

AD-A033 253

RIA-76-U665

A

TECHNICAL LIBRARY AD-A033 253

# **CORROSION FATIGUE BEHAVIOR OF COATED 4340 STEEL FOR BLADE RETENTION BOLTS OF THE AH-1G HELICOPTER**

MILTON LEVY and JOSEPH L. MORROSSI  
METALS RESEARCH DIVISION

October 1976

Approved for public release; distribution unlimited.

ARMY MATERIALS AND MECHANICS RESEARCH CENTER  
Watertown, Massachusetts 02172

The findings in this report are not to be construed as an official Department of the Army position, unless so designated by other authorized documents.

Mention of any trade names or manufacturers in this report shall not be construed as advertising nor as an official indorsement or approval of such products or companies by the United States Government.

#### DISPOSITION INSTRUCTIONS

Destroy this report when it is no longer needed.  
Do not return it to the originator.



Block No. 20

## ABSTRACT

The main rotor blade bolt for the 540 helicopter rotor system of the AH-1G, UH-1C, and UH-1M helicopters is proposed to be improved by substituting plasma-sprayed tungsten carbide coating on the outer shank of the 4340 steel bolt for the present cadmium or chromium plate. This study was undertaken to determine the effects of these coating systems on the fatigue behavior of 4340 steel in environments likely to be encountered in service. Both axial tension-tension and rotating bending fatigue testing of coated specimens were carried out in air and in 3.5% NaCl solution. The NaCl environment causes reductions in fatigue strength of bare and coated 4340 steel. The severity of the degradation depends on the coating applied and the type of fatigue test utilized.

## INTRODUCTION

The main rotor blade bolt for the 540 rotor system (AH-1G, UH-1C, UH-1M helicopter) is proposed to be improved by the substitution of plasma-sprayed tungsten carbide (WC) coating on the outer shank for the present cadmium (Cd) or chromium (Cr) plate. The WC coating was originally suggested to reduce the costly machining and plating operations required to recondition blade retention bolts for the AH-1G helicopter. Fretting-induced corrosion resulted in significant surface pitting on the bolts after less than 500 hours of service. Based on prior results with a similar configuration bolt for the 240 rotor system, a WC-coated bolt is expected to last over 3000 hours as compared to 300 hours for current Cd-plated new production bolts and about 500 hours for Cr-plated reworked bolts. A cost savings in excess of \$100,000 per year has been projected if bolt reconditioning and replacement were eliminated.

Although WC-coated bolts for the 240 rotor system have not experienced any fatigue problems, the U.S. Army Aviation Systems Command Value Engineering decided that coating effects on fatigue life should be investigated prior to approval of the WC coating for the following reasons: (1) the blade retention mechanism is subjected to fatigue conditions due to steady centrifugal loading; (2) stress analysis of the blade retention bolt provided no assurance that the part was not fatigue critical; (3) available data on the effect of the WC coating on the fatigue strength of 4340 steel was inadequate;<sup>1,2</sup> (4) the need for utilizing the Coricone sealer in conjunction with the WC coating for enhanced corrosion resistance had not been confirmed. This study was undertaken to determine the effects of these coating systems (WC, Cd, Cr) on the fatigue behavior of 4340 steel in environments likely to be encountered in service. In addition, the efficacy of the Coricone sealant (for enhanced corrosion resistance) in combination with the WC and the solid film lubricant was also determined. Full-scale axial fatigue tests of the coated blade retention bolts were carried out by AMRDL-Langley Research Center, Va., and the data obtained are reported elsewhere.<sup>3</sup>

## MATERIALS

The substrate alloy (bolt material) was VAR 4340 steel. Axial tension-tension and rotating beam fatigue specimens were rough machined (turned) and heat treated according to the following schedule: normalized at 1650 F for 1 hour; air cooled; austenitized at 1525 F for 1 hour; oil quenched (130 to 170 F); tempered at 900 F for 4 hours directly from the oil quench before reaching room temperature. The ultimate tensile strength of the alloy was 194 ksi. The fatigue specimens were subsequently finish machined (turned and polished to 8-16 rms finish), shot peened, and the following coating systems applied:

1. Cd plating plus chromate treatment
2. Cr plating plus solid film lubricant (SFL)

1. VIGLIONE, J., JANKOWSKY, E. J., and KETCHAM, S. J. *Effects of Metallic Coatings on the Fatigue Properties of High Strength Steels*. Materials Protection and Performance, v. 11, March 1972, p. 31-36.
2. LEVY, M., and MORROSSI, J. *Erosion and Fatigue Behavior of Coated Titanium Alloys for Gas Turbine Engine Compressor Applications*. Army Materials and Mechanics Research Center, AMMRC TR 76-4, February 1976.
3. LEVY, M., and SWINDLEHURST, C. E., Jr. *Evaluation of Tungsten Carbide Coated Blade Retention Bolts for the AH-1G Helicopter*. WASA TM in process.

