



ELECTROMAGNETIC COMPONENT RESEARCH

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14. ABSTRACT This report documents basic research in electromagnetic components performed by members of the Antenna Technology Branch of the Air Force Research Laboratory in collaboration with on-site contractors. The work includes modeling and measurement of III-V compound semiconductor devices for microwave and millimeter-wave transmit and receive applications, investigations of the factors that limit lifetime of radio frequency micro-electro-mechanical switches, creation of new micro-electro-mechanical and micromachined electromagnetic components, and investigation of the potential of emerging material systems for application to components. The research was supported by the Air Force Office of Scientific Research under Laboratory Tasks 92SN04COR and 07RY06COR.					
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PREFACE

This final report documents in-house research performed by members of the Air Force Research Laboratory, Sensors Directorate, Electromagnetics Technology Division, Antenna Technology Branch during the period 1 October 2001 to 30 September 2009. The authors of this report wrote most of the words contained herein, but the report represents the contributions of many more people. The scientific and technical team included the following Air Force Civilian employees, Air Force military members, and on-site contractors, collaborators, and students.

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Finally, we are grateful for the leadership and counsel of our late Division Chief, Dr. Richard Payne. His dedication to maintaining the basic research mission of the Electromagnetics Technology Division inspires us toward excellence.

1.0 SUMMARY

The broad objective of the Electromagnetic Component Research task was to improve our fundamental understanding of phenomena that enable or limit the future development of electromagnetic components for antenna applications. Overviews of the research accomplishments are presented in this report with additional detail provided in publications and patents. A list of publications, patents and presentations is included in the Appendix.

Over the eight year period of performance, we explored many different technologies. Our research in semiconductor devices included physics-based modeling and measurement of III-V devices made from GaAs, InP, GaN and various III-V alloys. The work was driven by the emergence of new material layer structures that enable high performance devices for microwave and millimeter-wave applications. Generally, our work included experimental measurement of devices to validate the models. In the area of Radio Frequency (RF) Micro-Electro-Mechanical Systems (MEMS), our research included both the creation of new MEMS components and the investigation of contact physics with the goal of understanding the failure mechanisms that plagued the early implementations of MEMS. We adapted emerging three-dimensional fabrications methods to create innovative microwave and millimeter-wave components such as couplers, filters, and diplexers. The resulting 3D circuits promise highly integrated, low-cost, high-performance alternatives to waveguide and microstrip components. Among the emerging material systems we investigated are multi-ferroic polymers for tunable RF components, GaN nanowires for sensors and transistors, and AlN pyroelectrics for millimeter-wave sensors. We investigated novel sensor components including fiber-optic based strain gages and thermal sensors for millimeter-wave imaging. We initiated research on micro-robots based on spherical shells of silicon and silicon dioxide. As they mature, the micro-sensors could detect a variety of phenomena including acoustic, electromagnetic activity, motion, and biological or chemical agents.

Several research areas have been transitioned to exploratory development, in particular, the MEMS and 3D circuit work. Two areas will be continued under new research tasks. The *Ensemble Based Matter* task will explore the creation and physics of dynamic matter formed from an ensemble of coordinated of sub-cubic millimeter robots. *Antenna-Coupled Millimeter-Wave Detectors using Pyroelectric Thin Films* will research a sensor consisting of an antenna, resistive load, pyroelectric capacitor, and interconnects supported by a micromachined silicon dioxide membrane on a silicon substrate.

2.0 INTRODUCTION

This report documents the Electromagnetic Component Research work unit over the period 01 October 2001 to 30 September 2009. The broad objective of the research was to improve our fundamental understanding of phenomena that enable or limit the future development of electromagnetic components for antenna applications. A number of scientific and technical areas were investigated during the course of this effort, and the sections are organized around these areas. Section 3 will discuss our research in physics based modeling of semiconductor devices. Section 4 covers microelectromechanical and micromachined devices. Section 5 reports on nanowires, multiferroics, and pyroelectrics. Novel sensor components are the subject of Section 6. Section 7 presents conclusions and recommendations. The effort has been documented in a number of journal articles, patents, and technical reports which are listed in the Appendix. These publications are available through the Defense Technical Information Center and in on-line repositories.

3.0 SEMICONDUCTOR DEVICE RESEARCH

Our research in semiconductor devices included physics-based modeling and measurement of III-V devices made from GaAs, InP, GaN and various III-V alloys. The work was driven by the emergence of new material layer structures that enable high performance devices for microwave and millimeter-wave applications. Generally, our work included experimental measurement of devices to validate the models.

Subsections 3.1 and 3.2 discuss High Electron Mobility Transistors (HEMTs) with InGaAs channel layers. These transistor structures use a high mobility indium alloy in the channel region and can achieve low noise figure with high gain in the millimeter-wave range, making them suitable for receive amplifiers in high performance electronic systems. However, along with the high mobility, the indium-based channel layer has a narrow bandgap, leading to the onset of impact ionization at relatively low voltages. Impact ionization is a noisy process and can limit the low-noise performance of the device. Modeling of the impact ionization provides insight into process and guides design of high performance HEMTs.

In subsection 3.3, we report on our research in AlGaIn/GaN HEMTs. GaN is a wide bandgap semiconductor and devices based on GaN can sustain high bias voltages and operate at higher temperatures compared to GaAs and InP based devices. These features make GaN HEMTs strong candidates for compact high power transistors in the microwave and millimeter-wave ranges.

3.1 Determination of Impact Ionization Gate Current InAlAs/InGaAs/InAlAs HEMTs

We devised a method to model the impact ionization gate current in InAlAs/InGaAs/InAlAs HEMTs. In the high electric field region of a HEMT, some electrons gain enough energy to produce holes and secondary electrons through impact ionization. Although the holes are confined to the channel by potential barriers of the HEMT quantum well, some fraction will tunnel through the barrier or have sufficient energy overcome the barrier and reach the gate. This hole current is proportional to the drain current, the impact ionization coefficient, the length of the high field region, and the probability of hole transmission through the gate potential barrier. We compute the ionization coefficient as a function of electric field using a Monte Carlo technique, calculating the fraction of electrons that participate in impact ionization as a function of electric field. The gate potential barrier is computed by solving the Schrödinger and Poisson's equations self-consistently to find the band bending and the two dimensional electron gas concentration in the channel as a function of the electric field at the heterointerface. The conduction band and valence band profiles can then be computed as a function of gate voltage. Next we use the Wentzel-Kramers-Broullion approximation to compute the hole transmission probability through the computed barrier. The ratio of the calculated channel hole current to the drain current follows the well known exponential dependence of the ionization constant when plotted as a function of reciprocal electric field. Fig. 1 shows the gate current due to impact ionization calculated from the current voltage characteristic shown in the inset and plotted as a function of source-drain bias (solid line). Experimental data points obtained using the extraction technique in [1] are shown for comparison. Theoretical calculations agree with the experimental measurements of the gate current in InP HEMTs. Further details of this technique will be submitted for publication as a journal article.

