Journal:** Chemical Reviews

Publication Identifier Type: DOI

Publication Identifier: 10.1021/acs.chemrev.8b00311

Volume: 119                      Issue: 1

First Page #: 478

Date Submitted: 1/25/19 12:00AM

Date Published: 1/3/19 5:00AM

Publication Location:

**Article Title:** Electrically-Transduced Chemical Sensors Based on Two-Dimensional Nanomaterials

**Authors:** Zheng Meng, Robert M. Stolz, Lukasz Mendecki, Katherine A. Mirica

**Keywords:** chemical sensors; 2D nanomaterials

**Abstract:** Electrically-transduced sensors, with their simplicity and compatibility with standard electronic technologies, produce signals that can be efficiently acquired, processed, stored, and analyzed. Two dimensional (2D) nanomaterials, including graphene, phosphorene (BP), transition metal dichalcogenides (TMDCs), and others, have proven to be attractive for the fabrication of high-performance electrically-transduced chemical sensors due to their remarkable electronic and physical properties originating from their 2D structure. This review highlights the advances in electrically-transduced chemical sensing that rely on 2D materials. The structural components of such sensors are described, and the underlying operating principles for different types of architectures are discussed. The structural features, electronic properties, and surface chemistry of 2D nanostructures that dictate their sensing performance are reviewed. Key advances in the application of 2D materials, from both a historical

**Distribution Statement:** 3-Distribution authorized to U.S. Government Agencies and their contractors

Acknowledged Federal Support: Y

## RPPR Final Report as of 31-Aug-2021

**Publication Type:** Journal Article      Peer Reviewed: Y      **Publication Status:** 1-Published

**Journal:** Journal of the American Chemical Society

Publication Identifier Type: DOI

Publication Identifier: 10.1021/jacs.8b11257

Volume:

Issue:

First Page #:

Date Submitted: 1/25/19 12:00AM

Date Published: 1/25/19 10:00AM

Publication Location:

**Article Title:** Welding Metallophthalocyanines into Bimetallic Molecular Meshes for Ultrasensitive, Low-Power Chemiresistive Detection of Gases

**Authors:** Zheng Meng, Aylin Aykanat, Katherine A. Mirica

**Keywords:** metal-organic frameworks, gas sensors

**Abstract:** This paper describes the first demonstration of using a series of isorecticular nickel phthalocyanine- and nickel naphthalocyanine-based bimetallic conductive two-dimensional (2D) metal-organic frameworks (MOFs) as active materials in chemiresistive sensing of gases. Devices achieve exceptional sensitivity at sub-part-per-million (ppm) to part-per-billion (ppb) detection limits toward NH<sub>3</sub> (0.31–0.33 ppm), H<sub>2</sub>S (19–32 ppb), and NO (1.0–1.1 ppb) at low driving voltages (0.01–1.0 V) within 1.5 min of exposure. The devices maintain their performance in the presence of humidity (5000 ppm of H<sub>2</sub>O). The isorecticular analogs enable modular control over selectivity and sensitivity in gas sensing through different combinations of linkers and metal nodes. Electron paramagnetic resonance spectroscopy and X-ray photoelectron spectroscopy studies suggest that the chemiresistive response of the MOFs involves charge transfer interactions triggered by the analytes adsorbed on MOFs.

**Distribution Statement:** 3-Distribution authorized to U.S. Government Agencies and their contractors  
Acknowledged Federal Support: Y

**Publication Type:** Journal Article      Peer Reviewed: Y      **Publication Status:** 1-Published

**Journal:** Journal of the American Chemical Society

Publication Identifier Type: DOI

Publication Identifier: 10.1021/jacs.9b03441

Volume: 141

Issue: 30

First Page #: 11929

Date Submitted: 8/30/19 12:00AM

Date Published: 6/1/19 4:00AM

Publication Location:

**Article Title:** Two-Dimensional Chemiresistive Covalent Organic Framework with High Intrinsic Conductivity

**Authors:** Zheng Meng, Robert M. Stolz, Katherine A. Mirica

**Keywords:** covalent organic frameworks

**Abstract:** This paper describes the synthesis of a novel intrinsically conductive two-dimensional (2D) covalent organic framework (COF) through the aromatic annulation of 2,3,9,10,16,17,23,24-octa-aminophthalocyanine nickel(II) and pyrene-4,5,9,10-tetraone. The intrinsic bulk conductivity of the COF material (termed COF-DC-8) reached  $2.51 \times 10^{-3}$  S/m, and increased by 3 orders of magnitude with I<sub>2</sub> doping. Electronic calculations revealed an anisotropic band structure, with the possibility for significant contribution from out-of-plane charge-transport to the intrinsic bulk conductivity. Upon integration into chemiresistive devices, this conductive COF showed excellent responses to various reducing and oxidizing gases, including NH<sub>3</sub>, H<sub>2</sub>S, NO, and NO<sub>2</sub>, with parts-per-billion (ppb) limits of detection (for NH<sub>3</sub> = 70 ppb, for H<sub>2</sub>S = 204 ppb, for NO = 5 ppb, and for NO<sub>2</sub> = 16 ppb based on 1.5 min exposure). Electron paramagnetic resonance spectroscopy and X-ray photoelectron spectroscopy studies suggested

**Distribution Statement:** 3-Distribution authorized to U.S. Government Agencies and their contractors  
Acknowledged Federal Support: Y

### CONFERENCE PAPERS:

**Publication Type:** Conference Paper or Presentation      **Publication Status:** 1-Published

**Conference Name:** North Eastern Section of the American Chemical Society

Date Received: 29-Aug-2018

Conference Date: 05-Oct-2017

Date Published: 05-Oct-2017

Conference Location: Milford, MA

**Paper Title:** Multifunctional Materials for Chemical Sensing and Microelectronics

**Authors:** Katherine A. Mirica

Acknowledged Federal Support: Y

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**Publication Type:** Conference Paper or Presentation **Publication Status:** 0-Other  
**Conference Name:** Sigma Xi Seminar at U.S. Army Natick Soldier Research  
Date Received: 29-Aug-2018 Conference Date: 13-Nov-2017 Date Published:  
Conference Location: Natick, MA  
**Paper Title:** Multifunctional Materials for Chemical Sensing and Microelectronics  
**Authors:** Katherine A. Mirica  
Acknowledged Federal Support: **Y**

**Publication Type:** Conference Paper or Presentation **Publication Status:** 0-Other  
**Conference Name:** Advanced Materials for Energy and Bioengineering Applications (AMEBA) Symposium  
Date Received: 29-Aug-2018 Conference Date: 04-Dec-2017 Date Published: 04-Dec-2017  
Conference Location: Burlington, VT  
**Paper Title:** Stimuli-Responsive Materials for Chemical Sensing and Gas Capture  
**Authors:** Katherine A. Mirica  
Acknowledged Federal Support: **Y**

**Publication Type:** Conference Paper or Presentation **Publication Status:** 0-Other  
**Conference Name:** 4. 2nd From Carbon-Rich Molecules to Carbon-Based Materials Conference  
Date Received: 29-Aug-2018 Conference Date: 08-Jun-2018 Date Published: 08-Jun-2018  
Conference Location: Nassau, Bahamas  
**Paper Title:** Porous Scaffolds for Electronically-Transduced Chemical Sensing and Capture  
**Authors:** Katherine A. Mirica  
Acknowledged Federal Support: **Y**

**PATENTS:**

**Intellectual Property Type:** Patent Date Received: **30-Aug-2018**  
**Patent Title:** Conductive Textiles and Uses Thereof in Functional Devices  
**Patent Abstract:** This application claims priority to U.S. Provisional Patent Application No. 62/489,851, filed on Apr  
**Patent Number:** 15/962,156  
Patent Country: USA  
Application Date: 25-Apr-2018 Application Status: 2  
Date Issued:

**Intellectual Property Type:** Patent Date Received: **30-Aug-2018**  
**Patent Title:** Metal-Organic Frameworks as Ion-to-Electron Transducers and Detectors  
**Patent Abstract:** This invention describes the use of conductive metal-organic frameworks as ion-to-electron tran  
**Patent Number:** 18/46,510  
Patent Country: USA  
Application Date: 13-Aug-2018 Application Status: 2  
Date Issued:

**Intellectual Property Type:** Patent Date Received: **30-Aug-2018**  
**Patent Title:** Formation of Metal-Organic Frameworks  
**Patent Abstract:** This invention describes a new method for growing metal-organic frameworks using metallized s  
**Patent Number:** 62/681,436  
Patent Country: USA  
Application Date: 06-Jun-2018 Application Status: 1

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Date Issued:

**Intellectual Property Type:** Patent

Date Received: **30-Aug-2018**

**Patent Title:** Conductive Metal-Organic Frameworks for Voltammetric Detection of Neurochemicals

**Patent Abstract:** This invention describes the use of conductive metal-organic frameworks as working electrodes

**Patent Number:** 62/688,261

Patent Country: USA

Application Date: 21-Jun-2018

Application Status: 1

Date Issued:

**Intellectual Property Type:** Patent

Date Received: **30-Aug-2019**

**Patent Title:** Two-dimensional Stimuli-Response Covalent Organic Frameworks with High Intrinsic Conductivity

**Patent Abstract:** This paper describes the synthesis of a novel intrinsically

**Patent Number:** 62/837,118

Patent Country: USA

Application Date: 22-Apr-2019

Application Status: 1

Date Issued:

**Intellectual Property Type:** Patent

Date Received: **30-Aug-2019**

**Patent Title:** Conductive Bimetallic Metal-Organic Frameworks for the Detection of Analytes

**Patent Abstract:** This application describes the first demonstration of using a series of isoreticular nickel phthaloc

**Patent Number:** 62/719,264

Patent Country: USA

Application Date: 17-Aug-2018

Application Status: 1

Date Issued:

## Partners

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We have established a collaboration with Gregory Peterson (Aberdeen Proving Ground, MD) related to this project.

I certify that the information in the report is complete and accurate:

Signature: Katherine Mirica

Signature Date: 8/31/21 12:20PM

**1) Major activities:**

- Molecular Design of Receptors
- Molecular Design of Stimuli-Responsive Materials
- Design and Synthesis of Stimuli-Responsive MOFs
- Design and Synthesis of Stimuli-Responsive COFs
- Studies of Electrical Properties of MOFs and COFs
- Studies of Magnetic Properties of MOFs and COFs
- Coupling Chemiresistance with Magnetoresistance

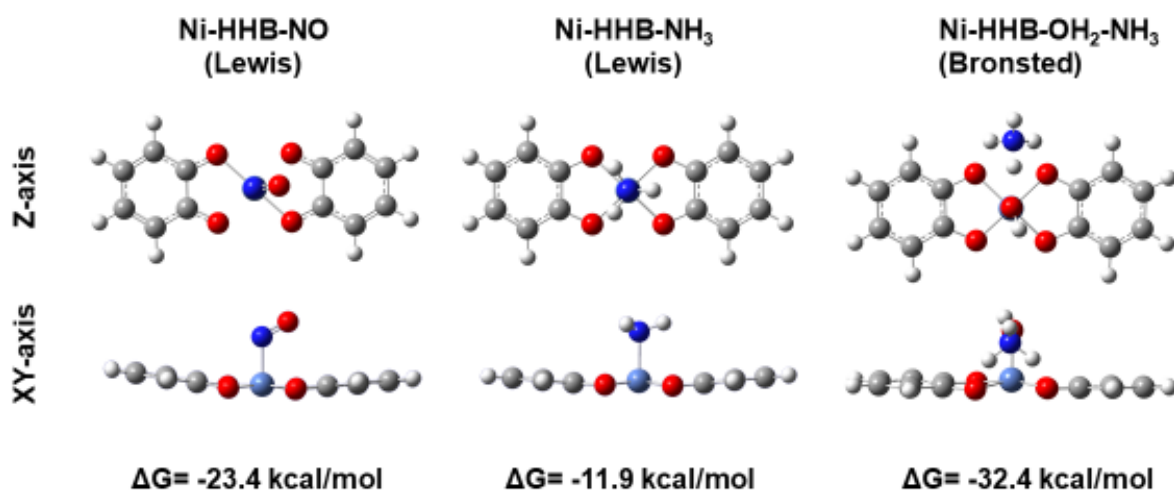
**2) Specific objectives:**

- Computational assessment of analyte-MOF and analyte-COF interactions
- Optimization of morphological control for producing free-standing MOF nanowires and nanosheets
- Synthesis and characterization of COF precursors
- Measurements of conductivity of MOFs having highly tuned morphological control
- Examination of the impact of the electrodes on device performance in MOFs
- EPR and SQUID magnetometry studies of MOFs
- Fabrication of setup for measuring magnetoresistance

**3) Significant results: see below**

**Figure 1. Progress towards Task 1.1.1.** Computational modeling of host sites reveals analyte-specific binding modes and energies. Shown are the computationally modeled interaction of Ni-bis(dioxolene) host-sites with analyte guests of NO and NH<sub>3</sub>. Ni-bis(dioxolene) binds NO favorably by 23.4 kcal/mol. The binding event does not interfere with the ligand-metal-ligand planarity. The interaction of NH<sub>3</sub> with Ni-bis(dioxolene) is favorable by 11.9 kcal/mol. Binding NH<sub>3</sub> causes the Ni to pucker out of the plane of the ligands. Hydration of the axial position of Ni-bis(dioxolene) host-sites causes it react as a Brønsted acid host site rather than a Lewis acid site. The change in acidity causes a more favorable host-guest interaction with ammonia by more than 20 kcal/mol.

**1.1.1. Computational assessment of MOF-analyte interactions using molecular analogs of metal bis(dioxolene) receptors**



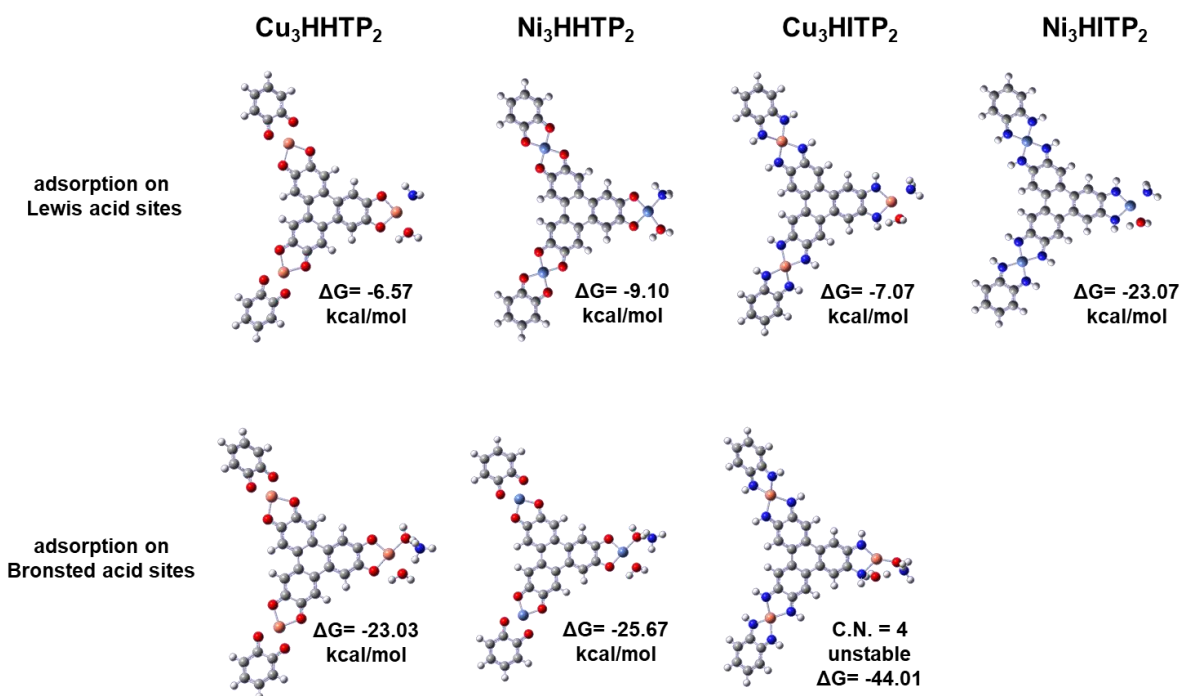
Method: UWB97XD (DFT method)

Basis set: 6-31G

All calculations performed using Gaussian09 under license to Dartmouth College

**Figure 2. Progress towards Task 1.1.1.** Computational modelling of acidic site interactions between  $\text{NH}_3$  and various MOFs demonstrates the stability of all binding modalities but a preference for adsorption to Brønsted acid interactions. The advancement in modelling full HHTP molecules instead of fractional ligands is expected to drastically improve calculation accuracy.