3. Plasma-sprayed WC plus solid film lubricant with Coricone and
4. Plasma-sprayed WC plus solid film lubricant without Coricone.

Detailed coating procedures are shown in Table 1. Coating thickness requirements are contained in Table 2. The coated test specimens were fatigue tested (both axial tension-tension and rotating bending) in air and 3.5% sodium chloride solution (to simulate marine atmosphere). Fatigue testing of the bare 4340 alloy was also carried out to obtain base-line data.

## EXPERIMENTAL PROCEDURES

### Fatigue Tests

Rotating Bending Fatigue: Stress versus cycles-to-failure studies of smooth fatigue specimens were carried out using a Krouse rotating bending fatigue

Table 1. COATING PROCEDURES

a. Cadmium Plating (0.0003-0.00049 in. Cd)	<ol style="list-style-type: none"> <li>1. Degrease and rinse in flowing water</li> <li>2. Cyanide dip 30 to 60 sec at RT</li> <li>3. Cd plate at 20 amps/sq ft</li> <li>4. Rinse in water</li> <li>5. Bake at 385 F, minimum 23 hours (within 1 hour of plating)</li> <li>6. Degrease</li> <li>7. Cyanide reactivation - 5 to 10 sec at RT</li> <li>8. Rinse in water</li> <li>9. 15 to 20 sec in chromate conversion bath</li> <li>10. Rinse in water and dry</li> </ol>
b. Chromium Plating (2-2.5 mil Cr)	<ol style="list-style-type: none"> <li>1. Degrease and rinse in flowing water</li> <li>2. Reverse etch in chromium plating bath (131 F)</li> <li>3. Cr plate 131 F</li> <li>4. Rinse</li> <li>5. Bake at 375 ± 25 F for minimum of 3 hours</li> <li>6. Dip plated samples into solid film lubricant (MIL-L-46010)</li> <li>7. Air dry for 30 minutes at RT</li> <li>8. Cure at 400 F for minimum of 1 hour (SFL thickness 0.35 mil)</li> </ol>
c. Plasma-Sprayed Tungsten Carbide Plating	<ol style="list-style-type: none"> <li>1. Grit blast fatigue specimens according to MIL-A-21380B</li> <li>2. Plasma spray WC-Co (METCO 72F-NS) 7 to 9 mil per side</li> <li>3. Surface grind to 3.25 to 4.0 mil per side</li> <li>4. Apply Coricone 1700 sealant by spraying - 0.2 mil per side (specimens were prepared with and without Coricone)</li> <li>5. Dry at RT</li> <li>6. Cure at 350 F for at least 20 minutes</li> <li>7. Dip specimens into SFL (0.35 mil per side)</li> <li>8. Air dry for 30 minutes at RT</li> <li>9. Cure at 400 F for minimum of 1 hour</li> </ol>

Table 2. THICKNESS OF COATINGS

Cadmium	0.3 - 0.5 mil/side
Chromium	2.0 - 2.5 mil/side
Tungsten Carbide (After Finish Grind)	3.25 - 4.0 mil/side
Coricone Sealer	0.2 mil/side
Dry Film Lubricant	0.35 mil/side

machine. The fatigue apparatus was operated at 3000 rpm, which is equivalent to a cyclic stress reversal frequency of 50 Hertz and a stress ratio  $R = -1$ . The standard Krouse fatigue machine was modified by the addition of a Lucite environmental chamber for corrosion fatigue studies, as shown schematically in Figure 1. A Schaar Sigmamotor tubing pump delivered the test fluid through Tygon tubing from a two-liter reservoir at the rate of 15 liters per hour. The pulsations of the pumping system were removed in a settling chamber, so that a steady stream of fluid would flow over the test section of the fatigue specimen. The fluid was then returned to the reservoir by gravity flow. At speeds up to 3000 rpm, the fluid maintained good contact with the rotating specimen as it flowed over and around the test section.

Rotating bending fatigue specimens measured 1/2" diameter by 5" long, with a 2" radius reduced section giving a minimum cross section of 1/4" diameter at the specimen center. Stressing of the specimen was accomplished using a sliding weight to apply a bending moment. A counter kept track of the number of stress cycles, and a limit switch shut off the machine at specimen failure. Run-out cycle count was  $10^7$  cycles.

