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High Critical Temperature Superconductor Substrate
and Buffer Layer Compounds, A_2MeSbO_6
(Where A=Ba and Sr; and Me=Sc, In and Ga)

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ARL-TR-930

May 1996

19960725 056

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1. AGENCY USE ONLY (Leave blank)		2. REPORT DATE May 1996	3. REPORT TYPE AND DATES COVERED Technical Report: Jan. 95 - Dec. 95	
4. TITLE AND SUBTITLE HIGH CRITICAL TEMPERATURE SUPERCONDUCTOR SUBSTRATE AND BUFFER LAYER COMPOUNDS, A ₂ MeSbO ₆ (Where A=Ba and Sr; and Me=Sc, In and Ga)			5. FUNDING NUMBERS PR: 611102 PE: H94	
6. AUTHOR(S) Arthur Tauber,*Steven C. Tidrow, Robert D. Finnegan and William D. Wilber				
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) US Army Research Laboratory (ARL) Physical Sciences Directorate ATTN: AMSRL-PS-PC Fort Monmouth, NJ 07703-5601			8. PERFORMING ORGANIZATION REPORT NUMBER ARL-TR-930	
9. SPONSORING/MONITORING AGENCY NAME(S) AND ADDRESS(ES)			10. SPONSORING/MONITORING AGENCY REPORT NUMBER	
11. SUPPLEMENTARY NOTES *Arthur Tauber is with Geo-Centers Inc., 615 Hope Road, Eatontown, NJ 07703				
12a. DISTRIBUTION/AVAILABILITY STATEMENT Approved for Public Release: Distribution is Unlimited			12b. DISTRIBUTION CODE	
13. ABSTRACT (Maximum 200 words) Compounds in the series A ₂ MeSbO ₆ , where A=Ba, Sr, and Me=Sc, In, and Ga, have been used as substrate buffer layers with YBa ₂ Cu ₃ O _{7-x} thin films. These materials were prepared by solid-state reaction of the oxides and carbonates. The compounds are ordered perovskites except for Ba ₂ InSbO ₆ . All compounds are cubic except Sr ₂ ScSbO ₆ and Sr ₂ GaSbO ₆ which are pseudo-cubic, tetragonal. Dielectric constant and loss tangent are reported for each bulk compound. Herein is described the successful deposition of thin films A ₂ MeSbO ₆ on (100) MgO and A ₂ MeSbO ₆ /(001) YBCO/(100) MgO by pulsed laser ablation.				
14. SUBJECT TERMS High Critical Temperature Superconductor, (HTSC), substrate, buffer, microwave equipment, Sr ₂ MeSbO ₆ , thin films, epitaxial growth; lattice parameter			15. NUMBER OF PAGES 13	
			16. PRICE CODE	
17. SECURITY CLASSIFICATION OF REPORT Unclassified	18. SECURITY CLASSIFICATION OF THIS PAGE Unclassified	19. SECURITY CLASSIFICATION OF ABSTRACT Unclassified	20. LIMITATION OF ABSTRACT UL	

CONTENTS

	<u>Page</u>
INTRODUCTION	1
EXPERIMENTAL	2
RESULTS	3
SUMMARY	6
REFERENCES	7

TABLES

TABLE 1. Conditions for Deposition of Multilayered Structures	2
TABLE 2. Properties of A_2MeSbO_6 Compounds	4
TABLE 3. Percent Misfit to YBCO	5
TABLE 4. Epitaxial Relationships Among A_2MeSbO_6 , YBCO and MgO	6

FIGURES

FIGURE 1. X-ray Spectra of A_2MeSbO_6 Bulk Target Samples	3
FIGURE 2. X-ray Spectra of A_2MeSbO_6 /YBCO/(100) MgO Bilayers	5

High Critical Temperature Superconductor Substrate and Buffer Layer Compounds,
 A_2MeSbO_6 (Where A=Ba, Sr and Me=Sc, In and Ga)