**1.1.1. Computational assessment of MOF-analyte edge-type interactions using molecular analogs of metal bis(dioxolene) receptors**



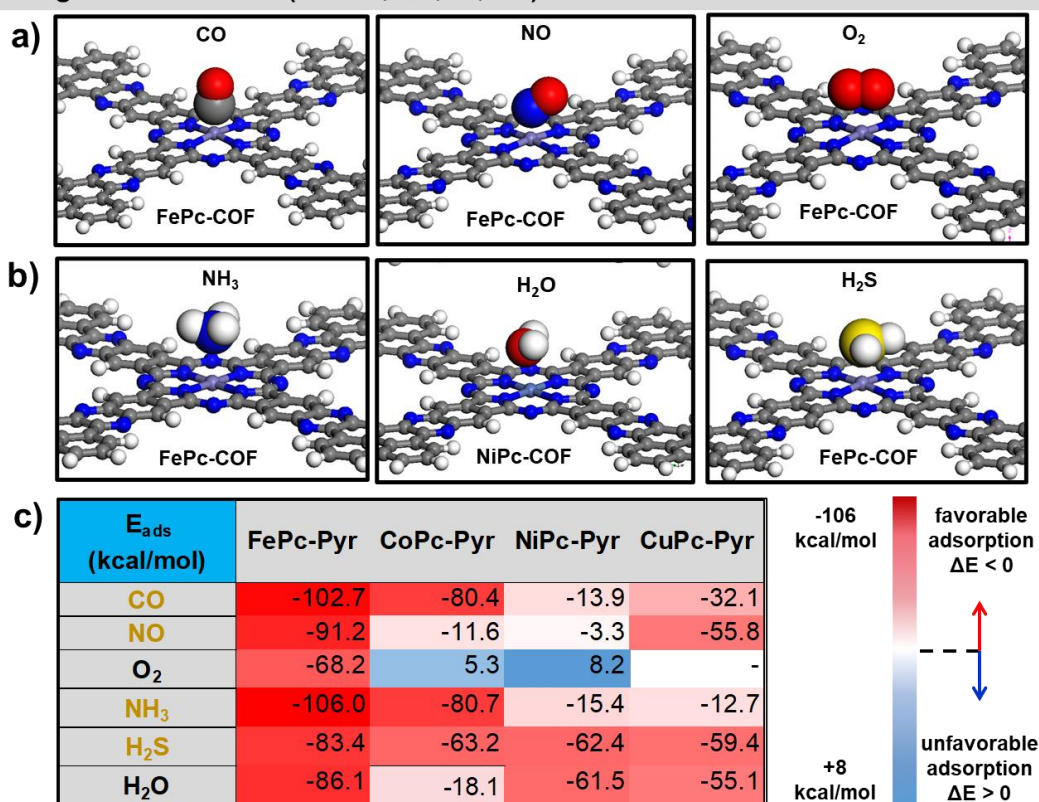
Method: UWB97XD (DFT method)

Basis set: 6-31G

All calculations performed using Gaussian09 under license to Dartmouth College

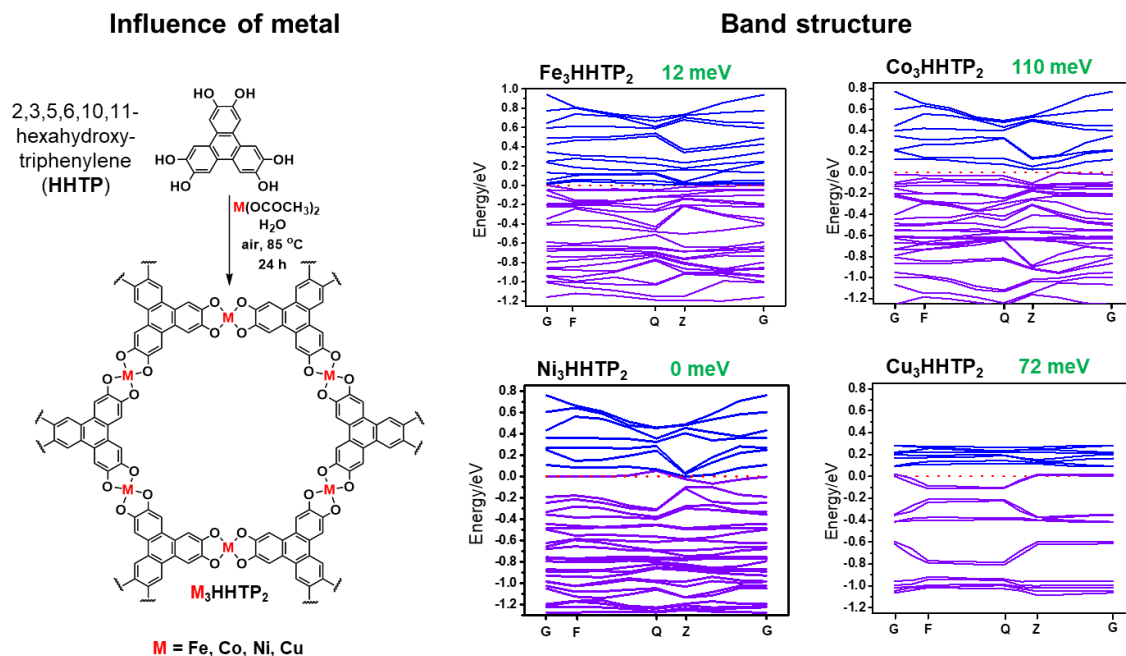
**Figure 3. Progress towards Task 1.1.2.** The Host–Guest interactions between MPc-Pyr-COFs and targeted gas analytes was examined using computational DFT techniques. The most favorable adsorption location was determined for two types of gases. Representative adsorption geometries are shown for **a)** diatomic electron withdrawing gases including NO, CO, and O<sub>2</sub>, and **b)** highly reduced electron donating gases NH<sub>3</sub>, H<sub>2</sub>O, and H<sub>2</sub>S. The most favorable host site was determined for each MPc-Pyr COF + gas pair to be the vacant axial coordination site of the MPc metal. The table **c)** shows calculated adsorption energy of MPc-Pyr-COF (M=Fe, Co, Ni, Cu) and gases (NO, CO, O<sub>2</sub>, NH<sub>3</sub>, H<sub>2</sub>O, and H<sub>2</sub>S). Adsorption sites were found using the Adsorption-Anneal locator in Materials Studio 2019. The most favorable adsorption location determined by Adsorption-Anneal was then optimized using DMol3 as part of Materials Studio 2019. Optimized geometries, single point energies, and physical/electronic parameters were modelled using the M11-L method and a DNP basis set.

### 1.1.2 Computational assessment of Host-Guest interactions MPc-Pyr-COFs and gasotransmitters (M = Fe, Co, Ni, Cu)



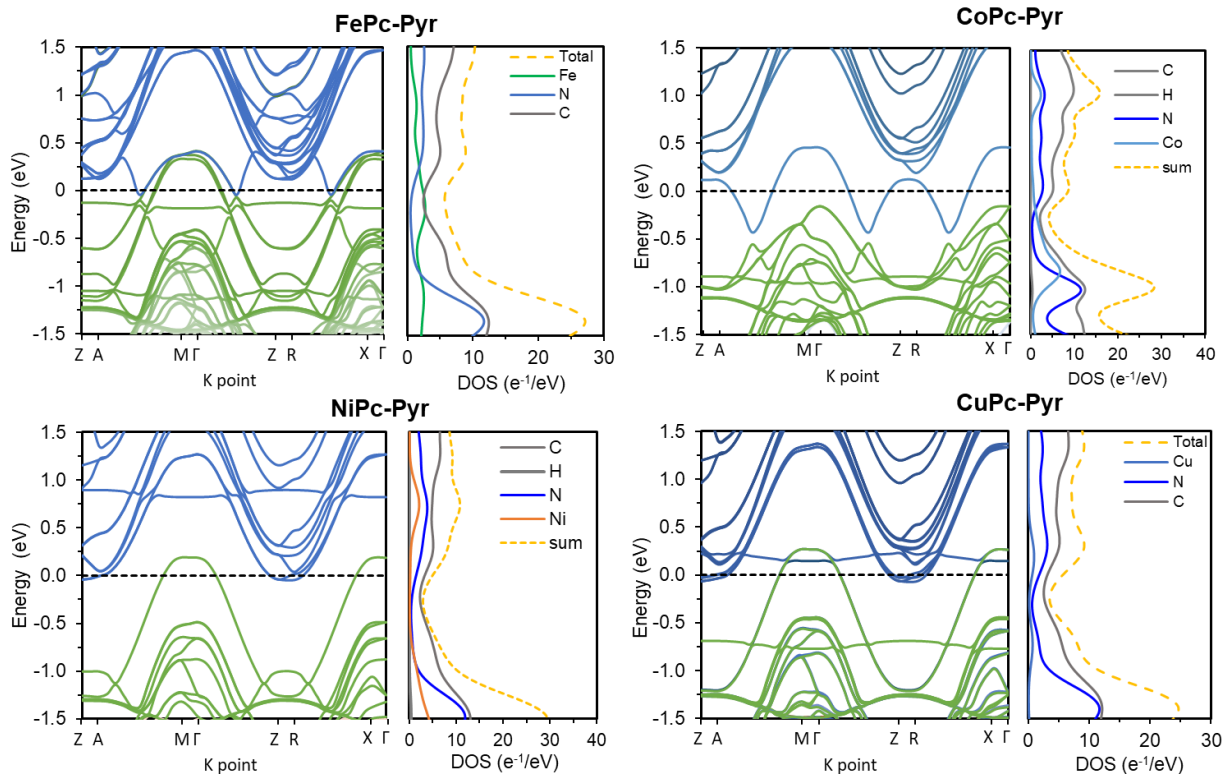
**Figure 4. Progress towards Task 1.2.1.** Synthetic scheme implemented for generating the proposed  $M_3\text{HHTP}_2$  MOFs ( $M=\text{Fe, Co, Ni, Cu}$ ). Electronic band structure for (A)  $\text{Fe}_3(\text{HHTP})_2$ , (B)  $\text{Co}_3(\text{HHTP})_2$ , (C)  $\text{Ni}_3(\text{HHTP})_2$ , and (D)  $\text{Cu}_3(\text{HHTP})_2$  calculated by CASTEP imbedded in Materials Studio 2017 (cutoff energy: 300-500 eV; k-point mesh:  $2\times 2\times 3$ ).

### 1.2.1. Computational examination of the influence of metal and ligand on band gap of MOFs monolayers



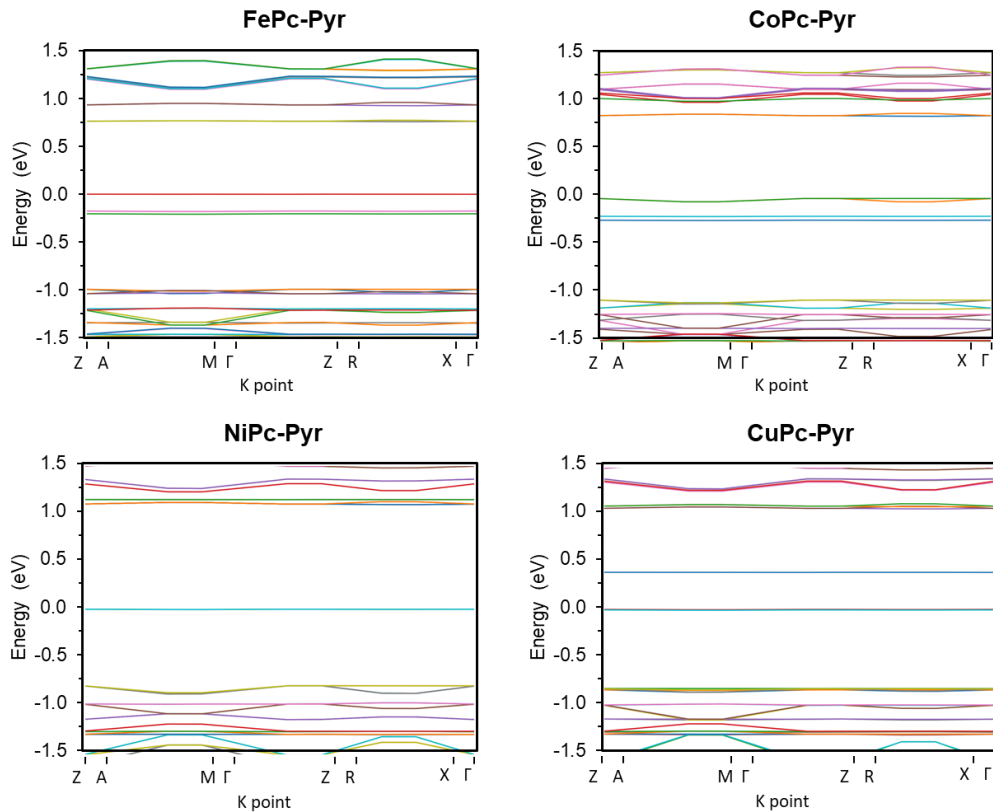
**Figure 5. Progress towards Task 1.2.2.** Calculated electronic band structure (left side) and density of state (right side) for bulk MPc-Pyr COFs (M=Fe, Co, Ni, Cu) having an eclipsed stacking structure. Geometry optimizations and property modelling was done using CASTEP as part of materials studio 2019 (cutoff energy: 300-500 eV; k-point mesh: 2×2×3). The GGA functional PBE was used. A Hubbard correction of 3.0 eV was used for all metal ions.

**1.2.2. Computational examination of the influence of metal and ligand on band gap of MPc-COFs (M = Fe, Co, Ni, Cu)**



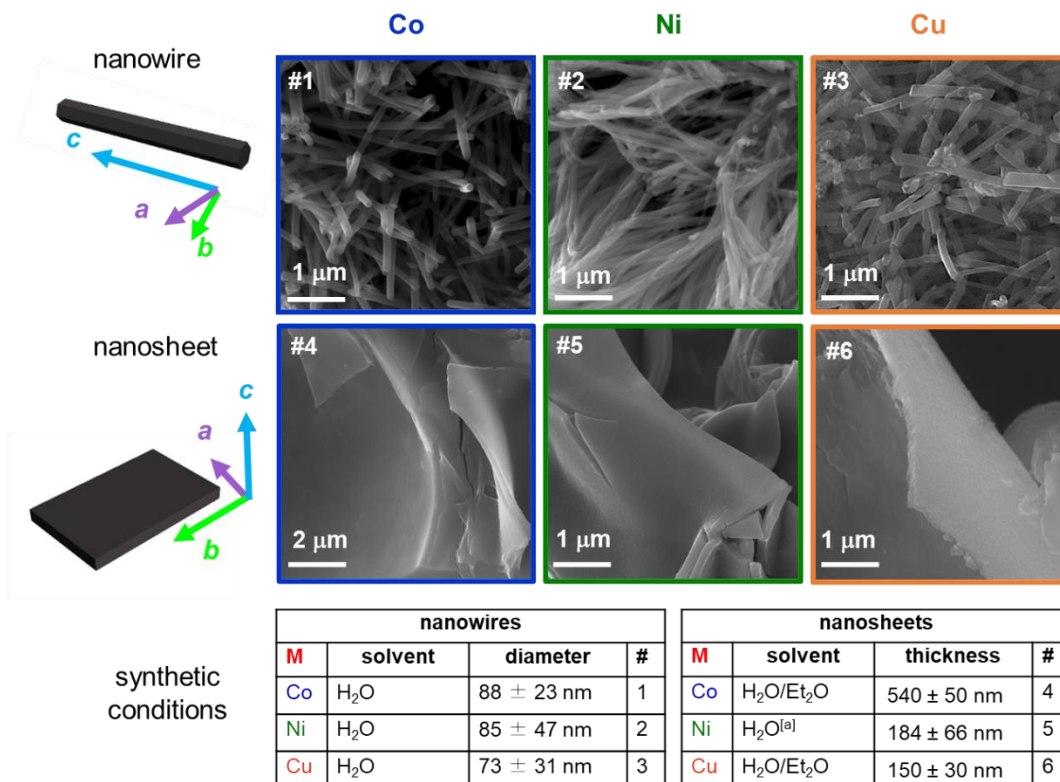
**Figure 6. Progress towards Task 1.2.2.a** Calculated electronic band structure for monolayers of MPc-Pyr (M=Fe, Co, Ni, Cu). A bandgap was observed for the monolayer models in contrast to the lack of bandgap observed for the bulk material calculations. Geometry optimizations and property modelling was done using CASTEP as part of materials studio 2019 (cutoff energy: 300-500 eV; k-point mesh:  $2 \times 2 \times 3$ ). The GGA functional PBE was used. A Hubbard correction of 3.0 eV was used for all metals.

### 1.2.2 Computational examination of MPc-COF monolayers (M = Fe, Co, Ni, Cu)



**Figure 7. Progress towards Task 1.3.1.** Scanning electron micrographs showing that control over synthetic conditions (conditions in table below) enables morphological tuning of layered MOF-based nanostructures through solvothermal synthesis.