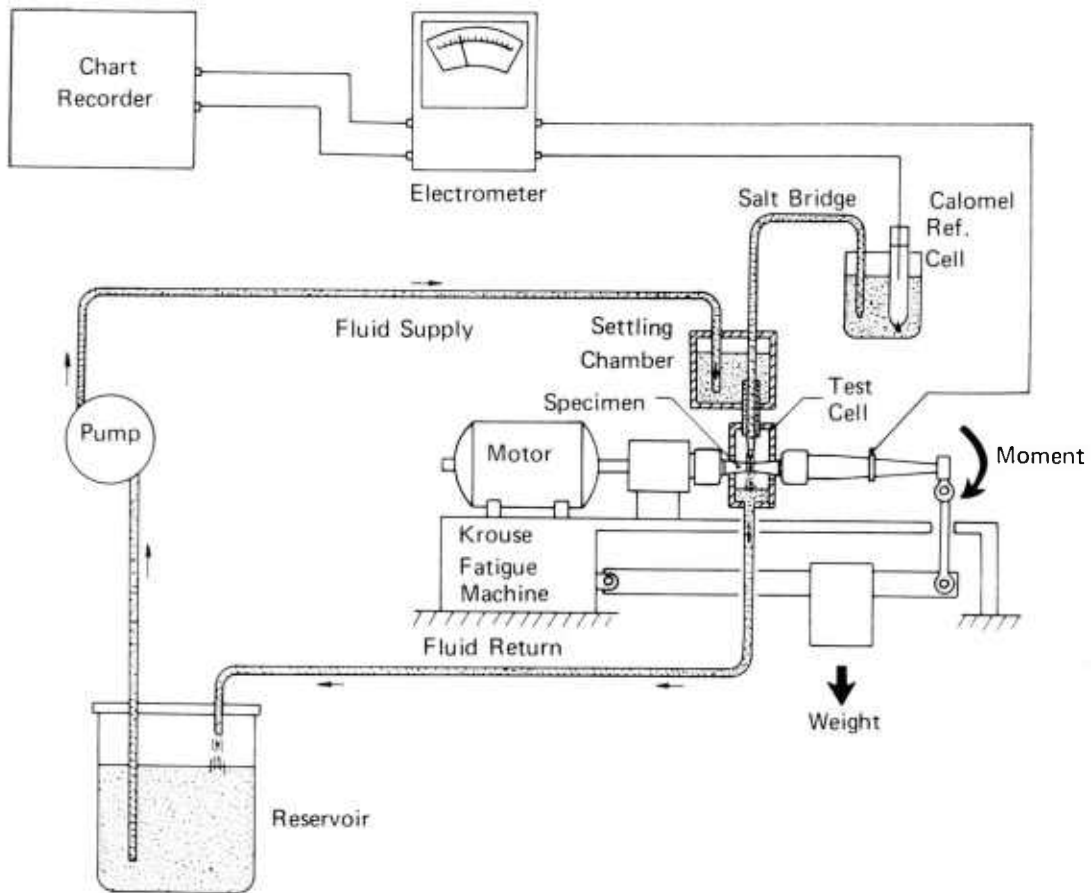


Figure 1. Schematic for corrosion fatigue apparatus.

## Axial Tension-Tension Fatigue Test

Fatigue testing of axial tension-tension smooth fatigue specimens was carried out using the Instron Dynamic Cycler Model 1211 system and sinusoidal loading. The apparatus was operated at 2000 rpm which is equivalent to a cyclic frequency of 33 Hertz and a stress ratio  $R = 0.8$ . For tests in 3.5% NaCl solution, a plastic cell containing this environment was attached to completely surround the gage length of the specimen. Unlike the corrosion cell of the rotating bending fatigue apparatus, the solution was quiescent. Axial tension fatigue specimens measured 1/2" diameter by 3" long, with a 1-1/2" radius reduced section giving a minimum cross section of 0.200" diameter at the specimen center.

## RESULTS

### Rotating Bending Fatigue

Figure 2a shows the deleterious effect of NaCl solution on the fatigue life of bare 4340 steel. The fatigue strength (value at  $10^7$  cycles) of the bare specimen decreases from 105 to 20 ksi, a reduction of 81%. Figure 2b contains S-N curves for the 4340 steel coated with electroplated cadmium plus chromate treatment. The air value of the bare material is identical to that of the coated material (105 ksi). Parallel to data for steels in general, the fatigue limit of the bare alloy in air is approximately one half the tensile strength. The NaCl solution reduced the fatigue strength of the coated alloy by 24% (from 105 ksi to 80 ksi). If we compare the fatigue strength of the coated alloy in NaCl solution with that of the bare alloy in the same environment, it is apparent that the cadmium plus chromate treatment significantly improves the fatigue strength of the alloy in NaCl solution (from 20 to 80 ksi). Figure 2c shows that 3.5% NaCl solution caused a 5.3% reduction in the fatigue strength of 4340 steel coated with electroplated chromium plus solid film lubricant (from 95 to 90 ksi). This coating system produced a 9.5% degradation in the fatigue life of the bare steel (Table 3) in air. The fatigue strength of the alloy coated with plasma-sprayed tungsten carbide plus solid film lubricant with or without the Coricone sealant was 90 ksi regardless of the environment (Figure 2d, Table 3), i.e., there was neither a deleterious effect of environment nor a beneficial effect of the Coricone sealant. A fatigue reduction of 14.3% was attributed to the coating system.

