

UNCLASSIFIED

***Review and Assessment of Failure Models
Applicable to Ballistic Impact
Simulations of Ductile Materials***

Sidney Chocron
Charles E. Anderson, Jr.

Southwest Research Institute®
P.O. Drawer 28510, San Antonio, TX 78238

Contract: W56HZV-06-C-0194

SwRI® Report 18.12544/027

Prepared for:

US Army RDECOM-TARDEC
RDTA-RS
Warren, MI 43897-5000

November 2010

UNCLASSIFIED

UNCLASSIFIED

UNCLASSIFIED



REPORT DOCUMENTATION PAGE			<i>Form Approved</i> OMB No. 0704-0188		
Public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching data sources, gathering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggestions for reducing this burden to Washington Headquarters Service, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suite 1204, Arlington, VA 22202-4302, and to the Office of Management and Budget, Paperwork Reduction Project (0704-0188) Washington, DC 20503.					
PLEASE DO NOT RETURN YOUR FORM TO THE ABOVE ADDRESS.					
1. REPORT DATE (DD-MM-YYYY) 09-11-2010		2. REPORT TYPE Technical		3. DATES COVERED (From - To) Oct. 2009 - Oct. 2010	
4. TITLE AND SUBTITLE Review and Assessment of Failure Models Applicable to Ballistic Impact Simulations of Ductile Materials			5a. CONTRACT NUMBER W56HZV-06-C-0194		
			5b. GRANT NUMBER		
			5c. PROGRAM ELEMENT NUMBER		
6. AUTHOR(S) Sidney Chocron and Charles E. Anderson, Jr.			5d. PROJECT NUMBER 18.12544		
			5e. TASK NUMBER		
			5f. WORK UNIT NUMBER		
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) Southwest Research Institute, P.O. Drawer 28510, San Antonio, TX 78238			8. PERFORMING ORGANIZATION REPORT NUMBER 18.12544/027		
9. SPONSORING/MONITORING AGENCY NAME(S) AND ADDRESS(ES) US Army Tank-Automotive Research, Development, and Engineering Center, Warren, MI 48397-5000			10. SPONSOR/MONITOR'S ACRONYM(S) RDECOM-TARDEC/RDTA-RS		
			11. SPONSORING/MONITORING AGENCY REPORT NUMBER		
12. DISTRIBUTION AVAILABILITY STATEMENT Approved for Public Release; Unlimited Distribution					
13. SUPPLEMENTARY NOTES The views, opinion, and/or findings contained in this report are those of the authors and should not be construed as an official Department of the Army position, policy, or decision, unless so designated by other documents.					
14. ABSTRACT A literature review was conducted to evaluate failure models for ductile materials. The focus was on failure models that could be used in numerical simulations of ballistic impact. The assessment examined the number of constitutive parameters required by the model, the types of characterization tests required, and the underlying physics of the model. The Johnson-Cook damage model because of its general acceptance and wide use, was used as a baseline for comparisons. A recommendation is made concerning possible improvement in replicating failure modes observed in ballistic experiments.					
15. SUBJECT TERMS failure models, invariants, numerical simulations, ballistic impact, model constants					
16. SECURITY CLASSIFICATION OF: Unclassified, Unlimited Distribution			17. LIMITATION OF ABSTRACT None	18. NUMBER OF PAGES 26	19a. NAME OF RESPONSIBLE PERSON Frederick C. Rickert
a. REPORT Unlimited	b. ABSTRACT Unlimited	c. THIS PAGE Unlimited			19b. TELEPHONE NUMBER (Include area code) 586-282-3914

UNCLASSIFIED



UNCLASSIFIED

Table of Contents

	Page
1.0 Introduction.....	1
2.0 Summary of Models.....	3
3.0 Evaluation of Models.....	9
4.0 Recommendations.....	13
5.0 Acknowledgements.....	15
6.0 References.....	17
Appendix A: Invariants.....	19

