

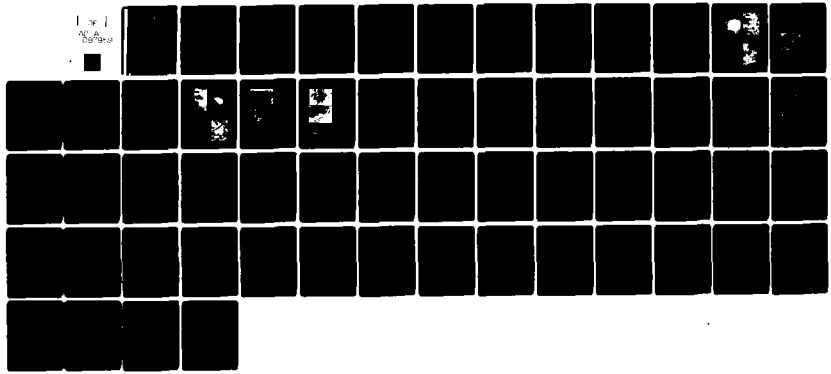
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**STUDY OF EROSION MECHANISMS
OF ENGINEERING CERAMICS**
Static Fatigue of Ceramics

by M.E. Gulden

SOLAR TURBINES INTERNATIONAL

An Operating Group of International Harvester
2200 Pacific Highway, P.O. Box 80966,
San Diego, California 92138

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20. ABSTRACT (Continue on reverse side if necessary and identify by block number) Two areas were investigated in this program: solid particle erosion of engineering ceramics and static fatigue of ceramics. In the erosion phase, the target materials investigated were several types of Si ₃ N ₄ , glass bonded Al ₂ O ₃ and magnesium fluoride. The particle types were SiC and natural quartz sand. Particle sizes ranged between 10 and 1000 microns and particle velocities were in the subsonic regime. The particle size and velocity dependence on erosion, effect of erosion on strength and effect of number of		

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impacts on erosion were determined. It was found that the type of erosion damage produced was more a function of relative target particle properties than of particle size and velocity for the erosion regime investigated. "Elastic-plastic" impact, which is characterized by plastic deformation in the particle contact area, and radial and lateral cracks propagating outward from the contact area was observed on all targets impacted with SiC particles. Damage from quartz particle impact ranged from elastic-plastic on the relatively soft targets like MgF_2 to shallow intergranular chipping with no secondary crack formation for the hard targets such as hot-pressed Si_3N_4 .

Static fatigue behavior of four ceramics was determined in various aqueous solutions at temperatures to $900^\circ C$. Hot-pressed Si_3N_4 did exhibit time-dependent failure and strength decreased with increase in available moisture. The glassy grain boundary phase is thought to be responsible for this behavior. Reaction bonded Si_3N_4 which does not contain a glassy phase did not exhibit time-dependent failure. A model was proposed for the mechanism of stress corrosion of hot-pressed Si_3N_4 based on ion exchange occurring in the glassy grain boundary phase.

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INTRODUCTION

This final report summarizes research for the Office of Naval Research under Contract N00014-73-C-0401, performed in the period from April 1973 through January 1979. Work was directed toward two principal areas: erosion mechanisms of ceramics and static fatigue of ceramics. The experimental approach has been to perform controlled experiments on ceramics of engineering interest which not only promotes basic understanding of behavior, but also generates data of direct use to engineered components.

Accomplishments under the contract have been reported in seven technical reports, six* publications in the open literature and five oral presentations. These are listed in Table 1. The results can be conveniently separated into six phases as follows:

- Phase I - Erosion of Si_3N_4 Materials
- Phase II - Correlation of Experimental Erosion Data with Elastic-Plastic Impact Models
- Phase III - Effect of Number of Impacts on Erosion in the Elastic-Plastic Response Regime
- Phase IV - Erosion of a Two-Phase Target - Alsimag 614 Al_2O_3
- Phase V - Effect of Erosion on Strength
- Phase VI - Static Fatigue of Ceramics

Each phase will be discussed separately in Section 2. The first two phases have not been published previously and will be discussed in detail. Results from the remaining phases have been published in technical reports and the open literature and will be summarized in this report. The experimental procedure and test equipment is described in Appendix A and erosion data is given in Appendix B.

* Two to be published

Table 1

Publications and Presentations

Publications

1. "Stress Corrosion of Silicon Nitride", J. Am. Cer. Soc., 59, p. 391, 1976.
2. "Impact Damage in Brittle Materials in the Elastic-Plastic Response Regime", Proc. Roy. Soc. Series A, 361, p. 343, 1978
3. "Solid Particle Erosion of High-Technology Ceramics (Si_3N_4 , Glass-Bonded Al_2O_3 and MgF_2)", ASTM - STP 664 Erosion: Prevention and Useful Applications, p. 101, 1979.
4. "Effect of Number of Impacts on Erosion of Polycrystalline MgF_2 in the Elastic-Plastic Response Regime", J. Am. Cer. Soc., p. 121, 1980.
5. "Solid Particle Erosion of Si_3N_4 Materials", to be published in Wear.
6. "Correlation of Erosion Data with Elastic-Plastic Impact Models", to be published in J. Am. Cer. Soc.

Technical Reports

1. Static Fatigue of Ceramic Materials, Feb. 1974.
2. Static Fatigue of Ceramic Materials, July 1974.
3. Stress Corrosion of Si_3N_4 , May 1975.
4. Study of Erosion Mechanisms of Engineering Ceramics, Apr. 1976.
5. Study of Erosion Mechanisms of Engineering Ceramics, Apr. 1977.
6. Study of Erosion Mechanisms of Engineering Ceramics, Aug. 1977.
7. Effect of Number of Impacts on Erosion in the Elastic-Plastic Response Regime, Mar. 1979.

Presentations

1. "Static Fatigue of Silicon Nitride", Program Review for ONR/NASC contracts on Silicon Nitride as a Bearing Material, King of Prussia, PA, 1974.
2. "Erosion Mechanisms of Brittle Materials", ONR Contractors Meeting on Impact Response and Erosion Mechanisms for Brittle Materials, La Jolla, CA, 1975.
3. "Stress Corrosion of Silicon Nitride", Am. Cer. Soc. Reg. Meeting, San Francisco, CA, Oct. 1976.
4. "Solid Particle Erosion of High Technology Ceramics, ASTM Symposium", Erosion: Prevention and Useful Applications, Vail, CO, Oct. 1977.
5. "Erosion of Si_3N_4 Materials", Am. Cer. Soc. Reg. Meeting, San Diego, CA, Oct. 1978.

2

RESEARCH ACCOMPLISHMENTS

2.1 PHASE I - EROSION OF Si_3N_4 MATERIALS

2.1.1 Introduction

The purpose of this phase of the investigation was to determine the effects of microstructural variation on erosion and incorporate these effects into the elastic-plastic impact models. A second objective was to further elucidate the conditions (particle size and velocity, relative target and particle properties) where elastic-plastic impact occurs. Elastic-plastic impact damage is characterized by a heavily deformed contact area between particle and target, with radial cracks propagating outward from the contact zone, and with subsurface lateral cracks propagating outward on planes nearly parallel to the surface (Ref. 1). The elastic-plastic models, based on isotropic materials and single impacts, predict that erosion is a function of particle size, velocity and density; and target density, hardness and fracture toughness.

2.1.2 Materials and Test Procedure

Four commercially available silicon nitride materials were selected for target materials. Although the targets do not provide a systematic variation in microstructure, they do cover a wide range of structure and properties. The target materials and relevant mechanical properties are shown in Table 2. Four-point bend strength is not specifically related to erosion resistance, but is included in Table 2 to illustrate the large variation in mechanical properties encompassed by these targets. Properties of the impacting particles are also included in Table 2. Microstructure of the targets is shown in Figures 1 and 2. Hot-pressed Si_3N_4 is fine grained, fully dense and contains minor amounts of glassy phase at the grain boundaries. The pressureless sintered material is also fine grained but contains significant amounts of glassy phase at the grain boundaries and isolated regions of small porosity and second phase particles of silicon and molybdenum silicide. Both of the reaction bonded materials are approximately 75 percent dense. However, the porosity in the NC350 product is relatively small (1-7 microns) and uniformly distributed. The KBI product is very inhomogeneous and contains non-uniform porosity as large as 50 microns, unreacted silicon regions as large as 60 microns and other second phase particles.

Table 2

Si₃N₄ Target and Particle Properties

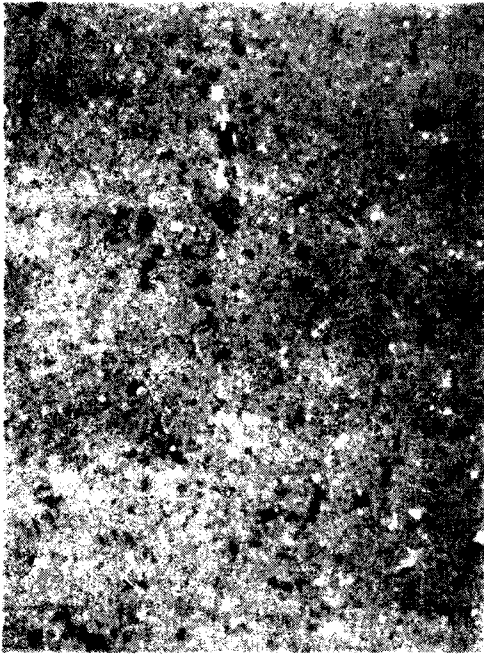
	Elastic Modulus (GPa)	Fracture Toughness (MPam ^{1/2})	Hardness (GPa)	Four-Point Bend Strength (MPa)
Hot-pressed (NC132)	320	5	16	725
Pressureless sintered (GTE 3502)	255	6	12	450
Reaction bonded (NC 350)	170	2.2	7.5	210
Reaction Bonded (KBI)	147	2.2	5	145
Quartz	95	0.7	6	
SiC	420	3	23	

The particles selected for impact and erosion testing also provide a variation in properties. SiC is harder than all of the targets, while quartz is softer than hot-pressed and pressureless sintered Si₃N₄ and has equivalent hardness to the two reaction bonded materials.

Six narrow size ranges between 10 and 385 microns average quartz and 8 to 940 microns average SiC particles were used. For each particle size up to five velocities between 24 and 285 m/sec were evaluated.

2.1.3 Erosion Weight Loss and Particle Size and Velocity Dependencies

Erosion weight loss at 90-degree impingement was measured on the four targets for impact with both SiC and natural quartz (SiO₂) particles. The erosion dependence on particle size and velocity was determined. The exponential functions are shown in Tables 3 and 4. For impact with SiC particles, erosion per impact of all targets was dependent on particle radius to the fourth power for all velocities. The radius exponent for quartz particle impact varied between three for hot-pressed Si₃N₄ to approximately four for the other targets. The power function for velocity dependence on erosion was not as consistent and varied from one for hot-pressed Si₃N₄ impacted with all particle sizes of quartz to as high as six for NC350 reaction bonded Si₃N₄ impacted with quartz, although the most common value was between three and four. Except for hot-pressed Si₃N₄ impacted with quartz, there was not a consistent variation in velocity exponent with target material, particle size or particle type. To compare erosion weight loss of the four targets the data was plotted as erosion volume loss/ impact versus particle radius



Hot Pressed (NC 132)



Reaction Bonded (NC 350)

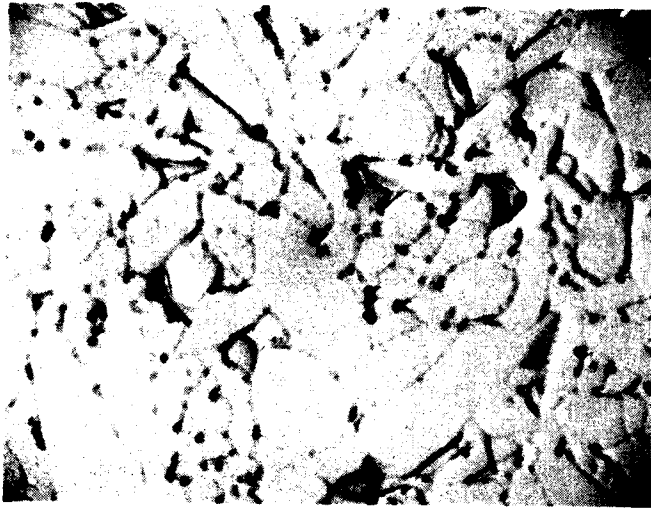


Pressureless Sintered (GTE 3502)



Reaction Bonded (KBI)

Figure 1. Microstructure of Target Materials
Magnification: 1000X



Hot Pressed (NC 132)



Pressureless Sintered
(GTE) 3502

Figure 2. Etched Microstructure of Hot Pressed and
Pressureless Sintered Si₃N₄
Magnification: 12,000X

Table 3
Exponential Function for Particle Size Dependence
on Erosion of Si₃N₄

	Quartz Particle Velocity (mps)			SiC Particle Velocity (mps)		
	80	160	250	50	100	150
NC 132	3.1	3.1	3.2	4.1	4.0	4.1
GTE 3502	-	3.9	3.9	4.0	3.9	4.0
NC 350	3.4	4.1	4.5	4.1	4.3	4.4
KBI	3.9	4.0	4.2	3.9	3.9	3.9

Table 4
Exponential Function for Particle Velocity Dependence
on Erosion of Si₃N₄

	Quartz Particle Size (microns)						SiC Particle Size (microns)					
	385	273	115	64	49	10	940	560	305	102	50	8
NC 132	-	1.1	1.1	1.1	1.0	1.0	3.1	4.1	4.9	3.0	5.6	3.6
GTE 3502	2.6	1.7	2.5	-	-	-	2.9	-	3.4	3.1	-	2.8
NC 350	5.8	4.8	2.9	3.4	3.9	4.3	4.1	-	3.0	2.9	-	2.1
KBI	3.1	2.9	2.8	-	-	2.4	2.5	-	2.8	3.1	-	2.9

to the fourth power (R^4) times velocity to the third power (V^3). This is shown in Figures 3 and 4. As can be seen in Figure 3, for impact with SiC particles, the data fit reasonably well on two parallel lines of slope = 1 for all particle sizes and velocities, suggesting the same mechanism is operative for this range of particle size and velocities. Erosion loss for the two reaction bonded targets is indistinguishable, as is that for the hot-pressed and pressureless sintered targets; and they are separated by approximately one order of magnitude of volume loss/impact for the same particle size-velocity test condition. There is less difference between the targets for impact with 8 micron particles. This is thought to be due to embedding of the small particles in the pores of the reaction bonded materials which, in effect, reduces the measured weight loss. (Examination of the eroded surfaces does reveal embedding of 8 micron particles, which does not occur significantly with the larger particles.) Erosion for quartz particle impact versus R^4V^3 is shown in Figure 4. Data for the reaction bonded targets fits a straight line relation of slope 1 (log.log basis) fairly well. However, there is a separation in volume loss of approximately one order of magnitude between the two targets for the same impact conditions.