### Axial Tension-Tension Fatigue

Figure 3 contains S-N curves obtained in air for the alloy, bare and coated. Fatigue data at  $10^7$  cycles showed that the cadmium and chromium electroplates, particularly the chromium, improved the fatigue strength of the bare alloy. The tungsten carbide coating reduced fatigue strength from 160 to 140 ksi which represented a fatigue reduction of 12.5%. The Coricone sealant had no effect on fatigue strength of the alloy in air. Table 3 contains fatigue data at  $10^7$  cycles for the uncoated and coated alloy in 3.5% NaCl solution. This environment caused a 31.2% reduction in the fatigue strength of the bare alloy and a 45 to 60% reduction in the fatigue strength of the WC-coated alloy and the Cr-plated alloy. The Coricone again provided no beneficial effect on the corrosion fatigue resistance of the WC-coated alloy. The Cd-plated alloy was unaffected by the

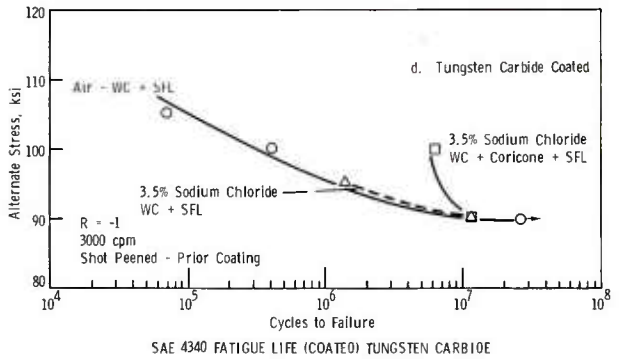
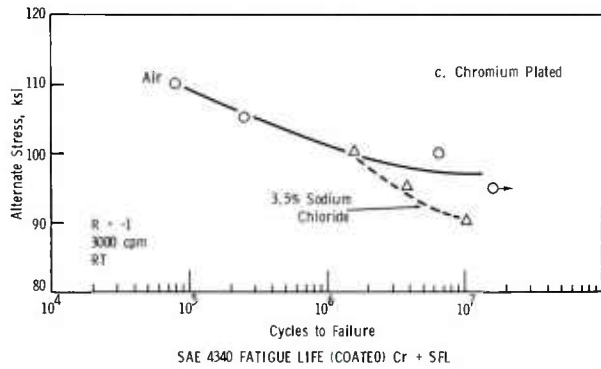
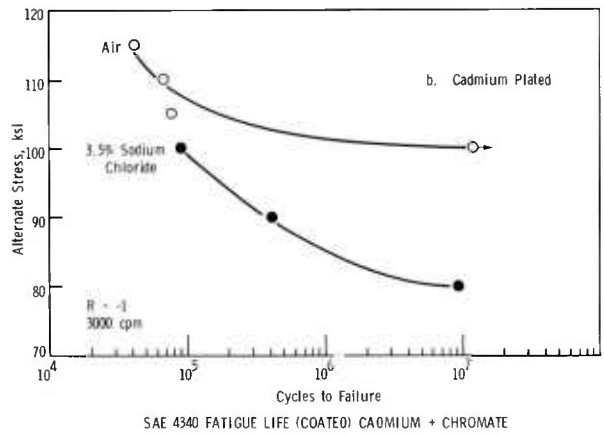
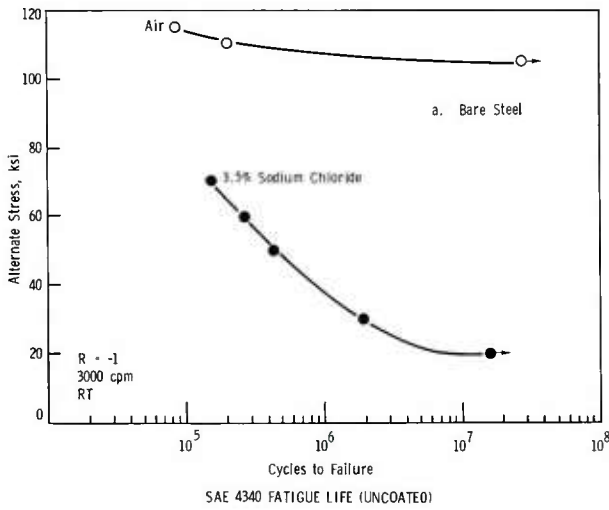


Figure 2. S-N curves (rotating bending) of 4340 steel showing effect of environment.

