

ARMY RESEARCH LABORATORY

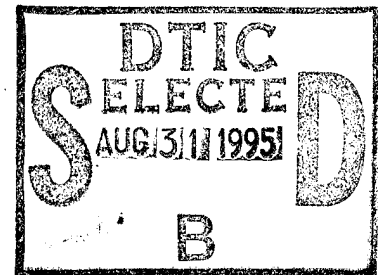


On Finite Strain Plasticity Constitutive Modeling in Computational Mechanics

Norris J. Huffington, Jr.

ARL-TR-789

July 1995



19950830 084

APPROVED FOR PUBLIC RELEASE; DISTRIBUTION IS UNLIMITED.

DTIC QUALITY INSPECTED 5

NOTICES

Destroy this report when it is no longer needed. DO NOT return it to the originator.

Additional copies of this report may be obtained from the National Technical Information Service, U.S. Department of Commerce, 5285 Port Royal Road, Springfield, VA 22161.

The findings of this report are not to be construed as an official Department of the Army position, unless so designated by other authorized documents.

The use of trade names or manufacturers' names in this report does not constitute endorsement of any commercial product.

REPORT DOCUMENTATION PAGE			Form Approved OMB No. 0704-0188	
Public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching existing data sources, gathering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggestions for reducing this burden, to Washington Headquarters Services, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suite 1204, Arlington, VA 22202-4302, and to the Office of Management and Budget, Paperwork Reduction Project(0704-0188), Washington, DC 20503.				
1. AGENCY USE ONLY (Leave blank)		2. REPORT DATE July 1995	3. REPORT TYPE AND DATES COVERED Final	
4. TITLE AND SUBTITLE Finite Strain Plasticity Constitutive Modeling in Computational Modeling			5. FUNDING NUMBERS PR: 1L162618AH80	
6. AUTHOR(S) Norris J. Huffington, Jr.				
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) U.S. Army Research Laboratory ATTN: AMSRL-WT-TD Aberdeen Proving Ground, MD 21005-5066			8. PERFORMING ORGANIZATION REPORT NUMBER ARL-TR-789	
9. SPONSORING/MONITORING AGENCY NAMES(S) AND ADDRESS(ES)			10. SPONSORING/MONITORING AGENCY REPORT NUMBER	
11. SUPPLEMENTARY NOTES				
12a. DISTRIBUTION/AVAILABILITY STATEMENT Approved for public release; distribution is unlimited.			12b. DISTRIBUTION CODE	
13. ABSTRACT (Maximum 200 words) The utility of employing discrete point input constitutive data (effective stress vs. effective plastic strain) for finite elastoplastic response problems is examined. A difficulty that occurs when the response effective plastic strain passes points of slope discontinuity in the input data is noted, and an algorithm that circumvents this difficulty is introduced. This algorithm has been successfully incorporated in the DYNA3D hydrocode and the NIKE3D structural mechanics code.				
14. SUBJECT TERMS finite strain plasticity, hydrocode, constitutive modeling			15. NUMBER OF PAGES 23	
			16. PRICE CODE	
17. SECURITY CLASSIFICATION OF REPORT UNCLASSIFIED	18. SECURITY CLASSIFICATION OF THIS PAGE UNCLASSIFIED	19. SECURITY CLASSIFICATION OF ABSTRACT UNCLASSIFIED	20. LIMITATION OF ABSTRACT UL	

INTENTIONALLY LEFT BLANK.

TABLE OF CONTENTS

		<u>Page</u>
	LIST OF FIGURES	v
1.	INTRODUCTION	1
2.	DISCRETE PAIRS CONSTITUTIVE INPUT	1
3.	THE "STRADDLE" ALGORITHM	4
4.	IMPLEMENTATION OF THE ALGORITHM IN COMPUTER CODES	6
5.	CONCLUDING REMARKS	7
6.	REFERENCES	9
	APPENDIX: STRADDLE ALGORITHM IN DYNA3D MATERIAL MODEL 10 ..	11
	NOMENCLATURE	19
	DISTRIBUTION LIST	21

Accession For	
RTIS GRA&I	<input checked="" type="checkbox"/>
DTIC TAB	<input type="checkbox"/>
Unannounced	<input type="checkbox"/>
Justification	
By	
Distribution	
Availability Codes	
Dist	Avail and/or Special
A-1	

INTENTIONALLY LEFT BLANK.

LIST OF FIGURES

<u>Figure</u>		<u>Page</u>
1.	Effective stress vs. effective plastic strain curves	2
2.	"Before and After" stress-time solution	3
3.	Geometry of the straddle analysis	5

INTENTIONALLY LEFT BLANK.

1. INTRODUCTION

In classical elastoplasticity (Hill 1950; Malvern 1969), it is customary to assume the existence of a plastic potential function in stress space, which is usually taken to be the yield function. Consistent with this assumption is the associated flow rule, which specifies that each component of plastic natural strain increment is proportional to the partial derivative of the potential function with respect to the corresponding stress component. For materials that work harden, the effect of inelastic straining on the "size" of the yield function must be considered. One method of accomplishing this is to introduce the concept of a universal plastic stress-strain curve that relates two scalar quantities: an "effective" stress and an accumulation of "effective" plastic strain increments (defined in sequel). This concept gives good agreement with physical experiments for proportional loading, but less satisfactory predictions for combined stresses deviating significantly from proportional loading. Nevertheless, this concept for modeling work hardening is widely used, both in analysis and in computational methodology.

An example of a universal stress-strain curve is the popular Johnson-Cook (1983) strength model, particularized to the rate-independent, isothermal case.* While the parameters for this model were derived by fitting experimental data, it will be shown that such a fit can only be satisfactory for a limited range of finite strains. An alternate approach is to provide the universal stress-strain curve as a discrete set of data points over as wide a range of effective plastic strains as desired and to use an interpolation procedure for intermediate values. This report will elaborate on the latter alternative and certain refinements.