UNCLASSIFIED



UNCLASSIFIED

List of Tables

	Page
Table 1. Essential Features of Failure Models Reviewed	5
Table 2. Model Constants, Characterization Tests Required, and Availability	7
Table 3. Specimen Geometries	8
Table 4. Model Evaluation Relative to Johnson-Cook Failure Model	12

UNCLASSIFIED



UNCLASSIFIED

1.0 Introduction

A literature review has been conducted to evaluate failure models for ductile materials. The focus was on failure models that could be used in numerical simulations of ballistic impact. The review is by no means exhaustive with regard to the extensive failure model field, but was limited to models that showed promise in ballistic applications, and in particular, might be able to reproduce ballistic limits and failure patterns. Simple criteria like maximum principal stress or strain were omitted since they provide accurate failure patterns only for very specific examples. Similarly, very "sophisticated" failure models that have many "fundamental" parameters that are difficult to ascertain (and which tend to be data-fit parameters) were not included. Rather, the intent was to examine models that have material parameters that can be obtained/measured in fundamental laboratory characterization experiments.

The Johnson-Cook (J-C) failure model [1] was selected as the reference model for this study owing to its widespread use and the fact that constants have been determined for many materials. Additionally, the laboratory tests for obtaining the material model constants are relatively straightforward. Consequently all the other models are evaluated relative to the J-C model in terms of ease of numerical implementation, ability and ease of obtaining material model constants, etc.

Models not widely used in the community but that, from the authors' perspective, could be considered as appropriate candidates for implementation in hydrocodes are also discussed. The primary objective of this report is to summarize succinctly the models and relevant fundamental features, the model parameters and the experiments required to determine the parameters, and maturity of the model for ballistic applications. Then, we conclude with a recommendation on a path forward that might lead to more accurate modeling of failure processes in ballistic applications.

UNCLASSIFIED



UNCLASSIFIED

2.0 Summary of Models

Essential features or characteristics of the failure models examined are summarized in the tables below. Table 1 provides an overall description of the failure models reviewed, and theoretical features that are implicitly or explicitly included into the model framework. Brief descriptions of the columns of the table are provided below.

- Column 1. *Model*: The first column lists the name of author or authors of the model together with the bibliographic reference for the model.
- Column 2. *Year*: This column lists the year the model was published.
- Column 3. *Model Equations*: A succinct summary of the governing equations is provided. It is not the objective to describe each model in detail; readers are referred to the respective publications. However, it was felt that a succinct summary of the equations highlights the model constants that need to be determined. Examination of the governing equations also shows the commonality and/or relationships to other models.
- Column 4. *Soft*: Some failure models like J-C use a damage parameter D as a “switch” to suddenly fail material (a computational element) when $D = 1.0$, but without changing material properties while damage is evolving. Failure of a computational element means that the element no longer can support shear or tension, but it can still support compression through the equation of state. Other failure models couple damage with the plasticity model, leading to softening (i.e., decrease in flow stress) as damage evolves. A “Y” in this column means that the flow surface of the model softens as damage increases. Although damage softening might be physically realistic, it adds constants to the calibration process, adding complexity to both the experimental and numerical procedure. Further, while damage softening may be applicable/important in creep (for example), some researchers believe it is minor importance in very rapid loading events such as ballistic impact.
- Column 5. *Triaxiality*: In isotropic homogeneous materials, both the plasticity and failure models should only depend on the three invariants of the stress tensor¹ (I_1 , I_2 , and I_3). In metal plasticity a further simplification is to assume that flow only depends on J_2 , an invariant of the deviatoric stress tensor. It is commonly accepted in failure models, since Bridgman's [13] pioneering studies, that the larger the triaxiality:

$$\sigma^* = \frac{I_1/3}{\sqrt{3J_2}} = \frac{-P}{\sigma_{eff}} \quad (1)$$

the smaller the strain to failure of a material; or in other words, the larger the pressure applied, the larger the strain required to fail the

¹ The Appendix provides a brief synopsis of the invariants of the stress and stress deviator tensors.

material. Models where failure does not depend on triaxiality have a "No" in this column.