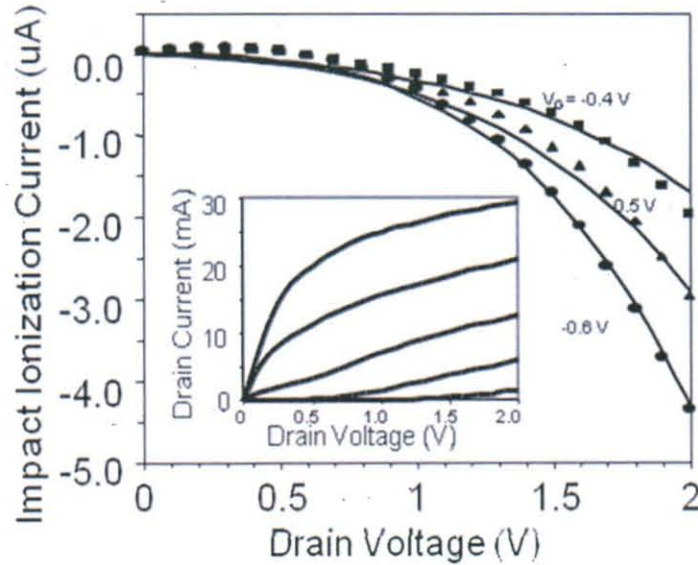


Figure 1: Gate Current due to Impact Ionization

3.2 AlGaAsSb/InGaAs/AlGaAsSb Metamorphic HEMTs

The impact ionization coefficient of AlGaAsSb/InGaAs/AlGaAsSb Metamorphic HEMTs (MHEMT) was experimentally determined by extracting the hole component of the gate current measured at DC. Impact ionization occurs when electrons in the HEMT are accelerated in the high field region between the gated channel and the drain electrode. The electrons gain sufficient energy to ionize lattice atoms, creating free electrons and holes. The newly created electrons add to the drain current, while the holes are attracted to the gate, which is at a more negative potential than the drain. Measurements were carried out over a temperature range extending from 4K to 296K. DC measurements were made with an Agilent 4156C Semiconductor Parameter Analyzer with the devices placed in a Desert Cryo probing station to control the measurement temperature. At each temperature, gate and drain currents were measured while the gate voltage was stepped from -0.8V to +0.2 V in 0.1V increments and the drain voltage was stepped from 0.0V to 1.2 V in 0.05 V increments. The source was held at 0.0 V. The hole component of the gate current was extracted using the technique described in [1]. The hole current of a $0.15 \mu\text{m} \times 40 \mu\text{m}$ AlGaAsSb/In_{0.8}Ga_{0.2}As HEMT is plotted in Fig. 2 as a function of gate voltage at a drain bias of 1.2V. The characteristic bell shaped curve is evident at all temperatures. This shape occurs because at large negative gate voltages the channel is depleted and there are few carriers to participate in impact ionization, while at positive gate voltages, the electric field between the gate and the positively biased drain is decreasing, and impact ionization becomes less likely. At moderately negative gate voltages, these effects balance and a maximum hole current flows.

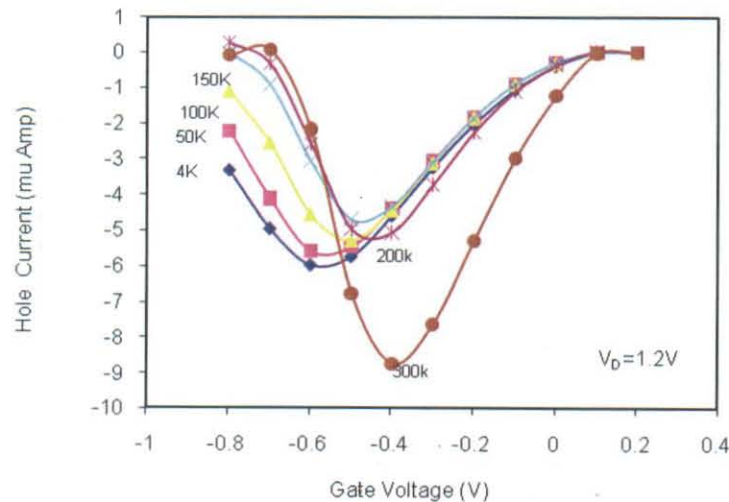


Figure 2: Extracted Hole Component of Gate Current versus Temperature

Fig. 3 shows the DC transconductance, g_D , as a function of gate to drain voltage. The onset of impact ionization occurs at a gate-drain voltage of 0.4 as evident from the dip in g_D . The

extraction of the ionization coefficient follows by plotting the log of the ratio of the hole component of the gate current to the drain current, I_{Hole}/I_D , as a function of reciprocal $(V_D - V_{D,sat})$ as shown in Fig. 4. Here, $V_{D,sat}$ is the drain voltage where the drain current saturates. The ionization coefficient is estimated from the intercept of the trend line (shown dashed for -0.7V) as $\alpha_n = \frac{1}{L} e^{Intercept}$, [1] where the length of the high field region, L , is estimated to be $0.5\mu m$.

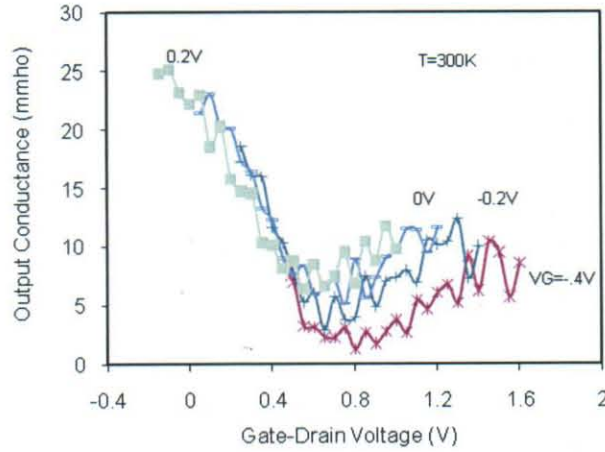


Figure 3: Room Temperature Output Conductance versus V_{GD}

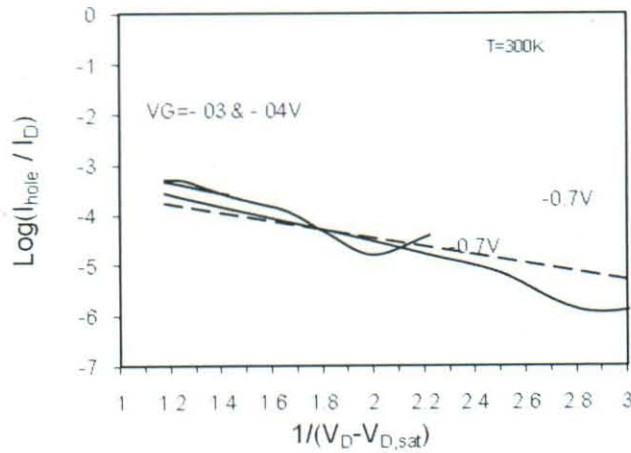


Figure 4: $\text{Log}(I_{Hole}/I_D)$ versus Reciprocal $(V_D - V_{D,sat})$, Gate Bias as Parameter

The extracted electron ionization coefficient as a function of temperature is shown in Fig. 5. For gate voltages above threshold ($>-0.7V$) the initial increase in α at lower temperatures is believed to be due to limited scattering that allows electrons to absorb energy from the applied field

facilitating impact ionization. Fig. 5 is replotted as a function of V_G with temperature as a parameter in Fig. 6. The initial decrease in α at lower gate bias is due to the increase in effective band gap, defined as the separation of the eigen energies, resulting from the quantum well becoming more triangular with increasing 2DEG concentration. A journal article is in preparation for this work and will provide additional details on the analysis of this data.

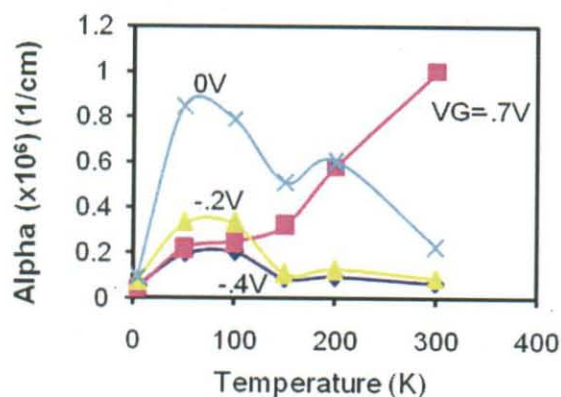


Figure 5: Impact Ionization Coefficient versus Temperature and V_G

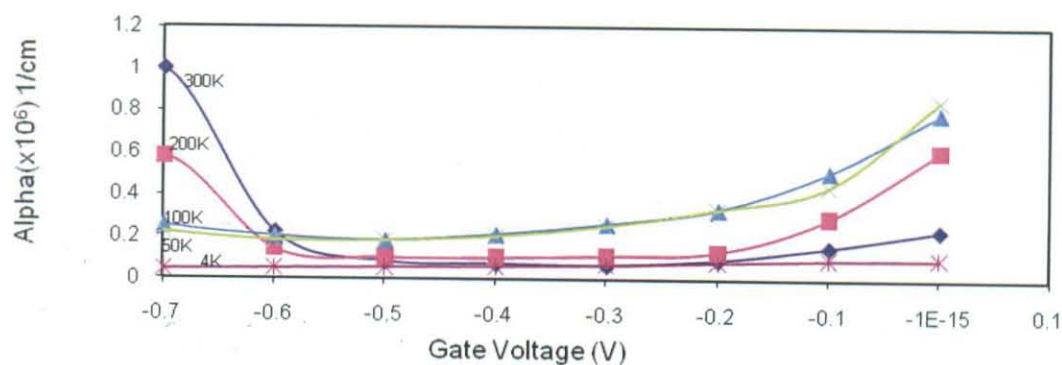


Figure 6: Impact Ionization Coefficient α versus V_G and Temperature

Noise in AlGaAsSb/InGaAs/AlGaAsSb Metamorphic HEMTs (MHEMTs) was determined through measurement and physics based modeling. A minimum noise figure of 0.82 dB at 15 GHz obtained for $0.15 \mu\text{m} \times 64 \mu\text{m}$ gates is an encouraging given the early stage of development for this material system. Additional details are available in [2].