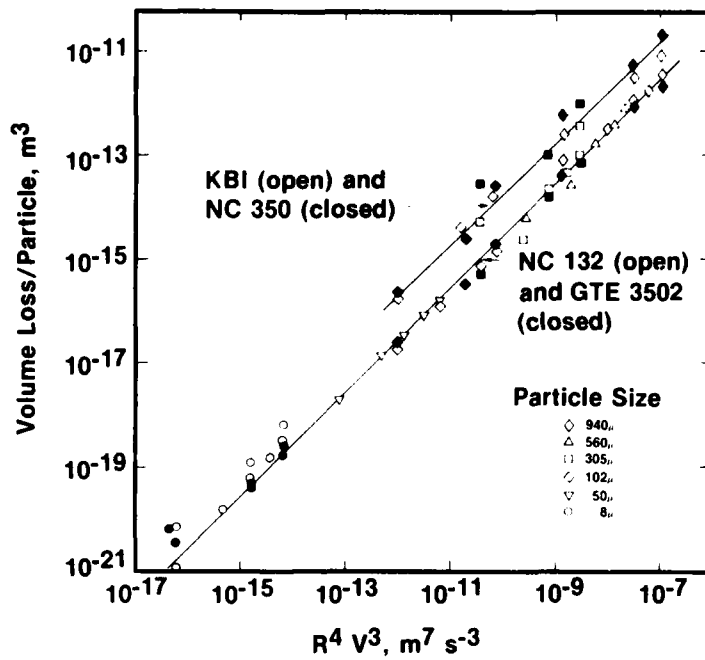


Figure 3. Erosion of Four Si₃N₄ Materials Impacted with SiC Particles

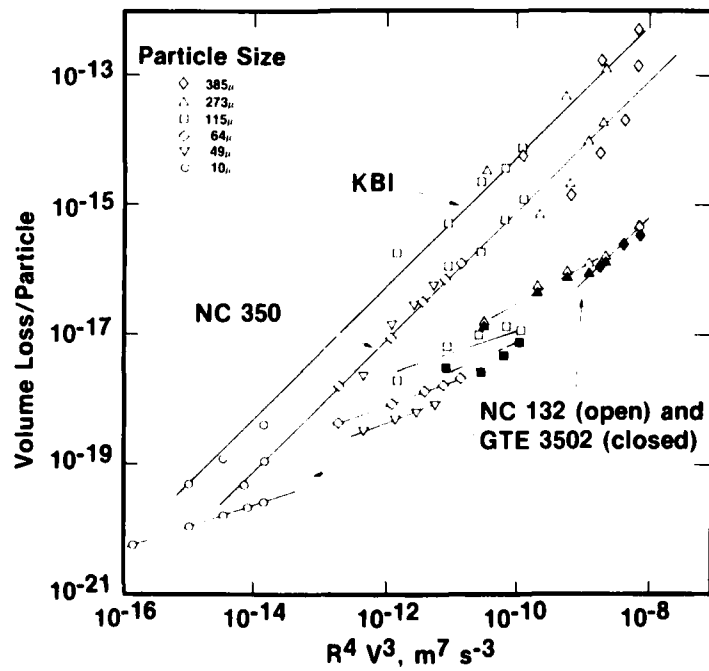


Figure 4. Erosion of Four Si₃N₄ Materials Impacted with Quartz Particles

The erosion data for hot-pressed and pressureless sintered Si_3N_4 do not follow a straight line relationship, which is due primarily to the lower velocity exponent dependence on erosion (Table 4) for these targets. Again, for a given particle size-velocity test condition, erosion weight loss was the same for hot-pressed and pressureless sintered Si_3N_4 . At higher velocities there are several orders of magnitude difference between reaction bonded and sintered (both pressureless sintered and hot-pressed Si_3N_4). The data suggest there is a change in erosion mechanism for impact with quartz particles on hot-pressed and pressureless sintered Si_3N_4 . The data further suggest that the relative target/particle properties are more important with regard to controlling erosion mechanism than variation in particle size and velocity for the test conditions investigated.

2.1.4 Characterization of Erosion Damage

Both heavily eroded surfaces and single impacts were examined to characterize damage and material removal processes. All targets impacted with SiC particles exhibited elastic-plastic type impact damage (Ref. 1). Typical examples for each target are shown in Figure 5. Although all of the targets impacted with SiC particles exhibited elastic-plastic impact, there was minor variation in the damage. Details of impact damage are shown in Figure 6. Figure 5A shows single impact damage on hot-pressed Si_3N_4 . The radial and lateral cracks are long and tight. None of the laterally cracked material in this example has been removed. The secondary cracks on the pressureless sintered target are more open and intergranular or through the glassy grain boundary phase, Figure 5B and 6A. The radial cracks in the hot-pressed Si_3N_4 were transgranular, but the lateral cracks were intergranular. Cracking in NC350 reaction bonded Si_3N_4 was wide and followed the porosity. Crack branching was observed. The impact area in KBI reaction bonded Si_3N_4 (Fig. 5C) was relatively large and secondary cracks, both lateral and radial, were minimized (Fig. 5D). It was difficult to differentiate impact areas from the large pores inherent in this target. The appearance of the actual impact areas for all the targets impacted with SiC was not one of brittle fracture, but rather of plastic flow. No cleavage markings or intergranular facets were observed. A typical example of the bottom of an impact crater is shown in Figure 6B.

The impact damage produced by quartz particles was quite different from that produced by SiC particles. The damage produced on hot-pressed and pressureless sintered Si_3N_4 was characterized by shallow intergranular chipping, as shown in Figure 7A and B and no cracking was observed away from the impact area. The damage on NC350 reaction bonded Si_3N_4 was characterized by concentric ring cracks. This type of damage has been observed previously, usually under low velocity or static impact, and is considered to be due to elastic interaction between particle and target. On the KBI reaction bonded Si_3N_4 , the quartz particle impacts were not distinguishable from the porosity.

100μ

a) NC 132 Hot Pressed Si_3N_4 Particle
Velocity = 176 mps



c) KBI Reaction Bonded Si_3N_4 Particle
Velocity = 79 mps



b) GTE 3502 Pressureless Sintered Si_3N_4 Particle
Velocity = 176 mps



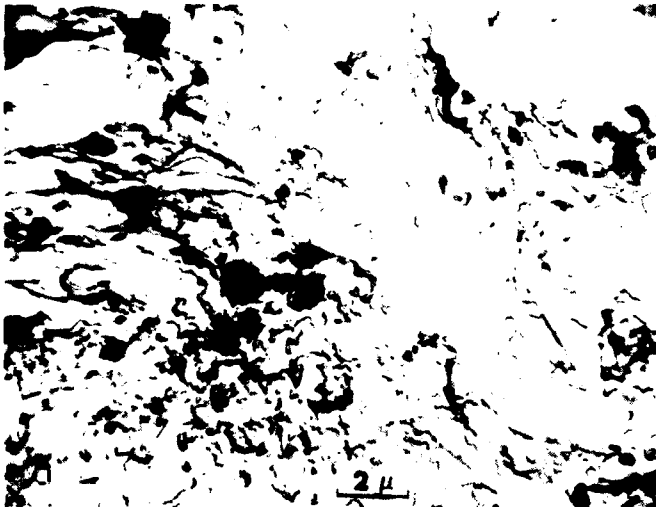
d) NC 350 Reaction Bonded Si_3N_4 Particle
Velocity = 79 mps

Figure 5. Typical Impacts with 300 Micron SiC Particles



GTE 3502 Pressureless
Sintered Si_3N_4

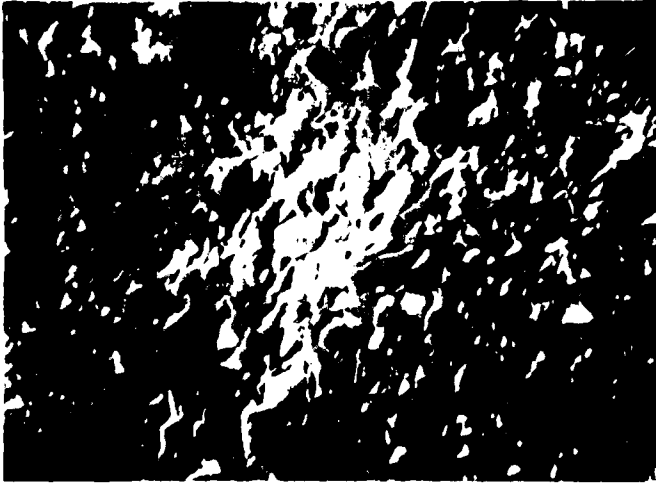
a) intergranular radial
crack



GTE 3502 Pressureless
Sintered Si_3N_4

b) plastic flow in
impact crater

Figure 6. Details of Damage from SiC Particle Impact
(102 Micron Particle, 52 mps Velocity)



A. NC132 Hot Pressed
 Si_3N_4

Particle size = 115 microns

Particle velocity = 219 mps



B. GTE 3502 Pressureless
Sintered Si_3N_4

Particle size = 49 microns

Particle velocity = 241 mps



C. NC 350 Reaction Bonded
 Si_3N_4

Particle size = 273 microns

Particle velocity = 188 mps

Figure 7. Typical Impact Damage Produced by Quartz Particles

2.1.5 Summary

All four Si_3N_4 targets impacted with SiC particles exhibited elastic-plastic impact which is characterized by a highly deformed impact crater and radial and lateral cracks propagating from the impact area. Erosion weight loss per impact was proportional to the fourth power of particle radius. The exponential velocity dependence varied apparently randomly between three and four. Erosion weight loss for the two forms of reaction bonded Si_3N_4 was the same and approximately one order of magnitude greater than hot-pressed and pressureless sintered Si_3N_4 under the same particle size/velocity test conditions. Weight loss on the hot-pressed and pressureless sintered targets was also indistinguishable.

Damage produced by quartz particles did not conform to characteristic elastic-plastic impact damage. Damage on hot-pressed and pressureless sintered Si_3N_4 was characterized by shallow intergranular chipping and no secondary crack formation, and both the radius and velocity exponents for erosion dependence were lower than for erosion with SiC particles. Erosion weight loss between the two targets was again indistinguishable. The reaction bonded silicon nitrides did exhibit some secondary cracking on impact with quartz, but the nature of the cracking was circumferential rather than radial or lateral.

The type of damage produced during erosion of the Si_3N_4 targets was more a function of particle type or relative target/particle properties than of variation in particle size or velocity. Models for elastic-plastic impact relate erosion to a function of target hardness and fracture toughness. The correlation of erosion weight loss with the models is discussed in the next section. It will be shown that for elastic-plastic impact, the microstructural variation, such as grain size and amount of second phase including porosity, of these targets is sufficiently accounted for in the fracture toughness and hardness values and does not have to be incorporated specifically in a model to explain or predict elastic-plastic erosion behavior. Microstructural variation may be a significant variable below the elastic-plastic threshold (Si_3N_4 impacted with quartz particles) because impact damage in this regime is primarily intergranular or inter-pore.

2.2 PHASE II - CORRELATION OF EXPERIMENTAL EROSION DATA WITH ELASTIC-PLASTIC IMPACT MODELS

2.2.1 Introduction

Recent investigations have shown that a number of erosion mechanisms for ceramics can exist and that erosion and impact is a complex process (Refs. 1, 2, 3 and 4). Essentially, two types of models have been proposed for solid particle impact (single particle) and erosion (multi particle) of brittle materials. The earlier models were based on elastic interaction between target and particle and predicted that material removal occurs by

the intersection of ring cracks on the target surface. This process has been observed on several materials under static and low velocity impact conditions with relatively large spherical particles (Refs. 3 and 4). More recent analysis has treated static and dynamic plastic indentation, which is characterized by plastic deformation of the contact area between the particle and the target, with radial cracks propagating outward on planes nearly parallel to the surface. This type of damage, termed elastic-plastic, is observed for impact with angular particles of generally greater hardness than the target (Refs. 1 and 2).

Two expressions have been derived for elastic-plastic impact (Refs. 6 and 7). Details of the derivations are given in the references. Generally, the expressions relate erosion to the depth of damage and area of lateral cracking. The expressions are as follows:

$$V \propto \frac{(\rho_p \rho_t \mu_p \mu_t)^{2/3}}{(\sqrt{\rho_p \mu_p} + \sqrt{\rho_t \mu_t})^{8/3}} \rho_p^{19/12} R_p^{11/13} v_p^{19/6} K_C^{-4/3} H^{-1/4} \quad (1)$$

after Evans (Ref. 6)

$$V \propto \frac{(\rho_p \rho_t \mu_p \mu_t)^{2/3}}{(\sqrt{\rho_p \mu_p} + \sqrt{\rho_t \mu_t})^{8/3}} \rho_p^{11/19} R_p^{11/3} v_p^{22/9} K_C^{-4/3} H^{1/9} \quad (2)$$

after Ruff and Weiderhorm (Ref. 7)

where

- V = erosion volume loss per impact
- ρ_p = particle density
- ρ_t = target density
- μ_p = particle shear modulus
- μ_t = target shear modulus
- R_p = particle radius
- v_p = particle velocity
- K_C = target fracture toughness
- H = target hardness

The major differences between the two expressions are in the particle density, particle velocity and target hardness dependencies. These expressions are based on single impacts and were developed for isotropic materials under idealized conditions. The validity of the above expressions to predict erosion can be assessed by comparing a plot of experimental erosion data as a function of the expressions.

2.2.2 Results

Five target materials were eroded with SiC and quartz particles. The targets were NC132 hot-pressed Si_3N_4 , GTE 3502 pressureless sintered Si_3N_4 , NC350 reaction bonded Si_3N_4 , KBI reaction bonded Si_3N_4 and Intran I MgF_2 . Properties relevant to erosion are shown in Table 5. Particle properties are also given.

Table 5

Target and Particle Properties

	Shear Modulus* (GPa)	Fracture Toughness (MPam ^{1/2})	Hardness (GPa)	Microstructure
Hot-pressed Si ₃ N ₄ NC 132	131	5	16	~2μ grain size-fully dense. Minor grain boundary phase
Pressureless-sintered Si ₃ N ₄ GTE 3502	105	6	12	~2μ grainsize-98% dense. Significant grain boundary phase.
Reaction-sintered Si ₃ N ₄ NC 350	70	2.2	7.5	75% dense-porosity 0.5-7μ. Multi-phase grain boundary phase.
Reaction-sintered Si ₃ N ₄ KBI	60	2.2	5	75% dense-porosity 2-50μ. Multi-phase-60μ Si grains.
Hot-pressed MgF ₂ Irtan 1	48	1	6	Fully dense-single phase. ~2μ grain size.
SiC	172	0.7	23	
Quartz	39	3	6	

* Shear modulus calculated from the expression $G = E/2(1+\gamma)$ where E = elastic modulus and γ is Poisson's ratio assumed to be 0.22 for all materials.

Six particle sizes between 10 and 385 microns average quartz particles and 8 and 940 microns average SiC particles were used for erosion testing. Between three and five particle velocities for each particle size were evaluated. All tests were performed at 90-degree impingement and ambient temperature. Erosion test data is given in Appendix B.

All of the targets impacted with SiC particles exhibited elastic-plastic impact. MgF₂ impacted with quartz particles also exhibited elastic-plastic impact. The other targets impacted with quartz exhibited very little if any damage outside of the particle-target contact area and for hot-pressed and pressureless sintered silicon nitride, damage was characterized by shallow intergranular chipping.