1. INTRODUCTION

Considerable progress has been achieved in the variety and quality of microwave devices that have been fabricated using superconductors. Because most of these devices are in the form of thin films, they require dielectric substrates and/or buffer layers with very exacting properties [1-4]. The most useful dielectrics are lanthanum-aluminate ($LaAlO_3$), magnesium-oxide (MgO), buffered yttrium-stabilized-zirconia (YSZ) and buffered sapphire, all of which have some drawbacks. We have been investigating new substrates in an attempt to overcome various deficiencies in these dielectrics. Compounds in the series A_2MeSbO_6 , where A=Ba, Sr and Me=Sc, In and Ga, have been prepared by solid state reaction between oxides and carbonates. When A=Ba and Me = rare earth, an analogous series results, as previously reported [5]. Yet another series where A=Sr and Me=rare-earth, has been reported [6]. All of the reported Ba compounds are cubic and all of the Sr compounds are tetragonal, except for Me=Lu or La which are cubic. Goldschmidt's tolerance factor [7], $t = (r_{a+} + r_0) / \sqrt{2} (r_{b+} + r_0)$, predicts that $0.8 < t < 0.9$ perovskite compounds should be cubic. For the barium rare-earth compounds which are cubic, $0.8 \leq t \leq 0.9$. For the Sr rare-earth compounds which are mostly non-cubic, $0.96 \leq t \leq 0.99$ [7]. In the present instance, A_2MeSbO_6 , the tolerance factor predicts all the compounds to be cubic.

We have previously reported the successful pulsed-laser deposition (PLD) on thin films of strontium-rare-earth-antimonate (Sr_2ReSbO_6) on yttrium-barium-copper-oxide ($YBa_2Cu_3O_{7-\delta}$ or YBCO), [$Sr_2ReSbO_6/(001)YBCO/(100)MgO$] [7]. Similar results have been achieved with barium-rare-earth-antimonate (Ba_2ReSbO_6) on YBCO, [$Ba_2ReSbO_6/(001)YBCO/(100)MgO$] [8]. Herein is described the successful deposition of thin film A_2MeSbO_6 on YBCO, $A_2MeSbO_6/(001)YBCO/(100)MgO$.

2. EXPERIMENTAL

The A_2MeSbO_6 compounds were prepared by mixing reagent grade strontium-carbonate ($SrCO_3$), barium-carbonate ($BaCO_3$), antimony-trioxide (Sb_2O_3), indium-trioxide (In_2O_3), gallium-trioxide (Ga_2O_3) and scandium-trioxide (Sc_2O_3) in stoichiometric proportions by grinding together in a mortar, pressing the mixtures into discs, heating the discs to $1100^\circ C$ (rate = $180^\circ C/h$) for 15 h in air, then cooling to $100^\circ C$ (same rate) and removing from the furnace. Each disc was ground to a powder (100 μm particle size), pressed into 1 and 1/4 inch discs in a steel die, repressed at 60 000 psi isostatically, sintered between 1300 and $1600^\circ C$ for 20-50 h, and cooled to $100^\circ C$ and removed from the furnace.

If x-ray diffractometer scans revealed the presence of a second phase, the discs, were reground and the last step was repeated until a single phase was achieved. When second phases were present they often appeared only on the surface and could be removed by grinding. Diffractometer

scans were taken from $15-155^\circ 2\theta$ using $\text{CuK}\alpha$ radiation ($\lambda = 1.5405 \text{ \AA}$). Lattice parameters were obtained by least squares fit to the Nelson-Riley function [9].

Density measurements were made on bulk samples by He gas pycnometry and compared with calculated x-ray densities in Table 2. The discs were employed as targets for deposition of the films by pulsed-laser deposition (PLD) technique. Deposition parameters were pulse-repetition rate of 10 Hz, laser fluence of $1-2 \text{ J/cm}^2$ at the target, and oxygen partial pressure and substrate temperature as given in Table 1.