2. DISCRETE PAIRS CONSTITUTIVE INPUT

Computer programs developed at Lawrence Livermore National Laboratory have provided options to input constitutive data as coordinate pairs of points on an effective stress vs. effective plastic strain curve. Specifically, these options apply to Material Types 10 and 24 of DYNA3D (Hallquist 1988) and Material Types 3 and 24 of NIKE3D (Hallquist 1984). An illustration of this form of input is shown in Figure 1, where experimentally determined values (Weerasooriya and Swanson 1991) for annealed OFHC copper are displayed as asterisks. The computer programs then calculate values of effective stress for intermediate values of the accumulated effective plastic strain by linear interpolation.

* The author believes that any rate-/temperature-dependant constitutive model should provide agreement with quasi-static experimental data when so particularized.

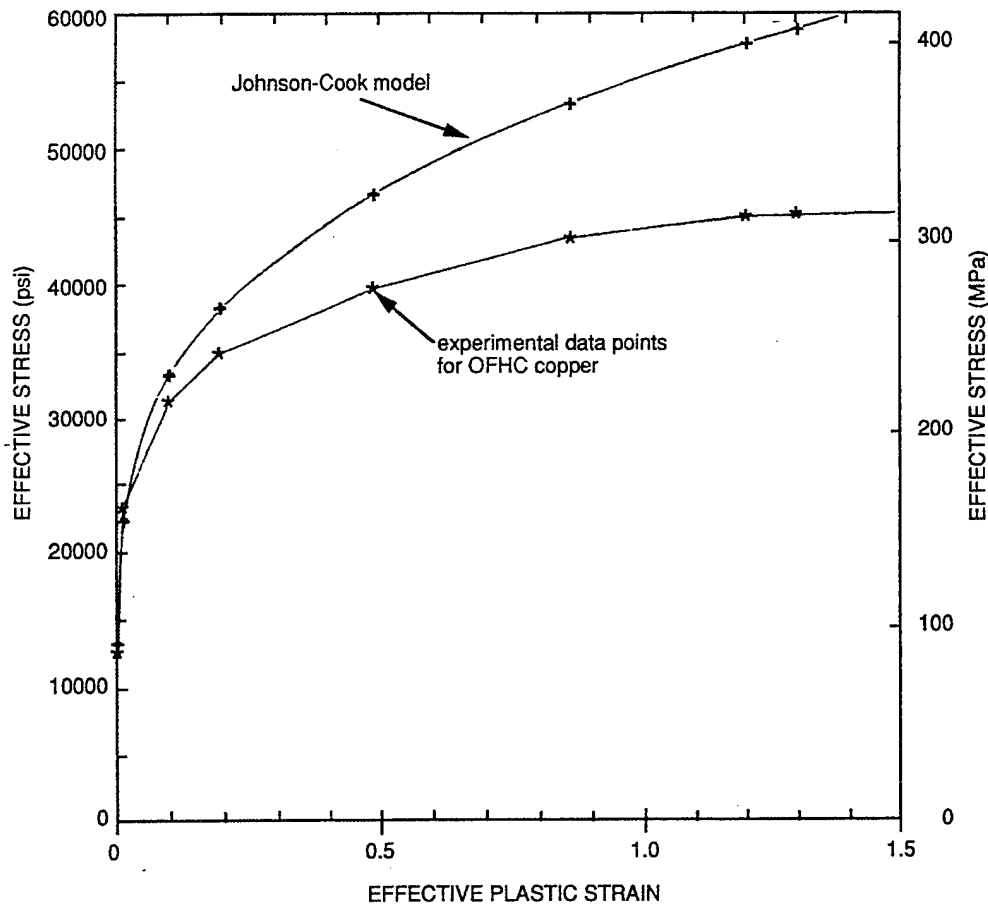


Figure 1. Effective stress vs. effective plastic strain curves.

Figure 1 also shows a corresponding curve for the Johnson-Cook model, based on their published parameters for OFHC copper. The differences between the two curves for small to moderate effective plastic strains are not regarded as significant, since there may have been differences in heat treatments between the two sets of experiments. However, the Johnson-Cook model, which uses a single power law term, cannot represent the stress saturation that is observed physically. The same limitation also applies to the Zerilli-Armstrong (1987) model.

The discrete point representation of constitutive data can also give rise to undesired effects. When faced with a limited number of input points (16 for DYNA3D, 8 for NIKE3D), one would pick points from available experimental data, not necessarily at uniform increments of effective plastic strain, which appear to include the essential features of these data. If the experimental data are known to represent a smoothly continuous phenomenon, the first (and possibly the second) differences of the selected input points should be checked for proper smoothness.

Another difficulty can arise, at least with the computational procedure for stress incrementation employed by the DYNA/NIKE codes. This is illustrated in Figure 2, which was obtained from an NIKE3D computation for the linearly increasing twist of a tube, using the copper data from Figure 1. The large "spike" that is observed at 0.1 s occurred when the computed effective plastic strain exceeded one of the input values (where there is a slope discontinuity). The problem is that the constitutive subroutine uses the previous slope of the yield function to evaluate the new yield stress and is not aware of the slope change until the next time step. In the case of NIKE3D, large time increments can be employed, which accentuates the obvious discrepancy.

