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HIGH PRESSURE AND MOLECULAR WEIGHT VARIATION OF
ELECTRICAL RELAXATION IN POLY(ETHYLENE OXIDE)(U) NAVAL
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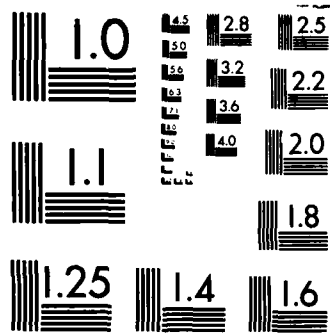
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Task No. NR 627-793

TECHNICAL REPORT NO. 15

HIGH PRESSURE AND MOLECULAR WEIGHT VARIATION OF ELECTRICAL RELAXATION IN
POLY(ETHYLENE OXIDE)

by

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Prepared for Publication

in

Journal of Applied Physics

U. S. Naval Academy
Department of Physics
Annapolis, MD 21402

June 1985

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REPORT DOCUMENTATION PAGE		READ INSTRUCTIONS BEFORE COMPLETING FORM
1. REPORT NUMBER 15	2. GOVT ACCESSION NO. A157660	3. RECIPIENT'S CATALOG NUMBER
4. TITLE (and Subtitle) HIGH PRESSURE AND MOLECULAR WEIGHT VARIATION OF ELECTRICAL RELAXATION IN POLY(ETHYLENE OXIDE)		5. TYPE OF REPORT & PERIOD COVERED Interim technical report
7. AUTHOR(s) JOHN J. FONTANELLA & MARY C. WINTERSGILL		6. PERFORMING ORG. REPORT NUMBER
9. PERFORMING ORGANIZATION NAME AND ADDRESS Physics Department U. S. Naval Academy Annapolis, MD 21402		8. CONTRACT OR GRANT NUMBER(s) N00014-85-AF-00001
11. CONTROLLING OFFICE NAME AND ADDRESS Office of Naval Research Attn. Code 413, 800 N. Quincy St. Arlington, VA 22217		10. PROGRAM ELEMENT, PROJECT, TASK AREA & WORK UNIT NUMBERS NR No. 627-793
14. MONITORING AGENCY NAME & ADDRESS (if different from Controlling Office)		12. REPORT DATE June 1985
		13. NUMBER OF PAGES 23
		15. SECURITY CLASS. (of this report)
		15a. DECLASSIFICATION/DOWNGRADING SCHEDULE
16. DISTRIBUTION STATEMENT (of this Report) Approved for public release and sale. Distribution unlimited.		
17. DISTRIBUTION STATEMENT (of the abstract entered in Block 20, if different from Report)		
18. SUPPLEMENTARY NOTES		
19. KEY WORDS (Continue on reverse side if necessary and identify by block number) Solid electrolytes, polymer electrolytes, poly(ethylene oxide), electrical relaxation, high pressures, molecular weight.		
20. ABSTRACT (Continue on reverse side if necessary and identify by block number) Vacuum audio frequency complex admittance measurements have been performed on poly(ethylene oxide) using a fully automated dielectric spectrometer. Measurements have been made over the temperature range 5.5-320 K for molecular weights from 1.85×10^4 to 4×10^6 . The τ relaxation is essentially unchanged over the range of molecular weights studied. This represents evidence against assigning the τ relaxation to the motion of		

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High Pressure and Molecular Weight Variation of Electrical Relaxation in
Poly(ethylene oxide)

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ABSTRACT

Vacuum audio frequency complex admittance measurements have been performed on poly(ethylene oxide) using a fully automated dielectric spectrometer. Measurements have been made over the temperature range 5.5-320K for molecular weights from 1.85×10^4 to 4×10^6 . The γ relaxation is essentially unchanged over the range of molecular weights studied. This represents evidence against assigning the γ relaxation to the motion of end groups. Next, previously reported data for the γ relaxation is reanalyzed using a reduced plot. The activation volume is found to be 3-4 cm^3/mol in good agreement with the previous work and the small value implies that the relaxation is associated with the motion of very small segments of the polymer chain. Next, the relaxation associated with the glass transition, α_a , is only observed in temperature for some samples but is observed in both frequency and temperature for others. The strength of α_a decreases monotonically as molecular weight decreases. Finally, α_a was studied up to pressures of about 0.2 GPa for 5×10^6 molecular weight material. The results for the shift of α_a with pressure are in qualitative agreement with recent theories.