Column 6. J_3 : J_3 is another invariant of the deviatoric stress tensor. Recently, some papers analyze carefully how J_3 (or, equivalently, the Lode angle) affects the strain to failure. These models are identified with a "Y" in this column.

Column 7. *Comments*: These are comments by the authors of this report that highlight significant features or distinguishing characteristics of the model.

Table 2 focuses on the model constants that must be determined. A brief explanation of the information in each of the columns of Table 2 is provided below. The models are listed in chronological order, i.e., the order is the same as for Table 1.

Column 1. *Model*: Model name, from Table 1.

Column 2. *Constants*: The constants needed by the failure model.

Column 3. *Characterization Tests*: These are the tests needed to calibrate the constants of the model. A model with many constants does not necessarily mean that many tests are required. For example the Gurson-Tveergard-Needleman (GTN) model has 8 constants and only one type of test is required for calibration. When evaluating the model the preference is for a small number of characterization tests, i.e., a relatively inexpensive procedure for determining the constitutive parameters. The symbols, designating various types of tests, are defined in Table 3.

Column 4. *Hydrocodes*: Four codes were considered in this column: AUTODYN (now ANSYS), CTH, EPIC and LS-DYNA. If the model is implemented in the code, it is listed in this row. Additionally, for LS-DYNA, the corresponding material number is listed.

Table 1. Essential Features of Failure Models Reviewed

Model	Year	Model Equations	Soft. ²	Tri-axiality	J ₃	Comments
Cockcroft-Latham [2]	1968	$W = \int \langle \sigma_I \rangle d\epsilon_{eq} \leq W_{crit}$	N	N	N	Essentially a plastic work (energy) criterion.
Tuler-Butcher [3]	1968	$D = \int (\sigma - \sigma_o)^n dt > D_{crit}$	N	N	N	Energy per unit area criterion; developed for spall.
Gurson [4]	1975	$\phi = \frac{\sigma_{eq}^2}{Y^2} + 2f \cosh\left(\frac{\sigma_{kk}}{2Y}\right) - 1 - f^2$ $f = \text{void volume fraction}$	Y	Y	N	Evolution of ductile voids that reduce the strength of the yield surface; void growth occurs in tension. Coalescence of voids not realistic.
Wilkins [5]	1980	$D_p = \int w_1 w_2 d\epsilon_{eq}^p \geq D_c, w_1 = \left(\frac{1}{1+aP}\right)^\alpha; w_2 = \left[2 - \max\left(\frac{s_2}{s_3}, \frac{s_2}{s_1}\right)\right]^{-\beta}$	N	Y	Y	Damage as a function of hydrostatic pressure and shear stress state.
Johnson-Cook [1]	1983	$\epsilon_f = (D_1 + D_2 e^{D_3 \sigma^*})(1 + D_4 \ln \dot{\epsilon})(1 + D_5 T^*)$	N	Y	N	Critical strain for failure as a function of hydrostatic pressure (triaxiality), strain rate, and temperature.
Gurson-Tvergaard-Needleman (GTN) [6]	1984	$\phi = \frac{\sigma_{eq}^2}{Y^2} + 2f^* q_1 \cosh\left(\frac{\sigma_{kk}}{2Y}\right) - \left\{1 + (q_1 f^*)^2\right\}$ $f^* = f \text{ if } f \leq f_c; f^* = f_c + K(f - f_c) \text{ if } f > f_c, K = (f_U - f_c)/(f_f - f_c)$ f_U ultimate void vol. fraction, f_c critical, f_f failure	Y	Y	N	Improved Gurson void growth model.
Borvik, et al. [7]	2001	$\epsilon_p = [D_1 + D_2 e^{D_3 \sigma^*}]^{D_4} [1 + \dot{\epsilon}_p]^{D_5} [1 + D_5 T^*]$	Y	Y	N	Similar to the J-C model, but with a different strain rate dependence.

² This is softening of the yield surface. None of the papers reviewed mentions softening of the stiffness matrix; however, some probably soften the elastic modulus as the material is damaged.