The validity of the expressions for elastic-plastic impact can be assessed by plotting erosion weight loss normalized by the appropriate particle size and velocity dependence versus the functions of particle and target particles predicted by the models. These plots are shown in Figures 8 and 9. The data points are average values of all data for a single target calculated on a logarithmic basis. The error bars correspond to \pm one standard deviation also calculated on a logarithmic basis. A line with a slope of one is drawn through the data points for those targets which exhibited elastic-plastic erosion damage. In both cases, the actual slope through the data points is 1.25 and 1.5 for Figures 8 and 9, respectively. The data for targets which did not exhibit elastic-plastic impact when impacted with quartz particles

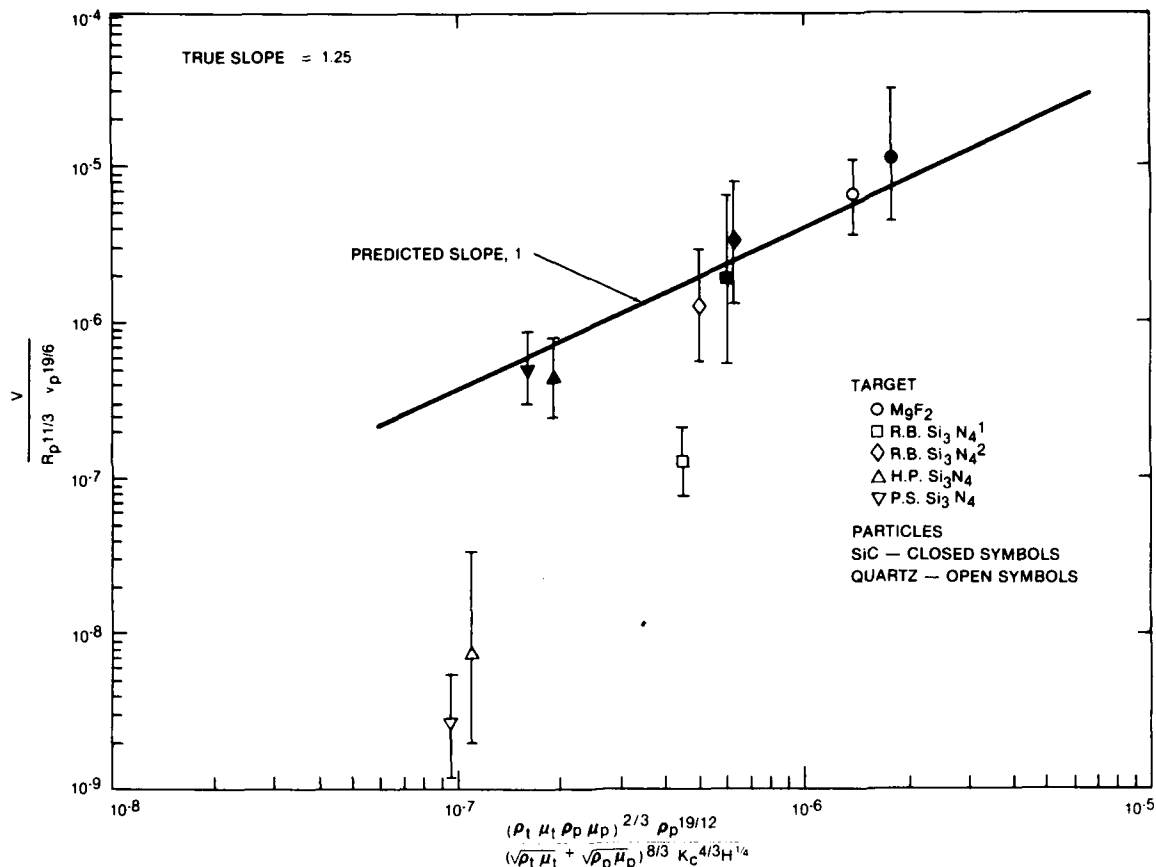


Figure 8. Erosion as a Function of Target and Particle Properties from Equation (1)

(hot-pressed, pressureless sintered and NC350 reaction bonded Si_3N_4) are in the correct order as predicted by both relationships, but fall well below the data line for elastic-plastic impact. The most remarkable result is that the experimental erosion data fit both models equally well although there are two major differences in the expressions; the dependence on particle velocity and target hardness. The power function of velocity from expression (1) is 3.17 which approximates the experimental value. The power function of velocity in expression (2) is lower at 2.44 and is also lower than observed experimentally in this work for elastic-plastic impact conditions. A lower velocity exponent has been observed in other work (Weiderhorn, private communication). The hardness exponents are $(-1/4)$ for expression (1) and $(+1/9)$ for expression (2). Although the targets exhibit a wide range of hardnesses, the small hardness dependence is not sufficient to separate the two expressions.

These results indicate that further experimental work is necessary before the model which most accurately describes elastic-plastic erosion can be ascertained, and that both models provide a reasonable fit with the experimental

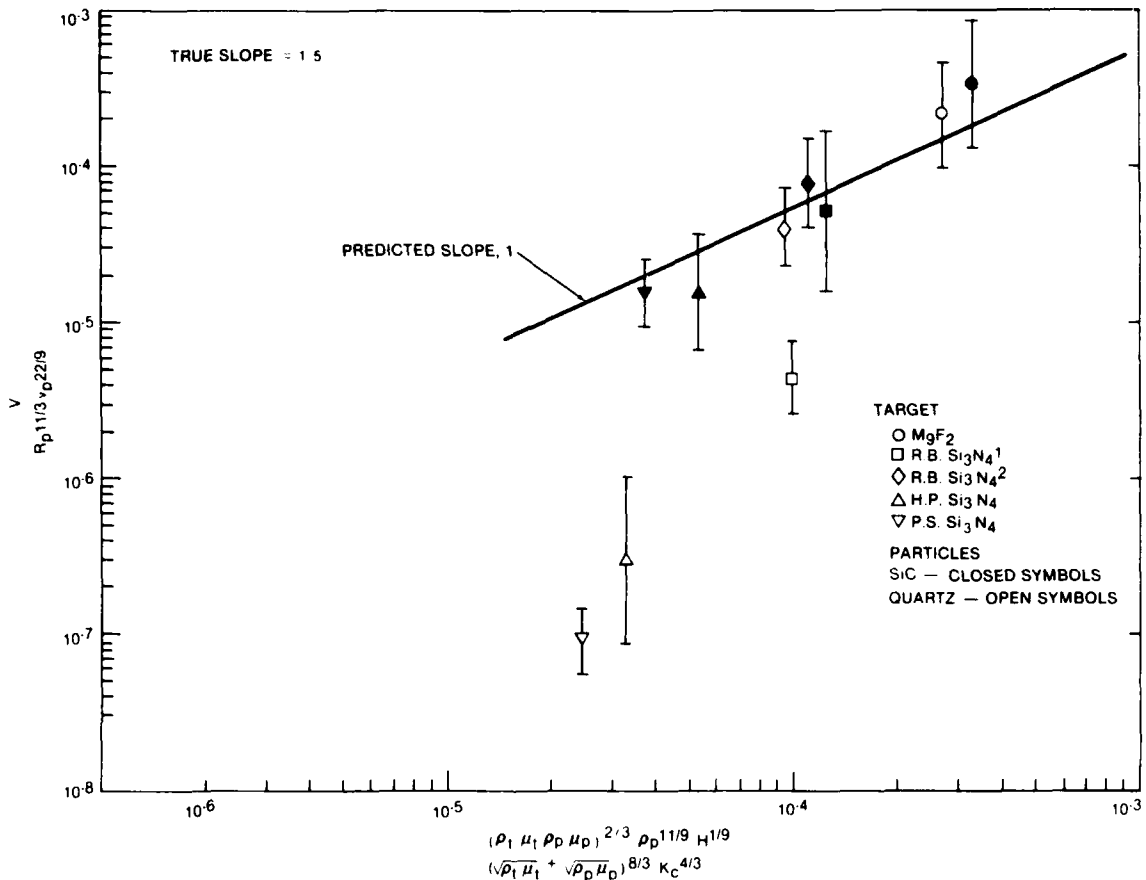


Figure 9. Erosion as a Function of Target and Particle Properties from Equation (2)

erosion data. The results further suggest that for the targets investigated, microstructural variation such as grain size and amount and type of second phase are sufficiently accounted for in the fracture toughness and hardness values and do not need to be considered separately in the models.

2.3 PHASE III - EFFECT OF NUMBER OF IMPACTS ON EROSION IN THE ELASTIC-PLASTIC RESPONSE REGIME

Models have recently been proposed to characterize the damage which occurs on impact with a relatively hard, angular projectile on a softer target surface. This damage system has been termed elastic-plastic and is characteristic of many situations in a dust erosion environment in the subsonic velocity regime. The laterally cracked material is responsible for most of the erosion loss. The models, which are based on single impact damage, predict that erosion is proportional to a function of particle size, velocity and density; and target fracture toughness and hardness. The usefulness of the models for explaining

and predicting actual erosion behavior is dependent in part on the effect of number of impacts on material loss. It is known that for rain erosion and for solid particle erosion occurring by elastic interaction, that an incubation period exists prior to the onset of uniform erosion (Refs. 3 and 4).

A MgF_2 target was subjected to impact conditions from single particle to 10^{10} impacts which simulated a natural dust environment (quartz sand) in the subsonic velocity regime. Six narrow particle size ranges varying between 10 and 385 microns average diameter and up to five velocities for each particle size were used for the investigation. For this target-particle system, erosion weight loss follows the function of particle size and velocity ($R^{3.7}v^{3.2}$) predicted by the elastic-plastic model (Ref. 6) and the single impact damage is characteristic of elastic-plastic impact. Three series of tests were performed to assess the effect of number of impacts on erosion. These can be conveniently separated into the following: weight loss changes as a function of number of impacting particles, effect of starting surface finish on erosion weight loss, and examination and measurement of single and second impacts on the damage area of the initial impact.

There was no consistent effect of number of impacts on erosion weight loss between 10^2 and 10^{10} total particle impacts. On a plot showing erosion normalized by $R^{3.7}v^{3.2}$ (the function of particle size and velocity predicted by the model) versus number of impacts, the data points fall in a one order of magnitude band. It was also shown, again on a weight loss basis that the amount of material removed by the first impact is not significantly different from that removed by the 50th impact on the same area.

The effect of starting surface finish on erosion was assessed by comparing results between a polished starting surface and a heavily pre-damaged surface. Initially, more material was lost from the pre-damaged surface than from the polished surface. However, this initial effect was not maintained and for heavy erosion conditions, the total weight loss was the same for both starting surface conditions. The results indicate that starting surface finish is not a significant variable for conditions of heavy erosion.

Examination and measurement of single impact damage revealed that for a given particle size-velocity condition the volume of material removed per impact varied over three orders of magnitude. This large variation is due primarily to differences in particle orientation during impact which results from the irregular angular natural quartz particles. The large variation precludes determination of significant differences between initial and second impacts using standard statistical procedures. However, a comparison between arithmetic means and standard deviations calculated from common logarithms of the measured volume removed indicate there is not a significant difference between the amount of material removed by the first impact and that removed by subsequent impacts on the initial impact damage area. The average values for volume loss by second impacts are within the one standard deviation range for volume loss by the initial impacts.

The results of this investigation indicate that there is not a significant incubation or damage enhancement effect for conditions which simulate a dust erosion environment in the subsonic velocity regime when the damage is

characterized by elastic-plastic impact. The results further indicate that the model developed for elastic-plastic impact, which was based on single impacts, is applicable to heavy erosion conditions.

This phase of the program is discussed in detail in Technical Report No. 7 and in the "Effect of Number of Impacts on Erosion of Polycrystalline MgF_2 in the Elastic-Plastic Response Regime" Journal of the American Ceramic Society, Vol. 63, No. 3-4, March-April, 1980, pp. 121-126.

2.4 PHASE IV - EROSION OF A TWO-PHASE TARGET - ALSIMAG 614 Al_2O_3

Alsimag 614 Al_2O_3 , a common radome material, was eroded with quartz particles to simulate a dust erosion environment. Three particle size ranges of quartz (1-30, 75-149 and 203-297 microns) accelerated to velocities between 30 and 313 m/sec were used for erosion tests. The target is sintered Al_2O_3 containing approximately four percent glass which resides at the grain boundaries. The Al_2O_3 grain size averages about 10 microns and the glassy phase can be as wide as 4 microns at grain boundary triple points. Erosion of this target did not exhibit a consistent velocity dependence for the three particle sizes. Erosion was proportional to particle radius cubed over the entire range of particle sizes, but the velocity exponent varied between three for the smaller particles and one for the larger particles at higher velocities. This variation can be related to the two-phase nature of the microstructure.

Examination of the eroded surfaces revealed that damage from 10 micron particle impacts was characterized by flow of the glassy grain boundary phase and minimal damage to the Al_2O_3 grains. Erosion occurs by flow and removal of the glass to the extent that entire Al_2O_3 grains are lost. An example of a surface heavily eroded by 10 micron quartz particles is shown in Figure 10A. The depth of erosion corresponds to approximately four grain diameters. As can be seen, little damage is sustained by the Al_2O_3 grains. As the particle size was increased, chipping of the Al_2O_3 grains occurred in addition to flow of the glass. An example is shown in Figure 10B. This chipping is similar to that which occurs on hot-pressed Si_3N_4 impacted with quartz particles.

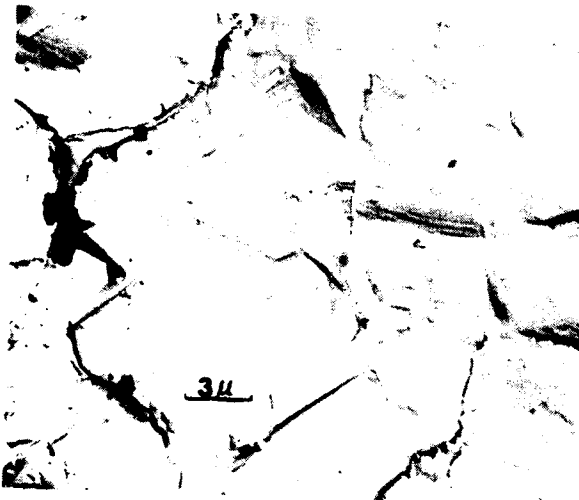
A more extensive discussion on this subject can be found in Technical Report No. 4 and in "Solid Particle Erosion of High Technology Ceramics (Si_3N_4 , Glass-Bonded Al_2O_3 and MgF_2)", ASTM STP 664, Erosion: Prevention and Useful Applications, pp. 101-122, 1979.

2.5 PHASE V - EFFECT OF EROSION ON STRENGTH

Strength in three-point bending after erosion with quartz particles was determined for NC132 hot-pressed Si_3N_4 , NC350 reaction bonded Si_3N_4 and Alsimag 614 glass bonded Al_2O_3 . Particle sizes ranging from 10 to 385 microns and particle velocities between 15 and 315 mps were used for the erosion



- A. 10 μ Particle size
285 m/sec Velocity
Contact radius $\sim 2\mu$
Depth of erosion $\sim 50\mu$
 $E \propto R^3 v^3$



- B. 250 μ Particle size
46 m/sec Velocity
Contact radius $\sim 13\mu$
Depth of erosion $\sim 6\mu$
 $E \propto R^3 v$

Figure 10. Glass Bonded Al_2O_3 Impacted with Quartz

tests. The strength results are given in Table 6. The baseline strengths (prior to erosion) plus and minus one standard deviation are given for each target material. An estimate of the critical flaw size is also given. This was calculated from fracture toughness values using the relationship for three-point bend specimens (Ref. 8). The strengths after erosion are shown as a function of the baseline strengths by giving the percentages which have strengths greater than +1, and less than -1 standard deviation of the non-eroded material strength.