I. INTRODUCTION

Poly(ethylene oxide) (PEO) is of current interest as the prototype polymeric solid electrolyte.¹⁻³ As a consequence, the basic properties of this material, and particularly the electrical properties, are being extensively studied. As a result, there is a considerable amount of literature relating to this material. Much of the early work has been reviewed by McCrum et al.⁴ and Hedvig.⁵ References to some of the more recent work can be found in the papers by Porter and Boyd,⁶ and Se et al.⁷ More recently, the authors have presented⁸⁻¹¹ the results of audio frequency electrical relaxation (DR) measurements on PEO and PEO complexed with a variety of alkali metal and alkaline earth salts. In addition, the effect of high pressure on the electrical conductivity in some of those materials has been presented.¹² In PEO to date, it appears that only the γ relaxation has been studied at high pressures.¹³ In the present paper, the effect of pressure on the glass transition relaxation, α_a , is reported. In addition, the previously reported data for the effect of pressure on the γ relaxation¹³ are reanalyzed in terms of a reduced plot. Finally, in order to gain further information concerning γ and α_a , different molecular weights have been studied.

II. EXPERIMENT

The PEO was obtained from Polysciences, with MW 1.85×10^4 to 5×10^6 . All but the 1.85×10^4 samples were melt-pressed at about 100°C using a Buehler Simplimet II press. The melt-pressed samples were about 25 mm in

diameter and 1.5 mm thick. Aluminum electrodes were evaporated onto the surfaces in a three terminal configuration. For the vacuum measurements, the guarded electrode was about 14 mm in diameter while for the high pressure measurements (MW 5×10^6) it was 9 mm in diameter. The remainder of the guarded side of the sample was the guard ring and the full face of the opposite side of the sample was evaporated as the high electrode. The guard gap was on the order of 0.1 mm wide.

Measurements of the capacitance and conductance divided by the angular frequency were performed in vacuum over the temperature range 5.5-320K as described elsewhere.⁹ In the present case, a new bridge constructed by one of the authors (C.G.A.) was used to perform the measurements. The new, fully automated bridge operates at seventeen frequencies over the range 10 - 10^5 Hz.

The data were transformed to the complex dielectric constant, $\epsilon^* = \epsilon' - j\epsilon''$ using procedures described in detail elsewhere.⁹ Once again no thermal expansion correction was included in the data reduction. The results of room temperature geometrical measurements and the capacitance data at 5.5K yielded $\epsilon' = 2.87, 3.01, 2.89$, and 2.95 for MW $4 \times 10^6, 6 \times 10^5, 3 \times 10^5$, and 1×10^5 respectively. These values agree to within the experimental uncertainty in that the precision is estimated to be about 3%. The values are, however, not accurate to 3% as thermal expansion of the samples was neglected in the data analysis.

The MW 1.85×10^4 samples were not melt-pressed as they were received as platelets. They were also evaporated in a three terminal configuration with a 9 mm diameter guarded electrode. The thickness was about 0.5 mm and the guard ring was about 5 mm wide. However, a geometrical measurement was

not used. The low temperature value of ϵ' was set equal to the average of the values quoted above for the higher molecular weight materials.

The high pressure measurements were carried out in the vessel described elsewhere.¹⁴ The pressure fluid was a mixture of pentane and Spinesstic 22 and the temperatures were achieved using a mechanically refrigerated ethylene glycol bath. The vacuum audio frequency complex impedance measurements were performed using a different, fully automated bridge constructed by one of the authors (C.G.A.) which operates with a more restricted frequency range.

IV. RESULTS AND DISCUSSION

Figure 1 shows typical vacuum electrical relaxation spectra at five frequencies between 10^2 and 10^5 Hz for PEO, in this case for a sample of molecular weight 4×10^6 . Figure 2 shows the data at 10^3 Hz for five molecular weights between 1.85×10^4 and 4×10^6 . The relaxation spectra at five frequencies for the 1.85×10^4 molecular weight material are shown in Figure 3. Two strong features are observed, the γ relaxation with peaks from about 140-200K and the α_a relaxation. It is interesting to note that at the highest frequency, the α_a and γ relaxations merge as predicted by Porter and Boyd.⁶ In addition, there is a weak relaxation region in all of the samples from 60-100K. The latter relaxation will not be discussed further in this paper. As no features were observed below 50 K, that region is not shown in the plots.

The first result of interest is that it is clear from Fig 2 that the γ relaxation does not vary much with molecular weight. If anything, there

is a slight decrease in the strength as the molecular weight decreases. This argues against assigning it to chain end hydroxyl groups.⁵ A similar result and conclusion was obtained by Se et al.⁷

In order to gain further information concerning the γ relaxation, high pressure measurements were performed and have been reported previously.¹³ In that paper, the data were analyzed using the "standard" technique of fitting a Cole-Cole expression to individual data sets. The relaxation time for each pressure and temperature was then determined from the peak position. In the present paper the data are reanalyzed in terms of a "reduced plot." The plot is shown in Fig. 4. All of the data for 17 frequencies, 3 temperatures, and 7 pressures have been "normalized" to a common curve using least square fitting computer techniques described elsewhere.¹³ In Ref. 13, this type of fitting procedure was applied to vacuum data using a Havriliak-Negami expression. In the present case, the empirical expression of Jonscher¹⁵ has been used:

$$\epsilon'' = \frac{\epsilon_s - \epsilon_\infty}{[(\omega/\omega_p)^{-m} + (\omega/\omega_p)^{(1-n)}]} \quad (1)$$