3.3 GaN HEMTs

Initially, we examined the transport properties of GaN devices through physics based modeling. The Monte Carlo method was used to determine the electron mobility and diffusion constant in short GaN structures. Using the best available material parameters from the literature, the transport problem was set up, taking into account the scattering due to acoustic phonons, polar optical phonons, ionized impurities, impact ionization, alloy effects, interface effects with equivalent and non-equivalent inter-valley scattering. The Γ -L-X valleys were considered and assumed to be spherical. The simulation tracks about 20,000 electrons as they progress through the channel, with each electron suffering about 10,000 collisions. Channel length was varied from 0.1 μm to 2.0 μm . Velocity-field characteristics were compiled for each channel length over a field range from 0 to 1000 kV/cm. Mobility was determined from the slope of the velocity field characteristic at 20 kV/cm and is inversely proportional to channel length for lengths below 1 μm . In addition, the peak velocity moves to higher fields as the length decreases, that is the carriers become ballistic. Further details are available in [3].

We investigated the dependence of microwave performance of GaN/AlGaIn High Electron Mobility Transistors (HEMTs), namely the unity gain current cut-off frequency (f_T) and the maximum oscillation frequency (f_{MAX}), as a function of the mole fraction of Al and the thickness of the barrier AlGaIn layer [4]. The parameters were computed using a physics-based model and compared to experimental results. Schrödinger and Poisson's equations were solved self-consistently to relate the applied gate bias to the channel electron concentration. The contributions of both spontaneous and piezoelectric polarizations to f_T were explored. Finally, because of interest in using this family of devices at elevated temperatures, each simulation was repeated between 300K and 500K for comparison.

A mechanism for current collapse in GaN Metal Semiconductor Field Effect Transistors (MESFETs) was proposed, which assumed the existence of acceptor traps with multiple states in the bandgap. Current collapse has been experimentally observed in the current-voltage characteristic after the drain voltage sweep had exceeded the threshold for impact ionization in a previous measurement. In the proposed model, electrons generated by impact ionization are captured by neutral acceptor trap states in the substrate at energy levels above the valence band. The charged trap states move to an energy level located near midgap, creating a positively charged depletion region in the channel, and causing current collapse. With increasing drain bias, the quasi Fermi level approaches the charged trap states at the drain end of the gate, initiating detrapping of the electrons and restoring the current. The calculated results show good agreement with published experimental data. Further details are available in [5].

We investigated the gate bias dependence of the charge due to piezoelectric polarization in AlGaIn/GaN HEMTs by implementing a fully coupled electro-mechanical formulation based upon the piezoelectric constitutive equations for stress and electric displacement. The coupled formulation results in lower charge due to piezoelectric polarization as compared to the

uncoupled formulation for a given Al-mole fraction. With increasing two-dimensional electron gas concentration, that is, for gate biases greater than threshold, the compressive strain along the c-axis in the barrier AlGaIn layer increases with a concomitant increase of in-plane stress. We showed how current collapse correlates with the resulting increase in source and drain resistances through their dependence upon surface charge. We also formulated an alternate explanation of current collapse using local charge neutrality. This work was reported in [6].

To further investigate the current collapse phenomenon and to relate it to traps in the AlGaIn/GaN HEMTs, we performed measurements of Transmission Line Method (TLM) structures. We can relate the pulse transient response of the TLM structures as a function of pulse rate and temperature to the energy levels and time constants of traps and potentially differentiate between effects occurring under the gate and effects occurring in the areas between the source, gate and drain contacts. Further details of this technique will be submitted for publication as a journal article.

A physics based model for GaN MESFETs was developed to determine the frequency dispersion of output resistance and transconductance due to traps. The equivalent circuit parameters were obtained by considering the physical mechanisms for current collapse and the associated trap dynamics. Detrapping time extracted from drain-lag measurements were 1.55 seconds and 58.42 seconds indicating trap levels at 0.69eV and 0.79eV, respectively. The dispersion frequency is in the range of MHz at elevated temperature, where a typical GaN power device may operate, although at room temperature it may be only a few Hz. For a 1.5 μm x 150 μm GaN MESFET with drain bias of 10 V and gate bias of -1V, 5% decrease in transconductance and 62% decrease in output resistance at RF frequencies from their DC values were observed. The dispersion characteristics were found to be bias dependent. A significant decrease in transconductance was observed when the device operated in the region where detrapping is significant. As gate bias approached cutoff, the difference in output resistance between DC and RF increases. For drain and gate biases of 10V and -5V, output resistance decreases from 60.2 K. at DC to 7.5 K. at RF for a 1.5 μm x 150 μm GaN MESFET. Further details are available in [7].

4.0 MICRO-ELECTRO-MECHANICAL SYSTEMS AND MICROMACHINED COMPONENTS

Radio Frequency Microelectromechanical Systems (RF MEMS) have generated a great deal of enthusiasm in the DOD community due to their potential impact on microwave and millimeterwave systems. RF MEMS switches have shown exceptional performance. In particular, these switches have a unique combination of low insertion loss (< 0.3 dB at 10 GHz), good isolation (> 20 dB at 10 GHz), low drive power (< 0.1 mW), wide-bandwidth (DC-60 GHz), and monolithic circuit integration. The impact of these switches in RF circuits is exemplified by their implementation in phase shifters. Phase shifters using RF MEMS switches

have been demonstrated from X-band through W-band. At every frequency, the MEMS based phase shifters have shown significantly lower loss than any other MMIC technology. The combination of exceptional RF performance, low drive power, and monolithic integration is not available from any other technology. This combination has direct impacts at the system level, particularly in the design of large-lightweight phased arrays.

However, when they were first introduced in the 1990's, RF MEMS switches were not viable for system implementation due to two key weaknesses: packaging and lifetime. These weaknesses severely impacted a number of DARPA programs including the MEMTenna, LCCMD, and Low Cost MEMS ESA programs. In all three cases the programs were terminated due to a lack of reliable MEMS switches. The lifetime of these switches is primarily driven by the physics of the contacting surfaces, and very long lifetimes can only be realized by a fundamental physical understanding of the contacting surfaces. The AFRL/RHYA basic research effort therefore focused on the problem of contact physics.

There are two primary types of RF MEMS switches: capacitive contact switches and metal contact switches. Capacitive contact switches are created by suspending a metal beam a short distance (typically 1-5 microns) above a dielectric coated lower metal. Applying a voltage between the two metals causes the suspended metal to collapse onto the dielectric, thus increasing the capacitance by a factor of 30-200, creating an effective short circuit at high frequencies. Metal contact switches remove the dielectric layer, and directly form a contact by driving one metal into the other. Our research investigated of the phenomena associated with bringing a metal surface into contact with either a dielectric or another metal.

4.1 Charging and Contact Physics in MEMS Switches

The primary focus of our work on capacitive contact switches was an investigation of dielectric charging. Capacitive switches operate by applying a large voltage between the two metal electrodes. This voltage pulls the suspended electrode down onto the dielectric. For most switches, the actuation voltage is on the order of 20-40 V. This voltage is dropped across the relatively thin dielectric (typically 0.2 micrometers) resulting in a field on the order of 10^8 V/m. Under such high fields, charge is injected into the dielectric where it becomes trapped and remains stored. To help us understand this mechanism, we performed simulations showing how charge in a thin film dielectric affects the performance of capacitive MEMS switches. The simulator incorporates fixed charge into the dielectric as a function of depth, and can readily be extended to handle charge as a function of both location and depth [8]. We validated these simulations with measurements from numerous switches. Direct observations of shifts in the Capacitance-Voltage (C-V) curves of capacitive switches correspond very well with our predictions. Using the measurement system created for verifying the C-V curve shift, we performed studies of charge build-up in the switches over extended cycling (up to 60 minutes) [9]. The data make it clear that the charging phenomenon is dynamic, with trapped charge concentration changing during the switch cycle.