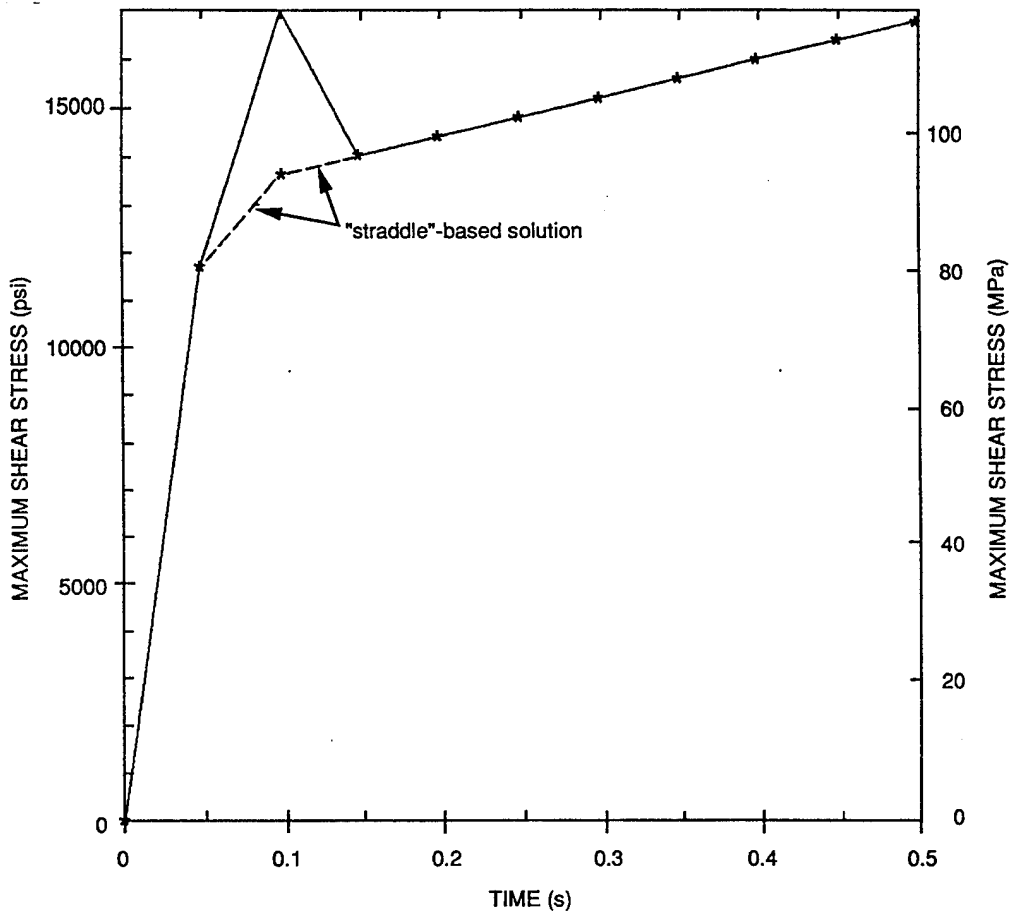


Figure 2. "Before and After" stress-time solution.

For the DYNA3D code, where the time increment is constrained to be small for numerical stability, the "spike" also appears (perhaps with a smaller amplitude), but is followed by an oscillatory response that does not rapidly die out. This behavior has been shown to be associated with the occurrence of spurious negative plastic work.

The foregoing phenomenon can be avoided by use of a special algorithm developed by the author that is invoked whenever the computed effective plastic strain reaches one of the input data values.

3. THE "STRADDLE" ALGORITHM

The discussion of this algorithm refers to the manner in which it was implemented in the DYNA3D code for the solid hexahedron element; however, for other applications, the procedure is essentially the same. The notation employed is defined in the Nomenclature section of this report.

Subroutine F3DM10 takes the stress components from the previous cycle and a new set of strain increments and calculates a set of trial deviatoric stress components on the basis of an elastic stress change. It recalls the previous effective plastic strain ϵ_j^P and calls the interpolation subroutine YIELDS to reconstruct the current yield stress σ_j^Y (see Figure 3). The trial stresses are used to calculate the von Mises effective (trial) stress σ^T . If $\sigma^T \leq \sigma_j^Y$, the stress change was elastic and the trial deviatoric stresses are accepted as valid. Otherwise, the stress change is plastic and the program calculates an incremental effective plastic strain (scalar) using

$$(\Delta\epsilon^P)^* = \frac{\sigma^T - \sigma_j^Y}{3G + H_k} \quad (1)$$

In the original version, of F3DM10 a new yield stress is determined using

$$(\sigma_{j+1}^Y)^* = \sigma_j^Y + H_k(\Delta\epsilon^P)^* \quad (2)$$

It is this procedure that results in a stress overshoot when ϵ_j^P and $\epsilon_j^P + \Delta\epsilon^P$ straddle a point such as $\bar{\epsilon}_{k+1}$ on the yield curve.

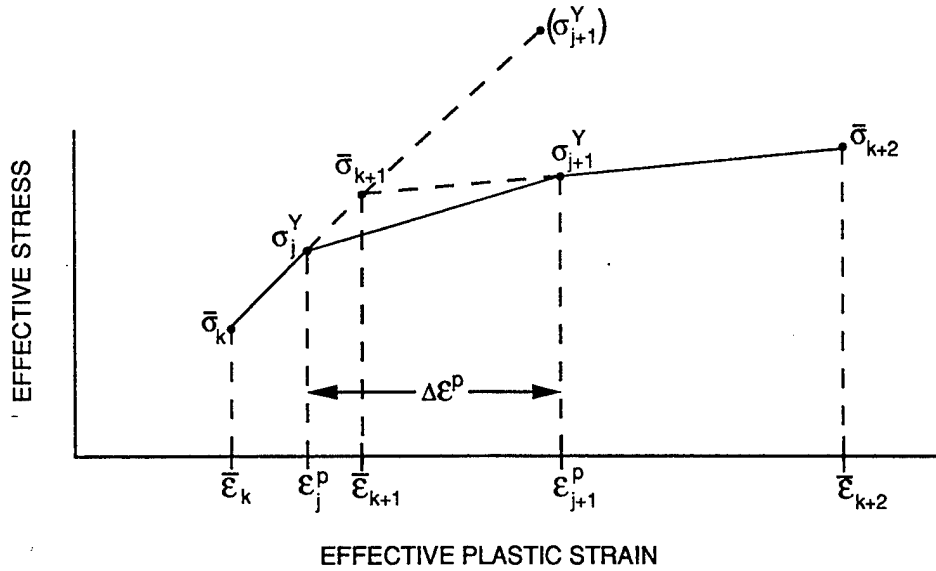


Figure 3. Geometry of the straddle analysis.