Model	Year	Model Equations	Soft. ²	Tri-axiality	J_3	Comments
Bao-Wierzbicki [8]	2004	Piecewise curve-fit for failure strain as a function of triaxiality	N	Y	Y	Empirical curve fit for the strain to failure vs. triaxiality curve.
Xue-Wierzbicki [9]	2005	$\bar{\epsilon}_f = c_1 e^{-c_2 \eta} - (c_1 e^{-c_2 \eta} - c_3 e^{-c_3 \eta}) (1 - \xi^{1/n})^n$ $\eta \text{ is triaxiality (i.e., } \sigma^*), \xi = \frac{27 J_3}{2 \bar{\sigma}^3}$	N	Y	Y	Failure depends both on triaxiality and Lode angle; also, replaced piecewise curve fit of Ref. [8] with analytic function.
Bai-Wierzbicki [10]	2008	$\hat{\epsilon}_f(\eta, \bar{\theta}) = \left[\frac{1}{2} (D_1 e^{-D_2 \eta} + D_5 e^{-D_5 \eta}) - D_3 e^{-D_4 \eta} \right] \bar{\theta}^2 + \frac{1}{2} (D_1 e^{-D_2 \eta} - D_5 e^{-D_5 \eta}) \bar{\theta} + D_3 e^{-D_4 \eta}$ $\bar{\theta} = 1 - \frac{6\theta}{\pi} \text{ where } \theta \text{ is the Lode angle}$	N	Y	Y	Both plasticity and failure depend on the Lode angle.
Xue-GTN [11]	2008	$\phi = \frac{\sigma_{eq}^2}{Y^2} + 2D \cosh\left(\frac{q_2 \sigma_{kk}}{2Y}\right) - (1 + D^2) = 0$ $D = K_D (q_1 f + \dot{D}_{shear})$	Y	Y	Y	Modification of GTN model to improve shear failure prediction.
Nahshon-Hutchinson [12]	2008	GTN with $f = (1 - f) D_{kk}^p + k_{\omega} f \omega(\bar{\sigma}) \frac{D_{ij}^p}{\sigma_{eq}}$	Y	Y	Y	Modification of GTN model to improve shear failure prediction.

Table 2. Model Constants, Characterization Tests Required, and Availability

Model	Constants	Characterization Tests ³	Availability in Hydrocodes
Cockcroft-Latham	W_{cr}	ST	LS-DYNA-107
Tuler-Butcher	σ_0, n, D_{cr}	Spall	LS-DYNA (Erosion)
Gurson	f_0, f_c, f_f	ST	LS-DYNA-120, CTH ⁴ , EPIC ³
Wilkins	a, α, b, D_o, t_c	ST, EN, FP, FPN, TO, TTO	LS-DYNA-81
Johnson-Cook	D_1, D_2, D_3, D_4, D_5	ST, EN, BN, TO	LS-DYNA-15, CTH, EPIC, AUTODYN
Gurson-Tvergaard-Needleman (GTN)	$q_1=1.5, q_2=1, f_0=0, f_c=0.15, f_f=0.25, f_N, \epsilon_N, S_N$	ST	LS-DYNA-120
Borvik et al.	$D_2, D_3, D_c, \epsilon_{threshold}$	ST, EN, BN,	LS-DYNA-104 & 107
Bao-Wierzbicki	$A, B, C, D_1, D_2, D_3, F, G$	CO, ST, EN, BN, TO	No
Xue-Wierzbicki	C_1, C_2, C_3, C_4, m	ST, EN, BN, TO	No ⁵
Bai-Wierzbicki	$D_1, D_2, D_3, D_4, D_5, D_6$	ST, EN, BN, TO, CO, FN	No
Xue-GTN	$q_1, q_2, q_3, q_4, f_0, f_c, f_f, f_N, \epsilon_N, S_N$	ST, TO	No
Nahshon-Hutchinson	$q_1, q_2, q_3, q_4, f_0, f_c, f_f, f_N, \epsilon_N, S_N, k_{\omega}$	ST, TO	No

³ See Table 3 for definition of the various types of characterization tests.

⁴The TEPLA model has a Gurson flow surface; the TEPLA model is implemented into CTH and EPIC.