We devised a new technique for measuring the voltage offset caused by charge in the dielectric layer of capacitive MEMS switches [10]. This technique does not require that the bridge come into contact with the dielectric. By avoiding contact during the measurements, the method measures the voltage shift caused by charge in the dielectric without introducing additional charge. Contact is avoided by driving the switch with a bias that always remains below the pull-in voltage. This low voltage results in only a small deflection of the bridge. The bridge deflection is dependent upon the applied bias and the charge in the dielectric. Although this deflection is small, it is sufficient to modulate a microwave signal. Therefore, by monitoring the modulated signal, the voltage shift can be directly observed. The technique is capable of measuring the charge induced-voltage shift with better than 0.2 V resolution.

An analytical calculation of microwave actuation for shunt capacitive MEMS switches was performed [11]. The calculation shows that the microwave signal deflects the beam according to the rms voltage of the signal. In addition, heating of the beam due to dissipated microwave power was shown to play a significant role in microwave actuation.

In metal contact switches, our efforts focused on understanding the metal-metal contact physics, in particular, the force versus voltage relationship for the switches. One of the primary design initiatives for metal-to-metal contact switches is to fabricate high force structures that exhibit low on-state contact resistances. Pull-in voltages can range from a few volts to 100 volts or more. The risk of having too low a pull-in voltage is that the beam can't provide enough restoring force to overcome contact forces and open the switch when the voltage is removed, thus the switch sticks closed. Too high a pull-in voltage causes difficulty in generating and distributing control voltages. With a high contact force, a better metal-to-metal contact is achieved which reduces the on-state resistance of the switch. For the metal contact switch, resistivity levels less than 1 ohm are desirable. An analysis of the contact and restoring forces for a series of metal contact micromechanical (MEMS) switches was performed. Several of the numerous tradeoffs required to increase contact and restoring forces were determined and reported in [12,13]. In collaboration with the Aerospace Components Division of the Sensors Directorate, we developed a new analytic contact resistance model for microswitches employing hemispherically-shaped upper contacts and sputtered contact metals [14]. The model was validated through the design, fabrication, and test of microswitches with metal alloy electric contacts. Overall, the results show increased microswitch reliability in exchange for a small increase in contact resistance for devices with bimetallic, binary alloy, and ternary alloy electric contacts.

In collaboration with the Air Force Institute of Technology, we investigated MEMS switch triboelectrification [15]. The goal of the work was to determine the degree of charge buildup due to the mechanical impingement of the switch onto the contacts and drive electrode. A dynamic model of switch deflection as a function of time was developed. This is a two-dimensional finite element beam model including electrostatic actuation. The model allows us to predict the bounce of the beam contact and the impact energy. We also developed a physics-based model of

charge injection resulting from the contact impacting the dielectric layer of the MEM actuating structure.

The residual stress and bending mechanisms in multi-layered nickel micro-cantilevers were examined through the use of interferometric microscope imaging. These cantilevers were constructed using layered, selective electrodeposition of Ni as a structural material and copper as the sacrificial material. Planarization of both the structural and sacrificial material occurs prior to subsequent layer depositions. Chemical etching of the sacrificial Cu layer is performed as a final step, releasing the structures. Each Ni deposition consists of a single layer of 8 μm thickness. The cantilevers were designed and fabricated with thicknesses ranging from 8 μm to 32 μm with corresponding lengths ranging from 200 μm to 500 μm . The cantilever array consisted of a series of 16 unique structures as illustrated in Fig. 7. Due to the planarization, these cantilevers had a near ideal fixed support instead of the typical stepped-up fixed support as generally observed in standard MEMS structures. In the four tested arrays, downward deflection was observed in all cantilevers. The measured downward deflection was most likely a result of either mismatch or the effects of planerization. Interpolation and error reduction techniques were used to account for sinusoidal variations in the interferometric measurements. Based on these data, analytic and computation approaches to the characterization of internal stresses were accomplished. Analytically, both pure bending and bending due to combined loading were considered in the characterization of the internal bending mechanisms and the calculation of the internal stress state. This behavior was examined as a function of both length and thickness. The Young's modulus used in the stress analysis was obtained from the cantilever fabricator, Microfabrica, Inc. Computationally, a finite element analysis software program (ANSYS, Ansys Inc.) was used to determine stresses resulting from an applied displacement equal to the maximum measured deflection. The result of this research on multi-layered Ni structures is important for both the successful design of RF MEMS components and for their performance. More specifically, this characterization enables the prediction of gaps, clearances, and behavior for electrostatically actuated devices.

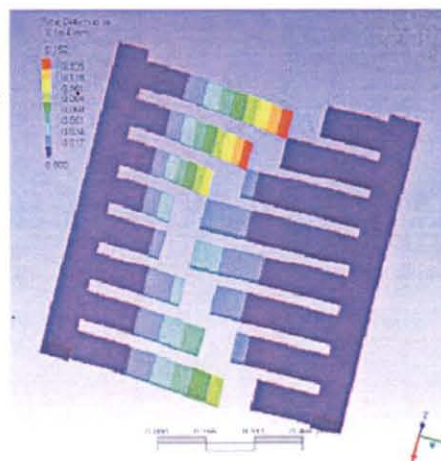


Figure 7: Deflection Model of Cantilever Array

4.2 New MEMS Devices and Packaging Concept

In addition to the studies of switch actuation and contacts, we researched new device and packaging concepts. A 2-bit digital variable MEMS capacitor was demonstrated with a capacitance range from 206 -335 fF [16]. The varactor consists of three cantilevered parallel plate capacitors. The bias voltage exerts electrostatic force to sequentially pull each cantilever down to a fixed stop, reducing the capacitor plate separation and increasing the capacitance. The cantilevers have different lengths and the longer ones collapse at lower voltage. Therefore, it can be biased with a single control voltage to allow simplified microwave system designs. The capacitor was designed so that the four capacitance states are approximately equally spaced. Fig. 8 shows a photomicrograph of the varactor and the measure capacitance-voltage curve. As a result of this work, United States patent number 7,283,347, *Low cost digital variable capacitor*, was issued October 16, 2007, with J. Robert Reid as the inventor.

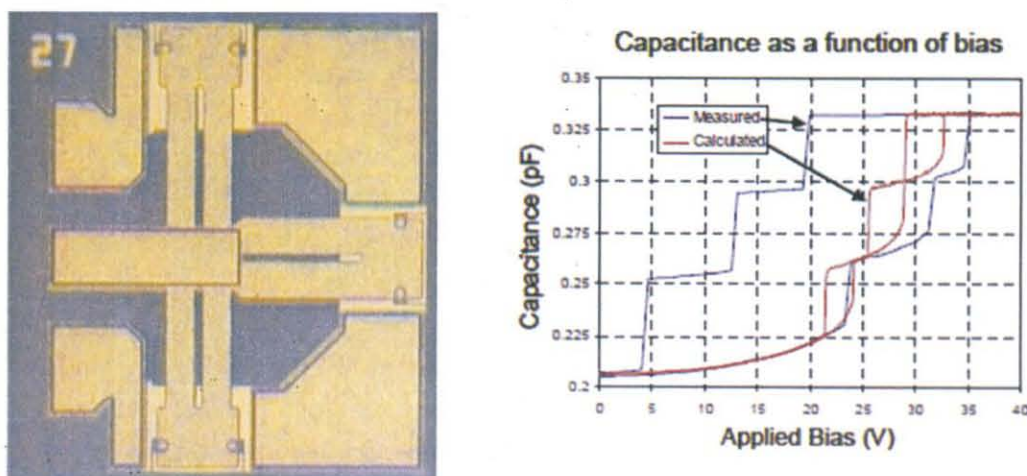


Figure 8: Optical Photo-micrograph of MEMS Varactor and Capacitance versus Applied Bias

We devised a method for contacting packaged MEMS switches that should result in higher performance, lower cost devices. Electrostatically actuated broadband micro-switches are prospective candidates as replacements for low-power mechanical relays or semiconductor switches and can be used in a variety of RF applications. These switches are fabricated using standard thin film processing allowing them to be integrated with active devices or designed into single-pole-single-throw (SPST), single-pole-double-throw (SPDT), or series-shunt switches. When compared with traditional electromechanical relays, MEMS RF switches provide significant savings in size and power consumption. However, as currently packaged, the individual switch die, shown in Fig. 9a, is fairly large at 1.45 mm square. Much of the top-side area is required for making electrical contacts to the device. In addition, the cap projects above the contact pads by 0.4 mm.