In the revised version of F3DM10, the new trial effective plastic strain

$$(\epsilon_{j+1}^p)^T = \epsilon_j^p + (\Delta \epsilon^p)^* \quad (3)$$

is calculated and compared to $\bar{\epsilon}_{k+1}$. If $(\epsilon_{j+1}^p)^T > \bar{\epsilon}_{k+1}$, the analysis branches to the straddle algorithm to deal with this special case. It is desired to determine a $\Delta \epsilon^p$ and an intermediate slope H_e that will result in a value of σ_{j+1}^Y which is consistent with the input data. The following conditions must be satisfied:

$$\epsilon_{j+1}^p = \epsilon_j^p + \Delta \epsilon^p \quad (4)$$

$$\Delta \epsilon^p = \frac{\sigma^T - \sigma_j^Y}{3G + H_e} \quad (5)$$

$$H_e = \frac{\sigma_{j+1}^Y - \sigma_j^Y}{\epsilon_{j+1}^p - \epsilon_j^p} \quad (6)$$

$$\sigma_{j+1}^Y = \sigma_{k+1}^Y + H_{k+1}(\epsilon_{j+1}^p - \bar{\epsilon}_{k+1}) \quad (7)$$

These simultaneous equations may be solved algebraically to yield

$$H_e = \frac{3GC_1 + H_{k+1}(\sigma^T - \sigma_j^Y)}{\sigma^T - \sigma_j^Y - C_1} \quad (8)$$

and

$$\Delta \epsilon^P = \frac{\sigma^T - \sigma_j^Y - C_1}{3G + H_{k+1}} \quad (9)$$

where

$$C_1 = \bar{\sigma}_{k+1} - \sigma_j^Y - H_{k+1}(\bar{\epsilon}_{k+1} - \epsilon_j^P) \quad (10)$$

Values of ϵ_{j+1}^P and σ_{j+1}^Y may then be obtained by use of equations (4) and (7). Certain tests to verify that ϵ_{j+1}^P falls within the range $\bar{\epsilon}_{k+1} < \epsilon_{j+1}^P < \bar{\epsilon}_{k+2}$ are required.

Following these calculations, the normal flow of this subroutine is rejoined and the stress components are scaled back (radial return method [Krieg and Key 1976]) so that the new total stresses will lie on the enlarged yield surface.

4. IMPLEMENTATION OF THE ALGORITHM IN COMPUTER CODES

Although the straddle algorithm was formulated for use with the solid element in DYNA3D, it is also directly applicable for the NIKE3D solid element, as well as for the elements in the 2-D versions of these codes, wherever the yield function is introduced as discrete points. The 3-D programs also feature beam and shell elements for which, for certain options, discrete yield data are accepted.

In DYNA3D, both the Hughes-Liu (1981) and Belytschko-Tsay (1981, 1984) shells have this feature and there is a choice between iterative and noniterative plane stress plasticity (Hallquist and Whirley 1989). In NIKE3D, the Hughes-Liu beam and shell formulations both provide for discrete yield data and both require iterative stress evaluation to satisfy the requirement that the stress component normal to the

lamina system be zero. This iteration process complicates the introduction of the straddle algorithm, but it has been successfully incorporated for the Hughes-Liu shell in subroutine s3mns. In fact, the dashed line solution in Figure 2 resulted from a NIKE3D calculation after this subroutine was modified.

To date, the straddle algorithm has only been incorporated in those portions of the DYNA and NIKE codes where a need has arisen. As an example, the inclusion of this algorithm in DYNA3D Material Model 10 is listed in the appendix.

5. CONCLUDING REMARKS

The straddle algorithm was developed to cope with nonphysical "spikes" and oscillations observed in solutions obtained using the Lagrangian hydrocodes NIKE3D and DYNA3D, in conjunction with discrete yield function input data. In many applications, these perturbations may be only a minor nuisance in the graphical output. However, when tracking material response to transient loading in conjunction with a material failure criterion, the decision as to whether survivability or failure will result may be altered by the presence of spurious spikes.

Consequently, it is argued that incorporation of the algorithm is an inexpensive means of avoiding unpleasant surprises (since the number of branches to the algorithm at each integration point cannot exceed the number of input data points minus one regardless of the number of cycles computed).

INTENTIONALLY LEFT BLANK.

6. REFERENCES

- Belytschko, T. B., and C. S. Tsay. "Explicit Algorithms for Nonlinear Dynamics of Shells." AMD-vol. 48, American Society for Mechanical Engineers (ASME), pp. 209-231, 1981.
- Belytschko, T. B., and C. S. Tsay. "Explicit Algorithms for Nonlinear Dynamics of Shells." Comp. Meth. Appl. Mech. Eng., vol. 43, pp. 251-276, 1984.
- Hallquist, J. O. "NIKE3D: An Implicit, Finite-Deformation, Finite Element Code for Analyzing the Static and Dynamic Response of Three-Dimensional Solids." University of California, Lawrence Livermore National Laboratory, Rept. UCID-18822, rev. 1, CA, 1984.
- Hallquist, J. O. "DYNA3D User's Manual (Nonlinear Dynamic Analysis of Structures in Three Dimensions)." University of California, Lawrence Livermore National Laboratory, Rept. UCID-19592, rev. 4, 1988.
- Hallquist, J. O., and R. G. Whirley. "DYNA3D User's Manual." Rept. UCID-19592, rev. 5, 1989.
- Hill, R. The Mathematical Theory of Plasticity. Oxford, 1950.
- Hughes, T. J. R., and W. K. Liu. "Nonlinear Finite Element Analysis of Shells: Parts I and II." Comp. Meths. Appl. Mech., vol. 27, pp. 167-181, 331-362, 1981.
- Johnson, G. R., and W. H. Cook. "A Constitutive Model and Data for Metals Subjected to Large Strains, High Strain Rates, and High Temperatures." Proceedings of the 7th International Symposium on Ballistics, The Hague, pp. 541-547, 1983.
- Krieg, R. D., and S. W. Key. "Implementation of a Time Independent Plasticity Theory Into Structural Computer Programs." AMD-vol. 20, American Society for Mechanical Engineers (ASME), pp. 125-137, 1976.
- Malvern, L. E. Introduction to the Mechanics of a Continuous Medium. Prentice-Hall, 1969.
- Weerasooriya, T., and R. A. Swanson. "Experimental Evaluation of the Taylor-Type Polycrystal Model for the Finite Deformation of an FCC Metal (OFHC Copper)." TR 91-20, U.S. Army Materials Technology Laboratory, Watertown, MA, 1991.
- Zerilli, F. J., and R. W. Armstrong. "Dislocation-Mechanics-Based Constitutive Relations for Material Dynamics Calculations." Journal of Applied Physics, vol. 61, no. 5, pp. 1816-1825, 1987.