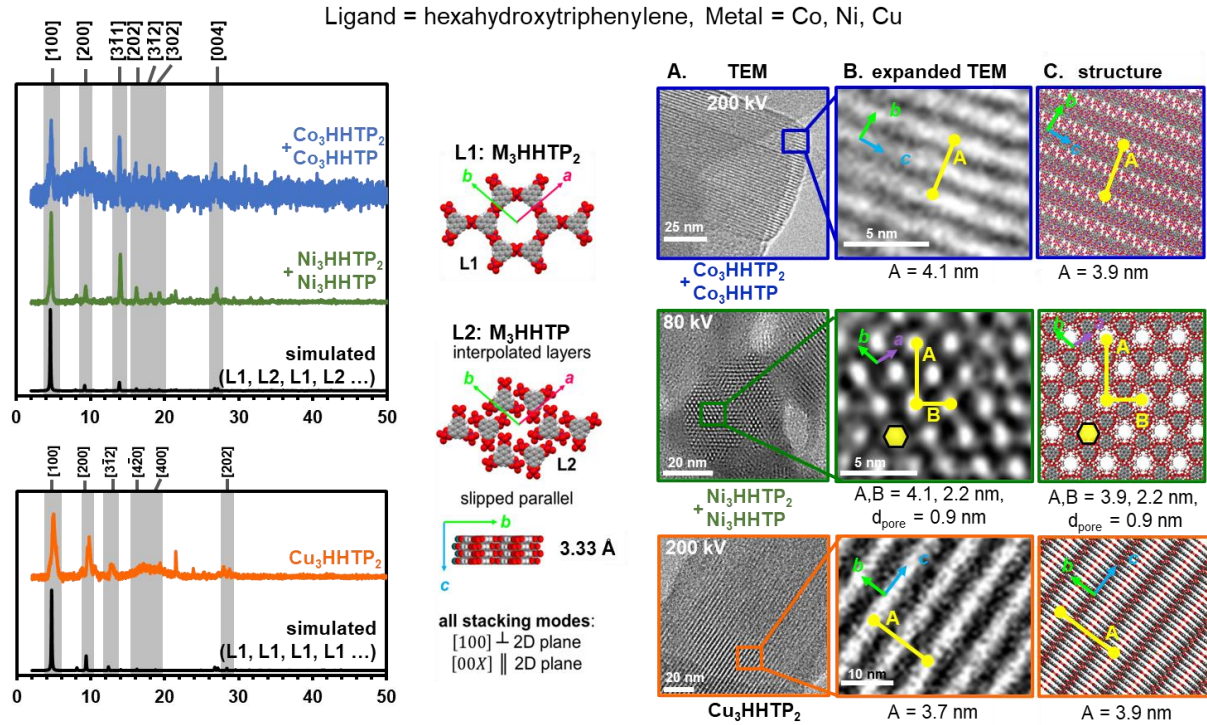
**1.3.1. Optimization of morphological control for producing free-standing MOF nanowires and nanosheets**



[a] slow addition

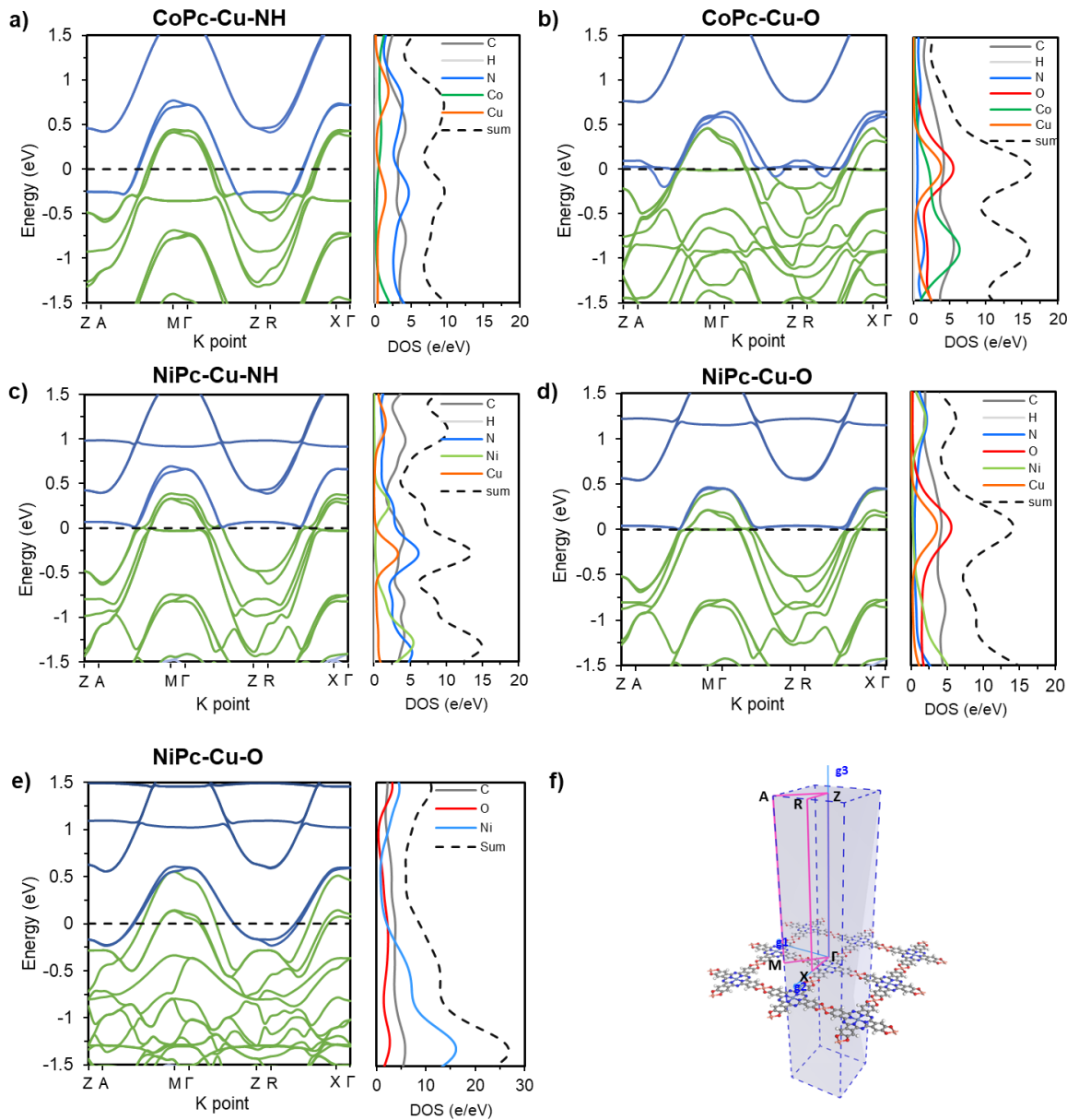
**Figure 8. Progress towards Task 1.3.1.** Powder X-Ray crystallography (pXRD) analysis confirms the structures of MOFs. Exfoliation of nanowire-based nanostructures by sonication enables visualization of the layered morphology by transmission electron microscopy (TEM).

**1.3.1 Optimization of morphological control for producing free-standing MOF nanowires and nanosheets**



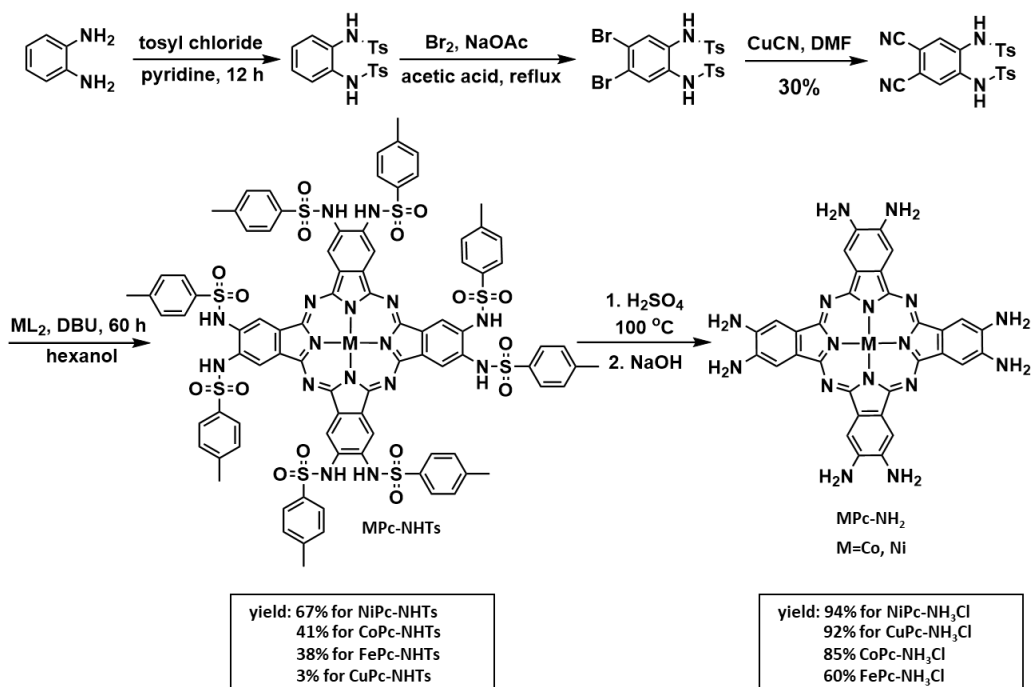
**Figure 9. Progress towards Task 1.3.2-3** Calculated electronic band structure (left) and density of state (right) for a) CoPc-Cu-NH, b) CoPc-Cu-O MOFs, c) NiPc-Cu-NH, and d) NiPc-Cu-O, e) NiPc-Ni-O, and f) Corresponding first Brillouin zone and high-symmetry K-points.

**1.3.2 Computational assessment of the influence of metal and ligand on the anisotropy of charge transport in bimetallic MOFs**



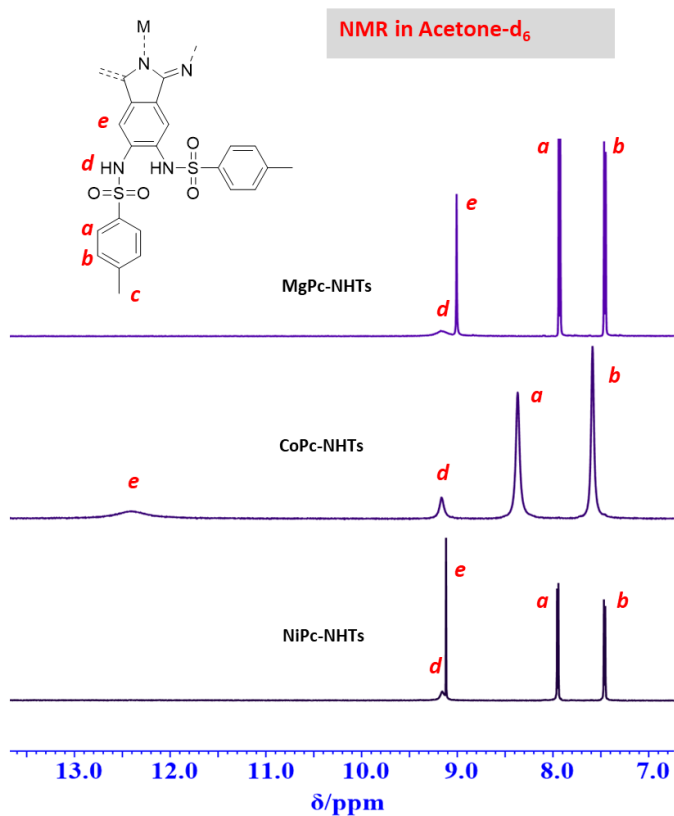
**Figure 10. Progress towards Task 1.4.1.** Synthetic scheme implemented to produce the proposed COF building blocks metal octaaminophthalocyanine (M = Fe, Co, Ni, Cu). The synthetic strategy, and resulting yields are shown.

### 1.4.1. Synthesis and Characterization of COF precursors



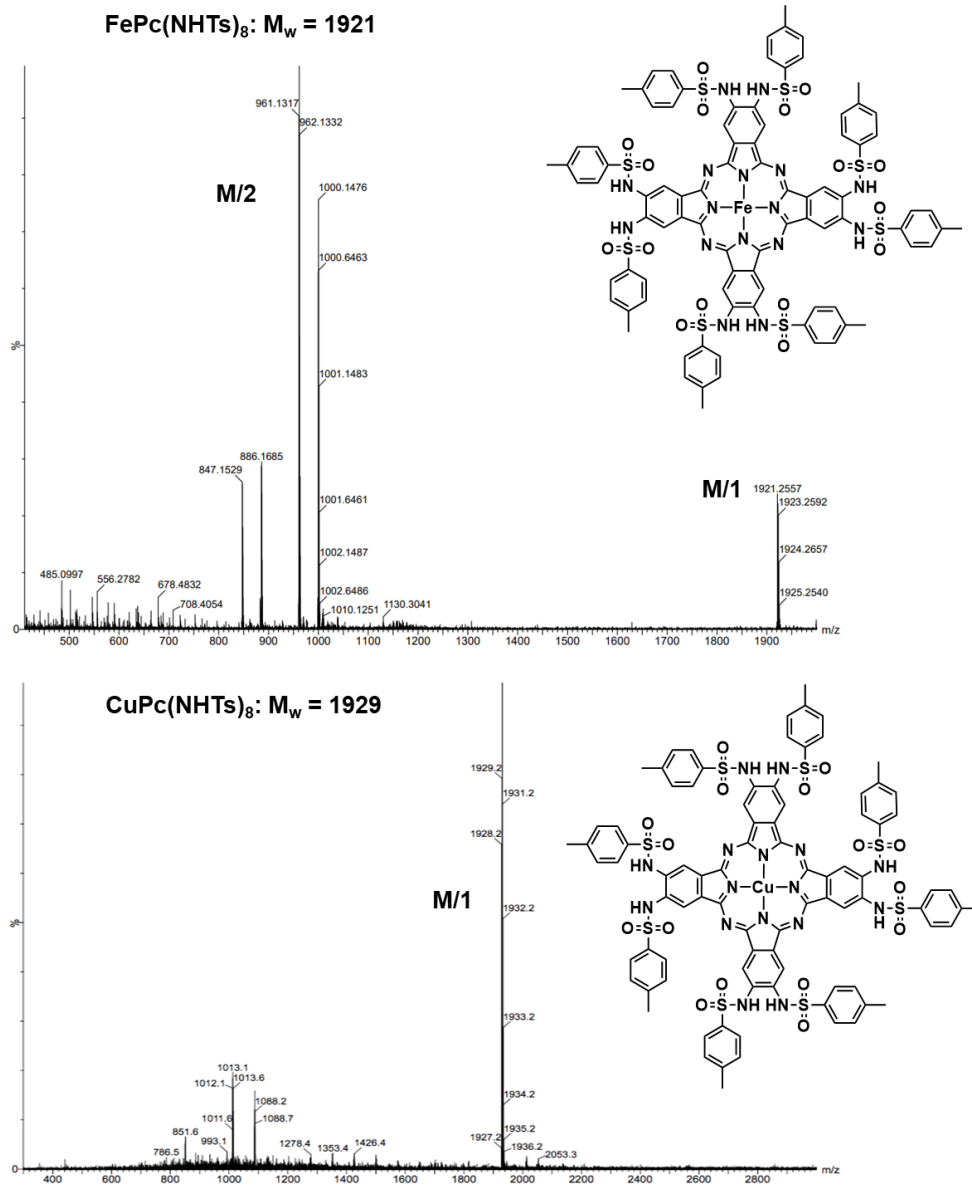
**Figure 11. Progress towards Task 1.4.1.** Characterization of ligand precursors for the MPc-Pyr COF was performed with  $H^1$ -NMR (acetone- $d_6$ , 300 MHz).

#### 1.4.1 $H^1$ -NMR characterization of paramagnetic MPc-Pyr COF precursors



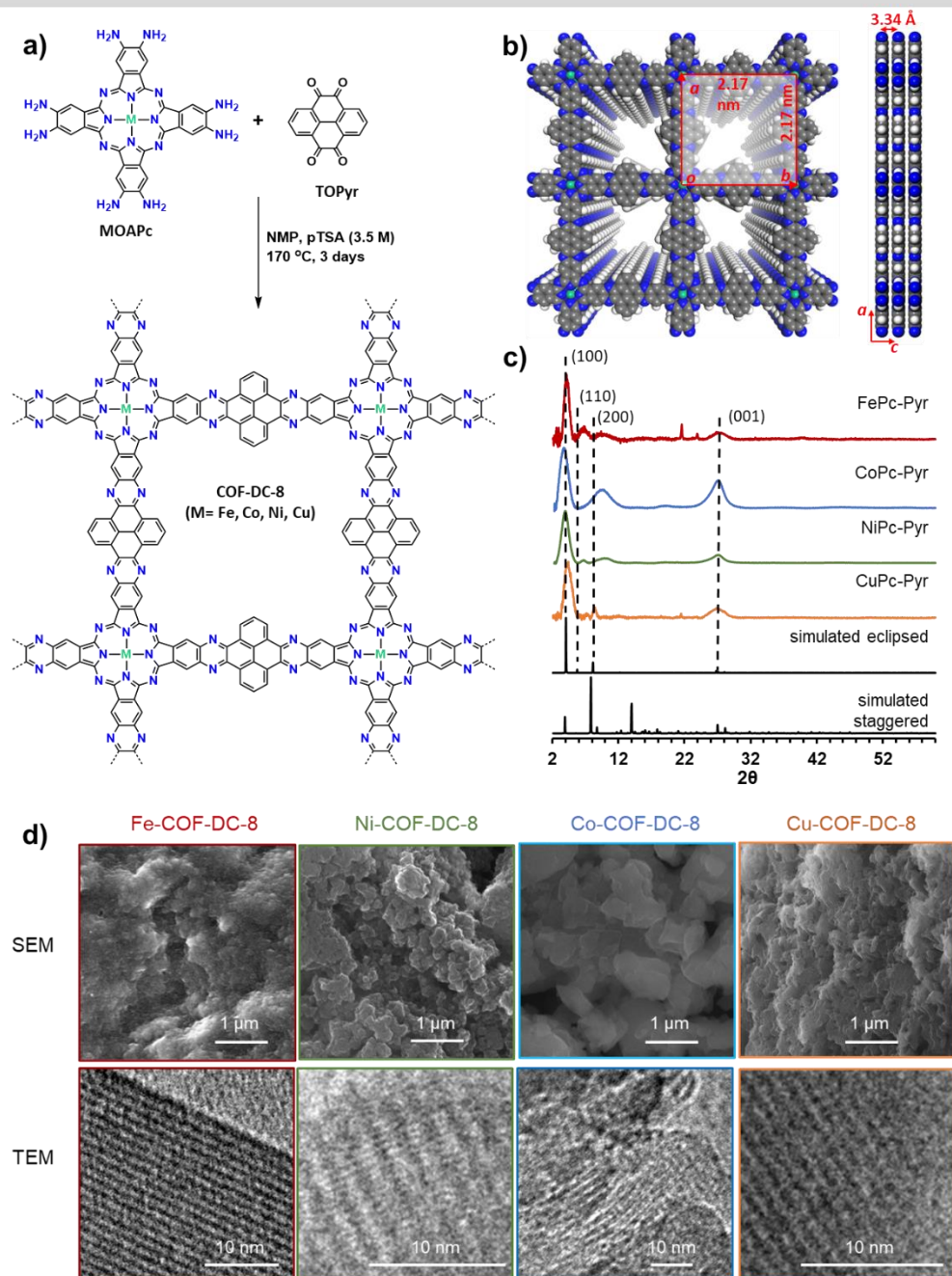
**Figure 12. Progress towards Task 1.4.1.b** Characterization of ligand precursors for the MPC-Pyr COF by HRMS (TOF ES+). The parent peaks are labeled with their M/z values where M is the molecular weight of the analyzed compound.