Hot-pressed Si_3N_4 and the GB Al_2O_3 did not exhibit a strength decrease under these test conditions for erosion depths up to $31 \mu\text{m}$. For hot-pressed Si_3N_4 this depth corresponds to 3×10^8 particle impacts or 600 grams of dust on a 0.71 cm^2 area. There was a trend toward strength increase (fracture stress for 50 percent of the specimens was greater than one standard deviation above

Table 6
Strength Distribution of Eroded Specimens
Relative to Standard Strength

	Pre-Erosion Baseline		Number of Tests	Maximum Erosion Depth (μ)	Post-Erosion Strength Distribution (%)		
	$\sigma_F \pm 1$ Standard Deviation (MPa)	Critical Flaw Size (μm)			Above +1 Standard Deviation	Within ± 1 Standard Deviation	Less Than -1 Standard Deviation
Hot-pressed Si_3N_4	798 ± 128	11	25	31	50	50	0
Glass-bonded Al_2O_3	252 ± 21	43	51	35	41	43	16
Reaction-bonded Si_3N_4	278 ± 10	17	17	360	6	12	82*

* Minimum strength = 81 MPa corresponding to critical flaw size of $\approx 200 \mu\text{m}$.

the baseline strength), which indicates that a "polishing" phenomenon may be occurring. No specimens failed at a stress lower than the one standard deviation band. The erosion depth is approximately three times the estimated critical flaw size, which indicates that under these test conditions the effective flaw size produced by erosion is no larger than pre-existing flaws characteristic of the "standard" machined surface. (Because large numbers of flaws are introduced by erosion or machining, the effective stress concentration will depend on both size and spacing of flaws. In contrast, critical flaw sizes were calculated for isolated flaws.) In confirmation, examination of the eroded surfaces in cross section did not reveal apparent subsurface damage; that is, the structure under the eroded area was indistinguishable from that below the as-machined surface and subsurface cracking was not apparent.

The trend toward a strength increase is not as conclusive with GB Al_2O_3 as with hot-pressed Si_3N_4 since 16 percent of the specimens failed at stresses

below the one standard deviation range. Furthermore, although the maximum depth of erosion (approx. 35 μm) is approximately four grain diameters, this also corresponds to the estimated critical flaw size of the baseline material. However, the results indicate that erosion under these test conditions does not produce flaws greater than those inherent to the assintered surface. There was no significant variation in strength with erosion test conditions (particle size and velocity) for either hot-pressed Si_3N_4 or GB Al_2O_3 .

Reaction bonded Si_3N_4 did exhibit a marked strength decrease under these erosion test conditions. Strength decreased rapidly for the first 100 μm erosion depth and remained essentially constant at 100 MPa up to the maximum depth tested (350 μm). The lowest strength (81 MPa) corresponds to an increase in estimated critical flaw size over the baseline material of one order of magnitude (17 μm compared with 200 μm). The depth of subsurface cracks perpendicular to the surface is also approximately 200 μm for this erosion condition.

According to the model for elastic-plastic impact response, the strength of the target (σ_F) should be dominated by radial crack formation by the following relation (1)

$$\sigma_F \propto \frac{K_C^{1.4}}{V^{0.5} R^{0.8}}$$

where K_C = target fracture toughness,
 V = particle velocity, and
 R = particle radius.

A log-log plot of strength versus ($V^{0.5} R^{0.8}$) for reaction bonded Si_3N_4 does result in a line with slope = 1 for erosion with particle sizes of 115, 273 and 385 microns.

Since quartz is the most erosive constituent of natural dusts and the particle size-velocity conditions are typical of airborne dust, these results have direct practical significance and indicate that a strength decrease does not necessarily occur for erosion conditions, which produce appreciable material removal (hot-pressed Si_3N_4 and Alsimag 614 glass bonded Al_2O_3). This would be expected to apply as long as the load-bearing volume is not reduced significantly.

A more detailed discussion of this subject is given in Technical Report Nos. 5 and 6, and "Solid Particle Erosion of High Technology Ceramics (Si_3N_4 , Glass-Bonded Al_2O_3 and MgF_2)", ASTM STP 664, Erosion: Prevention and Useful Applications, pp. 101-122, 1979.

2.6 PHASE VI - STATIC FATIGUE OF CERAMICS

Variable strain rate and deadweight load tests were conducted on four ceramics in various aqueous solutions at temperatures to 900°C. The ceramics were

single crystal Al_2O_3 , Al995 Al_2O_3 , NC132 hot-pressed Si_3N_4 and NC350 reaction reaction bonded Si_3N_4 . Most of the work was performed on the last two materials.

Hot-pressed Si_3N_4 did exhibit time-dependent failure and strength decreased with increase in available moisture. Fractures were intergranular. It is thought that the stress corrosion of the glassy grain boundary phase inherent in this material is responsible for the strength decrease. Reaction bonded Si_3N_4 in the as-fired condition also exhibited time-dependent strength reduction as a function of increasing moisture content. However, no decrease was observed when the oxide surface layer was removed prior to testing.

A model was proposed for the mechanism of stress corrosion of hot-pressed Si_3N_4 based on ion exchange occurring in the glassy grain boundary phase.

Further discussion on this subject is given in Technical Reports 1, 2 and 3 and in "Stress Corrosion of Silicon Nitride", Jr. Am. Cer. Soc., Vol. 59, No. 9-10, pp. 391-396, 1976.

REFERENCES

1. Evans, A.G., Gulden, M.E. and Rosenblatt, M., Proc. Roy. Soc., Lond. A. 361 (1978) 343.
2. Gulden, M.E., "Solid Particle Erosion of High Technology Ceramics", ASTM STP 664 - Erosion: Prevention and Useful Applications.
3. Adler, W.F. and Sha, G.T., "Analytical Modeling of Subsonic Particle Erosion", Report AFML-TR-72-144, WPAFB, Ohio (1972).
4. Oh, H.L., Oh, K.O.L., Vaidyanathan, S. and Finnie, I., "On the Shaping of Brittle Solids by Erosion and Ultrasonic Cutting", NBS Special Publication 348 (1972) 119.
5. Adler, W.F. and Hooker, S.V., "Characterization of Transparent Materials for Erosion Resistance", Report AFML-TR-76-16, WPAFB, Ohio (1976).
6. Evans, A.G., in Treatise on Materials Science and Technology, Vol. 16, Erosion, C.M. Preece Ed., Academic Press, NY (1979), P1.
7. Ruff, A.W. and Wiederhorn, S.M., *ibid*, p. 69.
8. Evans, A.G., in Fracture Mechanics of Ceramics 1, R.C. Bradt, D.P.H. Hasselman and F.F. Lange, Eds., Plenum Press, NY (1974), p. 17.

APPENDIX A

EROSION TEST FACILITY

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EROSION TEST FACILITY

TEST RIG

Erosion testing is performed with a stationary target impacted by particles accelerated in an air stream. Particles are injected into the air stream 3.05 metres from the target to provide sufficient distance for acceleration. High pressure, filtered and chemically dried air is used for the particle carrier gas. Particles are injected into the gas stream using either a precision, high-pressure, Plasmadyne powder feeder (for long-time tests of $>10^6$ particle impacts) or a smaller powder feeder which uses a weighted charge of particles. Most tests were performed using the smaller feeder.

Particle flow is regulated so that particle-particle interaction in the air stream and at the specimen surface can be considered negligible. This flow rate corresponds to ~ 1 gram per minute.

Exit air velocity is determined with a pitot tube connected to a mercury manometer to monitor total pressure (P_T) and to a water manometer to monitor static pressure (P_S). The pitot tube is placed 1.27 cm from the nozzle exit which corresponds to the target position. The following relationship is used to calculate air velocity:

$$\frac{V}{\sqrt{T}} = \sqrt{2g \cdot J \cdot C_p \left[1 - \left(\frac{P_S}{P_T} \right)^{\frac{\gamma-1}{\gamma}} \right]}$$

where V is velocity, T is the discharge temperature, g is the acceleration due to gravity, J is the mechanical equivalent of heat, C_p is the specific heat of the gas at constant pressure, and γ is the ratio of specific heats of the gas at constant pressure and constant volume. Current barometric pressure is used for the calculation, and the solution is computerized.

Maximum exit air velocity can be varied continuously between 15 and 343 (sonic velocity) m/sec at room temperature. The velocity variation across the 0.95 cm diameter nozzle is less than 5 percent.

In addition to exit gas velocity, the gas velocity where the particles enter the air stream is also determined by measuring static pressure and use of conventional Fanno line functions.

Target specimen size is greater than the nozzles to ensure that all particles impact the target.

PARTICLE VELOCITY

Considerable effort at Solar has been devoted to analyzing the acceleration characteristics of dust particles introduced into a high-velocity fluid. A general aerodynamic solution was developed assuming the particles react as Stokesian spheres in an accelerating gas stream. The derivation is given in pages 31 through 43.

The accuracy of this solution was determined by comparing the predicted particle velocities with measured particle velocities using the rotating double disc technique (Ruff, A.W. and Ives, L.R., *Wear*, 35 [1975] p. 195.)

The results are shown in Figure A-1 for quartz particles of various sizes and velocities. As can be seen, there is good agreement between the two methods of determining particle velocity. In view of the good correlation between measured and calculated particle velocities and the time-consuming process required to measure particle velocities, the calculated values were used for most of this work.

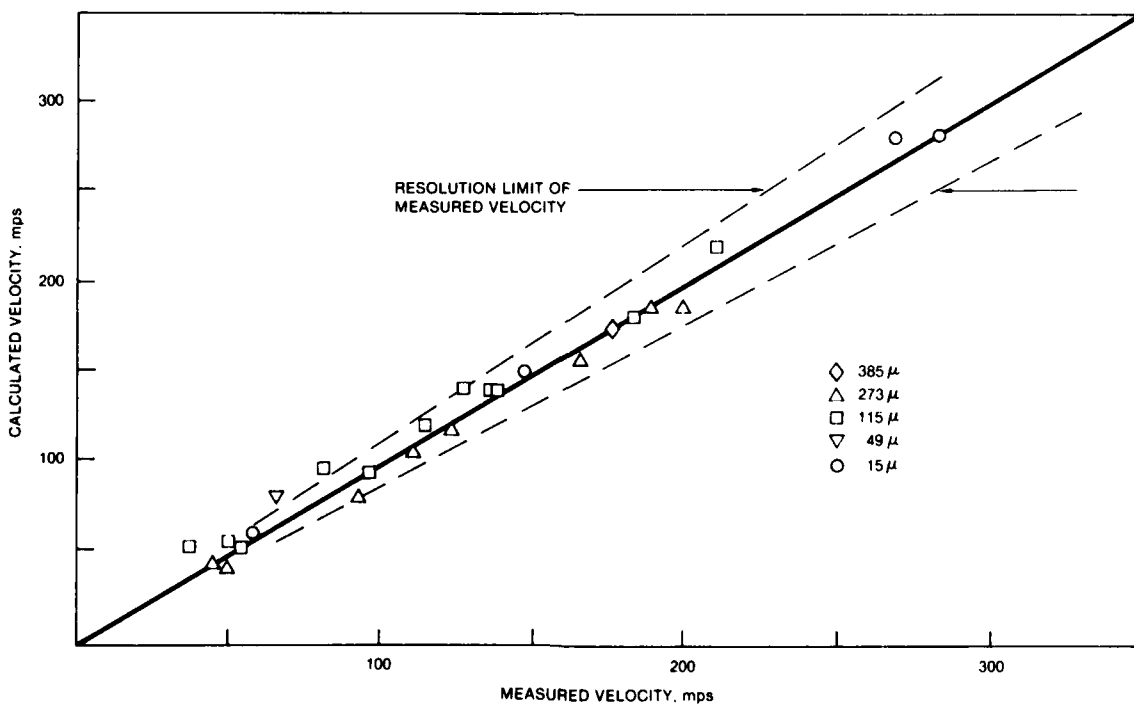


Figure A-1. Comparison of Quartz Particle Velocity Calculated from Aerodynamic Considerations and Measured by Rotating Double Disc Technique

FLOW PARAMETERS FOR PARTICULATE CLOUDS

A measure of the drag and inertia of a particle is the "range" or "carry". This is the distance the particle will travel in the direction of its initial velocity when projected into a motionless viscous fluid. The time to travel this distance is theoretically infinite, but the "range" concept is useful as a criterion for the threshold size for particle damage to surfaces. For example, if the range is less than the boundary layer thickness, then the particle will never reach the surface. Also, the range concept will give a measure (it is simply related to the inertia parameter discussed elsewhere) of the departure of the particle trajectories from the fluid streamlines in the vicinity of a solid body. The larger the range, the more the trajectories will depart from the streamlines in curving flow.

For spherical particles in the Stokesian motion regime ($Re \ll 1.0$), the drag coefficient $C_D = 24/Re$, where the particle Reynolds number, $Re = VD_p/\nu$,

and ν = kinematic viscosity - ft^2/sec

V = velocity - fps

D_p = diameter of particle - ft

The equation of motion of the particle in the direction of the initial velocity, assuming that the fluid drag is the only force acting in this direction, is:

$$m \frac{dV}{dt} = -C_D \frac{1}{2} \rho V^2 A \quad \text{with } V = V_0 \text{ when } t = 0,$$

$$\text{i.e.,} \quad \rho_p \frac{\pi D_p^3}{6} \frac{dV}{dt} = -\frac{24\nu}{VD_p} \frac{1}{2} \rho V^2 \frac{\pi D_p^2}{4}$$

$$\text{or} \quad \frac{dV}{dt} = -\frac{18\mu V}{\rho_p D_p^2} \quad (1)$$

where m = mass of particle - slugs

ρ_p = density of particle - slugs/ ft^3

A = projected area of particle - ft^2

ρ = density of fluid - slugs/ ft^3

μ = absolute viscosity of fluid = $\rho\nu$ slug/ $ft \text{ sec}$

Solutions of this equation are

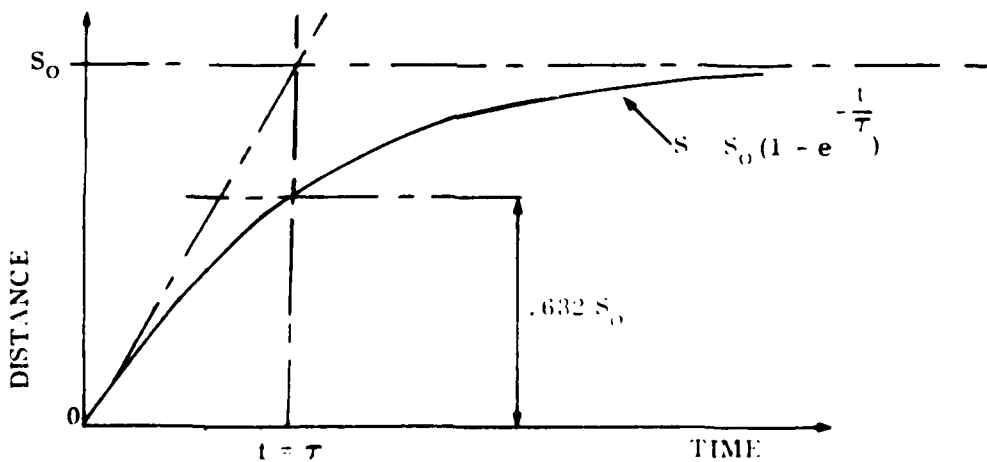
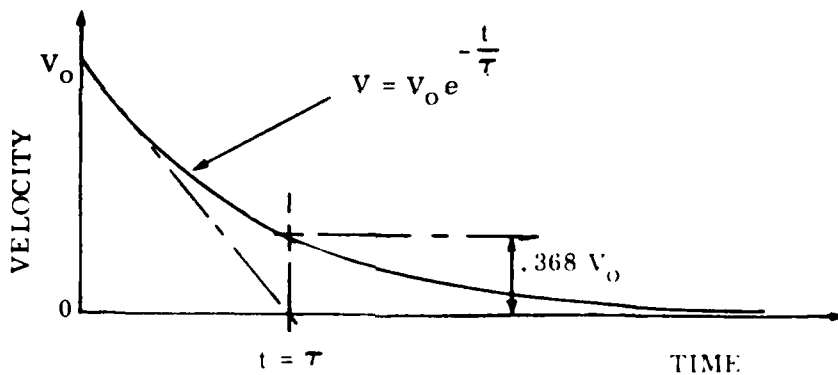
$$V = V_0 e^{-\frac{18\mu}{\rho_p D_p^2} t} = V_0 e^{-\frac{t}{\tau}} \quad (2)$$

$$S = V_0 \tau (1 - e^{-\frac{t}{\tau}}) = S_0 (1 - e^{-\frac{t}{\tau}}) \quad (3)$$

where $\tau = \rho_p D_p^2 / 18\mu$ is a time constant related to $S_0 = V_0 \tau$, the "range" or "carry".