TABLE 1. CONDITIONS FOR DEPOSITION OF MULTILAYERED STRUCTURES.

SAMPLE No.	COMPOUND	SUBSTRATE		OXYGEN PRESSURE (mTorr)	BLOCK TEMPERATURE ($^\circ\text{C}$)
		MgO	YBCO/MgO		
895	$\text{Ba}_2\text{InSbO}_6$	X	X	150	800
894	$\text{Ba}_2\text{ScSbO}_6$	X	X	150	800
883	$\text{Sr}_2\text{ScSbO}_6$	X	X	1	500
881	$\text{Sr}_2\text{ScSbO}_6$	X	X	1	400
880	$\text{Sr}_2\text{ScSbO}_6$	X	X	1	300
869	$\text{Sr}_2\text{ScSbO}_6$		X	15	550
868	$\text{Sr}_2\text{ScSbO}_6$		X	1	500
852	$\text{Sr}_2\text{InSbO}_6$		X	30	600
826	$\text{Sr}_2\text{GaSbO}_6$		X	170	800
815	$\text{Sr}_2\text{ScSbO}_6$	X	X	170	800
814	$\text{Sr}_2\text{GaSbO}_6$	X	X	170	800

Microwave measurements of the real and imaginary parts of the dielectric constant were performed at approximately 9.32 and 10.1 GHz and room temperature. A cavity perturbation technique was used with a reflection-type rectangular cavity excited in either the TE_{106} mode for 9.3 GHz or the TE_{107} mode for 10.1 GHz. The cavity was coupled to the waveguide by an adjustable iris. A 0.5 mm by 35 mm slot cut along the center of one of the broad sides of the cavity provides access for the thin, rectangular samples. The samples were held, such that their long dimension was parallel to the E-field of the cavity, and they were positioned at the E-field maximum, as determined by maximizing the shift of the cavity.

The real part of the dielectric constant is calculated from the shift in the resonance frequency of the cavity due to the sample, and the imaginary component is calculated from a change in the cavity Q. The accuracy of these measurements depends upon two general sources of error:

- (1) the accuracy of the cavity characterization; and
- (2) the material properties such as density and uniformity of shape.

The error due to the cavity characterization results in an accuracy of $\pm 2\%$ for the real part of the dielectric constant, and limits the resolution of the loss tangent (the imaginary component divided by the real component of the loss tangent) to approximately 0.001. The error due to material properties such as sample shape and material density can be considerably greater than the cavity characterization error. However, most sample densities were near 100% of theoretical density and thus should not introduce a large error.

3. RESULTS

All compounds in the series A_2MeSbO_6 , except barium-indium-antimonate (Ba_2InSbO_6), occur as ordered perovskites at room temperature. In an earlier investigation [11], Ba_2InSbO_6 was found to be ordered when prepared at $1100^\circ C$ using barium-dioxide (BaO_2). Only strontium-indium-antimonate (Sr_2InSbO_6) is cubic; the other Sr compounds are tetragonal (pseudo cubic). X-ray diffraction patterns are shown in Fig.1. Lattice parameters and microwave properties may be found in Table 2.