The significant best fit parameters are $n=0.84$ and $m=0.33$. The strength and peak position are arbitrary, however, further information can be gained from the amount of shift in $\log_{10}(\omega_p)$ necessary to reach the master curve. The results for these logarithmic frequency shifts of the data to the master curve for both pressure and temperature are shown in Fig. 5. The resultant slopes for the pressure data are given in Table I. These were then used to calculate the activation volume via:

$$\Delta V^* = - kT \frac{d \ln w_p}{dP} \quad (2)$$

The resultant activation volumes are also listed in Table I along with the results from the previous paper. The agreement is quite good considering the widely differing techniques used to arrive at the results. The activation volume, 3-4 cm³/mol, is very small which supports assigning the relaxation to either chain end hydroxyl groups or very small segments of the polymer chain. However, considering the above results of the molecular weight studies, the latter mechanism is to be preferred.

Finally, the variation of the glass transition with molecular weight is of interest. Since the strength of α_a decreases with decreasing molecular weight, it is apparent that the degree of crystallinity increases with decreasing molecular weight over the range of molecular weights used in the present study. This is a well known phenomenon in PEO. Also, the work of Faucher et al.¹⁶ indicates that there is a maximum in the glass transition temperature at a molecular weight of about 6000. This increase in T_g with decreasing molecular weight is evident in Figure 2, although the shift in peak positions between the 1×10^5 and the 1.85×10^4 molecular weight samples is enhanced by the change in the frequency dependence of the peak position which can be observed by comparison of this effect in figures 1 and 3.

The results for the high molecular weight material are particularly interesting in that a peak for α_a was never observed in frequency at any single temperature over the frequency range 10 to 10^5 Hz, although it is observed in the temperature sweeps. For the 18,500 MW material, however,

the material exhibited more normal behavior with the peak being observed in frequency as is apparent in Figure 3. However, as mentioned above, this is the only material which was not melt-pressed and thus the result may not solely be a consequence of the molecular weight. Experiments to determine the origin of this phenomenon are currently being undertaken.

As the effect of pressure on the glass transition in PEO does not appear to have been reported previously, those measurements were carried out for the high molecular weight material. Some of the data at 1000 Hz are shown in Fig. 6. For this peak, very strong shifts with pressure are observed as is expected for a glass transition. The shift of the peak position, T_{α} , with pressure is shown for two frequencies in Fig. 7 together with the best fit of the following equation to the data:

$$T_{\alpha} = T_{\alpha}^0 + BP + CP^2 \quad (3)$$

The best fit parameters for the two frequencies and those for a lower correlation fit at a third frequency are listed in Table II. The curvature which is apparent in the plots is consistent with the predictions of the retical treatments involving segmental relaxation phenomena¹⁷ and with pressure results for other polymers.¹⁸

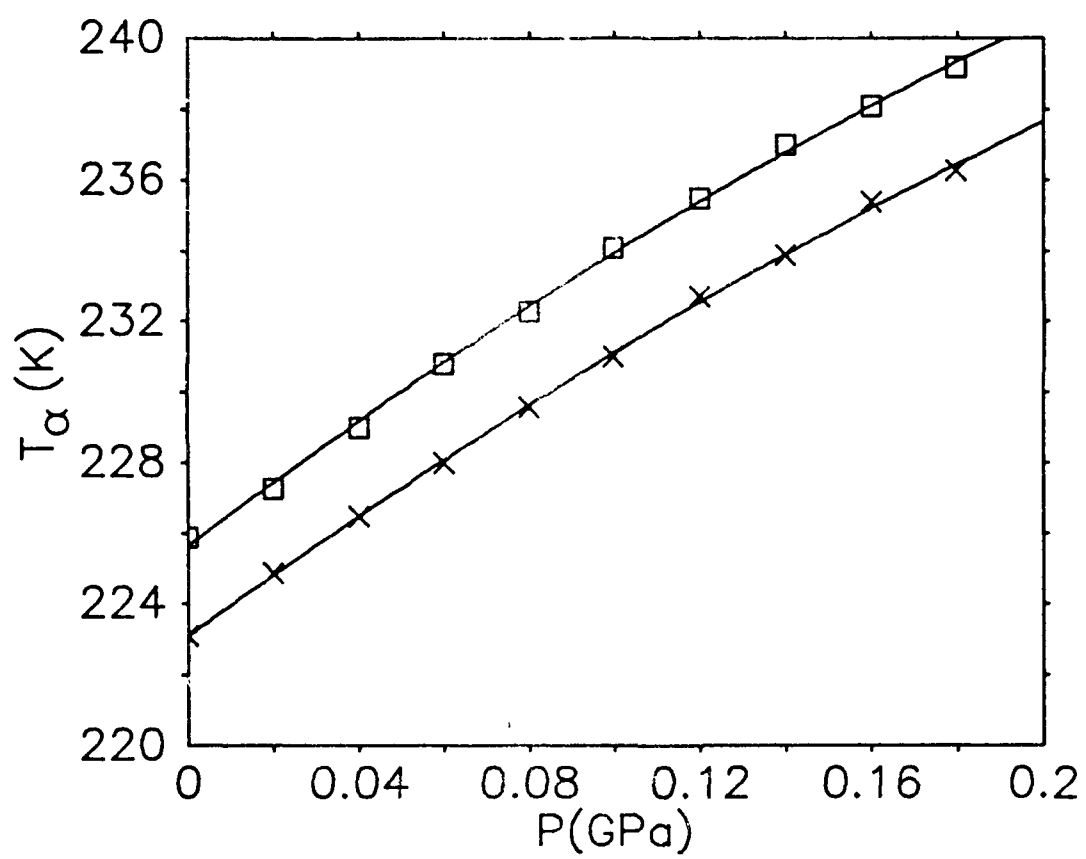
Since the peaks were not observed in frequency, a complete treatment in terms of the WLF equation cannot be carried out. However, both the frequency dependence of the effect of pressure on T_{α} and the magnitude of dT_{α}/dP agree qualitatively with the predictions of Arentas and Akrivos-Tubridi¹⁸ who show that:

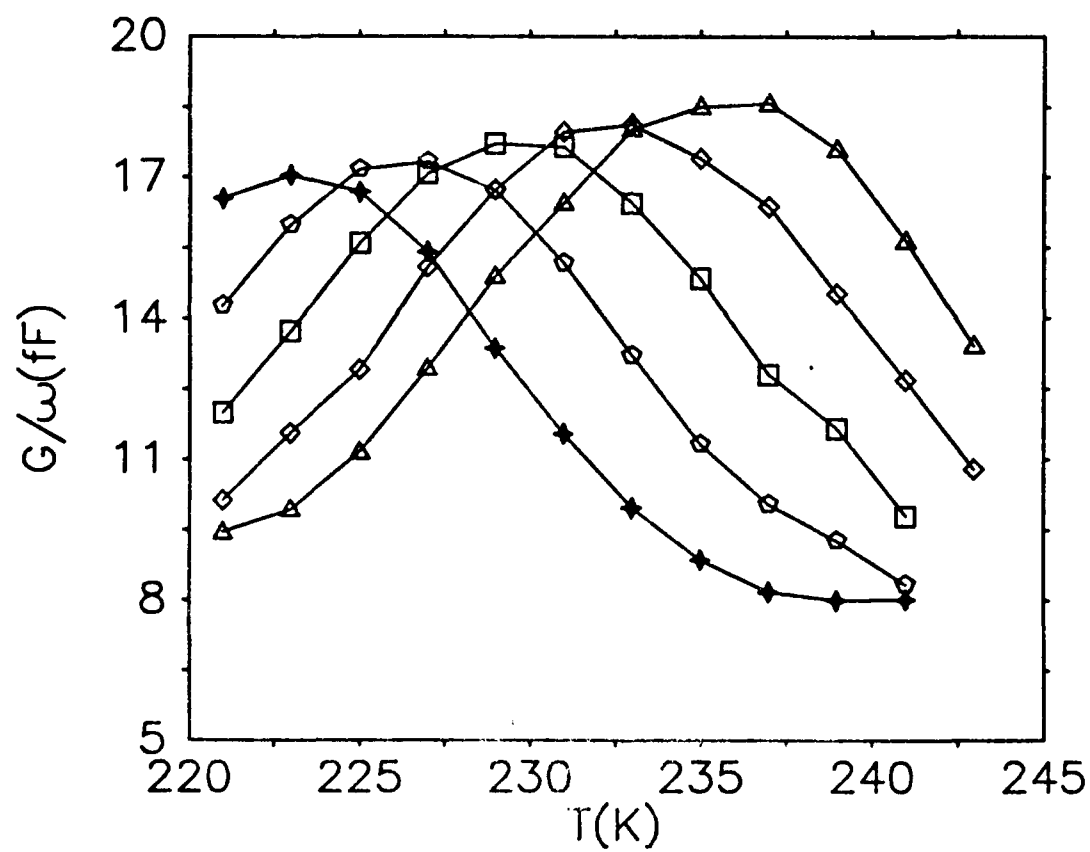
$$\frac{dT_{\alpha}}{dP} = \frac{\Delta V^*}{\Delta E^*} (T_{\alpha}^0 - T_r) \quad (4)$$