It is apparent that τ is the time for the velocity to drop to 36.8 percent of its initial value V_0 when the distance travelled will be 63.2 percent of S_0 .

The range S_0 can also be thought of as the distance that the particle would travel in time τ if its initial velocity V_0 were maintained. Plots of velocity and distance against time make these points clear.



For flow in the vicinity of a solid body, the trajectories of the particles will be determined by the ratio of the range to a characteristic length of the body. For a sphere or cylinder in crossflow, this length has (traditionally) been taken as the radius, giving rise to the so-called inertia parameter $K = 2S_o/D$

$$K = \frac{\rho_p D^2 V_o}{9\mu D} \quad (4)$$

The inertia parameter is clearly dimensionless, and the trajectories of all particles with the same value K will be geometrically similar so long as the particle can be regarded as a massive point (i.e., its dimensions are small compared to the impacting body) and its presence does not interfere with the flow pattern (i.e., the concentration is sufficiently small).

For flow of particles in a fluid falling under the influence of gravity, the particles will accelerate until the drag plus buoyancy forces just balance the gravitational force. The terminal (or settling) velocity so reached is given when

$$\frac{1}{2} \rho V_t^2 C_D A + mg \left(\frac{\rho}{\rho_p} \right) = mg \quad (5)$$

where m = mass of the particle - slugs

ρ_p = density of the particle - slugs/ft³

ρ = density of fluid - slugs/ft³

A = drag area of particle - ft²

C_D = drag coefficient

V_t = terminal velocity - fps

g = gravitational acceleration - ft/sec²

For a spherical particle in the Stokesian regime, this becomes

$$\frac{1}{2} \rho V_t^2 \frac{24}{Re} \frac{\pi D^2}{4} = \frac{\pi D^3}{6} (\rho_p - \rho) g$$

or

$$V_t = \frac{gD^2}{18\mu} (\rho_p - \rho)$$

$$= \frac{gD^2 \rho_p}{18\mu} \left(1 - \frac{\rho}{\rho_p}\right) \quad (6)$$

Substituting from (5) into (6), we get

$$V_t = \frac{gKD}{2V_o} \left(1 - \frac{\rho}{\rho_p}\right)$$

or

$$\frac{V_t V_o}{gD} = \frac{K}{2} \left(1 - \frac{\rho}{\rho_p}\right) \quad (7)$$

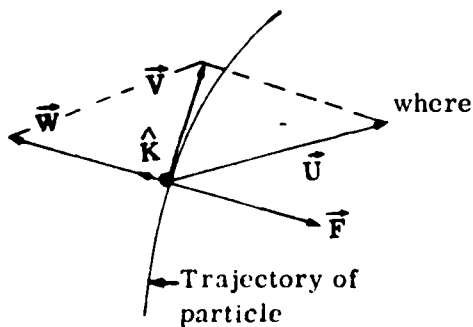
The quantity $V_t V_o / gD$ is called the separation number and is also dimensionless like the inertia parameter K . If the density of the particles is large compared to the density of the propelling fluid, then the separation number = 1/2 the inertia parameter. The separation number is related to a body with characteristic dimension D and is a measure of the separation (or collection) efficiency of that body when impacted by a particulate cloud.

DYNAMICS OF PARTICULATE CLOUDS

Equation of Motion - A particle moving through a fluid with instantaneous velocity \vec{V} will experience a resisting force given by

$$\vec{F} = -\frac{1}{2} \rho (\vec{V} - \vec{U})^2 C_D A \hat{K}$$

$$= -\frac{1}{2} \rho (\vec{W})^2 C_D A \hat{K} \quad (8)$$



\vec{U} = local velocity of the fluid

\vec{W} = velocity of the particle relative to the fluid = $\vec{V} - \vec{U}$

ρ = density of the fluid

C_D = drag coefficient of the particle

A = relevant projected area of the particle for the given drag coefficient

\hat{K} = unit vector in the direction of relative motion

Neglecting all other body forces (including gravity), the equation of motion of the particle will be

$$m \frac{d\vec{V}}{dt} = \vec{F} = -\frac{1}{2} \rho (\vec{W})^2 C_D A \hat{K} \quad (9)$$

where m = mass of the particle

or

$$\frac{d}{dt} (\vec{U} + \vec{W}) = -\frac{C_D A \rho}{2m} (\vec{W})^2 \hat{K} \quad (10)$$

SPHERICAL PARTICLE IN UNIFORM VELOCITY FIELD

If the particle is spherical of diameter D_p and density ρ_p , then

$$m = \rho_p \frac{D_p^3}{6} \pi \quad (11)$$

and a good approximation to the drag coefficient is

$$C_D = 0.4 + \frac{24}{Re} \quad (12)$$

with $A = \frac{\pi D^2}{4}$ the projected frontal area

Also,

$$Re = \frac{|\vec{W}| D_p \rho}{\mu} \quad (13)$$

where μ = absolute viscosity of the fluid

If additionally, \vec{V} is in the same direction as \vec{U} , then $-\vec{W}$ is also in this same direction, and with $\vec{U} = -U\hat{K}$, $\vec{V} = -V\hat{K}$, and $\vec{W} = W\hat{K}$, we get the scalar equations

$$W = U - V$$

and

$$\begin{aligned} \frac{d}{dt} (-U + W) &= -\frac{1}{2} \frac{6}{\pi \rho_p D_p^3} \left(1 + \frac{24\mu}{W D_p \rho} \right) \rho \frac{\pi D_p^2}{4} W^2 \\ &= -\frac{.3\rho}{\rho_p D_p} W^2 + \frac{18\mu}{\rho_p D_p^2} W \end{aligned} \quad (14)$$

The factor $18\mu/\rho_p D_p^2 = 1/\tau$ where τ is the Stokesian time constant associated with the range when $U = 0$. Also, if U is a constant velocity (i.e., a uniform fluid velocity field), then

$$\frac{d}{dt} (-U + W) = \frac{dW}{dt}$$

Putting

$$\frac{.3\rho}{\rho_p D_p} = \frac{1}{\delta}, \quad \text{we get}$$

$$\frac{dW}{dt} = -\frac{1}{\tau} \left(W + \frac{\tau}{\delta} W^2 \right) \quad (15)$$

which is the differential equation for the relative velocity W .

The solution of this equation is

$$\frac{t}{\tau} = \text{const} - \log\left(\frac{W}{1 + \frac{\tau}{\delta} W}\right)$$

If

$$W = W_0 \text{ when } t = 0,$$

then

$$\text{const} = \log\left[\frac{W_0}{1 + \frac{\tau}{\delta} W_0}\right]$$

and

$$t = \tau \log\left[\frac{W_0 \left(1 + \frac{\tau}{\delta} W\right)}{W \left(1 + \frac{\tau}{\delta} W_0\right)}\right] \quad (16)$$

giving the time to reach a relative velocity W from an initial relative velocity W_0 . It will be noted that the time for W to become zero is infinitely large. Also, when $t = \tau$ we have

$$\epsilon = \frac{W_0}{W} \left[\frac{1 + \frac{\tau}{\delta} W}{1 + \frac{\tau}{\delta} W_0} \right]$$

or

$$W = \frac{W_0}{\epsilon} \left\{ 1 + \frac{\tau}{\delta} W_0 \left(1 - \frac{1}{\epsilon} \right) \right\}^{-1}$$

$$= .368 W_0 \left\{ 1 + .632 \frac{\tau}{\delta} W_0 \right\}^{-1} \quad (17)$$

This shows the influence of the stronger drag coefficient as compared to the Stokesian solution. For Stokesian flow, τ is the time for the relative velocity to drop to 36.8 percent of the initial value, whereas with the stronger drag forces the relative velocity drops to somewhat less than 36.8 percent in time τ , depending on the value of $(\tau/\delta) W_0$. Also, for $1/\delta = 0$ or $W_0 \rightarrow 0$, the Stokesian equations are regained.

Now equation (15) may be rewritten as

$$W \frac{dW}{dS_{rel}} = -\frac{1}{\tau} (W + \frac{\tau}{\delta} W^2)$$

or

$$\frac{dW}{dS_{rel}} = -\frac{1}{\tau} \left(1 + \frac{\tau}{\delta} W \right) \quad (18)$$

where S_{rel} is the distance that the particle travels relative to the fluid.

The solution to equation (18) is

$$\frac{S_{rel}}{\tau} = \text{constant} - \frac{\delta}{\tau} \log \left(1 + \frac{\tau}{\delta} W \right)$$

when

$$S_{rel} = 0 \quad W = W_0 \quad \text{const} = \frac{\delta}{\tau} \left(1 + \frac{\tau}{\delta} W_0 \right)$$

Then,

$$S_{\text{rel}} = \delta \log \left(\frac{1 + \frac{\tau}{\delta} W_0}{1 + \frac{\tau}{\delta} W} \right) \quad (19)$$

The distance the particle will travel in time t is given by

$$S = \int V dt = \int (U - W) dt$$

So

$$\begin{aligned} S &= Ut - \int W dt \\ &= Ut - S_{\text{rel}} \end{aligned} \quad (20)$$

Substituting for t from equation (16) and S_{rel} from equation (19) gives

$$\begin{aligned} S &= U\tau \log \left[\frac{W_0}{W} \frac{1 + \frac{\tau}{\delta} W}{1 + \frac{\tau}{\delta} W_0} \right] - \delta \log \left[\frac{1 + \frac{\tau}{\delta} W_0}{1 + \frac{\tau}{\delta} W} \right] \\ &= U\tau \log \left[\frac{W_0}{W} \left(\frac{1 + \frac{\tau}{\delta} W}{1 + \frac{\tau}{\delta} W_0} \right)^{1 + \frac{\delta}{U\tau}} \right] \end{aligned} \quad (21)$$

and this is the distance travelled by the particle to attain a relative velocity W from an initial relative velocity W_0 in a uniform fluid velocity field of value U .

If the particle is introduced into the stream with initial velocity zero, then the initial relative velocity $W_0 = U$ and the instantaneous relative velocity $W = U - V$. Equation (21) becomes

$$S = U\tau \log \left[\left(\frac{U}{U - V} \right) \left(\frac{1 + \frac{\tau}{\delta} (U - V)}{1 + \frac{\tau}{\delta} U} \right)^{1 + \frac{\delta}{U\tau}} \right]$$

or putting

$$\lambda = \frac{V}{U},$$

$$S = U \tau \log \left[\left(\frac{1}{1-\lambda} \right) \left(\frac{1 + \frac{\tau U}{\delta} (1-\lambda)}{1 + \frac{\tau U}{\delta}} \right)^{1 + \frac{\delta}{U \tau}} \right] \quad (22)$$

Similarly, equation (16) for the time becomes

$$t = \tau \log \left[\left(\frac{1}{1-\lambda} \right) \frac{1 + \frac{\tau U}{\delta} (1-\lambda)}{1 + \frac{\tau U}{\delta}} \right] \quad (23)$$

Now $\tau U / \delta = \rho D_p U / 60 \mu = Re_o / 60$, where Re_o is the Reynolds number for the particle based on fluid velocity U . Also writing $U \tau = S_o$ - the "range" for Stokesian motion, equations (22) and (23) may be written

$$\frac{S}{S_o} = \log \left(\frac{1}{1-\lambda} \right) + \left(1 + \frac{60}{Re_o} \right) \log \left[1 - \frac{\lambda}{1 + \frac{60}{Re_o}} \right] \quad (24)$$

and

$$\frac{t}{\tau} = \log \left(\frac{1}{1-\lambda} \right) + \log \left[1 - \frac{\lambda}{1 + \frac{60}{Re_o}} \right] \quad (25)$$

For pure Stokesian motion, these equations reduce to the limiting equations as $Re_o \rightarrow 0$

$$\frac{S}{S_o} = \log \left(\frac{1}{1-\lambda} \right) - \lambda \quad (26)$$

and

$$\frac{t}{\tau} = \log \left(\frac{1}{1-\lambda} \right) \quad (27)$$

All four above equations are dimensionless and will therefore be valid for any spherical particle in any viscous fluid. The parameters Re_o , S_o and τ will, however, depend on the properties of the particular fluid and particle considered. It should be noted that

$$S_o = U \tau = U D_p^2 \frac{\rho_p}{18 \mu} = \left[\frac{D_p \rho_p}{18 \rho} \right] Re_o \quad (28)$$

SPHERICAL PARTICLE IN ACCELERATING VELOCITY FIELD

The following equation relates final particle velocity, V , as accelerated by a gas flow of uniform velocity, U , over a section, S , where the particle velocity at the entrance of the section, V_0 , is not necessarily zero.

$$\frac{S}{S_0} = \log \left[\left(\frac{1 - \frac{V_0}{U}}{1 - \frac{V}{U}} \right) \left(\frac{1 + \frac{R_{CO}}{60} \left(1 - \frac{V}{U} \right)}{1 + \frac{R_{CO}}{60} \left(1 - \frac{V_0}{U} \right)} \right) \right] \quad 29$$

$$\frac{60}{R_{CO}} \log \left[\frac{\left[1 + \frac{R_{CO}}{60} \left(1 - \frac{V}{U} \right) \right]}{\left[1 + \frac{R_{CO}}{60} \left(1 - \frac{V_0}{U} \right) \right]} \right]$$

This form is readily derivable from equation 21 given that:

$$W_0 = U - V_0$$

$$W = U - V$$

$$\frac{\delta}{U\tau} = \frac{60}{R_{CO}}$$

and

$$U\tau = S_0$$

The above equation can now be used to iteratively calculate final particle velocity at the exit of a gas stream where gas velocity varies with position in the stream. The iteration approximates the following explicit expression for particle velocity at tube exit V_f :

$$V_f = \int_0^L V (R_{CO}(L)) \cdot dL$$

i. e., an accelerating velocity field can be mathematically considered in small segments to estimate the integral of particle acceleration over its length. Variability of Reynolds number with tube length can be considered in this manner.

"Basic" language computer codes shown in Figures A-2a and A-2b have been written to compute particle velocity in an accelerating gas stream by the above method.