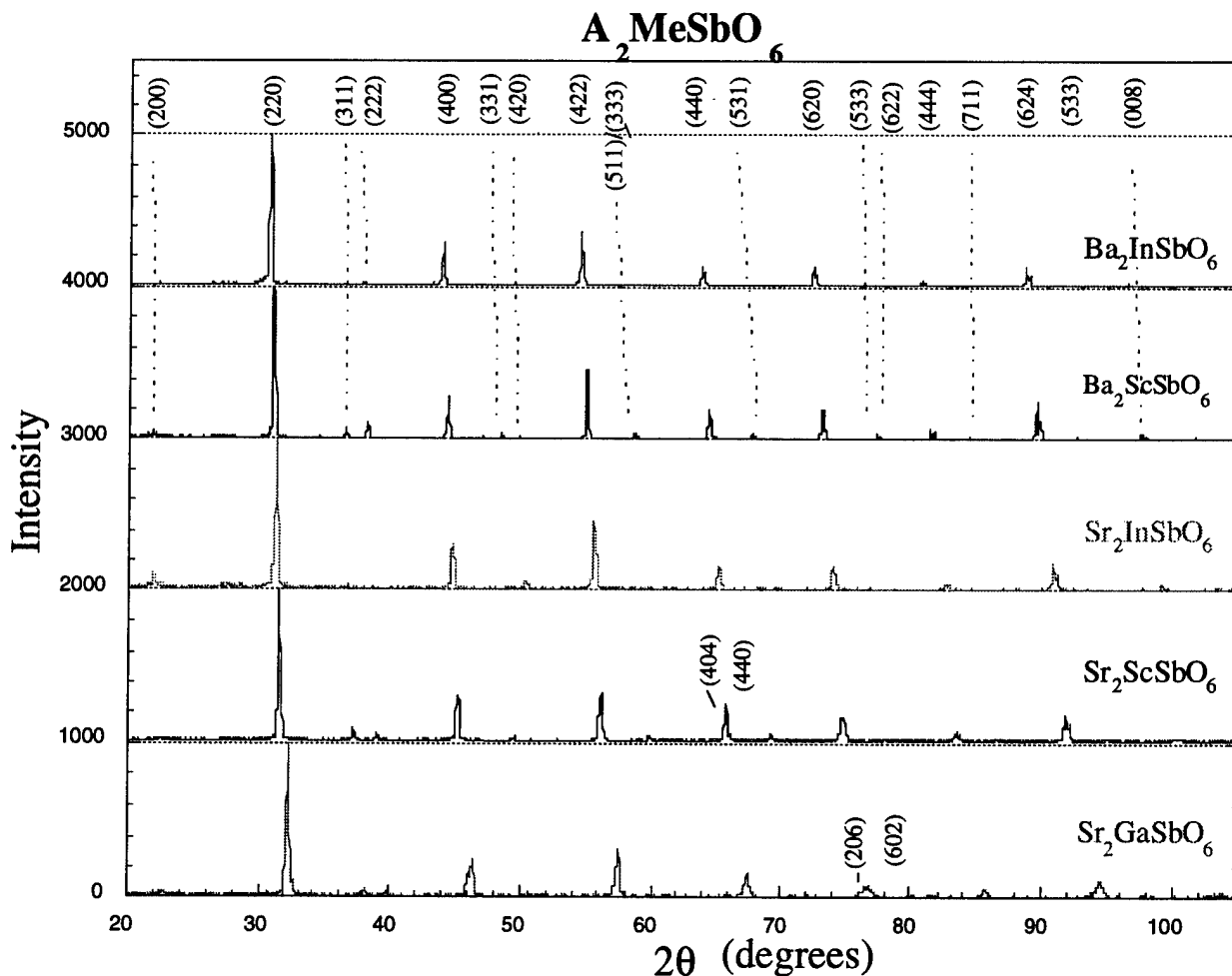


Figure 1. X-ray spectra of A_2MeSbO_6 bulk target samples. The x-ray lines were used to index the lattice parameters, Table 2. The FWHM of the 004 lines of the A_2MeSbO_6 compounds ranged from 0.19-29 2θ .

Cu K α was used as the radiation source.

TABLE 2. PROPERTIES OF A₂MeSbO₆ COMPOUNDS.