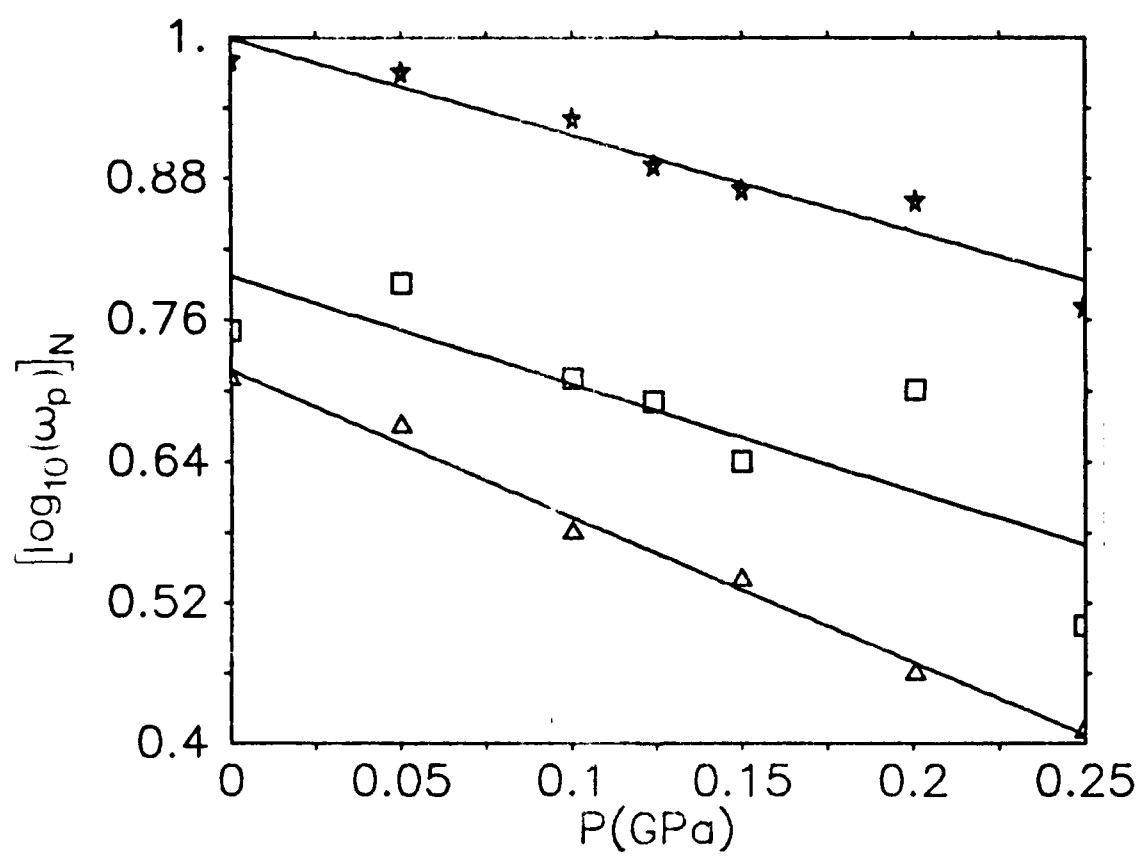
where T_{α}^0 represents the zero pressure peak position, ΔE^* and ΔV^* are the activation enthalpy and volume, respectively, and T_r is the minimum temperature at which the activated process is still possible. It is clear from the results that dT_{α}/dP and T_{α}^0 are larger for larger frequencies. This proportional dependence on frequency for both quantities is consistent with Eq. (4). Further, the values of T_{α}^0 and dT_{α}/dP for PEO would fit well on the plot in Fig. 5 of Quested and Oskooie-Tabrizi¹⁸ and thus PEO can be grouped with poly vinyl acetate (PVAc), acrylonitrile-butadiene-styrene (ABS), polyurethane elastomers (PU) and polyvinylidene fluoride (PVDF), in that it follows Eq. (4).