With the new method, we gain access from the backside of the switch wafer to the metal contacts of the MEMS switch. In this manner, contact can be made under the glass frit, and the die size could be reduced to well under 1 mm square. In addition, the devices could easily be flip-chip bonded, allowing hybrid integration of the devices. To implement the process, the switches are fabricated using the existing commercial process, except that the silicon starting wafer is replaced by a silicon-on-insulator (SOI) wafer as shown in Fig. 9b. Using the buried SiO_2 layer as an etch stop, we remove most of the backside silicon wafer. Via holes can then be etched through the SiO_2 , thin silicon, and oxide layers to make contact with the switch contacts under the glass wafer-bonding frit. This chip has shorter contact leads, lower contact parasitics, and is significantly smaller, so more chips can be fabricated on a wafer leading to lower cost per chip. We demonstrated this process on chips made by Radant MEMS, Inc on SOI wafers.

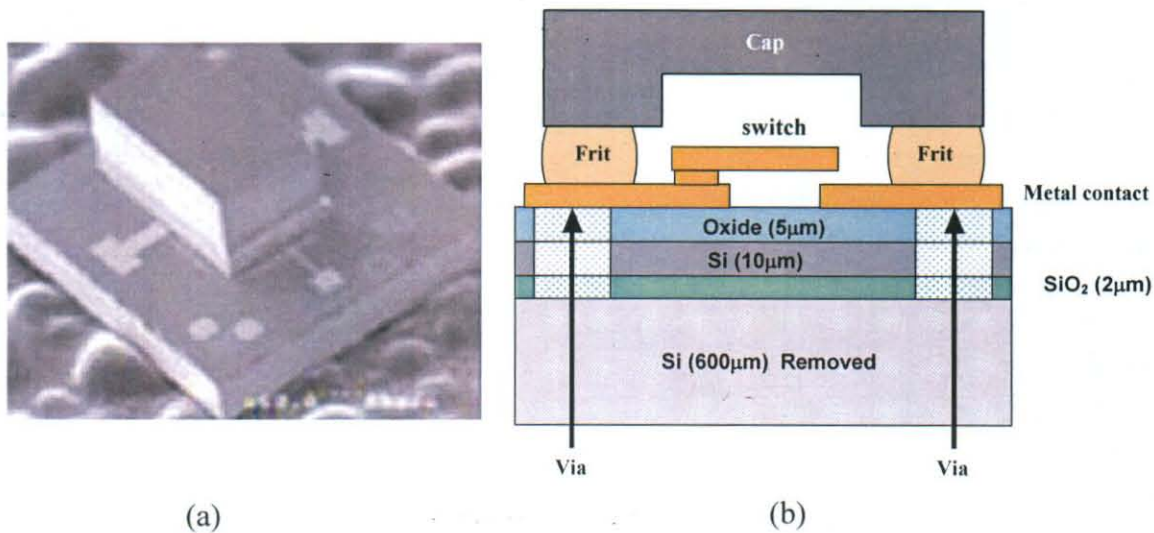


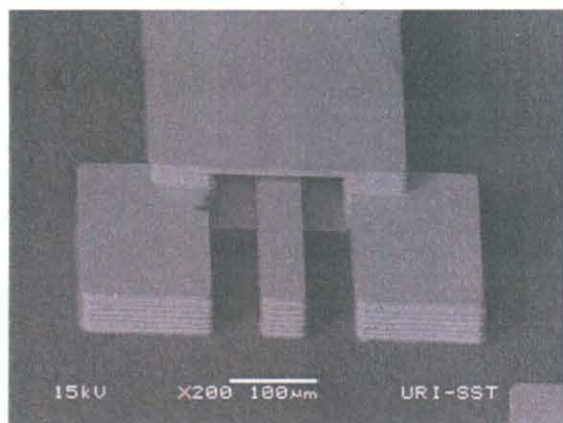
Figure 9: (a) SEM Micrograph of Commercial Capped Die Micro-switch (Radant MEMS, Inc.) (b) New Technique for Contacting Switch

4.3 Three Dimensional Millimeter-wave Components

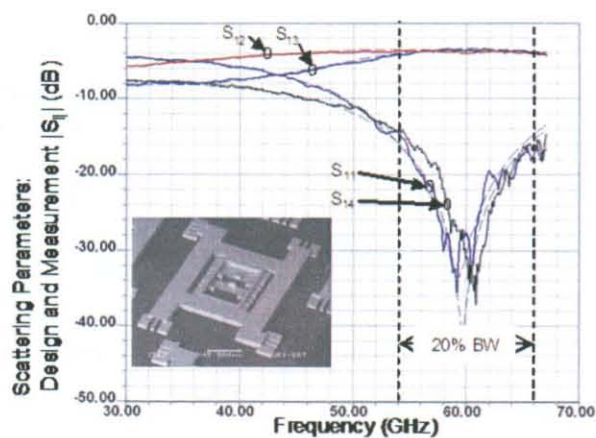
We created innovative millimeter-wave components using emerging three-dimensional micro-fabrication technologies. Micro-fabrication allows greater precision, more flexibility, and better reproducibility than previous millimeter-wave circuit technologies. A key feature of the technology is that shielded transmission lines can be densely packed to form previously unrealizable, complex millimeter-wave networks. When mature, this technology will incorporate active and passive devices, radiating elements, and control circuitry, enabling fully integrated millimeter-wave sub-systems.

In our initial investigations, we built millimeter-wave components using a micro-fabrication technique being developed by Microfabrica, Inc. of Burbank CA. In this process, the following three steps are repeated to generate a desired number of metal layers: 1) A metal pattern is electroplated onto the substrate; 2) A second metal is blanket deposited over the entire substrate; 3) The top surface is planarized to create a single layer of uniform thickness with two metals. After all of the layers have been fabricated, one of the two metals is selectively etched away leaving the other metal in place to create 3D mechanical structures. The process is fundamentally the same as a sacrificial surface micromachining process, with two critical distinctions. First, the layers in this process are typically much thicker (2-15 μm) than in surface micromachining (0.5-4 μm). Second, the planarization allows significantly more layers ($\gg 10$) than are common in surface micromachining. Fig. 10a shows a three dimensional shielded coaxial line with a coplanar transition for millimeter-wave probing. The metal structure is 92 μm high, and consists of twelve layers of nickel on an alumina substrate.

As an example of the excellent millimeterwave performance we achieved on our first build, Fig. 10b shows a 60 GHz branch line coupler fabricated using the rectangular coaxial transmission lines [17]. The coupler has a measured insertion loss of 3.4-4.2 dB over the 55-61 GHz frequency range. The device was fabricated successfully in a single design pass. We also built a V-band bandpass filter. The use of integrated coaxial lines allows the fabrication of a compact filter measuring 6.5 mm x 2.0 mm x 0.1 mm. In addition, the coaxial lines isolate the filter from the external environment allowing filters to be spaced close to other devices without electromagnetic coupling. The fabricated filter has a center frequency of 57.5 GHz and a bandwidth of 2.5 GHz [18]. We transitioned this 3D component work to exploratory development, and continue to develop advanced components including switch matrices, tunable filters, and diplexers.