⁵According to LSTC implementation into LS- DYNA is ongoing.

Table 3. Specimen Geometries

ST	Smooth Tensile
EN	E-Notch
BN	B-Notch
TO	Torsion
CO	Compression
FN	Flat-Notched
FP	Flat Plate Tension
FPN	Flat Plate Notched Tension
TTO	Tension-Torsion

3.0 Evaluation of Models

Table 4 compares each model with the reference model (J-C) and includes the authors' evaluation of the model. Thus, Table 4 will be used as a "decision" table that can highlight which failure models have potential for further exploration. We emphasize, however, that without actual implementation and thorough validation of each model, it is difficult to evaluate fully a specific model's ability to predict ballistic response accurately. Therefore, some of the evaluations presented below are based on the authors' experience. We acknowledge that until a model is fully implemented, exercised, and validated, there will exist some uncertainty in our evaluation, whether it be negative or favorable. The Recommendations Section will summarize the findings in this section.

An explanation of the information provided in the various columns of Table 4 is given below.

- Column 1. *Model*: Model names, listed in the same order as Tables 1 and 2.
- Column 2. *Validated*: All the models in the table have been validated to some extent. The most common validation consisted of reproducing, typically, tensile and/or shear tests that have been conducted. Since model constants are often adjusted so that the simulation reproduces the test, this is really an examination of self-consistency, and not a true validation. Not all the models have been used in ballistic impact applications. This column summarizes whether ballistic simulations, to the authors' knowledge, have been compared with ballistic test results. An "N" in this column means that no comparison with ballistic data was found in the literature. "Limited" implies that only a few comparisons have been conducted (maybe one or two papers). "Significant" is stated for models that have been used regularly by one or more researchers in the ballistics community. "Extensive" is reserved for the J-C model which has been used in hundreds of ballistic applications by many different researchers and groups worldwide.
- Column 3. *Maturity Level*: This column is a qualitative assessment how often the model has been used within the ballistics community and estimates the "maturity level" of the model with respect to ballistic applications. A model that has never been used will have in this column a "None" qualification. Models with minimal experience in ballistics will have a "Low" qualification. This column relates closely to the previous column.
- Column 4. *Additional Tests*. As mentioned above, the J-C model is the reference model in this report. Assuming that J-C model constants are already available for the material (as well as the data from tests performed to obtain the constants, see Tables 2 and 3), this column provides the number of *additional* characterization tests needed for the new

model. For example, a Gurson model⁶, in principle, only needs a tensile test to calibrate the constants. Since the tensile test is also needed in the J-C model, no additional tests should be required to build a Gurson model for the same material. Hence, the column lists "0" as the number of additional tests.

- Column 5. *Additional Computations.* This column assesses the *additional* computational runs required, given a J-C model of the material, to obtain constitutive constants for a different model. For example, the J-C model requires numerical simulations to estimate triaxiality on smooth-tensile and notched specimen tensile tests. Only the Wilkins [5] and Bai-Wierzbicki [11] models require additional computations to calibrate constants.
- Column 6. *Additional Work.* One of the objectives of this report is to answer the question: How much additional experimental and computational work is required to switch from the J-C model to a different model? This column summarizes the additional work estimated to migrate from a J-C model to a different failure model. The estimate is based on an assessment of additional experimental and numerical work required; thus, this column summarizes the two previous columns. But in addition, this column reflects an estimate of the effort to implement the model in a computer code; that is, write the material subroutine. The work involved in writing the code increases with the complexity of the model.
- Column 7. *Potential.* Failure modes in ballistic problems are very diverse. For example, targets can fail in pure shear (plugging), tensile extension, petalling, spall, compressive failure, etc. Similarly, when testing with an MTS machine in compression or tension, depending on the specific material, a variety of failure patterns can arise. An accurate model should be able to capture the rich patterns of failure modes seen in quasistatic and ballistic tests. This column provides our best estimate on how well the model might be able to replicate failure patterns. This estimate is based on the available papers and also on the fundamental framework of the model. For example, a failure model that depends on J_3 and triaxiality will have a higher likelihood at capturing/replicating more failure modes than a model that depends only on triaxiality. Nevertheless, it is not possible to know if the correct failure patterns are actually captured until the model is implemented into a computer code and tested against experimental data.
- Column 8. *Recommendation for Further Exploration.* The finding in this column is subjective, but it provides our summary recommendation