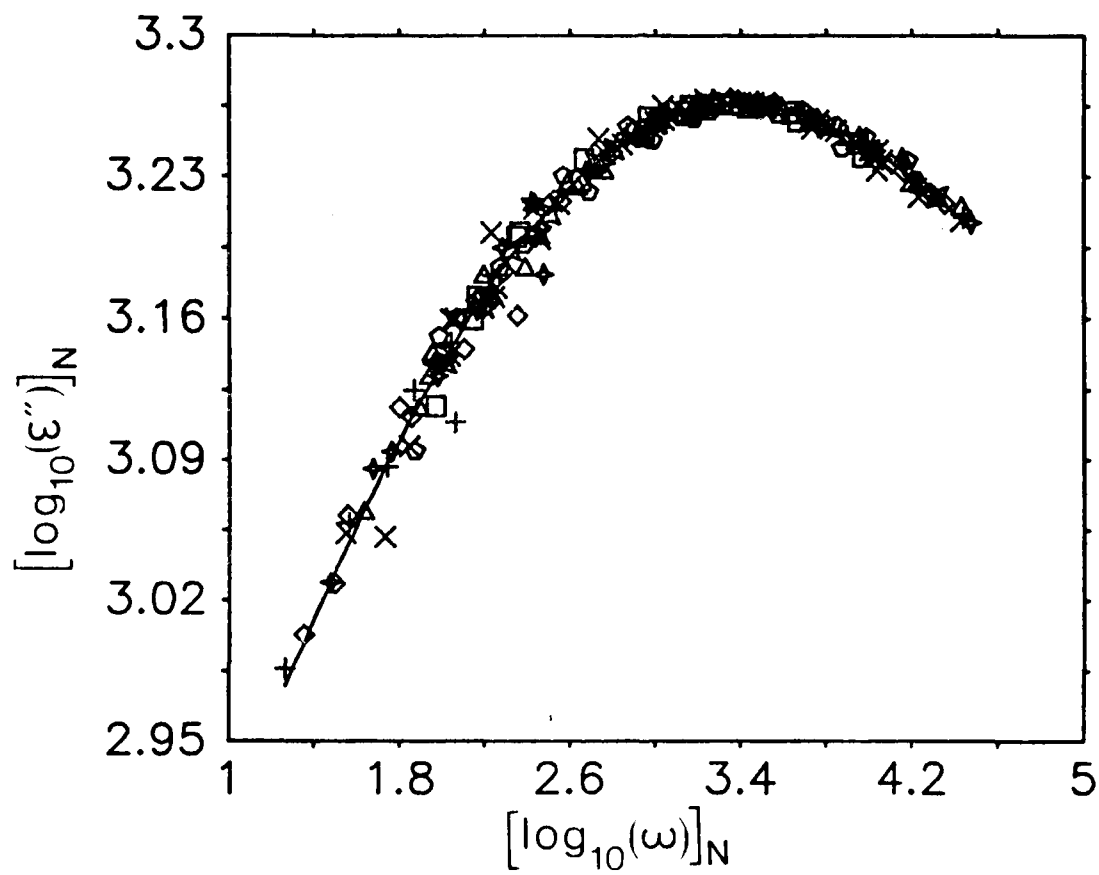
7. SUMMARY

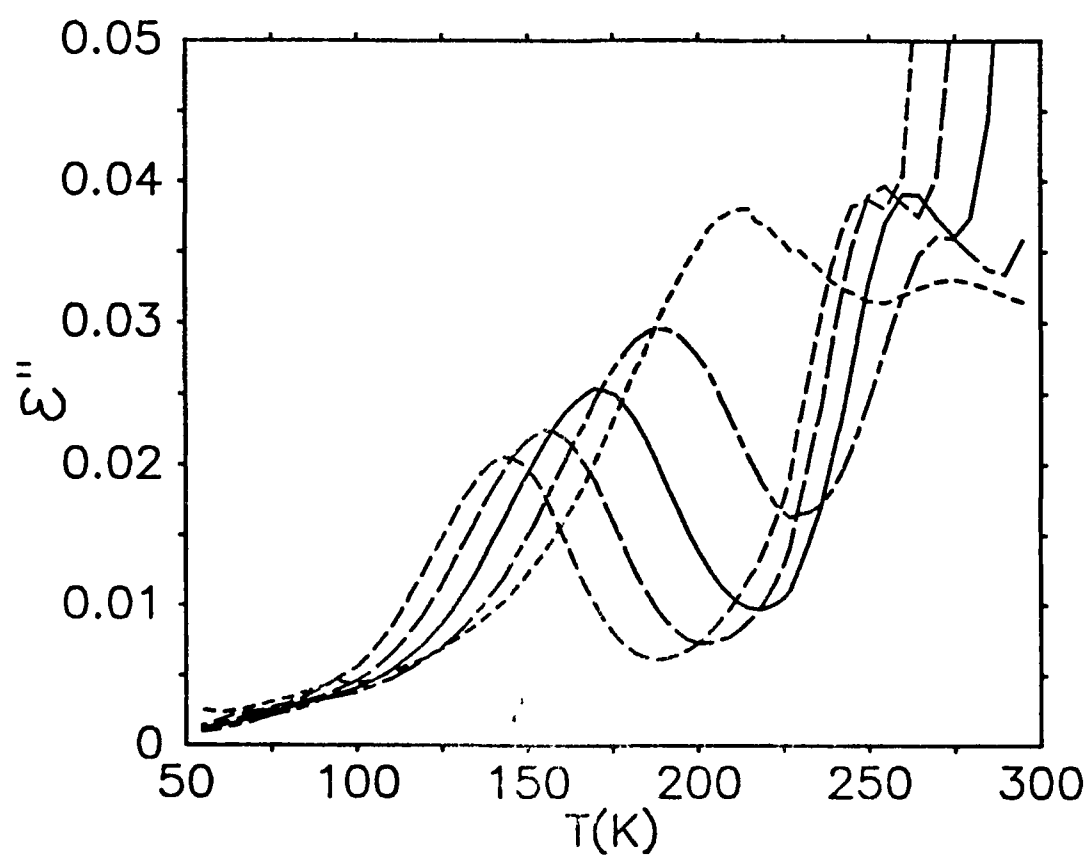
In summary, then, vacuum dielectric frequency complex admittance measurements have been carried out on PEO at low temperatures for a variety of molecular weights. The β relaxation varies very little over the range of molecular weights studied. This represents evidence against assigning the β relaxation to motion of end groups. Next, previously reported data for the γ relaxation are reanalyzed using a reduced plot. The activation volume obtained is in good agreement with the previous work and the fact that a very small value is found implies that the relaxation is associated with small segments of the polymer chain. It is found that the relaxation associated with the glass transition is only observed in temperature for