INTENTIONALLY LEFT BLANK.

APPENDIX:

STRADDLE ALGORITHM IN DYNA3D MATERIAL MODEL 10

INTENTIONALLY LEFT BLANK.

The changes made to DYNA3D to incorporate the straddle algorithm in the following two subroutines are indicated in bold lower case letters.

```
SUBROUTINE F3DM10 (CM)
COMMON/BK02/IBURN,DT1,DT2,ISDO
COMMON/AUX2/D1 (128),D2 (128),D3 (128),D4 (128),D5 (128),D6 (128),
1 WZZDT (128),WYYDT (128),WXXDT (128),EINC (128)
COMMON/AUX11/PO (128)
COMMON/AUX14/
&SIGN1 (128),SIGN2 (128),SIGN3 (128),SIGN4 (128),
&SIGN5 (128),SIGN6 (128),
&EPX1 (128),EPX2 (128),EPX3 (128),EPX4 (128),EPX5 (128),AUX (128,5)
COMMON/AUX18/DD (128),DF (128)
COMMON/AUX19/SP,BFAC (128),DR1V (128),DR2V (128),W1
COMMON/AUX20/
& AJ2 (128),SJ2 (128),SCALE (128),FJL (128),CC (128),P (128),
& POLD (128),DAVG (128),FJK (128),SPECEN (128),AK (128),AKT (128),
& YWH (128)
COMMON/AUX33/IX1 (128),IX2 (128),IX3 (128),IX4 (128),IX5 (128),
1 IX6 (128),IX7 (128),IX8 (128),MXT (128),NMEL
COMMON/AUX35/RHOA (128),CXXA (128)
COMMON/AUX36/LFT,LLT
COMMON/EOSD/PC (128),SHRM (128)
logical plastic (128)
common/state/plastic
DIMENSION CM (1),il (128)
DATA THIRD/.3333333333333333/
MX=48*(MXT(LFT)-1)
SP=0.0
G=CM (MX+1)
QH=CM (MX+3)
QS=CM (MX+2)
A1=CM (MX+5)
A2=CM (MX+6)
ISPALL=CM (MX+7)-1.
NPT=CM (MX+16)
G2=2.*DT1*G
DO 10 I=LFT,LLT
```

```

10 DAVG(I)=-THIRD*(D1(I)+D2(I)+D3(I))
   DO 20 I=LFT,LLT
   SHRM(I)=G
   AKT(I)=QH
   PC(I)=EPX5(I)
20 AK(I)=QS+QH*EPX1(I)+(A1+A2*PO(I))*MAX(0.,PO(I))
   DO 30 I=LFT,LLT
   SIGN1(I)=SIGN1(I)+PO(I)+G2*(D1(I)+DAVG(I))
   SIGN2(I)=SIGN2(I)+PO(I)+G2*(D2(I)+DAVG(I))
   SIGN3(I)=SIGN3(I)+PO(I)+G2*(D3(I)+DAVG(I))
   SIGN4(I)=SIGN4(I)+.50*G2*D4(I)
   SIGN5(I)=SIGN5(I)+.50*G2*D5(I)
30 SIGN6(I)=SIGN6(I)+.50*G2*D6(I)
   IF (ISPALL.NE.2) GO TO 80
   DO 40 I=LFT,LLT
40 DAVG(I)=PO(I)-EPX5(I)
   DO 50 I=LFT,LLT
50 SCALE(I)=.50*(1.+SIGN(1.,DAVG(I)))
   DO 60 I=LFT,LLT
   PC(I)=SCALE(I)*EPX5(I)
   SIGN1(I)=SCALE(I)*SIGN1(I)
   SIGN2(I)=SCALE(I)*SIGN2(I)
   SIGN3(I)=SCALE(I)*SIGN3(I)
   SIGN4(I)=SCALE(I)*SIGN4(I)
   SIGN5(I)=SCALE(I)*SIGN5(I)
60 SIGN6(I)=SCALE(I)*SIGN6(I)
   DO 70 I=LFT,LLT
70 EPX5(I)=PC(I)
80 IF (NPT.EQ.0) GO TO 100
   CALL YIELDS(NPT,CM(MX+17),CM(MX+33),EPX1,AK,AKT,LFT,LLT,il)
100 DO 110 I=LFT,LLT
   AJ2(I)=.5*(SIGN1(I)**2+SIGN2(I)**2+SIGN3(I)**2)+SIGN4(I)**2+SIGN5
1  (I)**2+SIGN6(I)**2
   SJ2(I)=SQRT(3.*AJ2(I))
110 CONTINUE
   do 115 i = lft,llt
   plastic(i) = .false.
   if(ak(i).le.sj2(i)) plastic(i) = .true.
115 continue

```

C.... CALCULATE INCREMENT IN EFFECTIVE PLASTIC STRAIN & NEW YIELD STRESS

THRG=THIRD/G

DO 120 I=LFT,LLT

if(plastic(i)) then

CC(I)=DIM(SJ2(I),AK(I))/(1.E-30+DIM(1.,-THRG*AKT(I)))

CC(I)=THRG*AMIN1(SJ2(I),CC(I))

epxt=epxl(i)+cc(i)

m=mx+il(i)+16

if(epxt.gt.cm(32)) go to 120

if(epxt.gt.cm(m) .and. epxt.le.cm(m+1)) then

hb=(cm(m+17)-cm(m+16))/(cm(m+1)-cm(m))

c1=cm(m+16)-ak(i)-hb*(cm(m)-epxl(i))

akt(i)=(3.*g*c1+hb*(sj2(i)-ak(i)))/(sj2(i)-ak(i)-c1)

cc(i)=(sj2(i)-ak(i)-c1)/(3.*g+hb)

ak(i)=max(cm(m+16)+hb*(epxl(i)+cc(i)-cm(m)),0.)

elseif(epxt.gt.cm(m+1)) then

write(13,1) i

call adios(2)

else

AK(I)=MAX(AK(I)+AKT(I)*CC(I),0.)

endif

else

cc(i)= 0.

endif

120 CONTINUE

C.... CALCULATE FACTOR TO SCALE STRESSES BACK TO YIELD SURFACE

SEPS=1.E-30*G

DO 130 I=LFT,LLT

SCALE(I)=(AK(I)+SEPS)/(MAX(AK(I),SJ2(I))+SEPS)