### 1.4.1 HRMS characterization of FePc-Pyr and CoPc-Pyr COF precursors



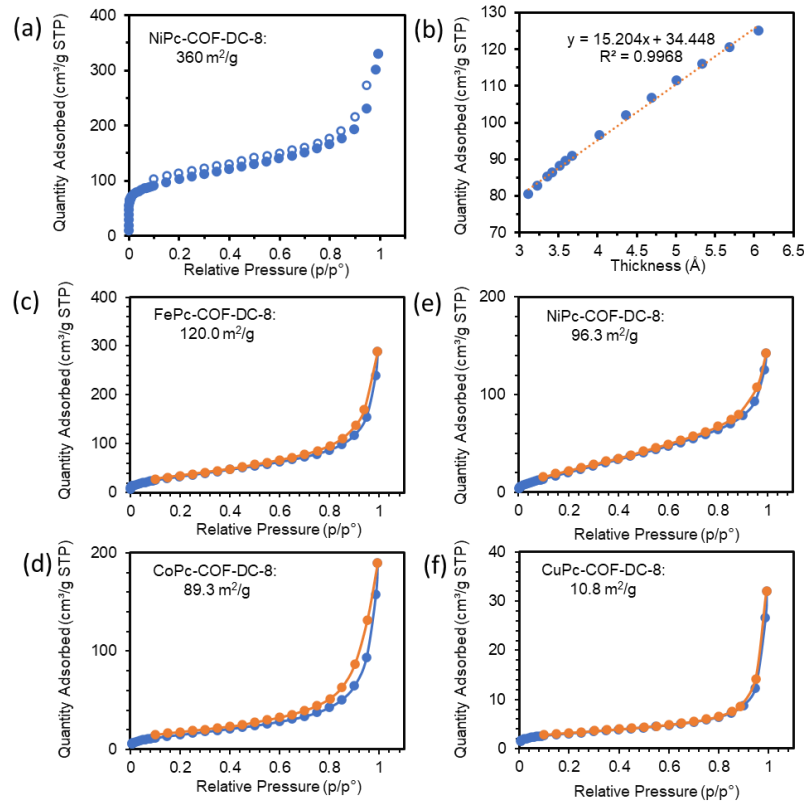
**Figure 13. Progress towards Task 1.4.2.** a) Synthetic scheme for conductive COF, named NiPc-Pyr. Structural characterization: b) Top and side view of the structure of M-COF-DC-8 with  $2 \times 2$  square grids in eclipsed stacking mode with a Lieb lattice. c) Comparison of the simulated and experimental PXRD patterns. d) SEM and TEM characterization of M-COF-DC-8 crystalline powders.

### 1.4.2 MPc-Based COFs PXRD



**Figure 14. Progress towards Task 1.4.2.** BET surface area analysis of the Ni-COPF-DC-8 COF. **a)** BET isotherm measured with N<sub>2</sub>. **b)** Quantity adsorbed versus thickness for Ni-COF-DC-8. (c) Isotherm and BET surface area of Fe-COF-DC-8. (d) Isotherm and BET surface area of Co-COF-DC-8. (e) Isotherm and BET surface area of Ni-COF-DC-8. (f) Isotherm and BET surface area of Cu-COF-DC-8. Further optimization is needed to maximize specific surface area for selected COFs.

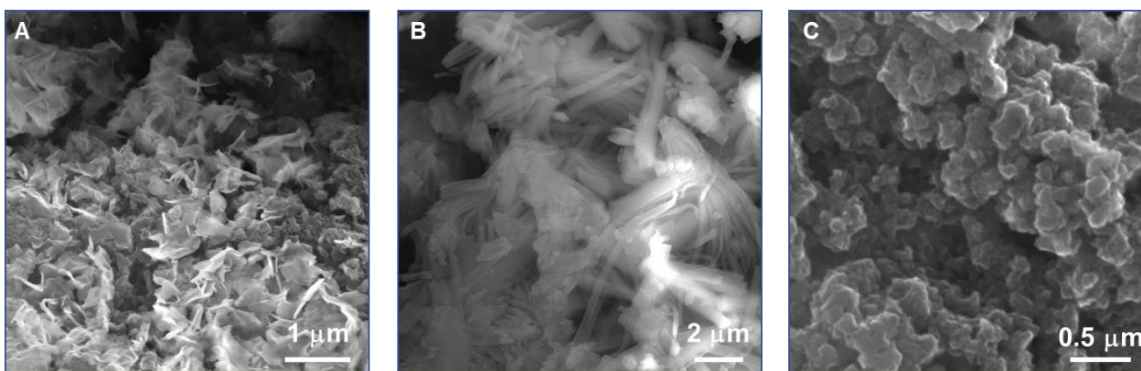
#### 1.4.2 BET characterization of MPc-based COFs



**Figure 15. Progress towards Task 1.4.3.** SEM obtained for NiPc-Pyr under different conditions which showed (a) flake, (b) rod and (c) grain like morphology.

### 1.4.3. Morphological control within MPC-based COFs

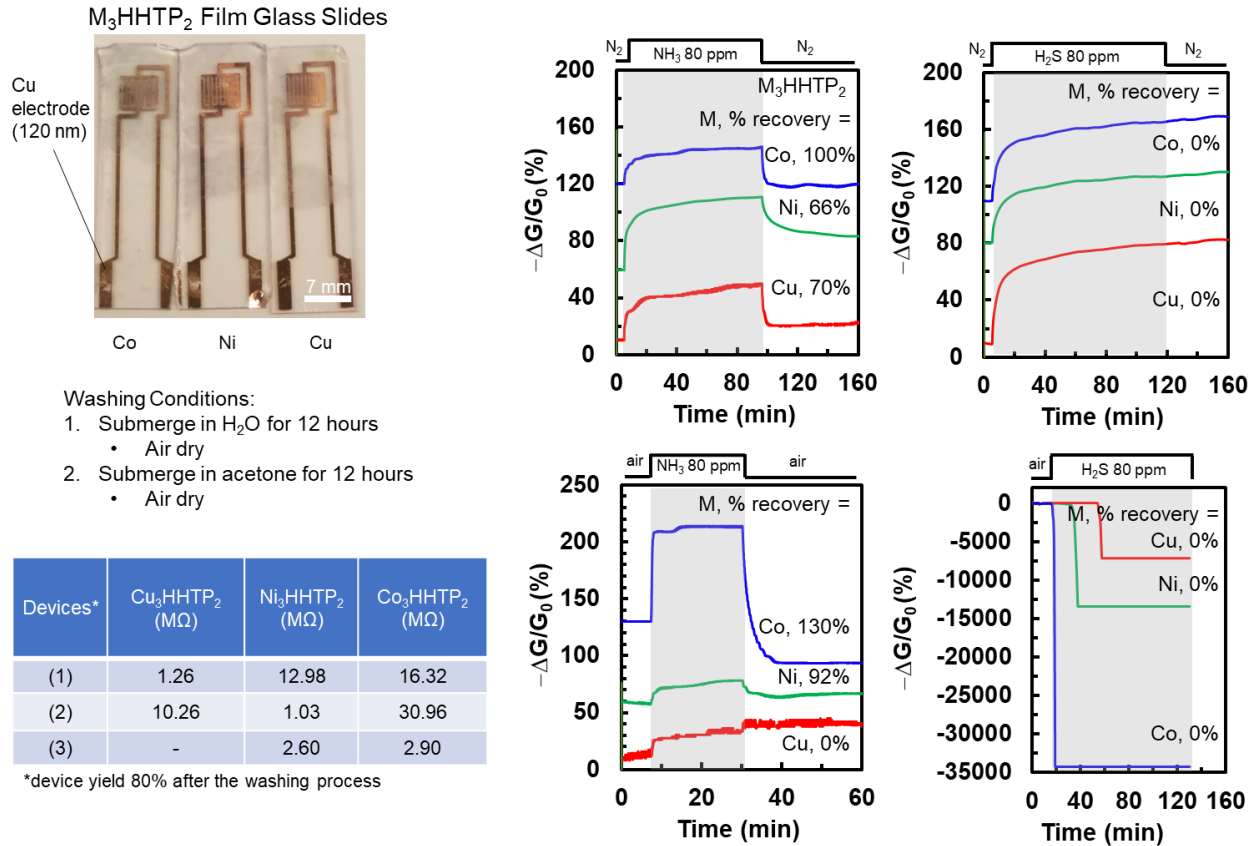
Entry	solvent	[NiOAPc] /mM	additives	Temperature /°C	time
A	DCB/DMAC (1/1)	2	--	202	10 d
B	DCB/DMAC (1/1)	2	H <sub>2</sub> SO <sub>4</sub> (9 eq)	202	10 d
C	NMP/6 M HOAc (1/0.25)	2	--	200	7 d



**Figure 16. Progress towards Task 2.1.1.** Images of  $M_3HHTP_2$  ( $M=Cu, Ni, Co$ ) thin films positioned in chemiresistive devices. Change in conductance over time plots reveals the normalized response of MOFs towards analytes.

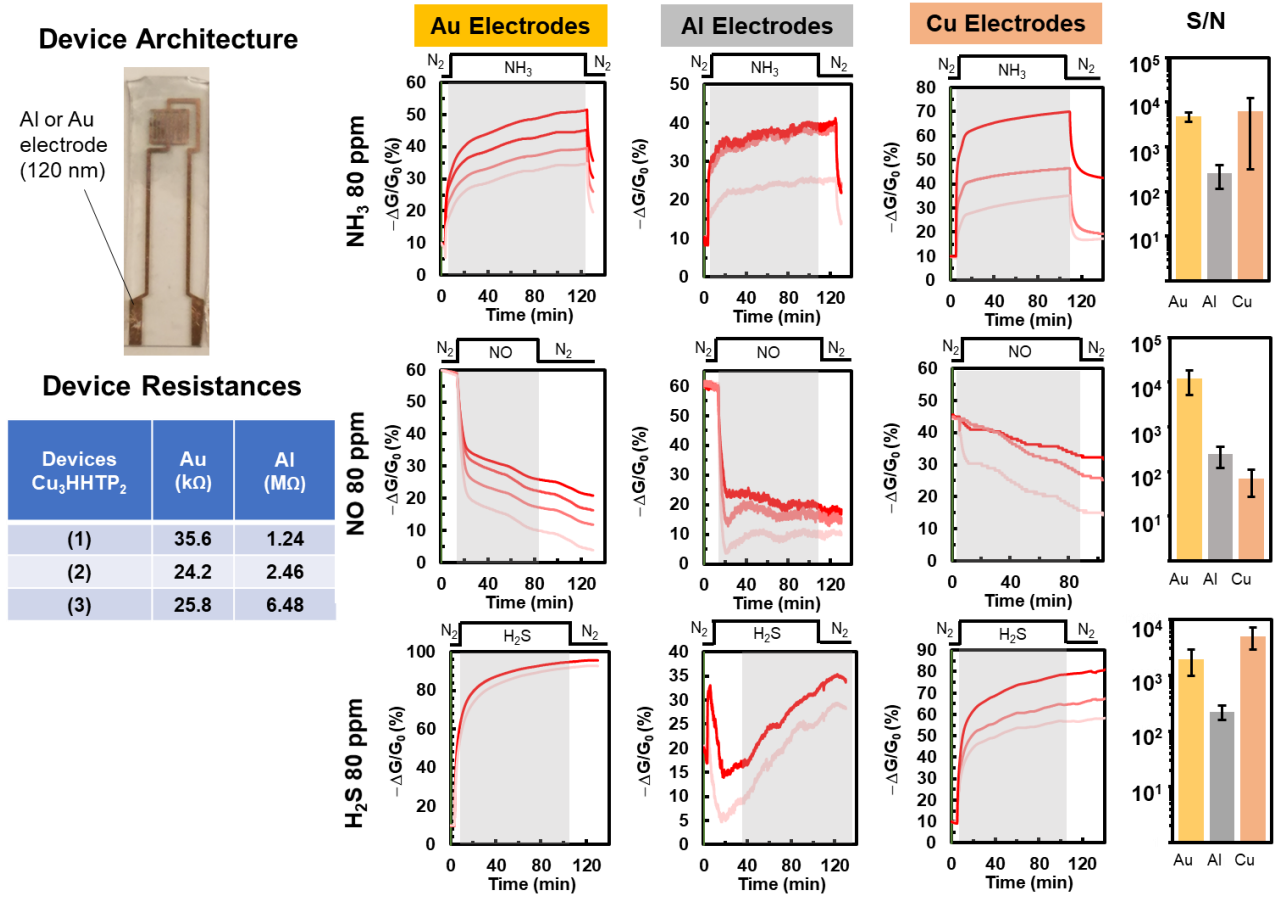
**2.1.1. Measurement of conductivity of MOFs having highly tuned morphological control**

Preliminary Results for Chemiresistive Detection of Gases Using Thin Films



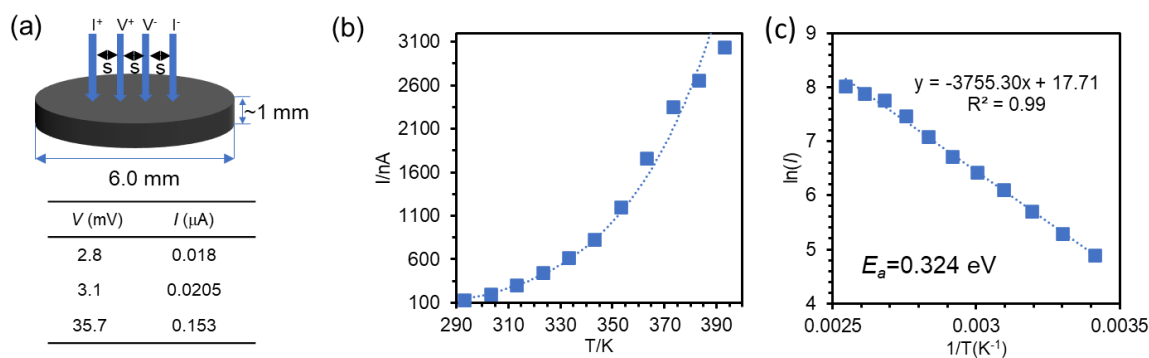
**Figure 17. Progress towards Task 2.1.2.** MOF chemiresistors fabricated using Langmuir-Schaefer deposition. The response of the MOFs towards analytes reveals the change in normalized conductance over time plots and signal to noise ratios as a function of electrode type.

**2.1.2. Examination of the impact of the electrodes on device performance in MOFs**



**Figure 18. Progress towards Task 2.1.3.** (a) Representation of configuration for the measurement of conductivity by four-point probe method and the voltage ( $V$ ) and current ( $A$ ) recorded. (b) The current change of a NiPc-Pyr pellet measured by a 2-point probe with as the temperature was increased from 293 to 393 K with an applied voltage of 1 V. (c) The Arrhenius plot of the current and temperature. (d) 2-point probe measurements of four M-COF-DC-8 as thin chemiresistive films. The applied voltage for for the thin film analysis was 1.0 V. (e) 4-point probe measurements performed on pressed pellets of M-COF-DC-8.

### 2.1.3. Electronic property study of MPC-based COFs



#### (d) 2-point probe thin film conductivity

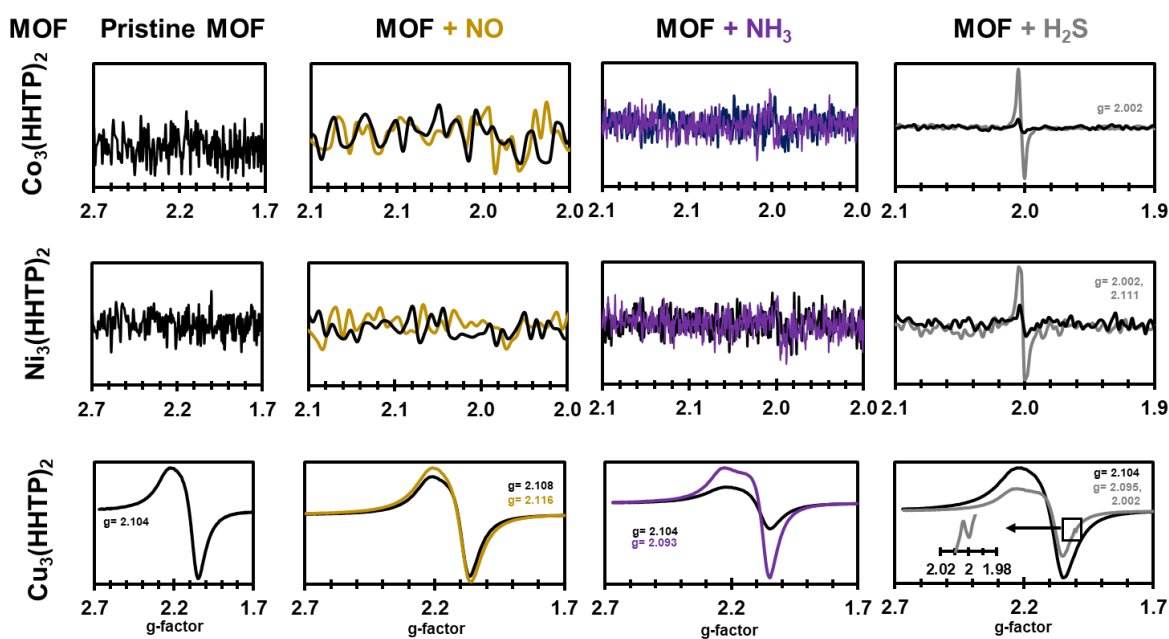
M-COF-DC-8	$\sigma$ (S/cm)	Error (S/cm)
Fe-COF-DC-8	$6.3 \times 10^{-5}$	$1.1 \times 10^{-5}$
Co-COF-DC-8	$2.5 \times 10^{-5}$	$5.9 \times 10^{-6}$
Ni-COF-DC-8	$2.8 \times 10^{-5}$	$7.52 \times 10^{-6}$
Cu-COF-DC-8	$3.2 \times 10^{-5}$	$5.5 \times 10^{-6}$

#### (e) 4-point probe pressed pellet conductivity

M-COF-DC-8	$\sigma$ (S/cm)
Fe-COF-DC-8	$1.1 \times 10^{-4}$
Co-COF-DC-8	$3.8 \times 10^{-5}$
Ni-COF-DC-8	$2.5 \times 10^{-5}$
Cu-COF-DC-8	no reading

**Figure 19. Progress towards Task 2.2.1.** Electron paramagnetic resonance spectroscopy (EPR) at 77 K was used to study the concentration and location of unpaired spins in HHTP-based MOFs. The pristine MOFs of Ni and Co show EPR-silent spectra indicating that any singularly occupied molecular orbitals (SOMO) are antiferromagnetically coupled. The large absorbance of  $\text{Cu}_3\text{HHTP}_2$  at  $g > 2$  indicates that the SOMO primarily resides on the Cu metal center. The interaction of  $\text{Co}_3\text{HHTP}_2 + \text{Co}_3\text{HHTP}$  and  $\text{Ni}_3\text{HHTP}_2 + \text{Ni}_3\text{HHTP}$  MOFs with NO showed no perturbation to the EPR spectrum. However, the interaction of  $\text{Cu}_3\text{HHTP}_2$  with NO showed an increase in the spin-density in the material. A similar trend was observed for all three MOFs with  $\text{NH}_3$ . The interaction of the bilayer MOFs with  $\text{H}_2\text{S}$  showed a dramatic increase in the spin-density in the MOFs. The spin-density of  $\text{Cu}_3\text{HHTP}_2$  showed a significant decrease but the EPR spectrum also detected a small  $g = 2.005$  peak indicating a new radical species in the system.