⁶ It might be assumed that micromechanical examination of void growth and coalescence would be required for the Gurson model and other void growth models. However, there is no indication in the papers that this was done. Rather, void growth and coalescence appear to be calculated from assumptions combined with theoretical considerations.

concerning the model. A "Low" consideration is given for a model that might be difficult to implement into a computer program, and/or requires many characterizations experiments, and/or requires many material constants, and/or cannot reproduce desired failure modes. A "High" consideration would reflect opposite characteristics. The assessment attempts to balance these various characteristics. Essentially, this column provides our assessment of whether it might be worthwhile to explore further a new failure model for ballistic applications.

Table 4. Model Evaluation Relative to Johnson-Cook Failure Model

Model	Validated w/ Ballistic Experiments	Maturity Level in Ballistics	Additional Tests	Additional Comps.	Additional Work to Migrate from J-C	Potential in Reproducing Failure Modes	Recommendation for Further Exploration
Cockcroft-Latham	Limited ⁷	Low	0	None	Low	Low	Low
Tuler-Butcher	N	Low	0	None	Low	Low	Low
Gurson	N	None	0	None	High	Low	Low
Wilkins	Significant	Medium	3	Many	High	High	Medium
Johnson-Cook	Extensive	High	-	-	-	High	-
GTN	N	None	0	None	High	Medium	Low
Borvik, et al.	Significant	Medium	0	None	Medium	High	Medium
Bao-Wierzbicki	Limited	Low	1	None	Medium	High	Medium
Xue-Wierzbicki	N	None	0	Few ⁸	Medium	High	High
Bai-Wierzbicki	N	None	2	Many	High	Very high	Medium
Xue-GTN	N	None	0	None	High	Very high	Medium
Nahshon-Hutchinson	N	None	0	None	High	Very High	Medium

⁷ See Borvik, et al. [16-17].

⁸ Only optimization of parameters

4.0 Recommendations

As mentioned previously, it is difficult to assess the applicability and merit of the models to ballistic impact without actually testing them and comparing them with ballistics data. Nevertheless, from the physics incorporated into the model, it is possible to assess if the model is capable of reproducing certain types of failure patterns. For example, if the plasticity and failure surfaces are von Mises, i.e., they depend only on the second invariant of the stress deviator tensor, J_2 , the model will not be able to reproduce slanted failure surfaces in simple compression or tensile tests. The reason is that a von Mises surface, which is a circle in the principal stress plane, has no preferential direction for plastic deformation or failure. However, a characteristic failure angle can be predicted if the plasticity or failure model depends on the Lode angle (i.e., the third invariant J_3).

The physics of the model, or in other words, its capability to capture failure modes, was one of the primary decision parameters for the recommendation provided in the last column of Table 4. With respect to two (or more models) that seem to incorporate similar physics, preference was given to the simpler model.

Finally, when providing our recommendation, the “maturity” level of the model as well as any data comparing the model with actual ballistics results were also taken into account. A model is expected to work better if many groups in the ballistics community have used it successfully to reproduce different ballistic tests.

There is only one model, Xue-Wierzbicki [9], among the models studied in this report that receives a “High” recommendation to be implemented and tried in hydrocodes. Actually the recommendation is to implement a slight variation of the model that will be explained below. Even though the model has not been validated with ballistics tests and hence has no “maturity,” it is thought that the potential of this model in the ballistics community is high for the following reasons:

1. The model is simple since it only introduces a Lode angle dependence on the failure model.
2. The model does not soften the yield surface with damage (that is, the plasticity model and the failure model are not explicitly coupled)⁹; this makes the model simpler with respect to numerical implementation and limits the number of parameters to be determined experimentally.
3. The model can satisfactorily reproduce the experimental data of strain to failure versus triaxiality, as well as differences in strain to failure as a function of stress state. Specifically, the model is able to capture the fact that torsion specimens often fail at smaller strains than smooth tensile specimens. The J-C model cannot replicate this observation.
4. Given the J-C plasticity and failure constants for a material, it is possible to construct readily a Xue-Wierzbicki model from the same set of tests and constants.