130 CONTINUE

C.... SCALE BACK STRESS AND INCREMENT PLASTIC STRAIN

DO 140 I=LFT,LLT

SIGN1(I)=SCALE(I)*SIGN1(I)

SIGN2(I)=SCALE(I)*SIGN2(I)

SIGN3(I)=SCALE(I)*SIGN3(I)

SIGN4(I)=SCALE(I)*SIGN4(I)

SIGN5(I)=SCALE(I)*SIGN5(I)

SIGN6(I)=SCALE(I)*SIGN6(I)

```

      EPX1(I)=EPX1(I)+CC(I)
140 CONTINUE

      IF(ISPALL.NE.1) RETURN
      DO 150 I=LFT,LLT
      AJ2(I)=.5*(SIGN1(I)**2+SIGN2(I)**2+SIGN3(I)**2)+SIGN4(I)**2+SIGN5
1      (I)**2+SIGN6(I)**2+1.E-12
150 SJ2(I)=SIGN1(I)*SIGN5(I)**2+SIGN2(I)*SIGN6(I)**2+SIGN3(I)*SIGN4(I)
1      **2-SIGN1(I)*SIGN2(I)*SIGN3(I)-2.*SIGN4(I)*SIGN5(I)*SIGN6(I)
      DO 160 I=LFT,LLT
      AKT(I)=-SQRT(27./AJ2(I))*SJ2(I)*0.5/AJ2(I)
      AKT(I)=SIGN(AMIN1(ABS(AKT(I)),1.),AKT(I))
160 YWH(I)=ACOS(AKT(I))*THIRD
      DO 170 I=LFT,LLT
170 SJ2(I)=2.*SQRT(AJ2(I)*THIRD)*COS(YWH(I))
      DO 180 I=LFT,LLT
180 DAVG(I)=PO(I)-SJ2(I)-EPX5(I)
      DO 190 I=LFT,LLT
190 SCALE(I)=.50*(1.+SIGN(1.,DAVG(I)))
      DO 200 I=LFT,LLT
      PC(I)  =SCALE(I)*EPX5(I)
      SIGN1(I)=SCALE(I)*SIGN1(I)
      SIGN2(I)=SCALE(I)*SIGN2(I)
      SIGN3(I)=SCALE(I)*SIGN3(I)
      SIGN4(I)=SCALE(I)*SIGN4(I)
      SIGN5(I)=SCALE(I)*SIGN5(I)
200 SIGN6(I)=SCALE(I)*SIGN6(I)
      DO 210 I=LFT,LLT
210 EPX5(I)=PC(I)
      RETURN
1  format(' EXCESSIVE PLASTIC STRAIN IN ELEMENT',i10,/)
      END

```

```

SUBROUTINE YIELDS(NPT, EPS, SIGE, EPX, AK, QH, LFT, LLT, il)

```

```

C      Rewritten by Glenn Randers-Pehrson, BRL, 6 April 1992, to
C      interpolate correctly within the beginning and end segments,
C      to extrapolate correctly, and to run faster.
C

```

```

INTEGER  NPT, LFT, LLT
REAL     EPS (16), SIGE (16), EPX (LLT), AK (LLT), QH (LLT)

C

INTEGER  I, L, IL (128)
REAL     SLOPE (16)

DO 10 I=LFT, LLT
  IL (I) = 0
  IF (EPX (I) .LT. EPS (2)) IL (I) = 2
  IF (EPX (I) .GT. EPS (NPT-1)) IL (I) = NPT
10 CONTINUE

DO 30 L=2, NPT
  DO 20 I=LFT, LLT
    IF (IL (I) .EQ. 0 .AND. (EPX (I) .GE. EPS (L-1) .AND. EPX (I) .LE. EPS (L)))
+     IL (I) = L
20 CONTINUE
    SLOPE (L) = (SIGE (L) - SIGE (L-1)) / (EPS (L) - EPS (L-1))
30 CONTINUE

DO 40 I=LFT, LLT
  QH (I) = SLOPE (IL (I))
  AK (I) = SIGE (IL (I) - 1) + QH (I) * (EPX (I) - EPS (IL (I) - 1))
40 CONTINUE

RETURN
END

```

INTENTIONALLY LEFT BLANK.

NOMENCLATURE

- $C_1 = \bar{\sigma}_{k+1} - \sigma_j^Y - H_{k+1}(\bar{\epsilon}_{k+1} - \epsilon_j^p)$
 G shear modulus
 H_e slope determined by the straddle algorithm
 $H_k = (\bar{\sigma}_{k+1} - \bar{\sigma}_k)/(\bar{\epsilon}_{k+1} - \bar{\epsilon}_k)$; slope of yield curve for interval $\bar{\epsilon}_k$ to $\bar{\epsilon}_{k+1}$
 $H_{k+1} = (\bar{\sigma}_{k+2} - \bar{\sigma}_{k+1})/(\bar{\epsilon}_{k+2} - \bar{\epsilon}_{k+1})$
 N number of input $(\bar{\epsilon}_m, \bar{\sigma}_m)$ pairs
 $\Delta\epsilon^p$ incremental change in the effective plastic strain
 $\bar{\epsilon}_m$ input values of effective plastic strain, $m = 1, 2, \dots, N$
 $\epsilon_j^p = \sum \sqrt{\frac{2}{3} \Delta\epsilon_{ij}^p \Delta\epsilon_{ij}^p}$; current value of effective plastic strain
 σ_j^Y current value of yield stress
 $\bar{\sigma}_m$ input values of effective stress, $m = 1, 2, \dots, N$
 $\sigma^T = \sqrt{\frac{3}{2} \tau_{ij}^T \tau_{ij}^T}$; trial value of von Mises effective stress
 τ_{ij} deviatoric components of Cauchy stress tensor
 $()^T$ trial value
 $()^*$ quantity calculated using original program

INTENTIONALLY LEFT BLANK.