### 2.2.1. EPR studies of analyte binding in MOFs

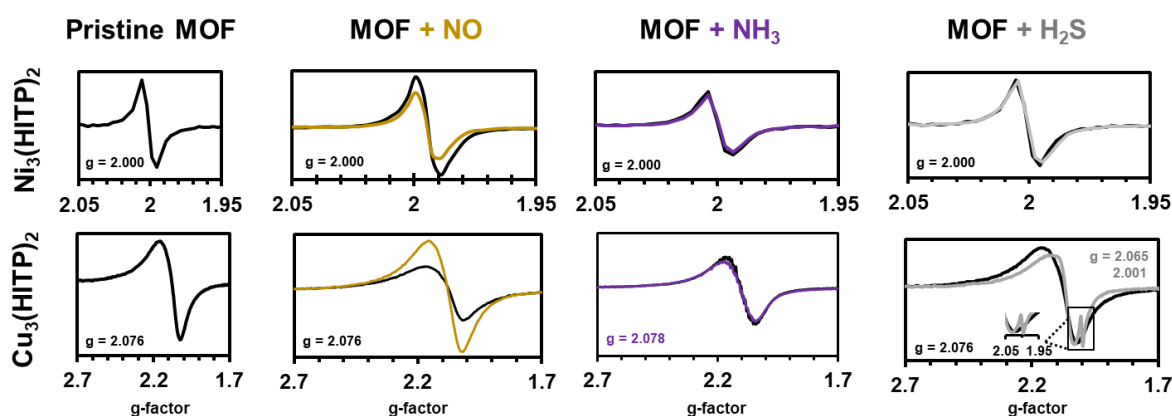


X-band EPR at 9.7 GHz, measurements taken at 77 K using 1-2 mg of sample

Samples degassed for 1 h with house  $\text{N}_2$ . The EPR spectrum was then taken (black trace). After initial analysis, the samples were exposed to a desired analyte gas at 1 % (balance  $\text{N}_2$ ) for 20 min. Then a second spectrum was taken. MOFs were washed with water then with acetone in a Soxhlet extractor for 24-48 h. MOFs were dried under high vacuum and stored under argon unless otherwise noted.

**Figure 20. Progress towards Task 2.2.1.** Electron paramagnetic resonance spectroscopy (EPR) at 77 K was used to study the concentration and location of unpaired spins in HITP-based MOFs. The pristine MOFs of Ni show strong  $g=2.00$  values. The large absorbance of  $\text{Cu}_3\text{HITP}_2$  at  $g>2$  indicates that the SOMO primarily resides on the Cu metal center. The interaction and  $\text{Ni}_3\text{HITP}_2$  with NO showed a small decrease in signal intensity suggesting that some radical residing on HITP ligand was quenched by the radical on NO.  $\text{Cu}_3\text{HITP}_2$  showed an increase in signal intensity suggesting that NO caused a conversion of  $\text{Cu}^{\text{I}}$  to  $\text{Cu}^{\text{II}}$  upon exposure. An opposite trend was observed for  $\text{Cu}_3\text{HITP}_2$  with  $\text{NH}_3$ , while  $\text{Ni}_3\text{HITP}_2$  showed no response to  $\text{NH}_3$ .  $\text{Ni}_3\text{HITP}_2$  responded similarly to  $\text{H}_2\text{S}$  showing no observable changes in radical spin density.  $\text{Cu}_3\text{HITP}_2$  showed a shift in  $g$ -value as well as a new radical species in response to  $\text{H}_2\text{S}$  that was not metal-centered.

### 2.2.1. EPR studies of analyte binding in MOFs

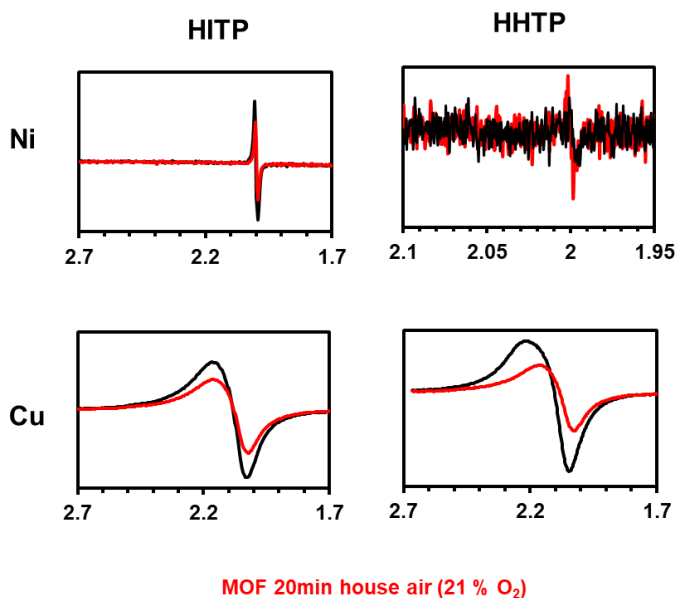


X-band EPR at 9.7 GHz,  
measurements taken at 77 K  
using 1-2 mg of sample

Samples degassed for 1 h with house  $\text{N}_2$ . The EPR spectrum was then taken (black trace). After initial analysis, the samples were exposed to a desired analyte gas at 1% (balance  $\text{N}_2$ ) for 20 min. Then a second spectrum was taken. MOFs were washed with water then with acetone in a Soxhlet extractor for 24-48 h. MOFs were dried under high vacuum and stored under argon unless otherwise noted.

**Figure 21. Progress towards Task 2.2.1.**

Electron paramagnetic resonance spectroscopy (EPR) at 77 K was used to study the concentration and location of unpaired spins in HHTP and HTP-based MOFs. For  $\text{Ni}_3\text{HITP}_2$ , oxygen caused a decrease in the intensity of the signal at  $g=2.00$ .  $\text{Ni}_3\text{HHTP}_2$  showed an increase in signal intensity at  $g=2.00$ . Both  $\text{Cu}_3\text{HITP}_2$  and  $\text{Cu}_3\text{HHTP}_2$  showed a reduction in signal intensity in response to oxygen exposure. These results indicate that oxygen plays a significant role in the oxidation state of the MOF components and the charge balance within the framework.

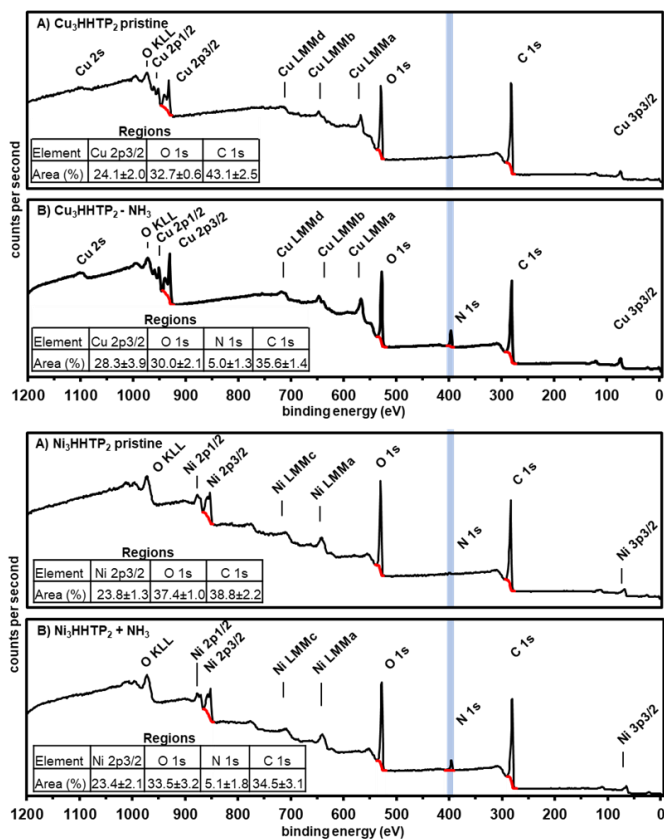
**2.2.1. EPR studies of analyte binding in MOFs**

X-band EPR at 9.7 GHz,  
measurements taken at 77 K  
using 1-2 mg of sample

Samples degassed for 1 h with house  $\text{N}_2$ . The EPR spectrum was then taken (black trace). After initial analysis, the samples were exposed to a desired analyte gas at 1 % (balance  $\text{N}_2$ ) for 20 min. Then a second spectrum was taken. MOFs were washed with water then with acetone in a Soxhlet extractor for 24-48 h. MOFs were dried under high vacuum and stored under argon unless otherwise noted.

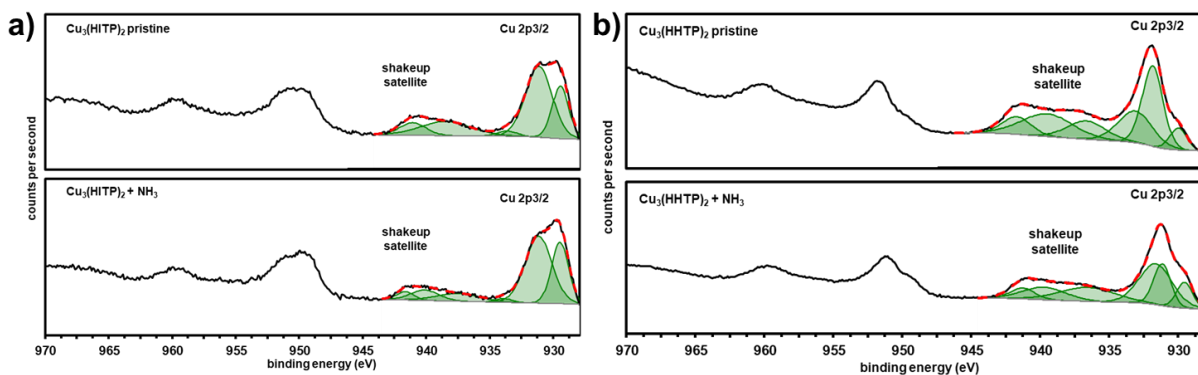
**Figure 22. Progress towards Task 2.2.1.** XPS was used to determine the effects of NH<sub>3</sub> binding to MOF samples. The presence of a persistent N 1s emission line for HHTP-Based MOFs demonstrated that NH<sub>3</sub> binds irreversibly under the high-vacuum conditions of XPS. Further characterization of XPS was performed using high-resolution regions corresponding to specific analytes.

**2.2.1. XPS studies of NH<sub>3</sub> confirms adsorption on MOF surface**



**Figure 23. Progress towards Task 2.2.1.**

Comparison of pristine and  $\text{NH}_3$ -exposed samples by XPS can reveal metal-oxidation state shifts by close examination of the metal  $2p_{3/2}$  regions. **a)** pristine  $\text{Cu}_3(\text{HITP})_2$  shows a higher ratio of shakeup satellite emission line to primary  $\text{Cu} 2p_{3/2}$  emission line when compared to  $\text{NH}_3$  exposed  $\text{Cu}_3(\text{HITP})_2$ . This indicates an increase in  $\text{Cu}^{\text{II}}$  and a concomitant decrease in  $\text{Cu}^{\text{I}}$ . **b)** pristine  $\text{Cu}_3(\text{HHTP})_2$  shows a lower ratio of shakeup satellite emission line to primary  $\text{Cu} 2p_{3/2}$  emission line when compared to  $\text{NH}_3$  exposed  $\text{Cu}_3(\text{HHTP})_2$ . **c)** These shifts in intensity indicated a decrease in  $\text{Cu}^{\text{II}}$  and a concomitant increase in  $\text{Cu}^{\text{I}}$  for  $\text{Cu}_3(\text{HITP})_2$  and, for  $\text{Cu}_3(\text{HHTP})_2$ , increase in  $\text{Cu}^{\text{II}}$  and a concomitant decrease in  $\text{Cu}^{\text{I}}$ .

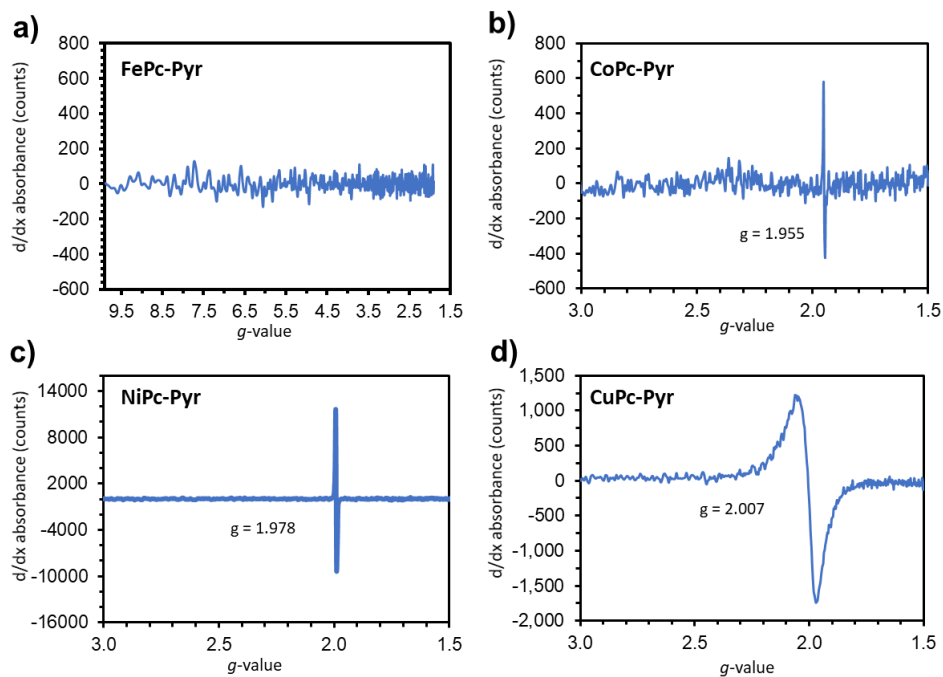
**2.2.1. XPS studies of  $\text{NH}_3$  binding, and oxidation state changes of Cu in MOFs**

**c)**

MOF-analyte	% $\text{Cu}^{\text{II}}$ by XPS	% $\text{Cu}^{\text{II}}$ by EPR	$\Delta\%$ $\text{Cu}^{\text{II}}$ by XPS	$\Delta\%$ $\text{Cu}^{\text{II}}$ by EPR
$\text{Cu}_3(\text{HITP})_2$	$57.7 \pm 1$	$36.1 \pm 4$	na	na
$\text{Cu}_3(\text{HITP})_2 + \text{NH}_3$	$51.5 \pm 1$	$33.4 \pm 4$	$-6.2 \pm 1$	$-2.7 \pm 2$
$\text{Cu}_3(\text{HHTP})_2$	$85.3 \pm 1$	$81.7 \pm 2$	na	na
$\text{Cu}_3(\text{HHTP})_2 + \text{NH}_3$	$95.3 \pm 3$	$93.5 \pm 2$	$+10.0 \pm 2$	$+12.9 \pm 3$

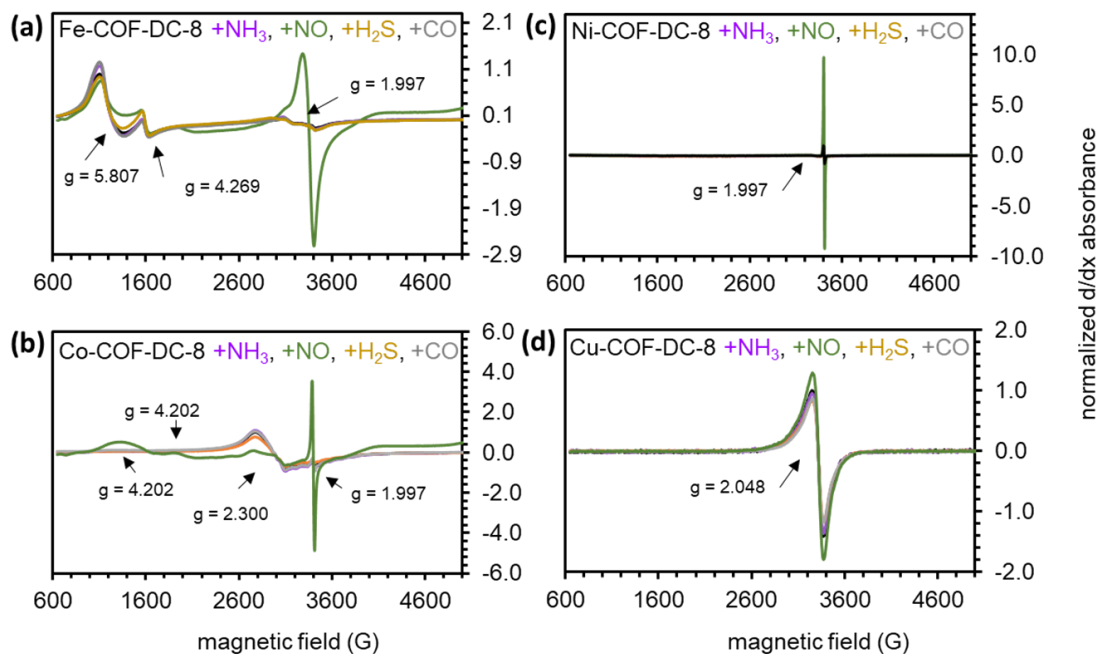
**Figure 24. Progress towards Task 2.2.2.** EPR characterization of a) FePc-Pyr COF, b) CoPc-Pyr COF, c) NiPc-Pyr-COF, and d) CuPc-Pyr COF. Measurements were taken at 77 K after the material was flushed with N<sub>2</sub> for 24 h.