Compound	Lattice Parameter (Å)		Sintering Temp. (°C)	X-Ray Density (g/cm ³)		Dielectric		Color in Bulk
	Literature	Measured		x-ray	exp	Constant	Loss tangent (10 ⁻³)	
Ba ₂ InSbO ₆	(4.128) ¹² (4.136) ¹³ (8.269) ¹¹	a=4.174 a=4.134	1300 1400	7.33	7.31	6.9	20	Black
Ba ₂ ScSbO ₆	(8.203) ¹² (4.096) ¹³ (8.197) ¹¹	a=8.172 a=8.196	1400 1600	6.54	6.59	8.1	6.5	White
Sr ₂ InSbO ₆	(8.095) ¹² (4.051) ¹³	a=8.094 a=8.096 a=8.086	1200 1400 1600	6.34	6.30	8.3	<0.1	White
Sr ₂ ScSbO ₆	(a=5.678) ¹² (b=5.691) ¹² (c=8.021) ¹² (4.011) ¹³	a=8.019 c=8.063	1400	5.49	5.08	8.8	2.1	White
Sr ₂ GaSbO ₆	(a=7.84) ¹¹ (c=7.91) ¹¹ (7.892) ¹² (3.924) ¹³	a=7.860 c=7.910 a=7.880 c=7.784	1400 1500	6.22	6.12	13.5	9.5	White

When the A₂MeSbO₆ compound which formed exhibits an ordered cubic perovskite structure, a subcell and a supercell were sometimes observed. The most probable space group in this case is Fm3m. The lattice constant frequently contracted as the sintering temperature increased. For the two tetragonal Sr compounds, strontium-scandium-antimonate (Sr₂ScSbO₆) and strontium-gallium-antimonate (Sr₂GaSbO₆), splitting of the pseudo-cubic peaks is visible above 2 θ > 65° (Fig.1).

LaAlO₃ and MgO are often used as substrates on which high critical temperature superconducting films, such as YBCO, are grown for device applications. The lattice mismatch for heteroepitaxial growth of YBCO on LaAlO₃ is outstanding, about 1%; for MgO it is about 7%. The mismatch in lattice parameter for A₂MeSbO₆ compounds epitaxially grown on YBCO is given in Table 3.

Table 3. PERCENT MISFIT TO YBCO

Compound	<i>a</i> Lattice Parameter (Å)	% misfit to YBCO	
		<i>a</i> -axis	<i>b</i> -axis
Ba ₂ InSbO ₆	a=4.174	8.2	6.2
Ba ₂ ScSbO ₆	a=8.196	7.2	5.3
Sr ₂ InSbO ₆	a=8.086	5.8	3.9
Sr ₂ ScSbO ₆	a=8.019	4.9	3.0
Sr ₂ GaSbO ₆	a=7.880	3.1	1.3

With regard to % misfit, only one compound, Sr₂GaSbO₆, compares favorably with LaAlO₃; all but Ba₂ScSbO₆ are superior to MgO. In addition, the compounds of the series contain no magnetic ions and should exhibit no magnetic loss at any temperature.

Analysis of x-ray diffraction data obtained from single layer films of A₂MeSbO₆ on (100) MgO reveals that the 400 reflection of the A₂MeSbO₆ perovskite phase, when ordered, or 200 when not, is the most intense (see Table 4). The high quality of the A₂MeSbO₆ layer on YBCO (001) / (100) MgO is exhibited not only by the sharpness of x-ray diffraction lines (FWHM ≈ 0.30 2θ for 004 reflection) but also by the presence of high-order reflection, 008 (see Fig. 2). Crystallization of these thin-film A₂MeSbO₆ materials is observed at deposition temperatures as low as 600°C.