⁹ Damage development can depend upon the stress state—for example, the strain to failure in the Johnson-Cook damage model depends on the triaxiality—but explicit coupling of the failure and plasticity models is where damage development softens the yield surface, which then can change the rate of damage evolution.

5. The Xue-Wierzbicki model was successfully implemented by Lidén [14] to model the fracture of a long rod impacting oblique steel plates.¹⁰

Although the recommendation is to implement the Xue-Wierzbicki (X-W) model, preliminary computations show that incorporating the Lode angle only on the failure part of the numerical model is not sufficient to capture all the failure patterns seen in the experiments. Consequently, the following modifications of the X-W model are also recommended:

1. Use the J-C strength model as the plasticity model, but modify it to incorporate the Lode angle.
2. X-W calculates material failure constants from the experimental data. We recommend use of the J-C failure model and constants, but modify the failure envelope by the Lode angle function used in the X-W model [9].
3. It is known that the strain hardening response can be different in tension and torsion, see for example Wilkins [5]. Preliminary computations show that taking into account the differences in the hardening responses leads to appropriate failure patterns.

A model based on the recommendations above was implemented as a user subroutine in LS-DYNA. Preliminary results are very promising because the model reproduces failure patterns like cup and cone failure in tensile specimens, and slanted failure in compression specimens. The model also satisfactorily reproduces final length on Taylor anvil tests and failure patterns observed in ballistic impact on aluminum 6061 plates [18].

¹⁰ Although this is a ballistic test, to the authors' knowledge, it is the only example of the application of the Xue-Wierzbicki model to ballistics. Further, the application was solely to the projectile, and not the target. For this reason the Xue-Wierzbicki model maturity was graded as "None".

5.0 Acknowledgements

The authors would like to thank Borja Erice from the Polytechnic University of Madrid and Ken Nahshon from the Naval Surface Warfare Center Carderock Division of the US Navy for very helpful comments and discussions.