Table 3. EFFECTS OF COATINGS AND ENVIRONMENT ON THE FATIGUE STRENGTH OF 4340 STEEL (UTS 180 TO 200 KSI)

Test	Condition	Air		3.5% NaCl	
		Stress, ksi	Reduction, %	Stress, ksi	Reduction, %
Rotating Bending <sup>1</sup>	Bare	105	-	20	-81
	Cd + Chromate	105	-	80	-24
	Cr + Dry Film*	95	-9.5	90	-14.3
	WC + Dry Film*	90	-14.3	90	-14.3
	WC + Coricone + Dry Film*	90	-14.3	90	-14.3
Axial Tension <sup>2</sup>	Bare	160	-	110	31.2
	Cd + Chromate	165	+3.1	165	0
	Cr + SFL*	175	+9.4	90	43.8†
	WC + SFL*	140	-12.5	60	48.6‡
	WC + Coricone + SFL*	140	-12.5	60	62.5‡

\*Shot peened

†Compared to bare alloy air value

‡Compared to coated alloy air value

1. R = -1, 3000 cpm; fatigue life 10<sup>7</sup> cycles

2. R = 0.8, 2000 cpm; min. stress/max. stress

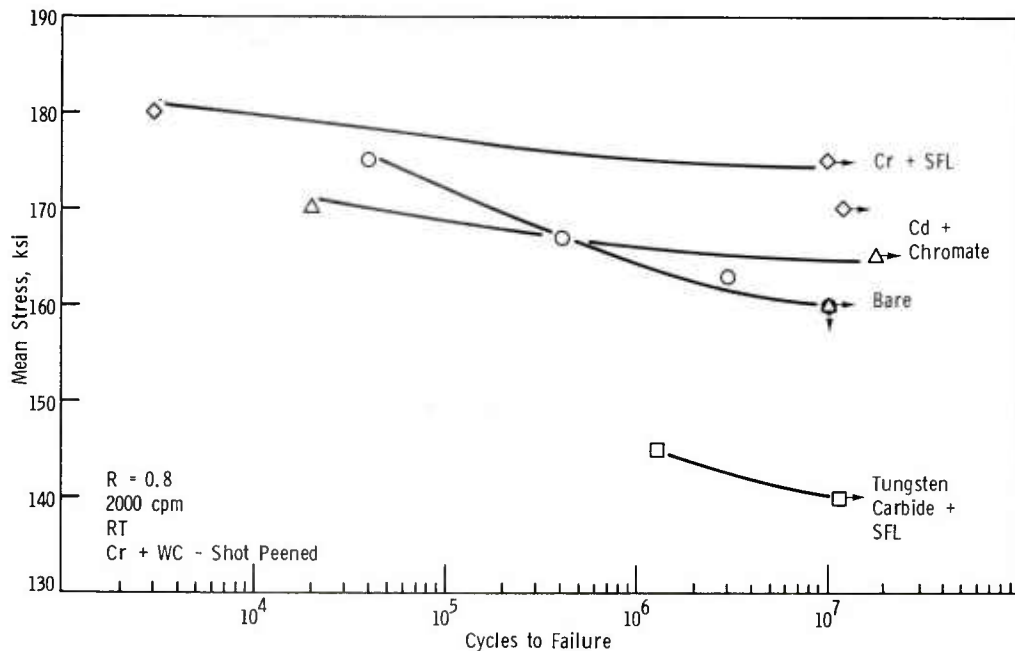
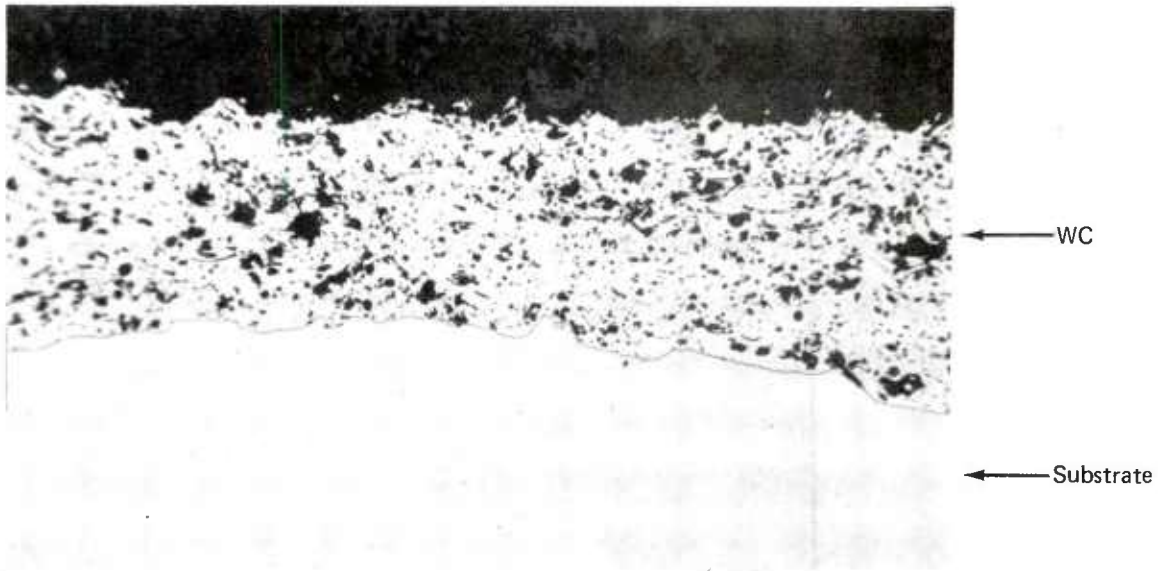