(a)



(b)

Figure 10: a) Three-dimensional Millimeter-wave Coaxial Line with Coplanar Transition for Probing (b) 60 GHz Branch Line Coupler, Measured S-parameters Agree with Design Curves

5.0 EMERGING MATERIALS FOR ELECTROMAGNETIC COMPONENTS

5.1 Thin Film Pyroelectrics

Pyroelectric materials produce a voltage that is proportional to the change in temperature and a room temperature sensor of millimeterwaves can be made from pyroelectric thin films. We investigated AlN as a candidate for pyroelectric sensors. We prepared thin films of AlN by DC reactive sputtering of aluminum in 3×10^{-3} Torr ammonia atmosphere onto polished n-type silicon wafers. The power density at the target was approximately 5 Watts cm^{-2} . The substrates were $0.1 \text{ }\Omega\text{-cm}$, n-type polished silicon with (111) orientation. The films were heated to above $200 \text{ }^\circ\text{C}$ and cooled to $<50 \text{ }^\circ\text{C}$ with the chamber at or below 1×10^{-8} Torr prior to the sputter deposition. The resulting AlN films were semi insulating to slightly p-type with typical through-film resistance of $25 \text{ M}\Omega/\mu\text{m}$ for 1 cm^2 of area. The thickness of films thus prepared, varied in the range $600 \text{ }\text{\AA}$ to $2500 \text{ }\text{\AA}$ controlled by sputtering time. Subsequent to the AlN deposition, a $400 \text{ }\text{\AA}$ layer of NiCr was applied to act as both a contact to the AlN and to increase the infrared (IR) absorbance. The n-type silicon substrate was used as the back contact. To measure the pyroelectric response, a broadband incident source was focused on the specimen after passing through a vane chopper. A silicon wafer etched and polished on both sides was interposed between the source and the specimen to insure that any electrical responses related to the AlN/Si substrate interface were suppressed. Response of a specimen to the chopped IR source is given in Fig. 11. As expected for a pyroelectric material, the AlN film responds only to the time rate of change of temperature (dT/dt) rather than to the absolute value of temperature per se. A first estimate of response time can be obtained from the exponential decay part of the trace. The roll-off part of the data rise was fitted to an exponential as shown in Fig. 12. From that a $1/e$ response time of 0.013 s is calculated. If one assumes a linear rise in temperature proportional to the rate of energy input and an exponential fall off proportional to the conductivity of the AlN and the Si substrate on which it was formed, then the time to maximum temperature rate of change is 0.75 s , which is close to the value observed in the figure. Additional information on this pyroelectric AlN work can be found in [19].

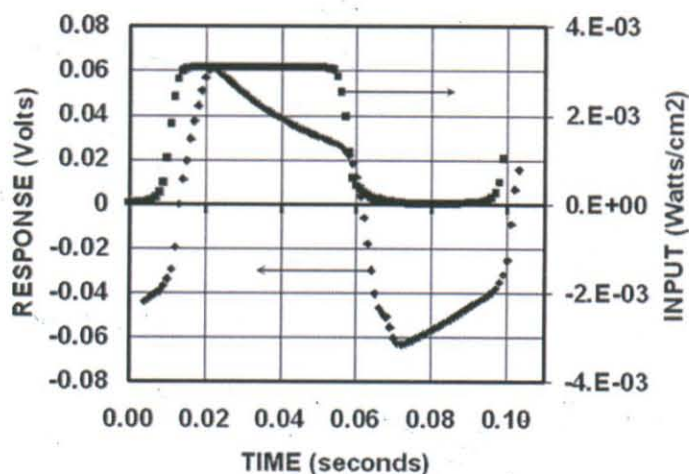


Figure 11: Response of AlN Film to Pulsed, Broadband, IR Light

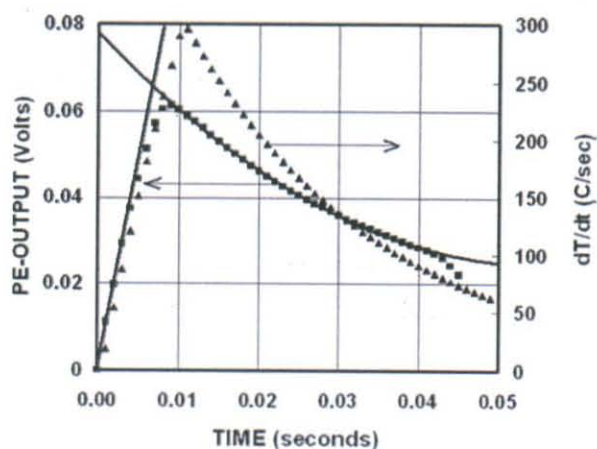


Figure 12: Positive Response Half Cycle of Fig. 11 compared to Rate of Temperature Change from Heat Transfer Model

5.2 Polymer Materials for Tunable Microwave Devices

Previous experience with polymer compounds indicate that poly (vinylidene) fluoride (PVDF) might be an appropriate candidate for creating tunable microwave devices such as phase shifters, time delay units, and reconfigurable antennas. We chose PVDF to investigate as the starting material because it is a well-documented polymer, extremely durable in solid form, readily dissolved, ultraviolet radiation resistant, and shown by researchers to have complicated polymorphism properties. Fig. 13 shows a model of PVDF created in the atomic chemical bonding computational computer graphics program HyperChem7.

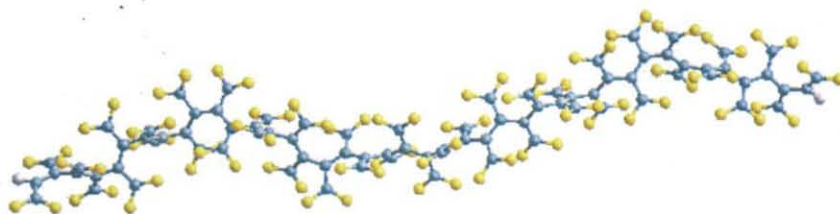


Figure 13: PVDF Alpha-Phase Model, Carbon = blue; Hydrogen = white; Fluorine = yellow

PVDF is a piezoelectric material. This means that when it is placed in an electric field it will change shape. Because fluorine is so much more electronegative than carbon, research has demonstrated that the fluorine atoms pull electrons away from the carbon atoms. This makes the CF_2 groups polar, with a partial negative charge on the fluorine atoms and a partial positive charge on the carbon atoms. When we place the molecule in an electrical field, the electrons will align causing the polymer sample to deform. It is already known that one of the polymorphs of PVDF (β -phase) is strongly ferroelectric. In consultation with Prof. William Euler, polymer expert from the University of Rhode Island, we explored ways to incorporate magnetic ions, such as cobalt and manganese, into the PVDF structure.

Solid PVDF-plus-ion specimens were prepared by retarded evaporation of the solvent from a quantity of solution contained in small ampoule bottles. Solidification was effected with and without the presence of polarizing 3 kG permanent magnets adjacent to the solidification chamber. Several chambers were constructed for specimen preparation including chambers that permit compressive stress to be applied after solidification. The specimens thus prepared were then measured in a SQUID magnetometer to determine their magnetic signature. Several specimens showed ferromagnetic response. In particular, the ones prepared from cobalt-based salts showed both the saturation and hysteresis character of ferromagnetic material. These were analyzed for optical transmission. Specimens of PVDF with gadolinium as the magnetic ion were also prepared for optical analysis. Because of the strong color tints of the liquid PVDF-plus-ion solutions, particularly when Co ions were used, the liquids were analyzed in the visible through near IR (380-1200nm). Strong absorption of the solvents precluded extending the measurements any further into the IR.

Thin films of the solutions were prepared by spinning onto 4 inch diameter polished silicon wafers. While the parameters necessary for consistent film production (spin speed, solution viscosity, and temperature) were not fully determined, portions of the films were measured by Fourier Transform Infrared spectroscopy (FTIR) and some interesting trends were observed. The following preliminary conclusions were drawn from our analyses:

1. The presence of the magnetic ions, regardless of specie, promotes the formation of the polar beta phase as opposed to the alpha phase that is usually more prevalent with the spin-casting and retarded evaporation methods of PVDF thin film preparation.
2. The largest magnetization effect observed was in a Co 'doped' specimen that contained 0.08% weight percent.
3. Both Gd and Mg containing PVDF specimens showed minimal magnetization response.
4. The IR transmission measurements are suitable for determining the ratios of the PVDF phases commonly formed by our preparation methods.
5. Scanning Electron Microscope/Electron Diffraction Spectroscopy (SEM/EDS) measurements of several films disclosed that the magnetic ion species (M^{+}) are generally evenly distributed throughout the PVDF specimens. Occasionally, in the specimens with the highest concentrations, islands of concentrated M^{+} were observed, but such localization did not show correlation with the SQUID response.

For microwave measurements, we prepared pellets with a diameter that is slightly larger than the coaxial waveguide used to measure permeability and permittivity of the materials. Final fit to the waveguide was performed by miniature lathe machining of the pellets. Measurements and analysis of PVDF-plus-Co specimens in the coaxial waveguide showed regions of reduced permeability ($\mu < 1$) centered at 13.6 GHz. This frequency is quite close to the reported value for electron paramagnetic resonance (EPR) of Co. However, deconvolution of the permittivity and permeability parameters from our preliminary measurements proved to be quite complex because of the extensive procedures required to calibrate the wave guide system for particular wave length regions. Thus, the correctness of these values is suspect. Our intent was to continue this work under a new research task, but that task has not been funded.