6.0 References

1. G. R. Johnson and W. H. Cook. "Fracture characteristics of three metals subjected to various strains, strain rates, temperatures and pressures," *Engng. Fract. Mech.*, **21**(1): 31-48, 1985.
2. M. G. Cockcroft and D. L. Latham, "Ductility and workability of metals, *J. Inst. Metals*, **96**: 33-39, 1968.
3. F. R. Tuler and B. M. Butcher, "A criterion for the time dependence of dynamic fracture," *Int. J. Fract. Mech.*, **4**(4): 431-437, 1968.
4. A. L. Gurson, "Continuum theory of ductile rupture by void nucleation and growth: Part I - Yield criteria and flow rules for porous ductile media, *ASME J. of Engng. Mat. & Tech.*, **99**: 2-15, 1977.
5. M. L. Wilkins, R. D. Streit, and J. E. Reaugh, "Cumulative-strain-damage model of ductile fracture: Simulation and prediction of engineering fracture tests," Technical Report UCRL-53058, Lawrence Livermore National Laboratories, Livermore, CA, 1980.
6. V. Tvergaard and A. Needleman, "Analysis of the cup-cone fracture in a round tensile bar," *Acta Metall.*, **32**(1): 157-169, 1984.
7. T. Borvik, O. S. Hopperstad, T. Berstad, and M. Langseth, "A computational model of viscoplasticity and ductile damage for impact and penetration," *Eur. J. Mech A/Solids*, **20**: 685-712, 2001.
8. Y. Bao and T. Wierzbicki, "On fracture locus in the equivalent strain and stress triaxiality space," *Int. J. of Mech. Sci.*, **46**: 81-98, 2004.
9. T. Wierzbicki, Y. Bao, Y. W. Lee, and Y. Bai, "Calibration and evaluation of seven fracture models," *Int. J. Mech. Sci.*, **47**: 719-743, 2005.
10. Y. Bai and T. Wierzbicki, "A new model of metal plasticity and fracture with pressure and Lode dependence," *Int. J. Plast.*, **24**: 1071-1096, 2008.
11. L. Xue, "Constitutive modeling of void shearing effect in ductile fracture of porous materials," *Eng. Frac. Mech.*, **75**: 3343-3366, 2008.
12. K. Nahshon and J. W. Hutchinson, "Modification of the Gurson model for shear failure," *Eur. J. Mech. A/Solids*, **27**: 1-17, 2008.
13. P. W. Bridgman, *Studies in Large Plastic Flow and Fracture*, (Robert F. Mehl, Consulting Editor), McGraw-Hill, NY, 1952. .
14. E. Lidén, "How to model fracture behaviour in long rod projectile," in *Proc. 24th Int. Symp. Ballistics* (Eds: S. Bless and J. Walker), **2**: 912- 919, DEStech Publications, Inc., Lancaster, PA, 2008.
15. A. S. Khan and S. Huang, *Continuum Theory of Plasticity*, John Wiley & Sons, Inc., NY, 1995.
16. T. Borvik, S. Dey, and A.H. Clausen, "Perforation resistance of five different high-strength steel plates subjected to small-arms projectiles," *Int. J. Impact Engng.*, **36**: 948-964, 2009.
17. T. Borvik, O. S. Hopperstad, and K. O. Pedersen, "Quasi-brittle fracture during structural impact of AA7075-T651 aluminum plates," *Int. J. Impact Engng.*, **37**: 537-551, 2010.
18. S. Chocron, B. Erice, and C. E. Anderson, Jr., "A new plasticity and failure model for ballistic applications," submitted for publication, 2010.



Appendix: Invariants

The following provides a synopsis of the stress invariants and the stress deviator invariants. Tensor notation is used. The reader is referred to Khan and Huang [15] or similar textbook on plasticity for a more detailed description.

Stress Invariants:

$$I_1 = \sigma_{kk} = \sigma_1 + \sigma_2 + \sigma_3 = -3P \quad (\text{A-1})$$

$$I_2 = \sigma_1\sigma_2 + \sigma_2\sigma_3 + \sigma_3\sigma_1 \quad (\text{A-2})$$

$$I_3 = \sigma_1\sigma_2\sigma_3 \quad (\text{A-3})$$

where σ_1 , σ_2 , and σ_3 are the principal stress, and P is the pressure (negative of the mean stress).

Stress Deviator Invariants:

It is common in plasticity theory to decompose the stress tensor into two parts,

$$\sigma_{ij} = -P\delta_{ij} + s_{ij} \quad (\text{A-4})$$

where s_{ij} is the deviator stress tensor and $P\delta_{ij}$ is the spherical or hydrostatic stress tensor. Then, the stress deviator invariants are (with s_1 , s_2 , and s_3 being the principal values of the stress deviator tensor):

$$J_1 = s_{ii} = s_1 + s_2 + s_3 = 0 \quad (\text{A-5})$$

$$\begin{aligned} J_2 &= \frac{1}{2}s_{ij}s_{ij} = \frac{1}{6}[(\sigma_1 - \sigma_2)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2] \\ &= \frac{1}{2}(s_1^2 + s_2^2 + s_3^2) = \frac{1}{3}(I_1^2 - 3I_2) \end{aligned} \quad (\text{A-6})$$

$$\sigma_{eff} = \sqrt{3J_2} \quad (\text{A-7})$$

where σ_{eff} is the effective (von Mises) stress.

$$\begin{aligned} J_3 &= \det(s_{ij}) = \left(\frac{1}{6}\right)e_{ijk}e_{mnl}s_{im}s_{jn}s_{kl} = s_1s_2s_3 \\ &= \frac{1}{27}(2I_1^3 - 9I_1I_2 + 27I_3) \end{aligned} \quad (\text{A-8})$$