Figure 3. Bare and coated SAE 4340 axial tension fatigue life air-media.

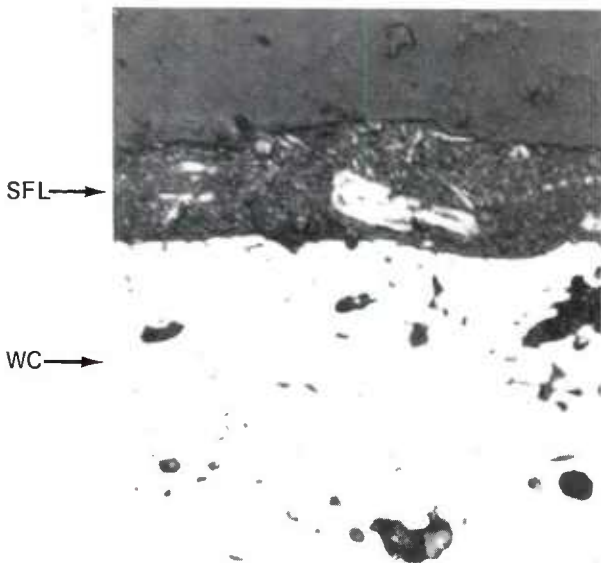
NaCl solution. It is evident that the axial tension fatigue data differs from the rotating bending fatigue data in the following manner: (a) degradation of fatigue strength of the bare alloy due to NaCl solution is considerably less in axial tension; (b) NaCl solution causes significantly greater reductions in axial tension fatigue strength for both the WC-coated alloy and the Cr-plated alloy. However, based on air values only, fatigue strength reductions due to the coatings were quite similar in both rotating bending and axial tension fatigue tests.

### Metallography

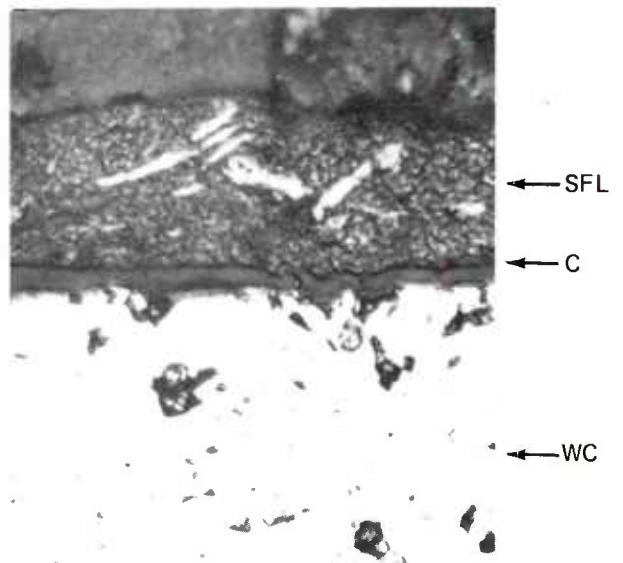
The WC-coated specimens were examined metallographically for an assessment of coating integrity, bonding, and the sealing capability of the solid film lubricant and Coricone. Figure 4 contains micrographs of the cross-sectional area of the coated specimens. Figure 4a demonstrates the good bonding between plasma-sprayed WC and the substrate 4340 steel. Note that some porosity is present and that the pores are discontinuous. Figure 4b shows good bonding between the solid film lubricant and the WC coating. Also shown is the capability of the SFL to fill surface pores present in the WC coating. Figure 4c demonstrates the good bonding that can be obtained between SFL and Coricone and between Coricone and WC coating. Note that the Coricone has filled any surface pores present in the WC coating. Since the pores in the WC coating are discontinuous, neither the SFL nor Coricone has infiltrated below the surface pores.



a. Tungsten carbide coating/substrate interface. Mag. 200X



b. Solid film lubricant/tungsten carbide coating interface. Mag. 1000X



c. Solid film lubricant/Coricone/tungsten carbide interface. Mag. 1000X

Figure 4. Micrographs of cross-sectional area of plasma-sprayed tungsten carbide coating interfaces.