In the form shown in Figure A-2b, it estimates gas velocity and density as linear functions of tube length from tube entrance and entrance conditions. The form in Figure A-2a accepts a nonlinear gas velocity and density versus length input.

```

1000 PRINT "THE 50000 LB/HR OF AIR FOR 10000"
1010 PRINT
1020 INPUT "TUBE INCREMENT SIZE IN FT" X
1030 INPUT "X"
1040 IF X < 5 THEN 1004
1050 GOTO 1005
1060 X=0.05
1070 GOTO 1004
1080 X=0.10
1090 PRINT "TUBE INCREMENT SIZE = 0.10 FT"
1100 INPUT "GAS VISCOSITY IN LB HR-FT" V
1110 INPUT "V"
1120 PRINT "GAS VISCOSITY = 0.01 LB HR-FT"
1130 INPUT "PARTICLE DIAMETER IN MICRONS" D
1140 INPUT "D"
1150 PRINT "PARTICLE DIAMETER = 10 MICRONS"
1160 INPUT "PARTICLE DENSITY IN G/CC" R
1170 INPUT "R"
1180 PRINT "PARTICLE DENSITY = 1.0 G/CC"
1190 PRINT
1200 GOTO 1004
1210 GOTO 1004
1220 GOTO 1004
1230 GOTO 1004
1240 GOTO 1004
1250 GOTO 1004
1260 GOTO 1004
1270 GOTO 1004
1280 GOTO 1004
1290 GOTO 1004
1300 GOTO 1004
1310 GOTO 1004
1320 GOTO 1004
1330 GOTO 1004
1340 GOTO 1004
1350 GOTO 1004
1360 GOTO 1004
1370 GOTO 1004
1380 GOTO 1004
1390 GOTO 1004
1400 GOTO 1004
1410 GOTO 1004
1420 GOTO 1004
1430 GOTO 1004
1440 GOTO 1004
1450 GOTO 1004
1460 GOTO 1004
1470 GOTO 1004
1480 GOTO 1004
1490 GOTO 1004
1500 GOTO 1004
1510 GOTO 1004
1520 GOTO 1004
1530 GOTO 1004
1540 GOTO 1004
1550 GOTO 1004
1560 GOTO 1004
1570 GOTO 1004
1580 GOTO 1004
1590 GOTO 1004
1600 GOTO 1004
1610 GOTO 1004
1620 GOTO 1004
1630 GOTO 1004
1640 GOTO 1004
1650 GOTO 1004
1660 GOTO 1004
1670 GOTO 1004
1680 GOTO 1004
1690 GOTO 1004
1700 GOTO 1004
1710 GOTO 1004
1720 GOTO 1004
1730 GOTO 1004
1740 GOTO 1004
1750 GOTO 1004
1760 GOTO 1004
1770 GOTO 1004
1780 GOTO 1004
1790 GOTO 1004
1800 GOTO 1004
1810 GOTO 1004
1820 GOTO 1004
1830 GOTO 1004
1840 GOTO 1004
1850 GOTO 1004
1860 GOTO 1004
1870 GOTO 1004
1880 GOTO 1004
1890 GOTO 1004
1900 GOTO 1004
1910 GOTO 1004
1920 GOTO 1004
1930 GOTO 1004
1940 GOTO 1004
1950 GOTO 1004
1960 GOTO 1004
1970 GOTO 1004
1980 GOTO 1004
1990 GOTO 1004
2000 GOTO 1004

```

Figure A-2a. Computer Code for Calculating Exit Particle Velocity From Accelerating Tube With Variable Gas Velocity

```

1000 PRINT TAB(15) "CALCULATION OF U0 FOR U0 0
1010 A1=0
1020 PRINT
1030 DISP "TUBE INCREMENT SIZE= H"
1040 INPUT H
1050 IF H<5 THEN 1070
1060 GOTO 1040
1070 H=0.05
1080 GOTO 1110
1090 H=0.02
1100 IF H=1 THEN 1410
1110 PRINT "TUBE INCREMENT SIZE= H"
1120 IF H=1 THEN 1430
1130 DISP "AIR VISCOSITY= LB/HR-FT"
1140 INPUT V
1150 PRINT "AIR VISCOSITY= V"
1160 DISP "PARTICLE DIAMETER= MICRONS"
1170 INPUT D
1180 PRINT "PARTICLE DIAMETER= D"
1190 DISP "PARTICLE DENSITY= GM/CC"
1200 INPUT R
1210 PRINT "PARTICLE DENSITY= R"
1220 M=0
1230 M=0
1240 M=0
1241 DISP "INITIAL AIR VELOCITY= U0 FEET"
1242 INPUT U0
1243 PRINT "INITIAL AIR VELOCITY= U0"
1244 DISP "FINAL AIR VELOCITY"
1245 INPUT U1
1246 PRINT "FINAL AIR VELOCITY= U1"
1247 DISP "INITIAL AIR DENSITY"
1248 INPUT A0
1249 PRINT "INITIAL AIR DENSITY= A0"
1250 M=0
1260 PRINT "CURRENT PARTICLE VELOCITY= V"
1270 IF M<0.9999 THEN 1670
1280 PRINT
1290 V=U1-U0-U0*10*(M+0.1)
1300 A=A0*(1-M)
1310 M=M+0.1
1320 F=U0+U1-U0-0.9999-0.94E-05 M
1330 GOTO 1350
1340 PRINT "REYNOLDS NUMBER= R"
1350 R=1.42E-07*A*V*D
1360 GOTO 1430
1370 PRINT "TIME CONSTANT= T"
1380 T=0.001
1390 S=0
1400 S=S+1
1410 V=V*(1+0.31-S)
1420 GOTO 1480
1430 V=V*(1+0.31-S)
1440 L=M*U
1450 IF L<1 THEN 1510
1500 GOTO 1550
1510 DISP "TUBE INCREMENT IS TOO LONG"
1520 WAIT 3000
1530 A7=1
1540 GOTO 1030
1550 L0=0
1560 S=S+LOG (1-L0)/(1-L) + 1+60 R -LOG (1+R 60)/(1-L) + 1+R 60/(1-L)
1570 IF S<1 THEN 1600
1580 DISP S
1590 GOTO 1610
1600 DISP " "
1610 IF S<1.00001 THEN 1450
1620 IF S<0.99999 THEN 1640
1630 IF S<1 THEN 1470
1640 PRINT "INCREMENT OF PART. TRAP= 1/3"
1650 PRINT "INCREMENTS= "
1660 IF M<0.1 THEN 1250
1670 DISP " "
1680 PRINT " "
1690 END

```

THIS DOCUMENT IS AVAILABLE FROM NTIS
 PB80-0014 - \$0.50 (including TV)

Figure A-2b. Computer Code for Calculating Exit Particle Velocity From Accelerating Tube With Assumed Linear Variable Gas Velocity and Density Over Length

APPENDIX B

EROSION TEST RESULTS

APPENDIX B

EROSION TEST RESULTS

The erosion data used to generate Figures 8 and 9 is given. Volume loss per impact is the average value of as many as ten tests. Number of impacts were calculated assuming spherical particles of the diameter given. All erosion tests were performed at 90-degree impingement angle and ambient temperature.

NC 132 Hot -Pressed Si₃N₄/SiC Particles

Part. Dia. (μm)	Velocity (mps)	Volume Loss Impact (m ³)	R3.67V3.17	Volume Loss R3.67V3.17	R3.67V2.44	Volume Loss R3.67V2.44
8	61	1.1 x 10 ⁻²¹	7.1 x 10 ⁻¹⁵	1.5 x 10 ⁻⁷	3.5 x 10 ⁻¹⁶	3.1 x 10 ⁻⁶
	122	1.5 x 10 ⁻²⁰	6.4 x 10 ⁻¹⁴	2.3 x 10 ⁻⁷	1.9 x 10 ⁻¹⁵	7.9 x 10 ⁻⁶
	183	5.9 x 10 ⁻¹⁹	2.3 x 10 ⁻¹³	2.6 x 10 ⁻⁷	5.1 x 10 ⁻¹⁵	1.2 x 10 ⁻⁵
	244	1.6 x 10 ⁻¹⁹	5.7 x 10 ⁻¹³	2.8 x 10 ⁻⁷	1.0 x 10 ⁻¹⁴	1.6 x 10 ⁻⁵
	292	3.4 x 10 ⁻¹⁹	1.0 x 10 ⁻¹²	3.4 x 10 ⁻⁷	1.6 x 10 ⁻¹⁴	2.1 x 10 ⁻⁵
50	58	1.4 x 10 ⁻¹⁸	5.0 x 10 ⁻¹²	2.8 x 10 ⁻⁷	2.6 x 10 ⁻¹³	5.4 x 10 ⁻⁶
	107	1.5 x 10 ⁻¹⁷	3.5 x 10 ⁻¹¹	4.3 x 10 ⁻⁷	1.15 x 10 ⁻¹²	1.3 x 10 ⁻⁵
	153	3.7 x 10 ⁻¹⁷	1.1 x 10 ⁻¹⁰	3.4 x 10 ⁻⁷	2.8 x 10 ⁻¹²	1.3 x 10 ⁻⁵
	198	8.1 x 10 ⁻¹⁷	2.5 x 10 ⁻¹⁰	3.2 x 10 ⁻⁷	5.2 x 10 ⁻¹²	1.6 x 10 ⁻⁵
	241	1.5 x 10 ⁻¹⁶	4.6 x 10 ⁻¹⁰	3.3 x 10 ⁻⁷	8.4 x 10 ⁻¹²	1.8 x 10 ⁻⁵
102	52	2.0 x 10 ⁻¹⁷	4.9 x 10 ⁻¹¹	4.1 x 10 ⁻⁷	2.7 x 10 ⁻¹²	7.4 x 10 ⁻⁶
	97	1.3 x 10 ⁻¹⁶	3.5 x 10 ⁻¹⁰	3.7 x 10 ⁻⁷	1.25 x 10 ⁻¹¹	1.0 x 10 ⁻⁵
	138	3.4 x 10 ⁻¹⁶	1.1 x 10 ⁻⁹	3.1 x 10 ⁻⁷	2.9 x 10 ⁻¹¹	1.2 x 10 ⁻⁵
	178	7.2 x 10 ⁻¹⁶	2.4 x 10 ⁻⁹	3.0 x 10 ⁻⁷	5.25 x 10 ⁻¹¹	1.4 x 10 ⁻⁵
	217	1.4 x 10 ⁻¹⁵	4.5 x 10 ⁻⁹	3.1 x 10 ⁻⁷	8.9 x 10 ⁻¹¹	1.6 x 10 ⁻⁵
305	79	2.25 x 10 ⁻¹⁵	1.0 x 10 ⁻⁸	2.25 x 10 ⁻⁷	4.2 x 10 ⁻¹⁰	5.35 x 10 ⁻⁶
	113	1.9 x 10 ⁻¹⁴	3.2 x 10 ⁻⁸	5.9 x 10 ⁻⁷	1.0 x 10 ⁻⁹	1.9 x 10 ⁻⁵
	145	4.4 x 10 ⁻¹⁴	7.0 x 10 ⁻⁸	6.3 x 10 ⁻⁷	1.8 x 10 ⁻⁹	2.4 x 10 ⁻⁵
	176	9.1 x 10 ⁻¹⁴	1.3 x 10 ⁻⁷	7.0 x 10 ⁻⁷	2.95 x 10 ⁻⁹	3.1 x 10 ⁻⁵
560	69	2.6 x 10 ⁻¹⁴	6.2 x 10 ⁻⁸	4.2 x 10 ⁻⁷	2.8 x 10 ⁻⁹	9.3 x 10 ⁻⁶
	98	1.65 x 10 ⁻¹³	1.9 x 10 ⁻⁷	8.8 x 10 ⁻⁷	6.6 x 10 ⁻⁹	2.5 x 10 ⁻⁵
	126	1.1 x 10 ⁻¹²	4.2 x 10 ⁻⁷	8.1 x 10 ⁻⁷	1.2 x 10 ⁻⁸	9.2 x 10 ⁻⁵
	152	2.3 x 10 ⁻¹²	7.5 x 10 ⁻⁷	9.6 x 10 ⁻⁷	1.9 x 10 ⁻⁸	1.2 x 10 ⁻⁴
940	31	7.2 x 10 ⁻¹⁵	3.3 x 10 ⁻⁸	2.2 x 10 ⁻⁷	2.7 x 10 ⁻⁹	2.7 x 10 ⁻⁶
	59	3.4 x 10 ⁻¹³	2.5 x 10 ⁻⁷	1.4 x 10 ⁻⁶	1.3 x 10 ⁻⁸	2.6 x 10 ⁻⁵
	85	9.7 x 10 ⁻¹³	8.0 x 10 ⁻⁷	1.2 x 10 ⁻⁶	3.1 x 10 ⁻⁸	3.1 x 10 ⁻⁵
	109	1.7 x 10 ⁻¹²	1.8 x 10 ⁻⁶	9.7 x 10 ⁻⁷	5.7 x 10 ⁻⁸	3.0 x 10 ⁻⁵
	132	3.4 x 10 ⁻¹²	3.2 x 10 ⁻⁶	1.0 x 10 ⁻⁶	9.1 x 10 ⁻⁸	3.7 x 10 ⁻⁵

NC 132 Hot-Pressed Si₃N₄/Quartz Particles

Part. Dia. (μm)	Velocity (mps)	Volume Loss Impact (m ³)	R3.67V3.17	Volume Loss R3.67V3.17	R3.67V2.44	Volume Loss R3.67V2.44
10	61	5.9 x 10 ⁻²¹	1.6 x 10 ⁻¹⁴	3.7 x 10 ⁻⁷	8.0 x 10 ⁻¹⁶	7.4 x 10 ⁻⁶
	120	1.1 x 10 ⁻²⁰	1.4 x 10 ⁻¹³	7.9 x 10 ⁻⁸	4.1 x 10 ⁻¹⁵	2.7 x 10 ⁻⁶
	175	1.7 x 10 ⁻²⁰	4.5 x 10 ⁻¹³	3.8 x 10 ⁻⁸	1.0 x 10 ⁻¹⁴	1.7 x 10 ⁻⁶
	229	2.2 x 10 ⁻²⁰	1.1 x 10 ⁻¹²	2.0 x 10 ⁻⁸	2.0 x 10 ⁻¹⁴	1.1 x 10 ⁻⁶
	285	2.9 x 10 ⁻²⁰	2.1 x 10 ⁻¹²	1.4 x 10 ⁻⁸	3.4 x 10 ⁻¹⁴	8.5 x 10 ⁻⁷
49	109	3.9 x 10 ⁻¹⁹	3.4 x 10 ⁻¹¹	1.15 x 10 ⁻⁸	1.2 x 10 ⁻¹²	3.25 x 10 ⁻⁷
	157	5.3 x 10 ⁻¹⁹	1.1 x 10 ⁻¹⁰	9.8 x 10 ⁻⁹	3.0 x 10 ⁻¹²	1.8 x 10 ⁻⁷
	202	6.6 x 10 ⁻¹⁹	2.4 x 10 ⁻¹⁰	2.75 x 10 ⁻⁹	5.5 x 10 ⁻¹²	1.2 x 10 ⁻⁷
	247	8.4 x 10 ⁻¹⁹	4.6 x 10 ⁻¹⁰	1.8 x 10 ⁻⁹	9.0 x 10 ⁻¹²	9.3 x 10 ⁻⁸
64	57	4.4 x 10 ⁻¹⁹	1.2 x 10 ⁻¹¹	3.7 x 10 ⁻⁸	6.1 x 10 ⁻¹³	7.2 x 10 ⁻⁷
	106	9.1 x 10 ⁻¹⁹	8.4 x 10 ⁻¹¹	1.1 x 10 ⁻⁸	2.8 x 10 ⁻¹²	3.2 x 10 ⁻⁷
	152	1.3 x 10 ⁻¹⁸	2.6 x 10 ⁻¹⁰	5.0 x 10 ⁻⁹	6.7 x 10 ⁻¹²	1.9 x 10 ⁻⁷
	196	1.7 x 10 ⁻¹⁸	5.9 x 10 ⁻¹⁰	2.9 x 10 ⁻⁹	1.25 x 10 ⁻¹¹	1.4 x 10 ⁻⁷
	239	2.3 x 10 ⁻¹⁸	1.1 x 10 ⁻⁹	2.1 x 10 ⁻⁹	2.0 x 10 ⁻¹¹	1.15 x 10 ⁻⁷
115	52	2.0 x 10 ⁻¹⁸	7.55 x 10 ⁻¹¹	2.6 x 10 ⁻⁸	4.2 x 10 ⁻¹²	4.8 x 10 ⁻⁷
	93	6.3 x 10 ⁻¹⁸	4.8 x 10 ⁻¹⁰	1.3 x 10 ⁻⁸	1.7 x 10 ⁻¹¹	3.7 x 10 ⁻⁷
	140	1.1 x 10 ⁻¹⁷	1.55 x 10 ⁻⁹	7.1 x 10 ⁻⁹	4.6 x 10 ⁻¹¹	2.4 x 10 ⁻⁷
	180	1.3 x 10 ⁻¹⁷	3.9 x 10 ⁻⁹	3.3 x 10 ⁻⁹	8.6 x 10 ⁻¹¹	1.5 x 10 ⁻⁷
	219	1.2 x 10 ⁻¹⁷	7.2 x 10 ⁻⁹	1.7 x 10 ⁻⁹	1.4 x 10 ⁻¹⁰	8.6 x 10 ⁻⁸
273	45	1.5 x 10 ⁻¹⁷	1.1 x 10 ⁻⁹	1.4 x 10 ⁻⁸	7.1 x 10 ⁻¹¹	2.1 x 10 ⁻⁷
	84	5.9 x 10 ⁻¹⁷	8.2 x 10 ⁻⁹	7.2 x 10 ⁻⁹	3.2 x 10 ⁻¹⁰	1.8 x 10 ⁻⁷
	120	9.7 x 10 ⁻¹⁷	2.55 x 10 ⁻⁸	3.8 x 10 ⁻⁹	7.7 x 10 ⁻¹⁰	1.3 x 10 ⁻⁷
	155	1.4 x 10 ⁻¹⁶	5.8 x 10 ⁻⁸	2.4 x 10 ⁻⁹	1.4 x 10 ⁻⁹	1.0 x 10 ⁻⁷
	188	1.3 x 10 ⁻¹⁶	1.1 x 10 ⁻⁷	1.2 x 10 ⁻⁹	2.3 x 10 ⁻⁹	5.6 x 10 ⁻⁸
385	175	4.7 x 10 ⁻¹⁶	3.0 x 10 ⁻⁷	1.6 x 10 ⁻⁹	6.7 x 10 ⁻⁹	7.0 x 10 ⁻⁸