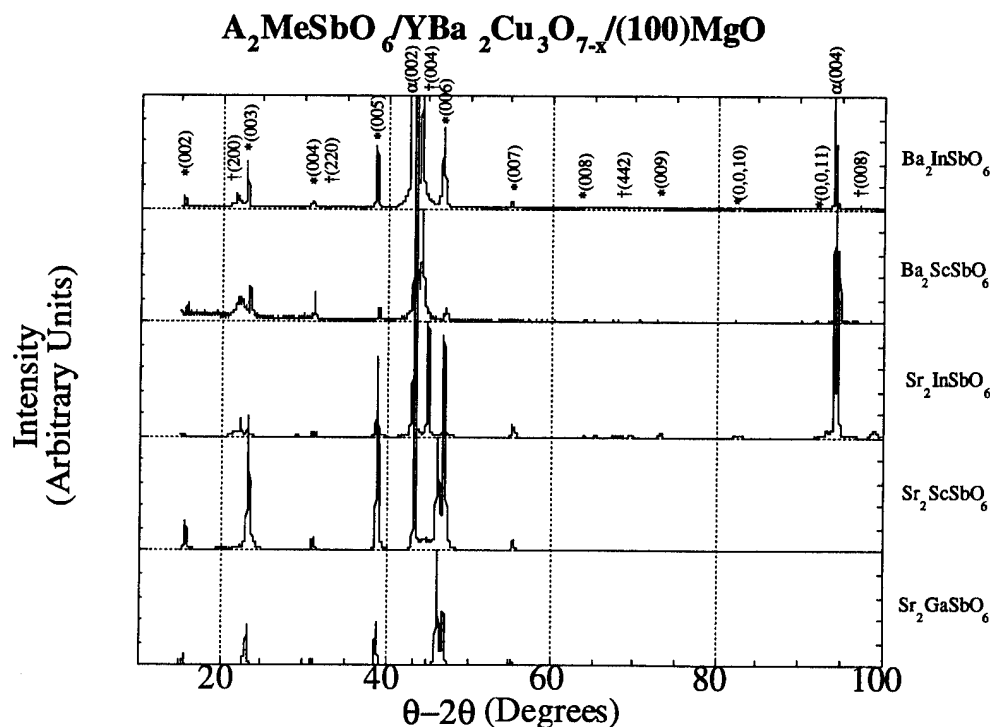


Figure 2. X-ray spectra of A₂MeSbO₆/YBCO/(100)MgO bilayers. The x-ray spectral lines for the YBCO, A₂MeSbO₆ films and MgO substrate are labeled using *, † and α, respectively, and appropriate (hkl)

indices. The FWHM of the 004 lines of the A_2MeSbO_6 layers ranged from 0.26-0.36 2θ . $Cu K\alpha$ was used as the radiation source.

TABLE 4. EPITAXIAL RELATIONSHIPS AMONG A_2MeSbO_6 , YBCO AND MgO.

Structure	Observed X-ray Reflections		
	MgO	Antimonate	
		Reflection	Intensity
Ba_2InSbO_6/MgO	200	100	very weak
	400	110	very weak
		200	very strong
		400	weak
Ba_2ScSbO_6/MgO	200	200	medium
		220	weak
		400	very strong
		422	very weak
		444	very weak
Sr_2InSbO_6/MgO	200	200	medium
		220	weak
		400	strong
		422	weak
		440	weak

4. SUMMARY

Five compounds in the series A_2MeSbO_6 , where $A = Ba, Sr$ and $Me = Sc, In$ and Ga , most of which are ordered perovskites, were successfully prepared by a solid-state reaction from oxide carbonate precursors. Epitaxial growth of thin-film A_2MeSbO_6 on YBCO by pulsed-laser ablation was demonstrated. X-ray diffraction lines for the epitaxial films were quite sharp and the added presence of a high-order reflection attested to the high quality of some of the films. Most of the A_2MeSbO_6 compounds possess dielectric constants smaller than substrate/buffer layers most commonly used, and most have quite acceptable loss tangents for microwave-device applications. The lattice misfit to YBCO is, in most instances, not comparable to $LaAlO_3$. Nevertheless, these compounds overcome many of the shortcomings of $LaAlO_3$, namely: most are cubic or almost cubic, do not undergo a phase transition on cooling, and have very desirable dielectric properties.

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