### 2.2.2. EPR spectroscopy of MPc-COFs (M= Fe, Co, Ni, Cu)



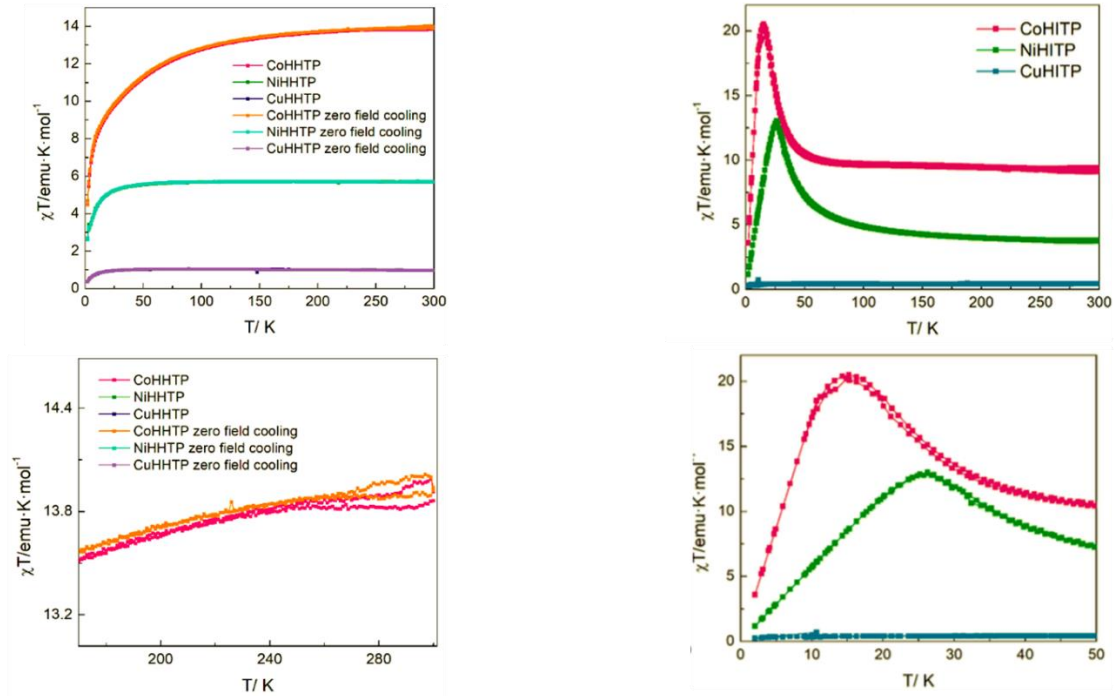
**Figure 25. Progress towards Task 2.2.2.** Comparison of the EPR spectra of (a) Fe-COF-DC-8, (b) Co-COF-DC-8, (c) Ni-COF-DC-8, and (d) Cu-COF-DC-8 before (black lines) and after (colored lines) dosing with (purple)  $\text{NH}_3$ , (green)  $\text{NO}$ , (gold)  $\text{H}_2\text{S}$ , and (grey)  $\text{CO}$ . EPR spectra were obtained at 4.2 K.

### 2.2.2. EPR studies of analyte binding in COFs



**Figure 26. Progress towards Task 2.2.3.** For each MOF, the temperature dependence of the molar magnetic susceptibility was measured from 300 K to 2 K.

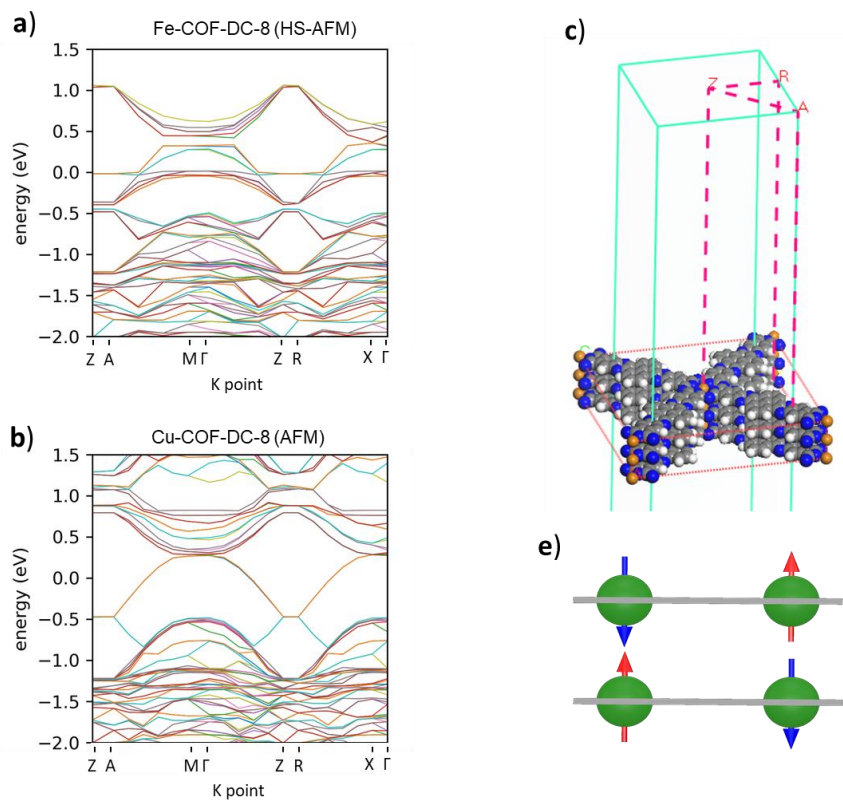
**2.2.3. SQUID magnetometry of MOFs**



DC Magnetization of HHTP and HITP-based MOFs at 2 – 300K at 10,000 Oe measured with MPMS

**Figure 27. Progress towards Task 2.2.4.** Computational modelling of spin-ordering in M-COF-DC-8 materials. (a) The band structure of Fe-COF-DC-8 in a high spin antiferromagnetically ordered spin state. (b) The band structure of Cu-COF-DC-8 in an antiferromagnetically ordered spin state. (c) The first Brillouin zone of COFs have I4-MMM symmetry enforced by spin orientation. (d) a graphical depiction of the antiferromagnetic ordering between metal centers in adjacent layers of the COFs.

### 2.2.4 Computational study of spin orientation and band structure



**Table 1. Progress towards Task 3.1.1.** MOFs are listed with their observed conductivities and their percent resistance changes normalized to their starting resistance in response to specified concentrations of gas. <sup>a</sup> 4-point probe method was used to obtain value. <sup>b</sup> 2-point probe method was used to obtain value

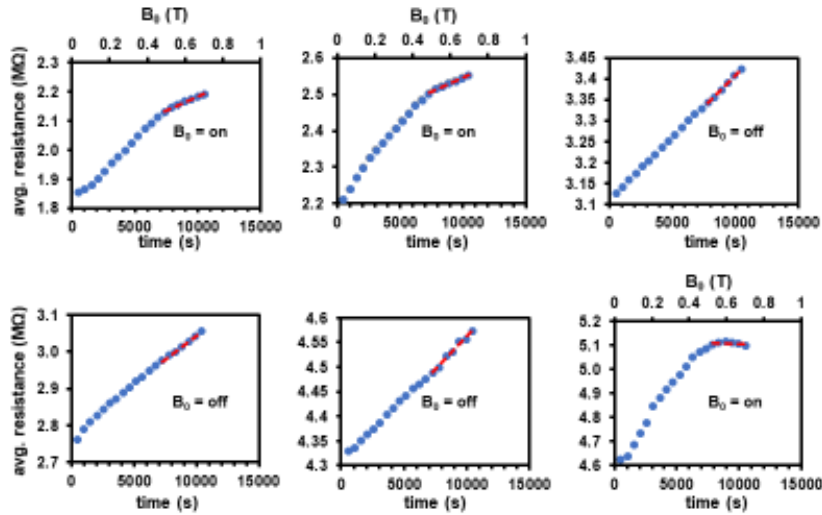
MOF	$\sigma$ (S/cm)	$\text{NH}_3$ ( $-\Delta G/G^\circ$ ) 80 ppm	$\text{H}_2\text{S}$ ( $-\Delta G/G^\circ$ ) 80 ppm	$\text{NO}$ ( $-\Delta G/G^\circ$ ) 80 ppb	$\text{CO}$ ( $-\Delta G/G^\circ$ ) 80 ppm
CoPc-Cu-NH	$5.79 \times 10^{-2}$ <sup>a</sup>	N/A	N/A	N/A	+ 30%
CoPc-Cu-O	$2.12 \times 10^{-2}$ <sup>a</sup>	N/A	N/A	N/A	+ 25%
NiPc-Cu-NH	$2.73 \times 10^{-3}$ <sup>a</sup>	N/A	N/A	N/A	+ 18%
NiPc-Cu-O	$1.43 \times 10^{-2}$ <sup>a</sup>	+ 45 %	+ 100 %	-120 %	+ 28%
NiPc-Ni-O	$7.22 \times 10^{-4}$ <sup>a</sup>	+ 43 %	+ 25 %	-80 %	N/A
$\text{Cu}_3(\text{HHTN})_2$	$9.01 \times 10^{-8}$ <sup>a</sup>	N/A	N/A	N/A	N/A
$\text{Ni}_3(\text{HHTP})_2/\text{cotton}$	$1.6 \times 10^{-4}$ <sup>b</sup>	- 80 %	+ 100 %	- 45 %	
$\text{Ni}_3(\text{HITP})_2/\text{cotton}$	$2.6 \times 10^{-4}$ <sup>b</sup>	0 %	+ 98 %	+ 80 %	

**Table 2. Progress towards Task 3.1.2.** COFs are listed with their observed conductivities (obtained from thin films and 2-point probe at and applied voltage of 1.0 V) and their percent resistance changes normalized to their starting resistance in response to specified concentrations of gas.

COF	$\sigma$ (S/cm)	$\text{NH}_3$ ( $-\Delta G/G^\circ$ ) 80 ppm	$\text{H}_2\text{S}$ ( $-\Delta G/G^\circ$ ) 80 ppm	$\text{NO}$ ( $-\Delta G/G^\circ$ ) 80 ppb	$\text{CO}$ ( $-\Delta G/G^\circ$ ) 80 ppb
Fe-COF-DC-8	$6.3 \times 10^{-5}$	+ 29.1 %	+ 70.0 %	- 1644 %	+ 17.5 %
Co-COF-DC-8	$2.5 \times 10^{-5}$	+ 87.8 %	+ 72.8 %	- 5665 %	0 %
Ni-COF-DC-8	$2.8 \times 10^{-5}$	+ 35.2 %	+ 76.3 %	- 3049 %	0 %
Cu-COF-DC-8	$3.2 \times 10^{-5}$	+ 77.8 %	+ 99.3 %	- 7235 %	0 %

**Figure 28. Progress towards Task 3.1.1** Magneto-transport studies using the 4-point probe devices was carried out using  $\text{Ni}_3(\text{HHTP})_2$ . Resistance values were obtained over a range of  $B_0$  field strengths ranging from 0 T to 0.8 T. Control studies were run using a  $B_0$  strength of 0 T for the duration of the experiment. The relative increase in resistance values observed above  $B_0 > 0.6$  T correlate with field strength when compared to control experiments.

**3.1.1. Measurements of field effect in MOFs**



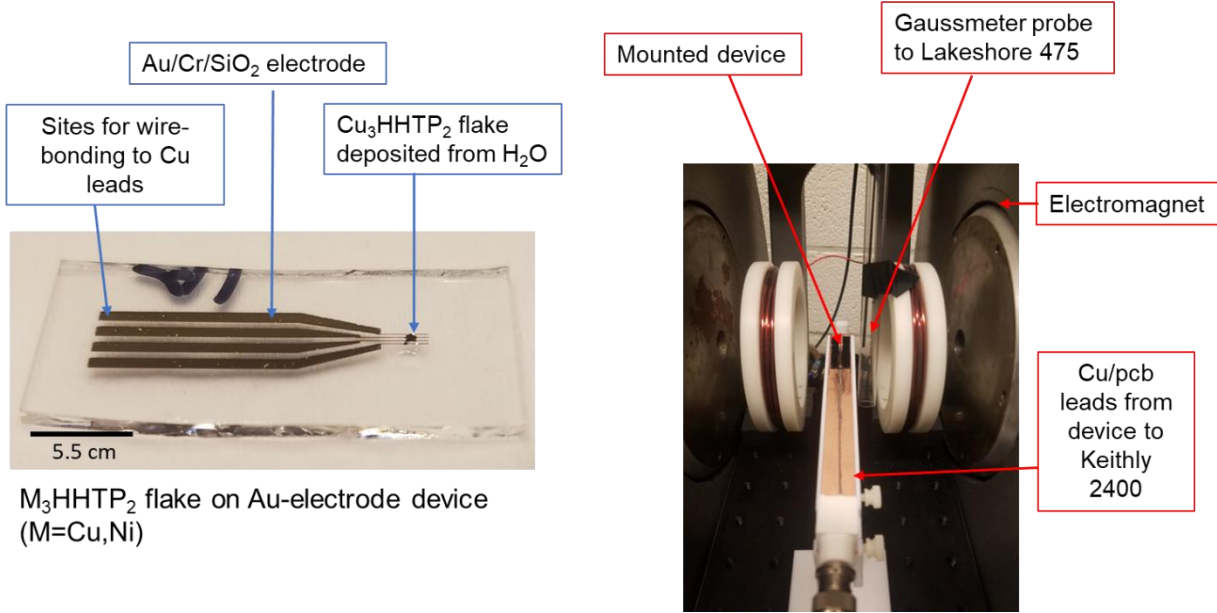
Device:  
 $\text{Ni}_3(\text{HHTP})_2$  nanosheets deposited on gold electrode.

Experiment:  
 The  $B_0$  field was scanned from 0 – 1 T with I/V curves taken at every 0.05 T increment. Average resistance obtained from the I/V curves plotted versus  $B_0$  (for  $B_0 = \text{on}$ ) or versus time (for  $B_0 = \text{off}$ ) possible transition observed at  $B_0 = 0.6$  T

**Figure 29. Progress towards Task 3.2.1-3** Photographs of devices capable of interfacing small MOF sheets with 4-point probe measurements and magneto-electronic instrumentation.

- 3.2.1. Fabrication of Hall Bar devices with MOFs
- 3.2.2. Measurements of charge mobility of MOFs
- 3.2.3. Measurements of field effect in MOFs

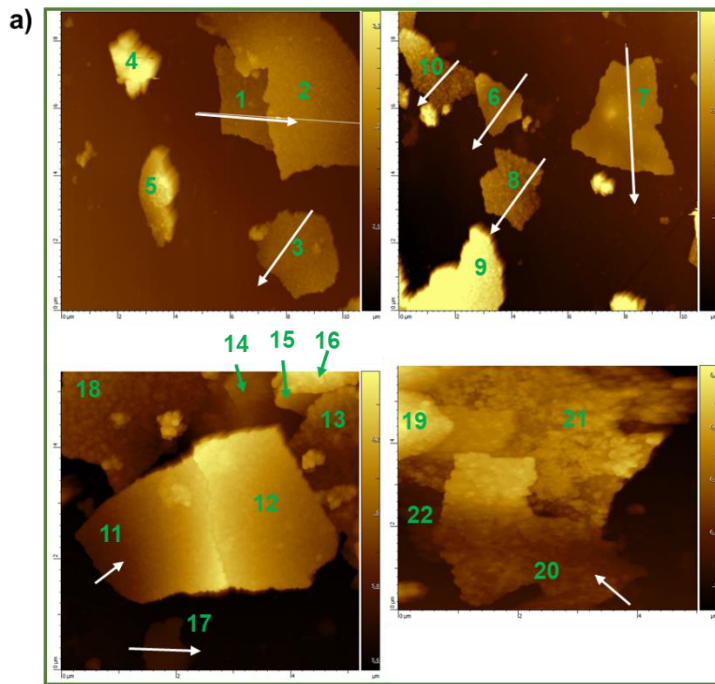
**Magnetoresistance device and instrument setup**



**Figure 30. Progress towards Task 3.2.1.** AFM analysis is a powerful method of characterizing thin freestanding films. The films used for magneto-transport studies were imaged by AFM (a) to determine (b) average thickness and average roughness.