19-066-777/AMC-75

## CONCLUSIONS

1. NaCl solution significantly degrades fatigue strength of the bare 4340 steel. The degradation is much more severe under conditions of rotating bending fatigue.

2. Although the fatigue strength of WC-plus-SFL-coated 4340 steel is unaffected by NaCl solution in rotating bending fatigue, it is significantly reduced by this environment in axial tension fatigue testing. Regardless of the method of testing, the Coricone sealant does not impart additional resistance to corrosion fatigue and the WC coating reduces the air fatigue strength of the 4340 steel by about 14%.

3. NaCl solution further degrades the rotating bending fatigue strength of both Cd- and Cr-plated 4340 steel, but in axial tension fatigue testing, further degradation is limited to the Cr-plated alloy.

4. Regardless of the method of test, the sprayed WC and the plated Cr coatings exhibit comparable fatigue behavior in NaCl solution.

5. Despite the reduced fatigue strength of the WC-coated steel, full-scale fatigue tests of the blade retention bolts (carried out at AMRDL-Langley) produced no failures through the equivalent of four life-times (approximately 14,000 flight hours of loading). Note that the full-scale component fatigue tests were carried out in laboratory air.

6. Rigorous inspection procedures for rotor blade bolts should be followed, particularly after operations in marine environments. It is recommended that inspection of the WC-coated blade retention bolt be made after 250 hours of operation, or at a lesser time if in consonance with inspection of a neighboring component.

## ACKNOWLEDGMENT

This work was supported by the U.S. Army Aviation Systems Command, St. Louis, Missouri.

DISTRIBUTION LIST

No. of Copies	To
1	Office of the Director, Defense Research and Engineering, The Pentagon, Washington, D. C. 20301
12	Commander, Defense Documentation Center, Cameron Station, Building 5, 5010 Duke Street, Alexandria, Virginia 22314
1	Metals and Ceramics Information Center, Battelle Memorial Institute, 505 King Avenue, Columbus, Ohio 43201
	Chief of Research and Development, Department of the Army, Washington, D. C. 20310
2	ATTN: Physical and Engineering Sciences Division
	Commander, Army Research Office, P. O. Box 12211, Research Triangle Park, North Carolina 27709
1	ATTN: Information Processing Office
	Commander, U. S. Army Materiel Development and Readiness Command, 5001 Eisenhower Avenue, Alexandria, Virginia 22333
1	ATTN: DRCDE-TC
	Commander, U. S. Army Electronics Command, Fort Monmouth, New Jersey 07703
1	ATTN: DRSEL-GG-DD
1	DRSEL-GG-DM
	Commander, U. S. Army Missile Command, Redstone Arsenal, Alabama 35809
1	ATTN: Technical Library
1	DRSMI-RSM, Mr. E. J. Wheelahan
	Commander, U. S. Army Armament Command, Rock Island, Illinois 61201
2	ATTN: Technical Library
	Commander, U. S. Army Natick Research and Development Command, Natick, Massachusetts 01760
1	ATTN: Technical Library
	Commander, U. S. Army Satellite Communications Agency, Fort Monmouth, New Jersey 07703
1	ATTN: Technical Document Center
	Commander, U. S. Army Tank-Automotive Research and Development Command, Warren, Michigan 48090
2	ATTN: DRDTA, Research Library Branch
	Commander, White Sands Missile Range, New Mexico 88002
1	ATTN: STEWS-WS-VT
	Commander, Aberdeen Proving Ground, Maryland 21005
1	ATTN: STEAP-TL, Bldg. 305
	Commander, Frankford Arsenal, Philadelphia, Pennsylvania 19137
1	ATTN: Library, H1300, Bl. 51-2
1	SARFA-L300, Mr. J. Corrie
	Commander, Harry Diamond Laboratories, 2800 Powder Mill Road, Adelphi, Maryland 20783
1	ATTN: Technical Information Office
	Commander, Picatinny Arsenal, Dover, New Jersey 07801
1	ATTN: SARPA-RT-S
	Commander, Redstone Scientific Information Center, U. S. Army Missile Command, Redstone Arsenal, Alabama 35809
4	ATTN: DRSMI-RBLD, Document Section
	Commander, Watervliet Arsenal, Watervliet, New York 12189
1	ATTN: SARWV-RDT, Technical Information Services Office