GTE Pressureless Sintered Si₃N₄/SiC Particles

Part. Dia. Size (μ)	Velocity (mps)	Volume Loss Impact (m ³)	R3.67V3.17	Volume Loss* R3.67V3.17	R3.67V2.44	Volume Loss R3.67V2.44
8	61	3.8 x 10 ⁻²¹	7.1 x 10 ⁻¹⁵	5.35 x 10 ⁻⁷	3.5 x 10 ⁻¹⁶	1.1 x 10 ⁻⁵
	183	5.3 x 10 ⁻²⁰	2.3 x 10 ⁻¹³	2.3 x 10 ⁻⁷	5.11 x 10 ⁻¹⁵	1.0 x 10 ⁻⁵
	292	2.9 x 10 ⁻¹⁹	1.0 x 10 ⁻¹²	2.9 x 10 ⁻⁷	1.6 x 10 ⁻¹⁴	1.8 x 10 ⁻⁵
102	52	2.65 x 10 ⁻¹⁷	4.8 x 10 ⁻¹¹	5.5 x 10 ⁻⁷	2.7 x 10 ⁻¹²	9.8 x 10 ⁻⁶
	138	3.4 x 10 ⁻¹⁶	1.1 x 10 ⁻⁹	3.1 x 10 ⁻⁷	2.9 x 10 ⁻¹¹	1.2 x 10 ⁻⁵
	217	1.8 x 10 ⁻¹⁵	4.5 x 10 ⁻⁹	4.0 x 10 ⁻⁷	8.9 x 10 ⁻¹¹	2.0 x 10 ⁻⁵
305	42	5.3 x 10 ⁻¹⁶	1.4 x 10 ⁻⁹	3.8 x 10 ⁻⁷	9.0 x 10 ⁻¹¹	5.9 x 10 ⁻⁶
	113	1.7 x 10 ⁻¹⁴	3.2 x 10 ⁻⁸	5.3 x 10 ⁻⁷	1.0 x 10 ⁻⁹	1.7 x 10 ⁻⁵
	176	6.6 x 10 ⁻¹⁴	1.3 x 10 ⁻⁷	5.1 x 10 ⁻⁷	2.95 x 10 ⁻⁹	2.2 x 10 ⁻⁵
940	31	4.7 x 10 ⁻¹⁴	3.3 x 10 ⁻⁸	1.4 x 10 ⁻⁶	2.7 x 10 ⁻⁹	1.71 x 10 ⁻⁵
	85	8.4 x 10 ⁻¹³	8.0 x 10 ⁻⁷	1.0 x 10 ⁻⁶	3.1 x 10 ⁻⁸	2.7 x 10 ⁻⁵
	132	3.0 x 10 ⁻¹²	3.2 x 10 ⁻⁶	9.4 x 10 ⁻⁷	9.14 x 10 ⁻⁸	3.3 x 10 ⁻⁵

GTE Pressureless Sintered Si₃N₄/Quartz Particles

Part. Dia. (μm)	Velocity (mps)	Volume Loss Impact (m^3)	R3.67v3.17	Volume Loss R3.67v3.17	R3.67v2.44	Volume Loss R3.67v2.44
115	93	3.1×10^{-18}	4.8×10^{-10}	6.5×10^{-9}	1.7×10^{-11}	1.8×10^{-7}
	140	2.8×10^{-18}	1.55×10^{-9}	1.8×10^{-9}	4.6×10^{-11}	6.1×10^{-8}
	180	5.0×10^{-18}	3.9×10^{-9}	1.3×10^{-9}	8.6×10^{-11}	5.8×10^{-8}
	219	8.1×10^{-18}	7.2×10^{-9}	1.1×10^{-9}	1.4×10^{-10}	5.8×10^{-8}
273	45	1.4×10^{-17}	1.1×10^{-9}	1.3×10^{-8}	7.1×10^{-11}	2.0×10^{-7}
	84	4.7×10^{-17}	8.2×10^{-9}	5.7×10^{-9}	3.2×10^{-10}	1.5×10^{-7}
	120	1.2×10^{-16}	2.55×10^{-8}	4.7×10^{-9}	7.7×10^{-10}	1.6×10^{-7}
		8.4×10^{-17}		3.3×10^{-9}		1.1×10^{-7}
	155	9.7×10^{-17}	5.8×10^{-8}	1.7×10^{-9}	1.4×10^{-9}	6.9×10^{-8}
188	1.7×10^{-16}	1.1×10^{-7}	1.5×10^{-9}	2.3×10^{-9}	7.4×10^{-8}	
385	112	1.4×10^{-16}	7.2×10^{-8}	1.9×10^{-9}	2.3×10^{-9}	6.1×10^{-8}
	144	2.8×10^{-16}	1.6×10^{-7}	1.75×10^{-9}	4.2×10^{-9}	6.7×10^{-8}
	174	4.1×10^{-16}	3.0×10^{-7}	1.4×10^{-9}	6.7×10^{-9}	6.1×10^{-8}

NC 350 Reaction Sintered Si₃N₄/SiC Particles

Part. Dia. (μm)	Velocity (mps)	Volume Loss Impact (m^3)	R3.67v3.17	Volume Loss R3.67v3.17	R3.67v2.44	Volume Loss R3.67v2.44
8	61	7.1×10^{-21}	7.1×10^{-15}	1.0×10^{-6}	3.5×10^{-16}	2.0×10^{-5}
	183	4.2×10^{-20}	2.3×10^{-13}	3.2×10^{-7}	5.1×10^{-15}	8.2×10^{-6}
	292	1.8×10^{-19}	1.0×10^{-12}	1.8×10^{-7}	1.6×10^{-14}	1.1×10^{-5}
102	52	2.2×10^{-16}	4.8×10^{-11}	4.6×10^{-6}	2.7×10^{-12}	8.15×10^{-5}
	138	2.3×10^{-15}	1.1×10^{-9}	2.1×10^{-6}	2.9×10^{-11}	7.9×10^{-5}
	217	1.1×10^{-14}	4.5×10^{-9}	2.4×10^{-6}	8.9×10^{-11}	1.2×10^{-4}
305	42	1.1×10^{-14}	1.4×10^{-9}	7.8×10^{-6}	9.0×10^{-11}	1.2×10^{-4}
	113	1.9×10^{-14}	3.2×10^{-8}	5.9×10^{-7}	1.0×10^{-9}	1.9×10^{-5}
	176	4.6×10^{-13}	1.3×10^{-7}	3.5×10^{-6}	2.95×10^{-9}	1.55×10^{-4}
940	31	5.8×10^{-14}	3.3×10^{-8}	1.8×10^{-6}	2.7×10^{-9}	2.15×10^{-5}
	85	5.4×10^{-12}	8.0×10^{-7}	6.75×10^{-6}	3.1×10^{-8}	1.7×10^{-4}
	132	2.2×10^{-11}	3.2×10^{-6}	6.9×10^{-6}	9.1×10^{-8}	2.4×10^{-4}

NC 350 Reaction Sintered Si₃N₄/Quartz Particles

Part.	Velocity	Volume Loss		Volume Loss		Volume Loss
Dia. (μm)	(mps)	Impact (m ³)	R3.67v3.17	R3.67v3.17	R3.67v2.44	R3.67v2.44
10	229	5.0 x 10 ⁻²⁰	1.1 x 10 ⁻¹²	4.55 x 10 ⁻⁸	2.0 x 10 ⁻¹⁴	2.5 x 10 ⁻⁶
	285	1.2 x 10 ⁻¹⁹	2.1 x 10 ⁻¹²	5.7 x 10 ⁻⁸	3.4 x 10 ⁻¹⁴	3.5 x 10 ⁻⁶
49	109	2.6 x 10 ⁻¹⁸	3.4 x 10 ⁻¹¹	7.6 x 10 ⁻⁸	1.2 x 10 ⁻¹²	2.2 x 10 ⁻⁶
	157	1.5 x 10 ⁻¹⁷	1.1 x 10 ⁻¹⁰	1.4 x 10 ⁻⁷	5.5 x 10 ⁻¹²	2.7 x 10 ⁻⁶
	202	2.9 x 10 ⁻¹⁷	2.4 x 10 ⁻¹⁰	1.2 x 10 ⁻⁷	3.0 x 10 ⁻¹²	9.7 x 10 ⁻⁶
	247	5.4 x 10 ⁻¹⁷	4.6 x 10 ⁻¹⁰	1.2 x 10 ⁻⁷	9.0 x 10 ⁻¹²	6.0 x 10 ⁻⁶
64	57	2.2 x 10 ⁻¹⁸	1.2 x 10 ⁻¹¹	1.8 x 10 ⁻⁷	6.1 x 10 ⁻¹³	3.6 x 10 ⁻⁶
	106	9.2 x 10 ⁻¹⁸	8.4 x 10 ⁻¹¹	1.1 x 10 ⁻⁷	2.8 x 10 ⁻¹²	3.3 x 10 ⁻⁶
	152	3.7 x 10 ⁻¹⁷	2.6 x 10 ⁻¹⁰	1.4 x 10 ⁻⁷	6.7 x 10 ⁻¹²	5.5 x 10 ⁻⁶
	196	7.5 x 10 ⁻¹⁷	5.9 x 10 ⁻¹⁰	1.3 x 10 ⁻⁷	1.25 x 10 ⁻¹¹	6.0 x 10 ⁻⁶
	239	1.4 x 10 ⁻¹⁶	1.1 x 10 ⁻⁹	1.3 x 10 ⁻⁷	2.0 x 10 ⁻¹¹	7.0 x 10 ⁻⁶
115	93	1.2 x 10 ⁻¹⁶	4.8 x 10 ⁻¹⁰	2.5 x 10 ⁻⁷	1.7 x 10 ⁻¹¹	7.1 x 10 ⁻⁶
	140	3.0 x 10 ⁻¹⁶	1.55 x 10 ⁻⁹	1.9 x 10 ⁻⁷	4.6 x 10 ⁻¹¹	6.5 x 10 ⁻⁶
	180	7.9 x 10 ⁻¹⁶	3.9 x 10 ⁻⁹	2.0 x 10 ⁻⁷	8.6 x 10 ⁻¹¹	9.2 x 10 ⁻⁶
	219	1.2 x 10 ⁻¹⁵	7.2 x 10 ⁻⁹	1.7 x 10 ⁻⁷	1.4 x 10 ⁻¹⁰	8.6 x 10 ⁻⁶
273	84	7.5 x 10 ⁻¹⁶	8.2 x 10 ⁻⁹	9.1 x 10 ⁻⁸	3.2 x 10 ⁻¹⁰	2.3 x 10 ⁻⁶
	120	2.1 x 10 ⁻¹⁵	2.55 x 10 ⁻⁸	8.2 x 10 ⁻⁸	7.7 x 10 ⁻¹⁰	2.7 x 10 ⁻⁶
	155	1.0 x 10 ⁻¹⁴	5.8 x 10 ⁻⁸	1.7 x 10 ⁻⁷	1.4 x 10 ⁻⁹	7.1 x 10 ⁻⁶
	188	1.95 x 10 ⁻¹⁴	1.1 x 10 ⁻⁷	1.8 x 10 ⁻⁷	2.3 x 10 ⁻⁹	8.5 x 10 ⁻⁶
385	78	1.6 x 10 ⁻¹⁵	2.3 x 10 ⁻⁸	7.0 x 10 ⁻⁸	9.5 x 10 ⁻¹⁰	1.7 x 10 ⁻⁶
	112	6.2 x 10 ⁻¹⁵	7.2 x 10 ⁻⁸	8.6 x 10 ⁻⁸	2.3 x 10 ⁻⁹	2.7 x 10 ⁻⁶
	144	2.1 x 10 ⁻¹⁴	1.6 x 10 ⁻⁷	1.3 x 10 ⁻⁷	4.2 x 10 ⁻⁹	5.0 x 10 ⁻⁶
	174	1.5 x 10 ⁻¹³	3.0 x 10 ⁻⁷	5.0 x 10 ⁻⁷	6.7 x 10 ⁻⁹	2.2 x 10 ⁻⁶