**3.2.1. Characterization of nanosheets by AFM**

**Magnetoresistance device and instrument setup**

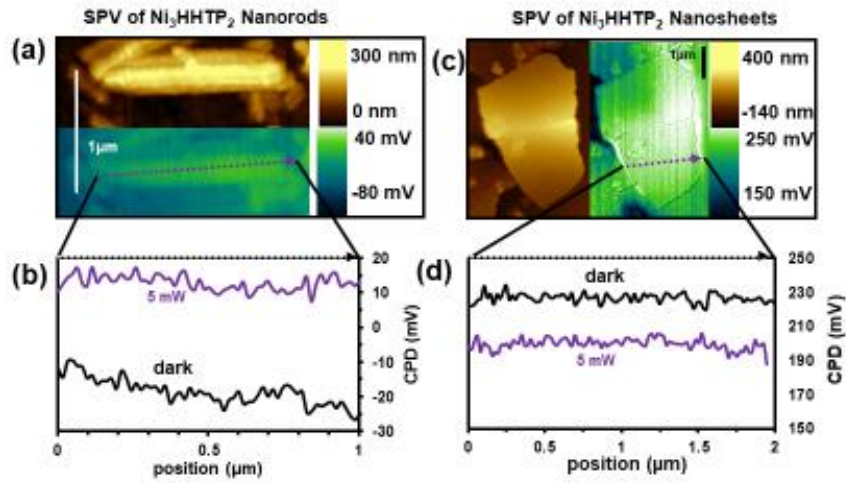


b)

NS number	Thickness (nm)	$\sigma$ (nm)
1	288.2	18.1
2	204.3	22.9
3	301.9	50.9
6	170.1	13.4
7	159.5	28.2
8	187.7	34.5
10	156.3	25.8
11	163.9	5.8
17	116.3	5.9
20	93.3	8.0
<b>Average thickness (nm)</b>	184.1	-
<b><math>\sigma</math> (nm)</b>	66.6	14.3

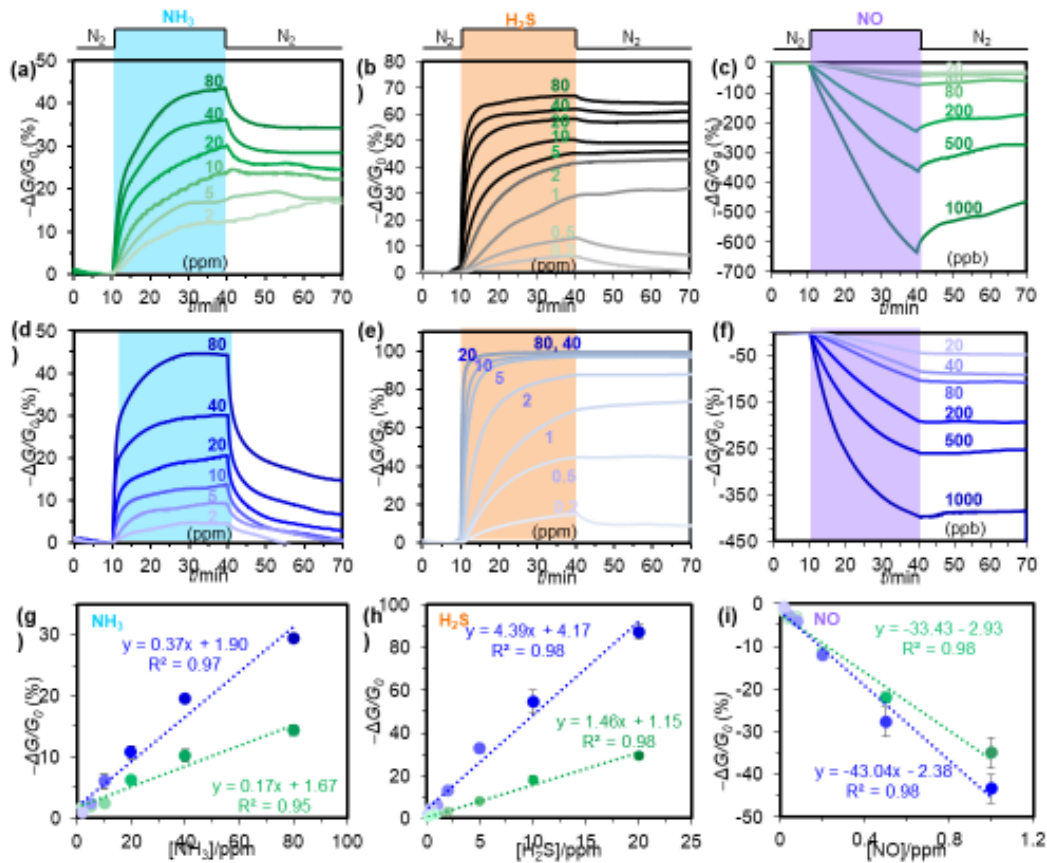
**Figure 31. Progress towards Task 3.2.2** CPD analysis of (a) nanorods and (b) nanosheets under illumination demonstrate that the photovoltage changes as a function of morphology. Nanorods exhibit a positive photovoltage response (c), while nanosheets exhibit a negative photovoltage response (d).

**3.2.1. Photovoltage response of different morphologies**



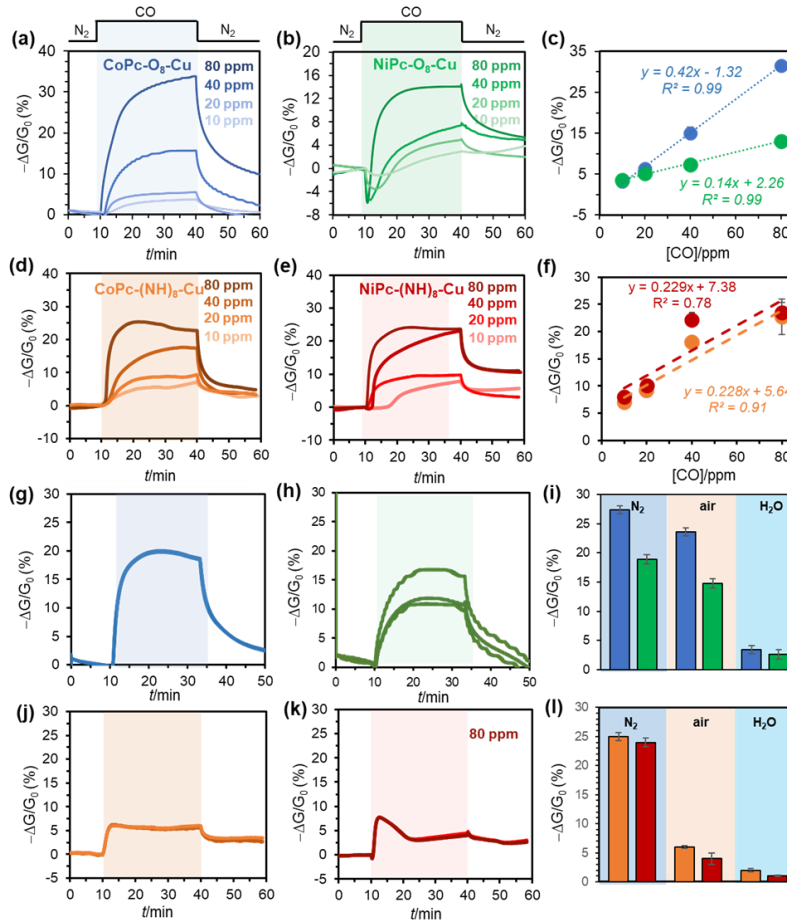
**Figure 32. Progress Towards Task 3.2.4.** Performance of chemiresistive device of NiPc-M MOFs. (a–c) Representative response of NiPc-Ni MOF to NH<sub>3</sub>, H<sub>2</sub>S, and NO at different concentrations. (d–f) Representative responses of NiPc-Cu MOF to NH<sub>3</sub>, H<sub>2</sub>S, and NO at different concentrations. (g–i) Responses ( $-\Delta G/G_0$ ) of NiPc-Ni (green) and NiPc-Cu (blue) after 1.5 min exposure versus concentration of NH<sub>3</sub>, H<sub>2</sub>S, and NO. Suspensions of NiPc-M MOF were drop cast onto prefabricated electrodes and allowed to air dry to form chemiresistive devices. Prefabricated devices were composed of interdigitated gold electrodes on glass with 5  $\mu\text{m}$  spacings between interdigitated electrode fingers

**3.2.4 Charge mobility in NiPc-M MOF modulated by gas adsorption**



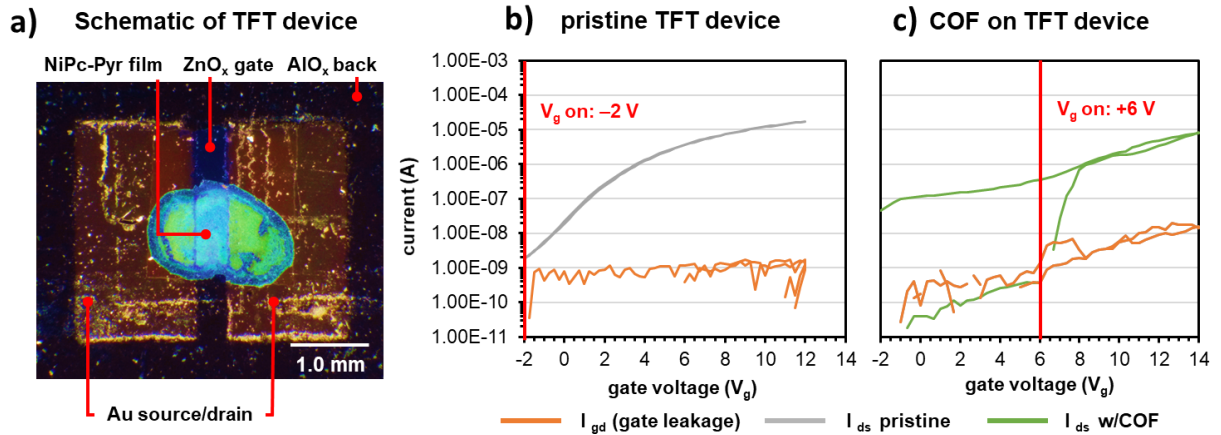
**Figure 33. Progress Towards Task 3.2.4.** Saturation sensing traces of (a) CoPc-O<sub>8</sub>-Cu, (b) , (d) CoPc-(NH)<sub>8</sub>-Cu, and (d) NiPc-(NH)<sub>8</sub>-Cu MOF after 30 min exposure to 80, 40, 20, and 10 ppm of CO. (c) Responses ( $-\Delta G/G_0$ ) of and CoPc-O<sub>8</sub>-Cu MOF and NiPc-O<sub>8</sub>-Cu after 30 min exposure versus concentration of CO. (c) (f) Responses of and CoPc-(NH)<sub>8</sub>-Cu MOF and NiPc-(NH)<sub>8</sub>-Cu after 30 min exposure versus concentration of CO. Sensing traces of (g) CoPc-O<sub>8</sub>-Cu, (h) , (j) CoPc-(NH)<sub>8</sub>-Cu, and (k) NiPc-(NH)<sub>8</sub>-Cu MOF after 30 min exposure to 80 of CO in air. (i) Response of CoPc-O<sub>8</sub>-Cu (blue) and NiPc-O<sub>8</sub>-Cu (green) to 80 ppm CO in N<sub>2</sub>, air, and 5000 ppm humidity. (l) Response of CoPc-(NH)<sub>8</sub>-Cu (orange) and NiPc-(NH)<sub>8</sub>-Cu (red) to 80 ppm CO in N<sub>2</sub>, air, and 5000 ppm humidity.

**3.2.4 Charge mobility in MPc-M MOF modulated by CO adsorption**



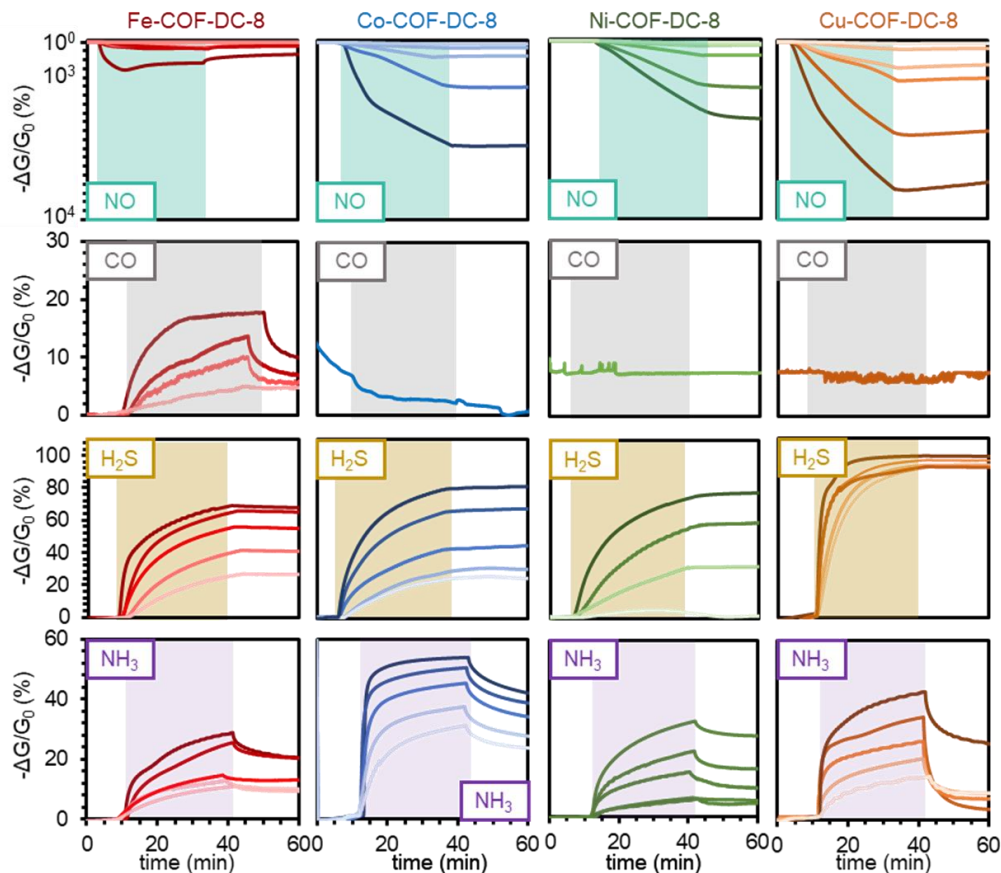
**Figure 34. Progress Towards Tasks 3.2.6 and 3.2.7.** Thin-film transistors (TFT) can probe materials for their charge transport properties, and performance under gated electric fields. NiPc-Pyr COF was drop cast onto TFT gates and the transistor performance was measured. a) optical image of a TFT device with NiPc-Pyr film covering the gate and connecting the Au source to drain. b) The on/off performance of a representative pristine TFT device showing a  $V_g$  on of -2 V. c) after NiPc-Pyr was drop cast on the device the  $V_g$  on shifted to +8 V.

### 3.2.6-7 Thin film transistors (TFT) containing COF films



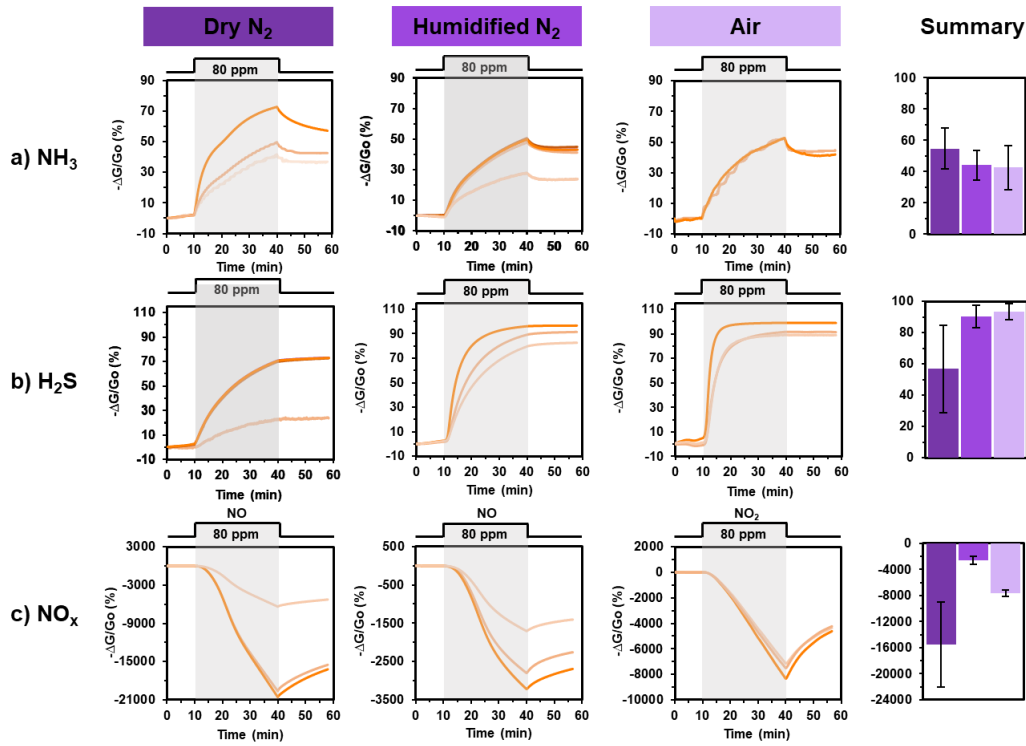
**Figure 35. Progress Towards Task 3.2.8.** Chemiresistive responses of devices with one of four unique MPc-Pyr COFs to (bottom row) NH<sub>3</sub>, (2<sup>nd</sup> from bottom row) H<sub>2</sub>S, (2<sup>nd</sup> from top row) CO, and (top row) NO under an applied voltage of 1.0 V and an atmosphere of dry nitrogen. Metal-dependent responses are observed for gas-MPc combinations. Suspensions of MPc-Pyr COF were drop cast onto prefabricated electrodes and allowed to air dry to form chemiresistive devices. Prefabricated devices were composed of interdigitated gold electrodes on glass with 5 μm spacings between interdigitated electrode fingers.