5.3 Nanowire Formation

We investigated the potential of reactive sputtering as a process for creating III-N nanowires. Simple reactive sputtering forms uniform, smooth, high density, polycrystalline films with c-axis orientation normal to the surface of the substrate. To promote filament growth out of the plane of the film growth, we added a driving force in the form of a high voltage negative bias onto the substrate with respect to the plasma. Typical temperature for growth is 350°C. We performed the experiment on silicon substrates with $\sim 100\text{\AA}$ of either SiO_2 or Al_2O_3 on the growth side of the substrate. Seeding of nanowires was promoted by depositing $\sim 50\text{\AA}$ gold layer on top of the SiO_2 or Al_2O_3 and then heating to 700°C. This processing uses to advantage the low wetting coefficient of the Au/oxide interface causing the Au to form submicron (nearly) circular dots which then act as the seed points for filament growth. The process was applied to the growth of GaN. Using -1000 V_{DC} bias, areas of filament growth were regularly formed. Both rod-like and articulated growths were observed but the processing parameters that control the various

morphologies are not understood as yet. The fact that neither unbiased nor positively biased substrates show any sign of filaments on their surfaces demonstrates that the negative bias field is important for III-N nanorod growth. Fig. 14 shows some examples of growth morphologies observed for the field enhanced reactive sputtering of GaN. On the left are commonly observed articulated loops. The largest loop in the picture is approximately 60 μ m long. In the center is a straight rod in proximity to an articulated loop. On the right is an example of pathological growth formation. In most of the growths the bright and/or bulbous end of the filament represents the Au seed from which the filament growth proceeds.

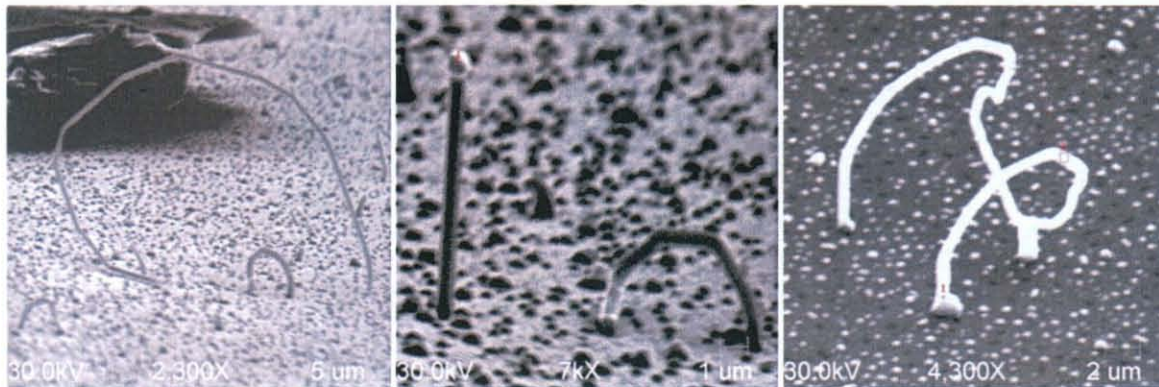


Figure 14: Examples of GaN Nanowires

6.0 NOVEL SENSORS

6.1 Optical Strain Gages

We performed experiments and analysis of bending strain sensors based on hollow wave-guide capillary coated with optically active thin films. The strain sensors were designed for compatibility with optical fiber installations and can be attached to structures to measure their bending strains. The sensors can be adjusted to maximize gage factor for predetermined strain ranges. The sensor consists of glass micro tubing coated on the outer surfaces with an optical absorbing layer followed by a reflecting layer. A mechanical strengthening layer was included in some experiments to extend the measurable range of strain. In operation, a source laser beam from an optical fiber is injected into one end of the gage. The light remaining in the beam after traveling through the gage is collected via another optical fiber. For a given thickness of the absorber layer, the detected light is proportional to the amount of bending. Thus by rigidly affixing the sensor to a structural member, the instantaneous strain and strain history experienced by the member can be monitored. Typical gage lengths are 3 to 10 cm. Applications envisioned for this device include monitoring the strain histories of buildings in earthquake prone areas,

ships on the open seas and airplanes during abrupt maneuvers. Several specimens were constructed with differing silicon layer thickness as the optically absorbing layer. Gage factors from -40 to -200 were measured for those specimens with the absolute value increasing as the thickness of the silicon absorber was increased. Maximum strains of 5000 micro-strain were observed for the gages prepared on bare glass capillaries. The maximum observable strain was increased to 10,000 micro-strain by the addition of a one micrometer thick polyamide strengthening coating on the capillary. This research project resulted in United States patent number 7,002,673, *Optical strain gage compatible with fiber optic systems*, E. E. Crisman, inventor.

6.2 Micro-machined millimeter-wave sensors

We combined MEMS micromachining with thin film technology as a step toward creating a simple low cost millimeter-wave sensor. The sensor consists of a planar millimeter-wave dipole antenna, terminated by a thin film resistor, thermally isolated from the underlying substrate by micromachining. Energy received by the antenna is dissipated in the resistor causing the resistor's temperature to rise. In our initial work, the change in temperature was measured using a thermopile. Thermal and millimeter-wave radiation calculations showed there is potential for the sensor to work actively and passively in an outside environment. We used the MOSIS fabrication service as an economical way to make the antenna, resistor, and interconnect circuitry. This is a standard CMOS process that we adapted to create millimeter wave antenna elements, resistors, thermopiles, and patterned glass structures for subsequent etching in our clean room. On the fabrication runs, we included test structures for each of these components as well as prototype sensors. A millimeter-wave antenna and a dc test structure are shown in Fig. 15a and 15b. The DC test results in Fig. 15c showed sensitivities in the range of 10 to 14 mV/mW. However, the sensitivity of the 94 GHz sensor was low due to interaction with the underlying silicon that was not properly accounted for in the initial design. We developed a process for release-etching the suspended structures using an initial etch in XeF_2 followed by a Tetra Methyl Ammonium Hydroxide (TMAH) wet etch which selectively removes silicon but not glass. The process includes a method for mounting the die to prevent backside etching. This work will continue under a new research task, using pyroelectric material as the temperature sensing element.

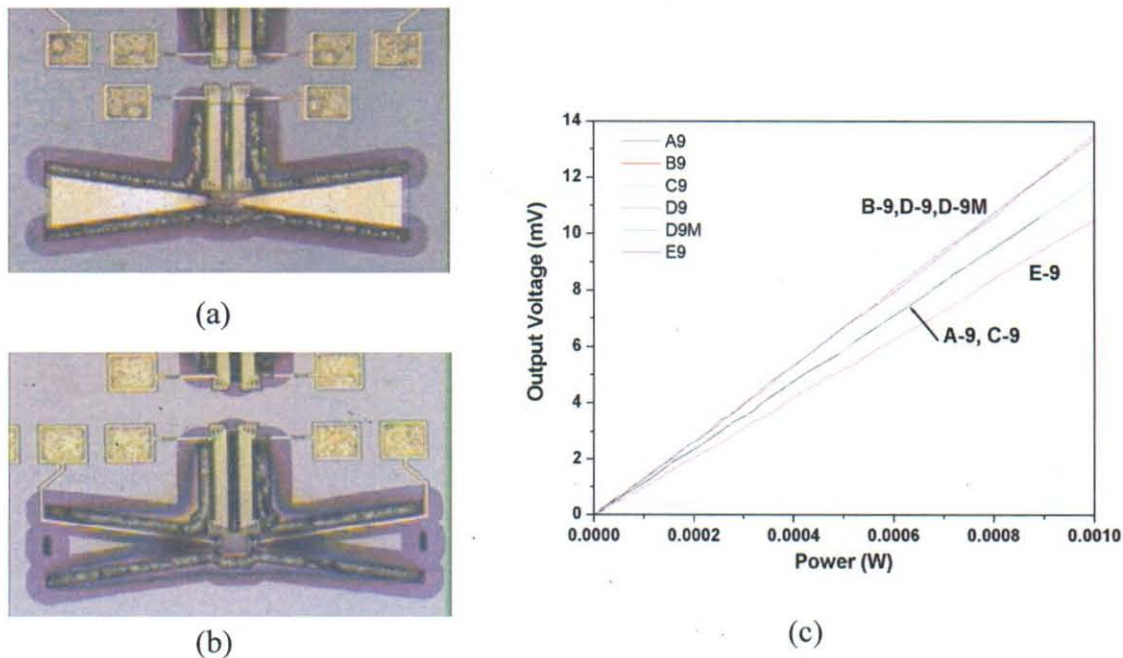


Figure 15: (a) 94 GHz Antenna Feeds Suspended Resistor (b) Thermopile Test Structure (c) Thermopile DC Voltage-Power Response

6.3 Micro-Robots

We envision the creation of autonomous robotic sensors for close-in and surface detection of physical and chemical phenomena. Our specific objective in this research was to provide proof of principle for three dimensional Silicon-On-Insulator (3D SOI) micro-spheres as a platform for autonomous sensors. Ultimately the devices would include a sensing mechanism, a means for autonomous motion, communication and control electronics, and a power source. With the help of advances that others are making in the theory and practice of robotics, these autonomous robots can act in ensembles, communicating with each other and with higher levels of the sensor network.