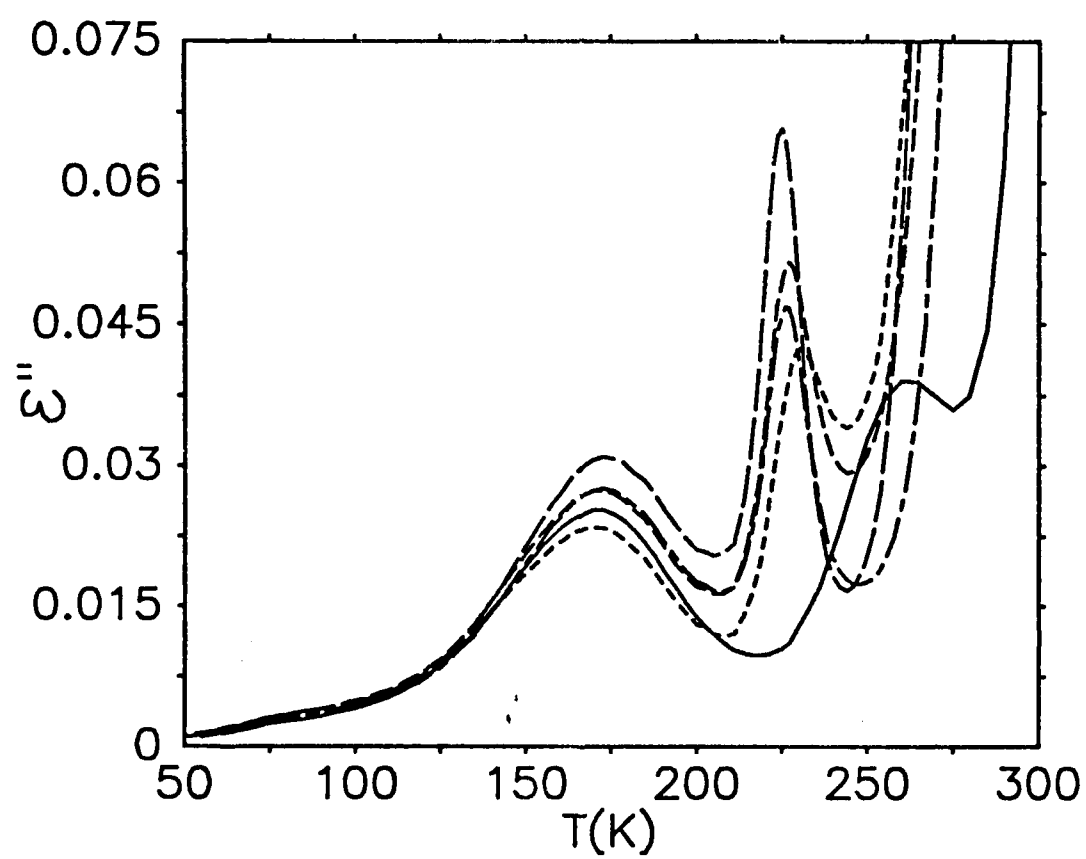


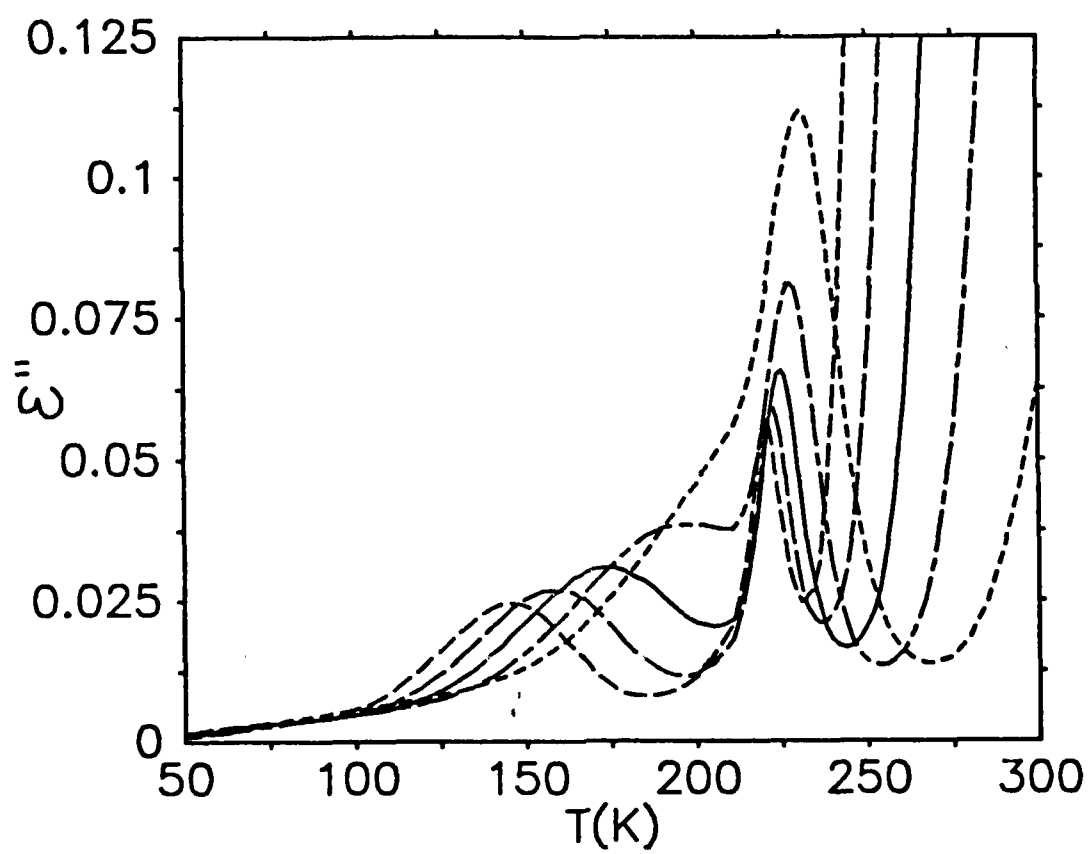












F₀'

Figure 6. G/ω (pF) vs. $T(K)$ at several pressures at 1000 Hz in the region of the glass transition for PEO (MW 5×10^6). The curves from left to right are 0.0001 (1 atm), 0.04, 0.08, 0.12, and 0.16 GPa. For clarity, straight line segments connect the datum points and data at other pressures have been omitted.

Figure 7. Peak position vs. $P(\text{GPa})$ at two frequencies, 1000 Hz (crosses) and 10,000 Hz (squares), for the α_a relaxation in PEO (MW 5×10^6). Also shown are the best fit quadratic curves.

FIGURE CAPTIONS

Figure 1. ϵ'' vs. $T(K)$ for PEO with molecular weight 4×10^6 . The curves (from left to right) are: medium dash- 10 Hz; long dash- 10^2 Hz; solid- 10^3 Hz; chain link- 10^4 Hz; short dash- 10^5 Hz. Straight line segments connect the datum points which are not shown.

Figure 2. ϵ'' vs. $T(K)$ for PEO at 1000 Hz with various molecular weights: long dash- 4×10^6 ; medium dash- 6×10^5 ; chain link- 3×10^5 ; short dash- 1×10^5 ; solid- 1.85×10^4 . Straight line segments connect the datum points which are not shown.