**3.2.8 NH<sub>3</sub>, H<sub>2</sub>S, CO sensing: 80 ppm**



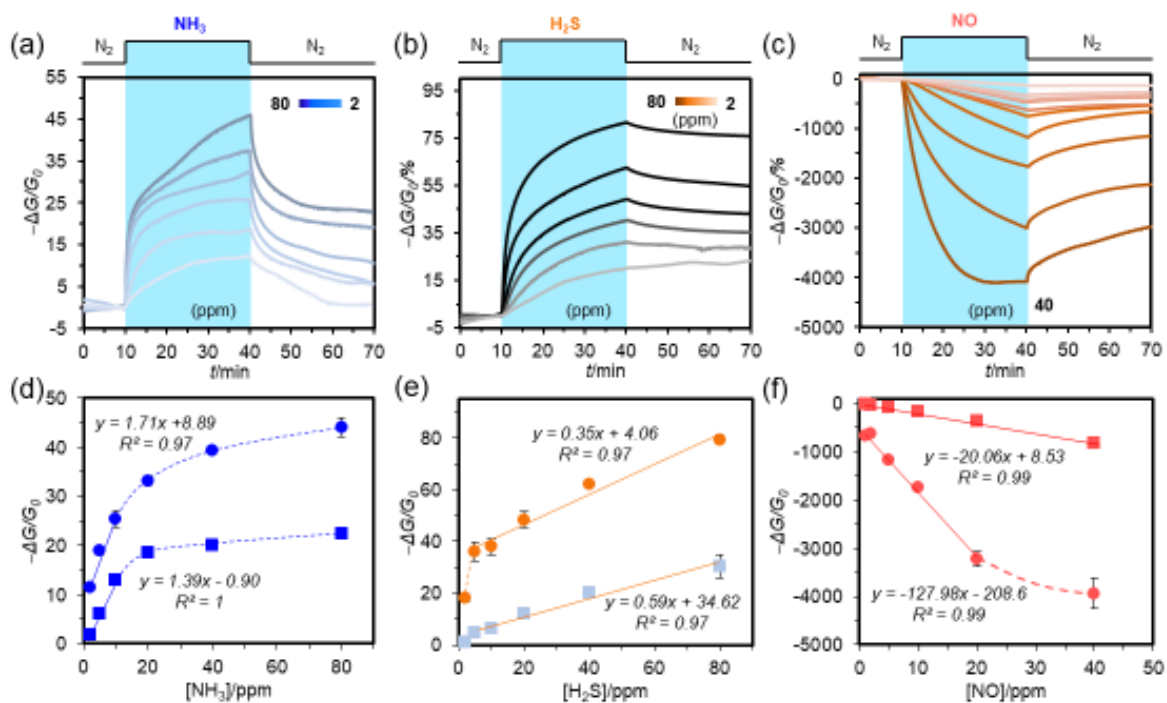
**Figure 36. Progress Towards Task 3.2.8.** Chemiresistive responses of devices with integrated NiPc-Pyr COF to (a) NH<sub>3</sub>, (b) H<sub>2</sub>S, and (c) NO<sub>x</sub>, under an applied voltage of 1.0 V. The carrier gas was changed to determine the sensing performance under dry nitrogen, humid nitrogen, and ambient air. Responses ( $-\Delta G/G_0$ ) of COF-DC-8. Suspensions of MPc-Pyr COF were drop cast onto prefabricated electrodes and allowed to air dry to form chemiresistive devices. Prefabricated devices were composed of interdigitated gold electrodes on glass with 5  $\mu\text{m}$  spacings between interdigitated electrode fingers.

**Chemiresistive Sensing of Gasotransmitters using NiPc-Pyr in Varied Carrier Gas Mixtures**



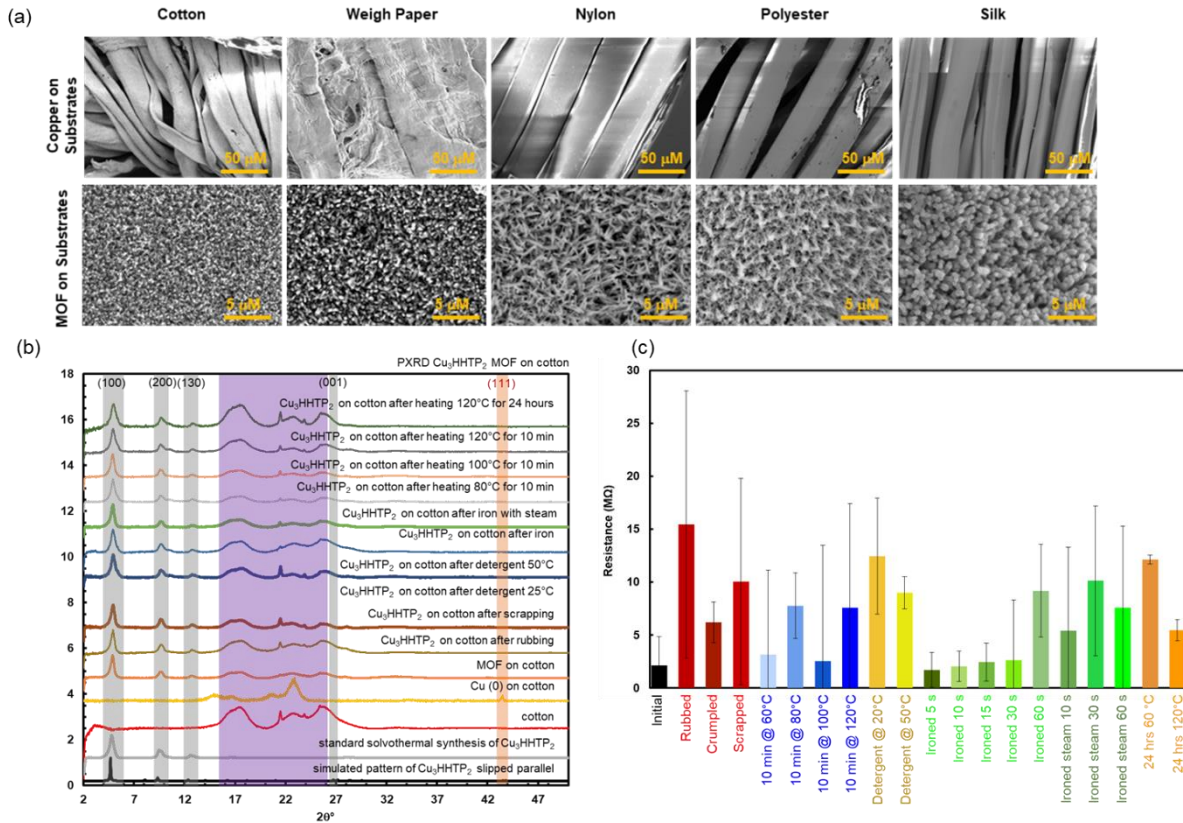
**Figure 37. Progress Towards Task 3.2.8.** Chemiresistive responses of devices with integrated NiPc-Pyr COF to (a)  $\text{NH}_3$ , (b)  $\text{H}_2\text{S}$ , and (c)  $\text{NO}$ , under an applied voltage of 1.0 V and an atmosphere of dry nitrogen. Responses ( $-\Delta G/G_0$ ) of COF-DC-8 after 1.5 min (squares) and 30 min (circles) exposure versus concentration of (d)  $\text{NH}_3$ , (e)  $\text{H}_2\text{S}$ , and (f)  $\text{NO}$ . The LODs for these gases was  $\text{NH}_3$ : 70 ppb,  $\text{H}_2\text{S}$ : 204 ppb, and  $\text{NO}$ : 5 ppb. Suspensions of MPC-Pyr COF were drop cast onto prefabricated electrodes and allowed to air dry to form chemiresistive devices. Prefabricated devices were composed of interdigitated gold electrodes on glass with 5  $\mu\text{m}$  spacings between interdigitated electrode fingers.

### 3.2.8 Charge mobility in NiPc-Pyr COF modulated by gas adsorption



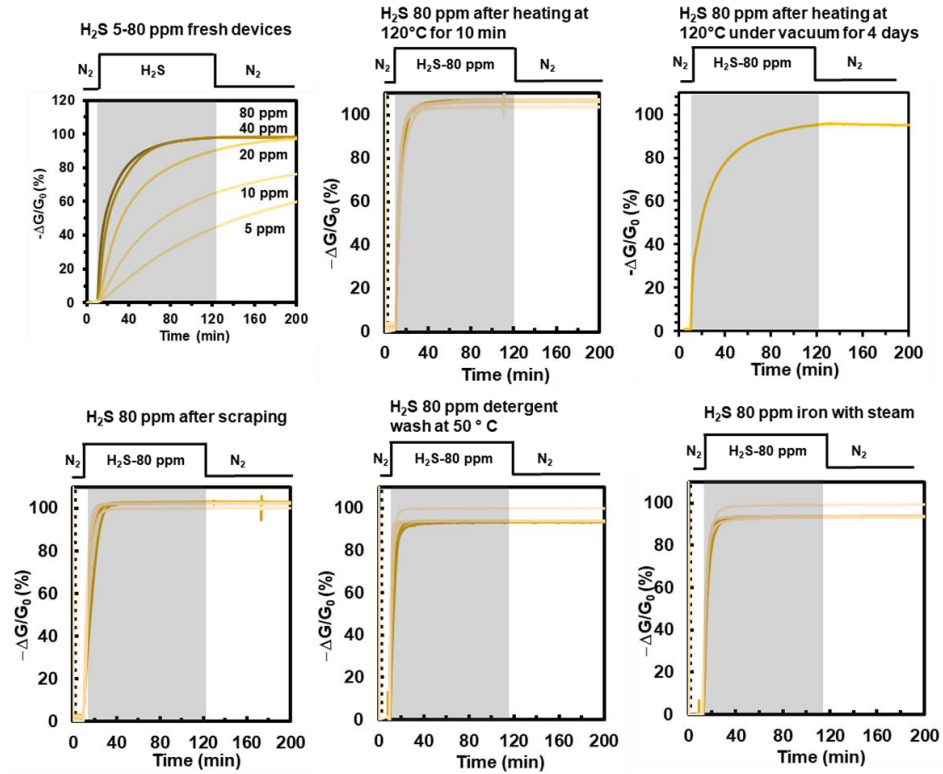
**Figure 38. Contributions from HSAP/URAP.** (a) Images of copper restructured into  $\text{Cu}_3\text{HHTP}_2$  MOF demonstrated on a variety of substrates. (b) Oxidative restructuring enables enhanced stability of the MOF on these substrates even through chemical and physical stability tests by maintaining the signature MOF diffraction peaks. 2-point probe resistance measurements of MOF on cotton before and after each stability test.

**HSAP/URAP: Oxidative restructuring of Cu metal into  $\text{Cu}_3\text{HHTP}_2$  MOFs with enhanced stability on cotton**



**Figure 39. Contributions from HSAP/URAP.** Chemiresistive sensing data using  $\text{Cu}_3\text{HHTP}_2$  on cotton for the detection of  $\text{H}_2\text{S}$  at 80 ppm before and after the stability test. The response of the devices after the stability tests to  $\text{H}_2\text{S}$  after 100 minutes are comparable to the response of the newly fabricated devices.

**HSAP/URAP: Chemiresistive sensing of  $\text{H}_2\text{S}$  with  $\text{Cu}_3\text{HHTP}_2$  MOFs on cotton**



**4) Key outcomes and other achievements.**

- EPR assessment of analyte binding in MOFs and COFs shows perturbations of analytes to spin density of these materials.
  - Synthetic access to novel COFs.
  - Detection of gasotransmitters (NO, CO, H<sub>2</sub>S) using a COF array
  - The choice of materials as electrodes in devices plays a critical in determining device performance (e.g., signal-to-noise ratio, sensitivity).
  - SQUID magnetometry of MOFs shows magnetic ordering of spins as a function of temperature.
  - Both proposed MOFs and COFs exhibit substantial conductivity and chemiresistive response to gaseous analytes at part-per-million and part-per-billion concentrations.
  - 13 papers published in peer reviewed journals acknowledging this grant since it has been awarded in 2017:
1. Ko, M.; Aykanat, A.; Smith, M. K.; **Mirica, K. A.\*** “Drawing Sensors with Ball-Milled Blends of Metal-Organic Frameworks and Graphite” *Sensors* **2017**, *17*, 2192. Invited paper for a special issue “Chemiresistive Sensors: Status and the Future” co-edited with Prof. Timothy M. Swager.
  2. Smith, M.; **Mirica, K. A.\*** “Self-Organized Frameworks on Textiles (SOFT): Conductive Fabrics for Simultaneous Sensing, Capture, and Filtration of Gases” *J. Am. Chem. Soc.* **2017**, *139*, 16759–16767.
  3. Mendecki, L.; **Mirica, K. A.\*** “Conductive Metal–Organic Frameworks as Ion-to-Electron Transducers in Potentiometric Sensors” *ACS Appl. Mater. Interfaces* **2018**, *10*, 19248–19257.
  4. Ko, M.; Mendecki, L.; **Mirica, K. A.\*** “Conductive two-dimensional metal–organic frameworks as multifunctional materials” *Chem. Commun.* **2018**, *54*, 7873-7891.
  5. Meng, Z.; Stolz, R. M.; Mendecki, L.; **Mirica, K. A.\*** “Electrically-Transduced Chemical Sensors Based on Two-Dimensional Nanomaterials” *Chem. Rev.* **2019**, *119*, 478–598.
  6. Meng, Z.; Aykanat, A.; **Mirica, K. A.\*** “Proton Conduction in 2D Aza-Fused Covalent Organic Frameworks” *Chem. Mater.* **2019**, *31*, 819–825.
  7. Meng, Z.; Aykanat, A.; **Mirica, K. A.\*** “Welding Metallophthalocyanines into Bimetallic Molecular Meshes for Ultrasensitive, Low-Power Chemiresistive Detection of Gases” *J. Am. Chem. Soc.* **2019**, *141*, 2046–2053.
  8. Meng, Z.; Stolz, R.; **Mirica, K. A.\*** “Two-Dimensional Chemiresistive Covalent Organic Framework with High Intrinsic Conductivity” *J. Am. Chem. Soc.* **2019**, *141*, 11929–11937.
  9. Aykanat, A.; Meng, Z.; Bendetto, G.; **Mirica, K. A.\*** “Molecular Engineering of Multifunctional Metallophthalocyanine-Containing Framework Materials” *Chem. Mater.* **2020**, *32*, 5372–5409.
  10. Ko, M.; Mendecki, L.; Eagleton, A.; Durbin, C. G.; Stolz, R. M.; Meng, Z.; **Mirica, K. A.\*** “Employing Conductive Metal–Organic Frameworks for Voltammetric Detection of Neurochemicals” *J. Am. Chem. Soc.* **2020**, *142*, 11717–11733
  11. Stolz, R. M.; Mahdavi-Shakib, A.; Frederick, B. G.\*; **Mirica, K. A.\*** “Host–Guest Interactions and Redox Activity in Layered Conductive Metal–Organic Frameworks” *Chem. Mater.* **2020**, *32*, 7639–7652.

12. Meng, Z.; Luo, J.; Li, W.\*; **Mirica, K. A.\*** “Hierarchical Tuning of the Performance of Electrochemical Carbon Dioxide Reduction Using Conductive Two-Dimensional Metallophthalocyanine Based Metal–Organic Frameworks” *J. Am. Chem. Soc.* **2020**, *142*, 21656–21669.
13. Meng, Z.; **Mirica, K. A.\*** “Two-Dimensional d- $\pi$  Conjugated Metal–Organic Framework Based on Hexahydroxytrinaphthylene” *Nano Research* **2021**, *14*, 369–375.
  - 6 patent applications have been filed acknowledging this grant:
    1. “Self-Assembly of Metal-Organic Frameworks on Textiles” **Mirica, K. A.**; Smith, M. K.; U.S. Provisional Patent Application No. 62/489,851 filed on 04/25/2017. U.S. Utility Patent Application No. 15/962,156 filed on 04/25/2018.
    2. “Metal-Organic Frameworks as Ion-to-Electron Transducers and Detectors” **Mirica, K. A.**; Mendecki, L.; U.S. Provisional Patent Application No. 62/544,355 filed on 08/11/2017. PCT Patent Application No. 18/46,510 filed on 08/11/2018.
    3. “Formation of Metal-Organic Frameworks” **Mirica, K. A.**; Ko, M.; Mendecki, L. U.S. Provisional Patent Application No. 62/681,436 filed on 06/06/2018. PCT Patent Application No. 19/35,727 filed on 06/06/2019.
    4. “Conductive Metal-Organic Frameworks for Electrochemical Detection of Analytes” **Mirica, K. A.**; Mendecki, L.; Ko, M.; Meng, Z.; Stolz, R. M.; Eagleton, A. U.S. Provisional Patent Application No. 62/688,261 filed on 06/21/2018. PCT Patent Application No. 19/38,492 filed on 06/21/2019.
    5. “Conductive Bimetallic Metal-Organic Frameworks for the Detection of Analytes” **Mirica, K. A.**; Meng, Z.; Aykanat, A. U.S. Provisional Patent Application No. 62/719,264 filed on 08/17/2018. PCT Patent Application No. 19/47,010 filed on 08/19/2019.
    6. “Two-dimensional Stimuli-Response Covalent Organic Frameworks with High Intrinsic Conductivity” **Mirica, K. A.**; Meng, Z.; Stolz, R. M.; U.S. Provisional Patent Application No. 62/837,118 filed on 04/22/2019.