No. of Copies	To
1	Commander, U. S. Army Foreign Science and Technology Center, 220 7th Street, N. E., Charlottesville, Virginia 22901 ATTN: DRXST-SD3
1	Director, Eustis Directorate, U. S. Army Air Mobility Research and Development Laboratory, Fort Eustis, Virginia 23604 ATTN: Mr. J. Robinson, SAVDL-EU-SS
1	Librarian, U. S. Army Aviation School Library, Fort Rucker, Alabama 36360 ATTN: Building 5907
1	Commander, U. S. Army Aviation Systems Command, P. O. Box 209, St. Louis, Missouri 63166 ATTN: Mr. A. Schaefer
1	Mr. F. Drosten
1	Mr. G. A. Gorline
1	Commander, USACDC Air Defense Agency, Fort Bliss, Texas 79916 ATTN: Technical Library
1	Commander, U. S. Army Engineer Waterways Experiment Station, Vicksburg, Mississippi 39180 ATTN: Research Center Library
1	Commander, U. S. Army Environmental Hygiene Agency, Edgewood Arsenal, Maryland 21010 ATTN: Chief, Library Branch
1	Commandant, U. S. Army Quartermaster School, Fort Lee, Virginia 23801 ATTN: Quartermaster School Library
1	Naval Research Laboratory, Washington, D. C. 20375 ATTN: Dr. J. M. Krafft - Code 8430
2	Dr. G. R. Yoder - Code 6382
2	Air Force Materials Laboratory, Wright-Patterson Air Force Base, Ohio 45433 ATTN: AFML/MXE/E. Morrissey
1	AFML/LC
1	AFML/LLP/D. M. Forney, Jr.
1	AFML/MBC/Mr. Stanley Schulman
2	NASA, Langley Research Center, Hampton, Virginia 23365 ATTN: SAVDL-LA-T, C. E. Swindlehurst, Jr.
1	National Aeronautics and Space Administration, Washington, D. C. 20546 ATTN: Mr. B. G. Achhammer
1	Mr. G. C. Deutsch - Code RR-1
1	National Aeronautics and Space Administration, Marshall Space Flight Center, Huntsville, Alabama 35812 ATTN: R-P&VE-M, R. J. Schwinghamer
1	S&E-ME-MM, Mr. W. A. Wilson, Building 4720
1	Ship Research Committee, Maritime Transportation Research Board, National Research Council, 2101 Constitution Ave., N. W., Washington, D. C. 20418
1	The Charles Stark Draper Laboratory, 68 Albany Street, Cambridge, Massachusetts 02139
2	Director, Army Materials and Mechanics Research Center, Watertown, Massachusetts 02172 ATTN: DRXMR-PL
1	DRXMR-AG
2	Authors

Army Materials and Mechanics Research Center,  
Watertown, Massachusetts 02172  
CORROSION FATIGUE BEHAVIOR OF COATED  
4340 STEEL FOR BLADE RETENTION BOLTS OF  
THE AH-1G HELICOPTER -  
Milton Levy and Joseph L. Morrossi

AD

UNCLASSIFIED  
UNLIMITED DISTRIBUTION

Key Words

Technical Report AMMRC TR 76-34, October 1976, 10 pp -  
illus-tables, O/A Project VN-MIPR L-5108A,  
Fiscal Code 80X0108

Rotor blades  
Protective coatings  
Tungsten carbide

The main rotor blade bolt for the 540 helicopter rotor system of the AH-1G, UH-1C, and UH-1M helicopters is proposed to be improved by substituting plasma-sprayed tungsten carbide coating on the outer shank of the 4340 steel bolt for the present cadmium or chromium plate. This study was undertaken to determine the effects of these coating systems on the fatigue behavior of 4340 steel in environments likely to be encountered in service. Both axial tension-tension and rotating bending fatigue testing of coated specimens were carried out in air and in 3.5% NaCl solution. The NaCl environment causes reductions in fatigue strength of bare and coated 4340 steel. The severity of the degradation depends on the coating applied and the type of fatigue test utilized.

Army Materials and Mechanics Research Center,  
Watertown, Massachusetts 02172  
CORROSION FATIGUE BEHAVIOR OF COATED  
4340 STEEL FOR BLADE RETENTION BOLTS OF  
THE AH-1G HELICOPTER -  
Milton Levy and Joseph L. Morrossi

AD

UNCLASSIFIED  
UNLIMITED DISTRIBUTION

Key Words

Technical Report AMMRC TR 76-34, October 1976, 10 pp -  
illus-tables, O/A Project VN-MIPR L-5108A,  
Fiscal Code 80X0108

Rotor blades  
Protective coatings  
Tungsten carbide

The main rotor blade bolt for the 540 helicopter rotor system of the AH-1G, UH-1C, and UH-1M helicopters is proposed to be improved by substituting plasma-sprayed tungsten carbide coating on the outer shank of the 4340 steel bolt for the present cadmium or chromium plate. This study was undertaken to determine the effects of these coating systems on the fatigue behavior of 4340 steel in environments likely to be encountered in service. Both axial tension-tension and rotating bending fatigue testing of coated specimens were carried out in air and in 3.5% NaCl solution. The NaCl environment causes reductions in fatigue strength of bare and coated 4340 steel. The severity of the degradation depends on the coating applied and the type of fatigue test utilized.