NO. OF
COPIES ORGANIZATION

2 ADMINISTRATOR
ATTN DTIC DDA
DEFENSE TECHNICAL INFO CTR
CAMERON STATION
ALEXANDRIA VA 22304-6145

1 DIRECTOR
ATTN AMSRL OP SD TA
US ARMY RESEARCH LAB
2800 POWDER MILL RD
ADELPHI MD 20783-1145

3 DIRECTOR
ATTN AMSRL OP SD TL
US ARMY RESEARCH LAB
2800 POWDER MILL RD
ADELPHI MD 20783-1145

1 DIRECTOR
ATTN AMSRL OP SD TP
US ARMY RESEARCH LAB
2800 POWDER MILL RD
ADELPHI MD 20783-1145

ABERDEEN PROVING GROUND

5 DIR USARL
ATTN AMSRL OP AP L (305)

<u>NO. OF COPIES</u>	<u>ORGANIZATION</u>
1	HQDA ATTN SARD TT F MILTON PENTAGON WASHINGTON DC 20310-0103
1	HQDA ATTN SARD TT J APPEL PENTAGON WASHINGTON DC 20310-0103
1	HQDA ATTN SARD TR R CHAIT PENTAGON WASHINGTON DC 20310-0103
2	DIRECTOR ATTN AMSRL VS W ELBER AMSRL VS S F BARTLETT USARL LANGLEY RSRCH CTR MAIL STOP 266 HAMPTON VA 23681-0001
1	DIRECTOR ATTN J RICHARDSON DARPA 3701 NORTH FAIRFAX DR ARLINGTON VA 22203-1714
1	COMMANDER DEFENSE NUCLEAR AGENCY 6801 TELEGRAPH RD ALEXANDRIA VA 22192
6	DIRECTOR ATTN I AHMAD K IYER J WU G ANDERSON J CHANDRA TECH LIB US ARMY RESEARCH OFFICE PO BOX 12211 4300 MIAMI BLVD RSRCH TRI PK NC 27709
5	COMMANDER ATTN W MCCORKLE M COLE DONALD LOVELACE MICHAEL SCHEXNAYDER W S HOWARD US ARMY MISSILE COMMAND REDSTONE ARSNL AL 35898-5250

<u>NO. OF COPIES</u>	<u>ORGANIZATION</u>
1	COMMANDER ATTN S G BISHOP US ARMY BELVOIR RD&E CTR FORT BELVOIR VA 22060-5166
6	COMMANDER ATTN T DAVIDSON V LINDNER J PEARSON E BAKER W EBIHARA TECH LIB US ARMY ARDEC PCTNY ARSNL NJ 07806-5000
1	DIRECTOR ATTN C W KITCHENS BENET WEAPONS LAB US ARMY ARDEC WATERVLIET NY 12189
1	COMMANDER ATTN C KOMINOS ASARDA THE PENTAGON WASHINGTON DC 20301
2	COMMANDER ATTN J THOMPSON K BISHNOI US ARMY TANK AUTOMOTIVE CMND WARREN MI 48397-5000
7	DIRECTOR ATTN L JOHNSON A RAJENDRAN S C CHOU J MCLAUGHLIN T WEERASOORIYA D DANDEKAR TECH LIB USARL WATERTOWN MA 02172-0001
3	COMMANDER ATTN DON THOMPSON CODE 3268 T J GILL TECH LIB NAVAL WEAPONS CTR CHINA LAKE CA 93555

<u>NO. OF</u> <u>COPIES</u>	<u>ORGANIZATION</u>	<u>NO. OF</u> <u>COPIES</u>	<u>ORGANIZATION</u>
2	COMMANDER ATTN WILLIAM MOCK TECH LIB NAVAL SURFACE WARFARE CTR DAHLGREN VA 22448-5000	9	DIRECTOR ATTN J M MCGLAUN P YARRINGTON E HERTEL M FORRESTAL D GRADY M KIPP S SILLING D GARDNER TECH LIB SANDIA NATIONAL LAB PO BOX 5800 ALBUQUERQUE NM 87185
5	COMMANDER ATTN H MAIR R GARRETT P WALTER F J ZERILLI TECH LIB NAVAL SURFACE WARFARE CTR 10901 NEW HAMPSHIRE AVE SILVER SPRING MD 20903-5000	20	DIRECTOR ATTN G E CORT T F ADAMS D MANDELL R KARPP J DIENES A ZUREK F ADDESSIO P FOLLANSBEE J N JOHNSON W KAWAHARA M W LEWIS D RABERN J REPA L SCHWALBE S SCHIFERL G T GRAY D TONKS P MAULDIN K HOLIAN TECH LIB LOS ALAMOS NATIONAL LAB PO BOX 1663 LOS ALAMOS NM 87545
1	COMMANDER ATTN S ZILLIACUS CODE 671 NSWC CARDEROCK DIV 9500 MACARTHUR BLVD BETHESDA MD 20084-5000		
2	DIRECTOR ATTN D BAMMANN J LIPKIN SANDIA NATIONAL LAB LIVERMORE CA 94550		
1	DIRECTOR ATTN T NICHOLAS AIR FORCE WRIGHT LAB MATERIALS LAB WRIGHT PATTERSON AFB OH 45433		
3	DIRECTOR ATTN MNSH W COOK MNMW J FOSTER TECH LIB WRIGHT LAB EGLIN AFB FL 32542		
1	COMMANDER ATTN R BECKER US ARMY BALLISTIC MISSILE DEFENSE SYSTEMS CMND PO BOX 1500 HUNTSVILLE AL 35807-3801	1	AEROJET PRECISION WEAPONS ATTN J CARLEONE DEPT 5131 T W 1100 HOLLYVALE AZUSA CA 91702
		1	BATTELLE ATTN B D TROTT 505 KING AVE COLUMBUS OH 43201

NO. OF
COPIES ORGANIZATION

12 DIRECTOR
ATTN R TIPTON
J GOUDREAU
R CHRISTENSEN
D BAUM
R COUCH
D LASSILA
M MURPHY
K SINZ
D STEINBERG
P RABOIN
W GOURDIN
TECH LIB
LAWRENCE LIVERMORE NATIONAL LAB
PO BOX 808
LIVERMORE CA 94550