KBI Reaction Sintered Si₃N₄/SiC Particles

Part.	Velocity	Volume Loss		Volume Loss		Volume Loss
Dia. (μm)	(mps)	Impact (m ³)	R3.67v3.17	R3.67v3.17	R3.67v2.44	R3.67v2.44
8	61	7.1 x 10 ⁻²¹	7.1 x 10 ⁻¹⁵	1.0 x 10 ⁻⁶	3.5 x 10 ⁻¹⁶	2.0 x 10 ⁻⁵
	183	1.3 x 10 ⁻¹⁹	2.3 x 10 ⁻¹³	3.1 x 10 ⁻⁵	5.1 x 10 ⁻¹⁵	2.5 x 10 ⁻⁵
	292	6.7 x 10 ⁻¹⁹	1.0 x 10 ⁻¹²	6.7 x 10 ⁻⁷	1.6 x 10 ⁻¹⁴	4.2 x 10 ⁻⁵
50	153	1.9 x 10 ⁻¹⁶	1.1 x 10 ⁻¹⁰	1.7 x 10 ⁻⁶	2.8 x 10 ⁻¹²	6.8 x 10 ⁻⁵
	217	1.1 x 10 ⁻¹⁵	3.3 x 10 ⁻¹⁰	3.3 x 10 ⁻⁶	6.5 x 10 ⁻¹²	1.7 x 10 ⁻⁴
102	52	1.9 x 10 ⁻¹⁶	4.8 x 10 ⁻¹¹	4.0 x 10 ⁻⁶	2.7 x 10 ⁻¹²	7.0 x 10 ⁻⁵
	138	3.6 x 10 ⁻¹⁵	1.1 x 10 ⁻⁹	3.3 x 10 ⁻⁶	2.9 x 10 ⁻¹¹	1.2 x 10 ⁻⁴
	217	1.4 x 10 ⁻¹⁴	4.5 x 10 ⁻⁹	3.1 x 10 ⁻⁶	8.9 x 10 ⁻¹¹	1.6 x 10 ⁻⁴
305	42	5.4 x 10 ⁻¹⁵	1.4 x 10 ⁻⁹	3.9 x 10 ⁻⁶	9.0 x 10 ⁻¹¹	6.0 x 10 ⁻⁵
	113	9.6 x 10 ⁻¹⁴	3.2 x 10 ⁻⁸	3.0 x 10 ⁻⁶	1.0 x 10 ⁻⁹	9.6 x 10 ⁻⁵
	176	3.5 x 10 ⁻¹³	1.3 x 10 ⁻⁷	2.7 x 10 ⁻⁶	2.95 x 10 ⁻⁹	1.2 x 10 ⁻⁴
940	31	2.9 x 10 ⁻¹³	3.3 x 10 ⁻⁸	8.8 x 10 ⁻⁶	2.7 x 10 ⁻⁹	1.1 x 10 ⁻⁴
	85	3.1 x 10 ⁻¹²	8.0 x 10 ⁻⁷	3.9 x 10 ⁻⁶	3.1 x 10 ⁻⁸	1.0 x 10 ⁻⁴
	132	9.2 x 10 ⁻¹²	3.2 x 10 ⁻⁶	2.9 x 10 ⁻⁶	9.1 x 10 ⁻⁸	1.0 x 10 ⁻⁴

KBI Reaction Sintered Si₃N₄/Quartz Particles

Part. Dia. (μm)	Velocity (mps)	Volume Loss Impact (m^3)	R3.67V3.17	Volume Loss R3.67V3.17	R3.67V2.44	Volume Loss R3.67V2.44
10	61	5.4×10^{-20}	1.6×10^{-14}	3.4×10^{-6}	8.0×10^{-16}	6.75×10^{-5}
	175	1.3×10^{-19}	4.5×10^{-13}	2.9×10^{-7}	1.0×10^{-14}	1.3×10^{-5}
	285	4.2×10^{-19}	2.1×10^{-12}	2.0×10^{-7}	3.4×10^{-14}	1.2×10^{-5}
115	52	1.8×10^{-16}	7.55×10^{-11}	2.4×10^{-6}	4.2×10^{-12}	4.3×10^{-5}
	93	5.4×10^{-16}	4.8×10^{-10}	1.1×10^{-6}	1.7×10^{-11}	3.2×10^{-5}
	139	2.4×10^{-15}	1.55×10^{-9}	1.55×10^{-6}	4.6×10^{-11}	5.2×10^{-5}
		1.2×10^{-15}	1.55×10^{-9}	7.7×10^{-7}	4.6×10^{-11}	2.6×10^{-5}
	180	3.8×10^{-15}	3.9×10^{-9}	9.7×10^{-7}	8.6×10^{-11}	4.4×10^{-5}
219	7.5×10^{-15}	7.2×10^{-9}	1.0×10^{-6}	1.4×10^{-10}	5.4×10^{-5}	
273	45	3.5×10^{-15}	1.1×10^{-9}	3.2×10^{-6}	7.1×10^{-11}	4.9×10^{-5}
	120	4.6×10^{-14}	2.55×10^{-8}	1.8×10^{-6}	7.7×10^{-10}	6.0×10^{-5}
	188	1.4×10^{-13}	1.1×10^{-7}	1.3×10^{-6}	2.3×10^{-9}	6.1×10^{-5}
385	41	6.5×10^{-15}	3.0×10^{-9}	2.2×10^{-6}	2.0×10^{-10}	3.25×10^{-5}
	112	1.8×10^{-13}	7.2×10^{-8}	2.5×10^{-6}	2.3×10^{-9}	7.8×10^{-5}
	174	5.4×10^{-13}	3.0×10^{-7}	1.8×10^{-6}	6.7×10^{-9}	8.1×10^{-5}

Irtran I Hot-Pressed MgF₂/Quartz Particles

Part. Dia. (μm)	Velocity (mps)	Volume Loss Impact (m^3)	R3.67V3.17	Volume Loss R3.67V3.17	R3.67V2.44	Volume Loss R3.67V2.44
10	61	6.0×10^{-20}	1.6×10^{-14}	3.75×10^{-6}	8.0×10^{-16}	7.5×10^{-5}
	120	7.3×10^{-19}	1.4×10^{-13}	5.2×10^{-6}	4.1×10^{-15}	1.8×10^{-4}
	175	2.2×10^{-18}	4.5×10^{-13}	4.9×10^{-6}	1.0×10^{-14}	2.2×10^{-4}
	229	4.2×10^{-18}	1.1×10^{-12}	3.8×10^{-6}	2.0×10^{-14}	2.1×10^{-4}
	285	8.0×10^{-18}	2.1×10^{-12}	3.8×10^{-6}	3.4×10^{-14}	2.35×10^{-4}
49	58	9.4×10^{-18}	4.7×10^{-12}	2.0×10^{-6}	2.6×10^{-13}	3.6×10^{-5}
	109	2.2×10^{-16}	3.4×10^{-11}	6.5×10^{-6}	1.2×10^{-12}	1.8×10^{-4}
	157	4.7×10^{-16}	1.1×10^{-10}	4.3×10^{-6}	3.0×10^{-12}	1.6×10^{-4}
	202	1.6×10^{-15}	2.4×10^{-10}	6.7×10^{-6}	5.5×10^{-12}	2.9×10^{-4}
	247	2.5×10^{-15}	4.6×10^{-10}	5.6×10^{-6}	9.0×10^{-12}	2.8×10^{-4}
64	57	3.3×10^{-17}	1.2×10^{-11}	2.75×10^{-6}	6.1×10^{-13}	5.4×10^{-5}
	106	4.4×10^{-16}	8.4×10^{-11}	5.2×10^{-6}	2.8×10^{-12}	1.6×10^{-4}
	152	2.5×10^{-15}	2.6×10^{-10}	9.6×10^{-6}	6.7×10^{-12}	3.7×10^{-4}
	196	5.0×10^{-15}	5.9×10^{-10}	8.5×10^{-6}	1.25×10^{-11}	4.0×10^{-4}
	239	9.2×10^{-15}	1.1×10^{-9}	8.4×10^{-6}	2.0×10^{-11}	4.6×10^{-4}
115	52	6.6×10^{-16}	7.55×10^{-11}	8.7×10^{-6}	4.2×10^{-12}	1.6×10^{-4}
	93	3.7×10^{-15}	4.8×10^{-10}	7.7×10^{-6}	1.7×10^{-11}	2.2×10^{-4}
	139	1.5×10^{-14}	1.55×10^{-9}	9.7×10^{-6}	4.6×10^{-11}	3.3×10^{-4}
	180	3.7×10^{-14}	3.9×10^{-9}	9.5×10^{-6}	8.6×10^{-11}	4.3×10^{-4}
	219	6.0×10^{-14}	7.2×10^{-9}	8.3×10^{-6}	1.4×10^{-10}	4.3×10^{-4}
273	45	6.25×10^{-15}	1.1×10^{-9}	5.7×10^{-6}	7.1×10^{-11}	8.8×10^{-5}
	84	3.4×10^{-14}	8.2×10^{-9}	4.1×10^{-6}	3.2×10^{-10}	1.1×10^{-4}
	120	2.3×10^{-13}	2.55×10^{-8}	9.0×10^{-6}	7.7×10^{-10}	3.0×10^{-4}
	155	5.7×10^{-13}	5.8×10^{-8}	9.8×10^{-6}	1.4×10^{-9}	4.1×10^{-4}
	188	9.4×10^{-13}	1.1×10^{-7}	8.5×10^{-6}	2.3×10^{-9}	4.1×10^{-4}
385	41	6.2×10^{-15}	3.0×10^{-9}	2.1×10^{-6}	2.0×10^{-10}	3.1×10^{-5}
	78	2.3×10^{-13}	2.3×10^{-8}	1.0×10^{-5}	9.5×10^{-10}	2.4×10^{-4}
	112	1.7×10^{-12}	7.2×10^{-8}	2.4×10^{-5}	2.3×10^{-9}	7.4×10^{-4}
	144	2.0×10^{-12}	1.6×10^{-7}	1.25×10^{-5}	4.2×10^{-9}	4.8×10^{-4}
	174	3.6×10^{-12}	3.0×10^{-7}	1.2×10^{-5}	6.7×10^{-9}	5.4×10^{-4}

Irran I Hot Pressed MgF₂/SiC Particles

Part.	Velocity	Volume Loss		Volume Loss		Volume Loss
Dia.	(mps)	Impact (m ³)	R3.67v3.17	R3.67v3.17	R3.67v2.44	R3.67v2.44
(μm)						
8	61	1.8 x 10 ⁻²⁰	7.0 x 10 ⁻¹⁵	2.6 x 10 ⁻⁶	3.5 x 10 ⁻¹⁶	5.1 x 10 ⁻⁵
	122	4.7 x 10 ⁻²⁰	6.4 x 10 ⁻¹⁴	7.3 x 10 ⁻⁷	1.9 x 10 ⁻¹⁵	2.5 x 10 ⁻⁵
	183	5.0 x 10 ⁻¹⁹	2.3 x 10 ⁻¹³	2.2 x 10 ⁻⁶	5.1 x 10 ⁻¹⁵	9.7 x 10 ⁻⁵
	244	1.0 x 10 ⁻¹⁸	5.7 x 10 ⁻¹³	1.75 x 10 ⁻⁶	1.0 x 10 ⁻¹⁴	1.0 x 10 ⁻⁴
	292	8.0 x 10 ⁻¹⁹	1.0 x 10 ⁻¹²	8.0 x 10 ⁻⁷	1.6 x 10 ⁻¹⁴	5.0 x 10 ⁻⁵
50	58	4.7 x 10 ⁻¹⁷	5.0 x 10 ⁻¹²	9.4 x 10 ⁻⁶	2.6 x 10 ⁻¹³	1.8 x 10 ⁻⁴
	107	2.8 x 10 ⁻¹⁶	3.5 x 10 ⁻¹¹	8.0 x 10 ⁻⁶	1.15 x 10 ⁻¹²	2.4 x 10 ⁻⁴
	153	8.7 x 10 ⁻¹⁶	1.1 x 10 ⁻¹⁰	7.9 x 10 ⁻⁶	2.8 x 10 ⁻¹²	3.1 x 10 ⁻⁴
	198	2.2 x 10 ⁻¹⁵	2.5 x 10 ⁻¹⁰	8.8 x 10 ⁻⁶	5.2 x 10 ⁻¹²	4.2 x 10 ⁻⁴
	241	4.1 x 10 ⁻¹⁵	4.6 x 10 ⁻¹⁰	8.9 x 10 ⁻⁶	8.4 x 10 ⁻¹²	4.9 x 10 ⁻⁴
102	52	6.0 x 10 ⁻¹⁶	4.9 x 10 ⁻¹¹	1.22 x 10 ⁻⁵	2.7 x 10 ⁻¹²	2.2 x 10 ⁻⁴
	97	2.7 x 10 ⁻¹⁵	3.5 x 10 ⁻¹⁰	7.7 x 10 ⁻⁶	1.25 x 10 ⁻¹¹	2.2 x 10 ⁻⁴
	138	1.7 x 10 ⁻¹⁴	1.1 x 10 ⁻⁹	1.55 x 10 ⁻⁵	2.9 x 10 ⁻¹¹	5.9 x 10 ⁻⁴
	178	3.1 x 10 ⁻¹⁴	2.4 x 10 ⁻⁹	1.3 x 10 ⁻⁵	5.25 x 10 ⁻¹¹	5.9 x 10 ⁻⁴
	217	5.3 x 10 ⁻¹⁴	4.5 x 10 ⁻⁹	1.2 x 10 ⁻⁵	8.9 x 10 ⁻¹¹	5.95 x 10 ⁻⁴
305	42	1.75 x 10 ⁻¹⁴	1.4 x 10 ⁻⁹	1.25 x 10 ⁻⁵	9.0 x 10 ⁻¹¹	1.9 x 10 ⁻⁴
	79	1.6 x 10 ⁻¹³	1.0 x 10 ⁻⁸	1.0 x 10 ⁻⁵	4.2 x 10 ⁻¹⁰	3.8 x 10 ⁻⁴
	113	6.7 x 10 ⁻¹³	3.2 x 10 ⁻⁸	2.1 x 10 ⁻⁵	1.0 x 10 ⁻⁹	6.7 x 10 ⁻⁴
	145	9.7 x 10 ⁻¹³	7.0 x 10 ⁻⁸	1.4 x 10 ⁻⁵	1.8 x 10 ⁻⁹	5.4 x 10 ⁻⁴
	176	1.8 x 10 ⁻¹²	1.3 x 10 ⁻⁷	1.4 x 10 ⁻⁵	2.95 x 10 ⁻⁹	6.1 x 10 ⁻⁴
560	36	2.4 x 10 ⁻¹³	7.8 x 10 ⁻⁹	3.1 x 10 ⁻⁵	5.7 x 10 ⁻¹⁰	4.2 x 10 ⁻⁴
	69	1.5 x 10 ⁻¹²	6.2 x 10 ⁻⁸	2.4 x 10 ⁻⁵	2.8 x 10 ⁻⁹	5.35 x 10 ⁻⁴
	98	4.4 x 10 ⁻¹²	1.9 x 10 ⁻⁷	2.3 x 10 ⁻⁵	6.6 x 10 ⁻⁹	6.7 x 10 ⁻⁴
	126	1.0 x 10 ⁻¹¹	4.2 x 10 ⁻⁷	2.4 x 10 ⁻⁵	1.2 x 10 ⁻⁸	8.3 x 10 ⁻⁴
	152	1.3 x 10 ⁻¹¹	7.5 x 10 ⁻⁷	1.7 x 10 ⁻⁵	1.9 x 10 ⁻⁸	6.8 x 10 ⁻⁴
940	31	2.9 x 10 ⁻¹²	3.3 x 10 ⁻⁸	8.8 x 10 ⁻⁵	2.7 x 10 ⁻⁹	1.1 x 10 ⁻³
	59	9.7 x 10 ⁻¹²	2.5 x 10 ⁻⁷	3.4 x 10 ⁻⁵	1.3 x 10 ⁻⁸	7.5 x 10 ⁻⁴
	85	2.2 x 10 ⁻¹¹	8.0 x 10 ⁻⁷	2.75 x 10 ⁻⁵	3.1 x 10 ⁻⁸	7.1 x 10 ⁻⁴
	109	4.4 x 10 ⁻¹¹	1.8 x 10 ⁻⁶	2.4 x 10 ⁻⁵	5.7 x 10 ⁻⁸	7.7 x 10 ⁻⁴
	132	4.7 x 10 ⁻¹¹	3.2 x 10 ⁻⁶	1.5 x 10 ⁻⁵	9.1 x 10 ⁻⁸	5.1 x 10 ⁻⁴

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