Figure 3. ϵ'' vs. $T(K)$ for PEO with molecular weight 1.85×10^4 . The curves (from left to right) are medium dash- 10 Hz; long dash- 10^2 Hz; solid- 10^3 Hz; chain link- 10^4 Hz; short dash- 10^5 Hz. Straight line segments connect the datum points which are not shown.

Figure 4. "Reduced plot" for the γ relaxation in PEO (MW 5×10^6). The best fit Jonscher curve and data points for 17 frequencies, 3 temperatures, and 7 pressures are shown. The peak position and strength are arbitrary.

Figure 5. Peak shift vs. pressure at three temperatures, 185, 182, and 179K from top to bottom, for the results shown in Figure 4. Also shown are the best fit straight lines.

Table II

Best fit parameters in Eq. 3 for peak position vs. pressure for the α_a relaxation in pure PEO.

f(Hz)	T_{α}^0	B(GPa ⁻¹)	C(GPa ⁻²)	RMS Deviation
10^3	223.1	87.8	-75.3	0.11
$10^{3.5}$	224.3	87.9	-56.4	0.21
10^4	225.6	91.8	-86.7	0.15

Table I

Pressure derivative of the relaxation time and activation volume
for the gamma relaxation at various temperatures for pure PEO.

T(K)	$\frac{d \ln \omega}{dP}$ (GPa) ⁻¹	ΔV^* (cm ³ /mol)	ΔV^* (cm ³ /mol) ^a
<u>Sample #2</u> (Cast film)			
179	2.9	4.3	4.3
182	2.1	3.2	-
185	1.9	2.9	4.7

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ACKNOWLEDGMENTS

This work was supported in part by the Office of Naval Research.

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