As they mature, the micro-sensors could detect a variety of phenomena including acoustic, electromagnetic activity; motion, and biological or chemical agents. Their sub-cubic-millimeter volume is significantly smaller than sensors in current practice. The small size means they can be deployed covertly or in highly confined or dangerous spaces such as building rubble. Since the sensors would be batch produced using semiconductor processing techniques, they should be very low cost.

The use of a large number of low cost, autonomous sensors acting as an ensemble is a significant change from the current practice, wherein a small number of sophisticated, high value sensors are used to detect or monitor phenomena. Although the current practice is effective and will

remain so in the future for many functions, the autonomous microsensor would provide an additional capability for covert, constricted, or dangerous environments.

Our research showed that we can use built-in layer stress to form micro-spherical shells of silicon sandwiched between oxide layers. We demonstrated a process for fabricating spheres and developed formulas for predicting the 3D shape based on the material properties. Fig. 16 shows scanning electron micrographs of spherical shells formed from silicon and oxide layers. The shell in Fig. 16a has a diameter of approximately 1 mm and was formed from a two dimensional flower pattern. A close up of a hemispherical structure showing the central hub is shown in Fig. 16b. Fig 16c shows a top view of an approximately 1.1 mm diameter shell formed with ribs to cover additional area. We demonstrated controlled motion of the spheres using a pattern of external electrodes excited by voltage pulses. Our analysis showed that we can incorporate electronics in the embedded silicon and supplement that with hybrid components enclosed within the 3D structure. Analysis further showed that we can provide and store enough power for sensing, computation, communication, and actuation, and that power dissipation will not have adverse thermal effects. We consulted experts in ensemble robotics who are convinced that sufficient computational capability can be incorporated in the 3D structure within the expected power and thermal budgets.

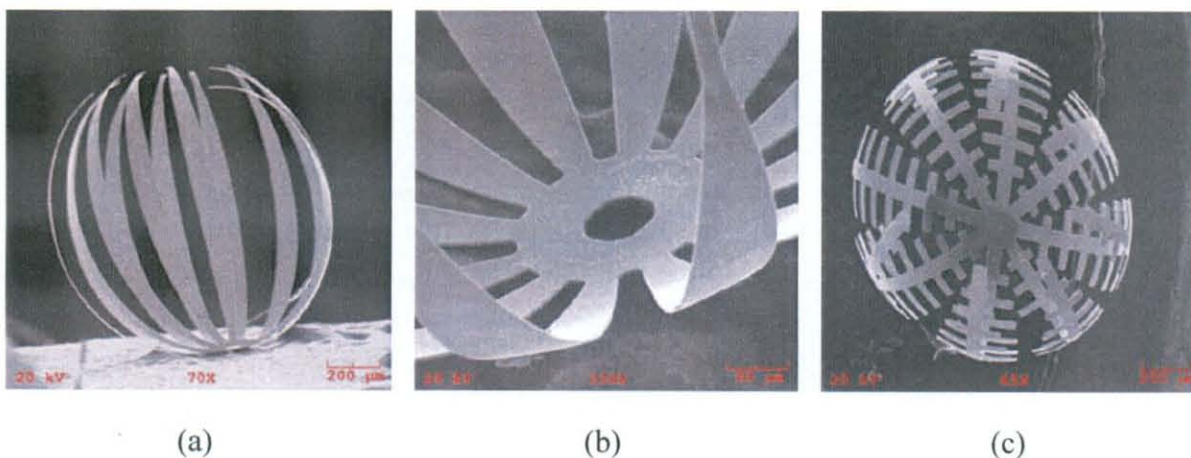


Figure 16: Scanning Electron Micrographs of Spherical Shells (a) Approx. 1 mm Diameter Shell (b) Close up of Hemispherical Structure (c) Approx. 1.1 mm Diameter Shell

The successful development of autonomous microsensors will provide a sensing capability that is not available today. An ensemble of small, low cost, expendable sensors that can be dispersed and maneuvered offers unique opportunity to increase situational awareness. The microsensors could give decision makers data on diverse phenomena such as the presence or movement of individuals or vehicles; the presence of explosive, chemical, or biological agents; the presence

and content of conversations or electromagnetic emissions. After our initial success with micro-robots, future work in this area will be performed under a separately funded research task.

7.0 CONCLUSIONS AND RECOMMENDATIONS

The broad objective of the Electromagnetic Component Research task was to improve our fundamental understanding of phenomena that enable or limit the future development of electromagnetic components for antenna applications. This objective was met through the investigation of a number of technical areas including semiconductor device physics, RF MEMS devices, 3D integrated circuits, multi-ferroic polymer materials for tunable RF components, nanowire formation, novel sensor components, and micro-robotics.

The research effort produced thirty-six publications, three patents, and large number of invited and contributed presentations. Several research areas have been transitioned to exploratory development, in particular, the MEMS and 3D circuit work.

Two areas have been recommended for continuation and will be funded under new research tasks. The *Ensemble Based Matter* task will explore the creation and physics of dynamic matter formed from an ensemble of coordinated of sub-cubic millimeter robots. *Antenna-Coupled Millimeter-Wave Detectors using Pyroelectric Thin Films* will research a sensor consisting of an antenna, resistive load, pyroelectric capacitor, and interconnects supported by a micromachined silicon dioxide membrane on a silicon substrate.

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APPENDIX -PUBLICATIONS, PATENTS, AND PRESENTATIONS

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LIST OF SYMBOLS, ABBREVIATIONS, AND ACRONYMS

2DEG	two dimensional electron gas
3D	Three Dimensional
Å	angstrom
α_n	impact ionization coefficient
C	Centigrade
cm	centimeter
CMOS	complementary metal oxide semiconductor
C-V	Capacitance-Voltage
dB	decibel
DC	Direct Current
dT/dt	time derivative of temperature
EPR	electron paramagnetic resonance
eV	electron volt
f _F	femtofarad
f _{MAX}	maximum oscillation frequency
f _T	current gain cutoff frequency
FTIR	Fourier Transform Infrared Spectroscopy
g _D	transconductance
GHz	gigahertz
HEMT	High Electron Mobility Transistor
Hz	hertz
I _D	drain current
I _{HOLE}	hole current
III-V	compound from columns three and five of the periodic table
IR	infrared
K	Kelvin

kG	kilogauss
kV	kilovolt
L	Length of high field region
M	meter
MEMS	Micro Electro Mechanical
MESFET	Metal Semiconductor Field Effect Transistor
MHEMT	Metamorphic High Electron Mobility Transistor
mm	millimeter
MMIC	monolithic microwave integrated circuit
mW	milliwatt
MΩ	megaohm
PVDF	poly vinylidene fluoride
RF	Radio Frequency
rms	root mean square
s	second
SEM/EDS	scanning electron microscope/electron diffraction spectroscopy
SOI	silicon on oxide
SPDT	single pole double throw
SPST	single pole single throw
UV	ultraviolet
V	Volts
V _D	drain voltage
V _{D,sat}	saturated drain voltage
V _{DC}	Voltage for direct current
V _G	gate voltage
V _{GD}	gate to drain voltage
Ω	ohm
μ	permeability
μm	micrometer