3 DYNA EAST CORPORATION
ATTN P C CHOU
R CICARELLI
W FLIS
3201 ARCH ST
PHILADELPHIA PA 19104

3 SOUTHWEST RESEARCH INSTITUTE
ATTN C ANDERSON
A WENZEL
U LINDHOLM
PO DRAWER 28255
SAN ANTONIO TX 78228-0255

2 ALLIANT TECHSYSTEMS INC
ATTN G R JOHNSON
F STECHER
MN 48 2700
7225 NORTHLAND DR
BROOKLYN PARK MN 55428

1 S CUBED
ATTN R SEDGWICK
PO BOX 1620
LA JOLLA CA 92038-1620

2 ORLANDO TECHNOLOGY INC
ATTN D MATUSKA
J OSBORN
PO BOX 855
SHALIMAR FL 32579

NO. OF
COPIES ORGANIZATION

1 LIVERMORE SOFTWARE TECH CORP
ATTN JOHN O HALLQUIST
2876 WAVERLY WAY
LIVERMORE CA 94550

1 RENSSELAER POLYTECHNIC INST
ATTN PROF E KREMPLE
TROY NY 12181

2 BROWN UNIVERSITY
ATTN R CLIFTON
B FREUND
DIVISION OF ENGINEERING
PROVIDENCE RI 02912

1 CARNEGIE MELLON UNIVERSITY
ATTN M E GURTIN
DEPT OF MATHEMATICS
PITTSBURGH PA 15213

4 THE JOHNS HOPKINS UNIV
ATTN PROF W SHARPE
PROF K T RAMESH
PROF A DOUGLAS
PROF R E GREEN
34TH AND CHARLES ST
BALTIMORE MD 21218

2 UNIV OF CALIFORNIA AT SAN DIEGO
ATTN PROF S NEMAT-NASSER
PROF M MEYERS
MECH AND AEROSPACE ENGRG DEPT
LA JOLLA CA 92093

1 UNIV OF DELAWARE
ATTN PROF J VINSON
DEPT OF MECHANICAL & AEROSPACE ENGRG
NEWARK DE 19711

2 UNIV OF FLORIDA
ATTN PROF L MALVERN
PROF D DRUCKER
GAINESVILLE FL 32601

1 VA POLYTECH INST & STATE UNIV
ATTN PROF C W SMITH JR
DEPT OF ENGRG SCIENCE & MECHANICS
BLACKSBURG VA 24061

<u>NO. OF COPIES</u>	<u>ORGANIZATION</u>
1	UNIVERSITY OF MARYLAND ATTN R ARMSTRONG DEPT OF MECHANICAL ENGRG COLLEGE PARK MD 20742
1	UNIVERSITY OF MISSOURI ROLLA ATTN R BATRA DEPT OF ME AE AND EM ROLLA MO 65401-0249
1	NORTHWESTERN UNIVERSITY ATTN T BELYTSCHKO DEPT OF CIVIL ENGRG EVANSTON IL 60208
1	UNIVERSITY OF VIRGINIA ATTN C O HORGAN DEPT OF APPLIED MATHEMATICS CHARLOTTESVILLE VA 22903
1	GEORGIA INSTITUTE OF TECHNOLOGY ATTN D L MCDOWELL SCHOOL OF MECHANICAL ENGRG ATLANTA GA 30332
1	STANFORD UNIVERSITY ATTN T J R HUGHES DEPT OF MECHANICAL ENGRG STANFORD CA 94305-4040

<u>NO. OF COPIES</u>	<u>ORGANIZATION</u>
	<u>ABERDEEN PROVING GROUND</u>
29	DIR, USARL ATTN AMSRL-WT-T, T. W. WRIGHT AMSRL-WT-TA, G. E. HAUVER AMSRL-WT-TB, N. M. GNIAZDOWSKI F. H. GREGORY AMSRL-WT-TC, R. S. COATES W. S. DE ROSSET F. I. GRACE K. D. KIMSEY M. L. LAMPSON L. S. MAGNESS W. P. WALTERS AMSRL-WT-TD, A. M. DIETRICH K. FRANK A. D. GUPTA J. T. HARRISON N. J. HUFFINGTON J. H. KINEKE M. N. RAFTENBERG G. RANDERS-PEHRSON J. M. SANTIAGO S. E. SCHOENFELD S. B. SEGLETES P. B. SIMMERS AMSRL-WT-PD, B. P. BURNS W. H. DRYSDALE S. A. WILKERSON D. HOPKINS T. BOGETTI G. GAZONAS

INTENTIONALLY LEFT BLANK.

USER EVALUATION SHEET/CHANGE OF ADDRESS

This Laboratory undertakes a continuing effort to improve the quality of the reports it publishes. Your comments/answers to the items/questions below will aid us in our efforts.

1. ARL Report Number ARL-TR-789 Date of Report July 1995
2. Date Report Received _____
3. Does this report satisfy a need? (Comment on purpose, related project, or other area of interest for which the report will be used.) _____

4. Specifically, how is the report being used? (Information source, design data, procedure, source of ideas, etc.) _____

5. Has the information in this report led to any quantitative savings as far as man-hours or dollars saved, operating costs avoided, or efficiencies achieved, etc? If so, please elaborate. _____

6. General Comments. What do you think should be changed to improve future reports? (Indicate changes to organization, technical content, format, etc.) _____

CURRENT
ADDRESS

Organization

Name

Street or P.O. Box No.

City, State, Zip Code

7. If indicating a Change of Address or Address Correction, please provide the Current or Correct address above and the Old or Incorrect address below.

OLD
ADDRESS

Organization

Name

Street or P.O. Box No.

City, State, Zip Code

(Remove this sheet, fold as indicated, tape closed, and mail.)
(DO NOT STAPLE)

DEPARTMENT OF THE ARMY

OFFICIAL BUSINESS

BUSINESS REPLY MAIL
FIRST CLASS PERMIT NO 0001,APG,MD

POSTAGE WILL BE PAID BY ADDRESSEE

**DIRECTOR
U.S. ARMY RESEARCH LABORATORY
ATTN: AMSRL-WT-TD
ABERDEEN PROVING GROUND, MD 21005-5066**



**NO POSTAGE
NECESSARY
IF MAILED
IN THE
UNITED STATES**

